Manganese redox cycling drives the epitaxial growth of dolomite on metastable kutnahorite templates

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Abstract

Fine-crystalline, fabric-preserving dolostones in deep-time successions defy high-temperature

burial models, implying an elusive low-temperature formation pathway hindered by the

kinetic hydration barrier of Mg^{2+} and the thermodynamic miscibility gap separating calcite

from ordered dolomite. Here, we demonstrate a kinetically facile route to self-assembling

dolomite driven by the synergy of manganese redox cycling and carboxyl functionalization.

Using a bio-inspired electrochemical reactor, we show that electrochemical valence-state

modulation selectively regulates Mn^{2+} co-precipitation. Unlike inorganic controls where

manganese is rapidly sequestered into non-templating phases, the functionalized system

stabilizes reactive Mn(III) intermediates. This sustained redox cycling prevents irreversible

oxide immobilization and templates the rapid nucleation of spheroidal magnesian-

kutnohorite. Nanostructural characterization reveals a core-shell architecture where this

metastable, isostructural precursor serves as a lattice-distorted scaffold, enabling the

heteroepitaxial growth of substitutionally disordered dolomite cortices. Mechanistically,

localized acidity from redox cycling triggers a "proton-driven cation pump," actively

liberating Mg^{2+} from the functionalized hydrogel reservoir to the mineralization front. This

electrochemical mechanism offers a unifying geological model that links the massive fabric-

retentive dolostones of the Precambrian to ancient Mn-stratified oceans, while explaining the

Phanerozoic scarcity of dolomite as a consequence of global oxygenation decoupling the

manganese redox shuttle from shallow marine environments.

1. Introduction

The massive accumulation of fabric-retentive, fine-crystalline dolostones in the geological

record—most notably in the Precambrian but also in specific Phanerozoic intervals like the

Triassic—implies a low-temperature formation pathway mechanistically distinct from the

burial diagenetic models often applied to replacive coarse-crystalline dolostones (Hood et al.,

54   2011, Bontognali, 2019; Chang et al., 2020). While the kinetic inhibition exerted by Mg^{2+}

55   hydration is well-established (Lippmann, 1973), direct dolomite nucleation is further

56   hindered by the miscibility gap separating the stability fields of calcite and the ordered

57   dolomite-group carbonates (Goldsmith and Graf, 1960). Direct nucleation of these ordered

58   phases requires traversing this thermodynamic discontinuity, where metastable Ca-rich

59   phases are structurally distinct from the final double-carbonate product.

60   To bridge this gap, we propose a redox-driven solution to the long-standing "dolomite

61   problem." We hypothesize that the primary driver for syndepositional dolomite is an active

62   manganese redox engine. Unlike the Ca-Mg system, the Ca-Mg-Mn system offers a

63   continuous solid solution series (Goldsmith and Graf, 1960). Specifically, redox oscillations

64   can drive the nucleation of metastable kutnahorite (Mucci, 1988; Peacor et al., 1987). This

65   phase acts as a distorted nanocrystalline scaffold, where the larger Mn^{2+} radius expands the

66   lattice, significantly lowering the activation energy for subsequent Mg^{2+} incorporation and

67   driving the chemical evolution of the lattice across the miscibility gap (Han et al., 2024). This

68   recursive process drives a continuous out-of-equilibrium precipitation mechanism that

69   ultimately leads to the neomorphism of the metastable, isostructural precursor into dolomite.

70   Here, we validate this hypothesis using a bio-inspired electrochemical reactor that

71   mimics the redox-stratified interfaces of ancient microbial mats, where redox-reactive Mn

72   accumulates (Petrash et al., 2015). We demonstrate that inducing quasi-reversible

73   Mn(II)↔Mn(III) redox cycles on carboxyl-functionalized surfaces enhances Mg

74   incorporation relative to passive controls. Mechanistically, this enhancement effect functions

75   as a proton-driven cation pump. Localized acidification driven by Mn(II) oxidation "depins"

76   Mg^{2+} and Ca^{2+} complexed in the hydrogel, concentrating them at the mineralization front.

77   This active mobilization, buffered by proton-consuming Mn-reduction cycles, drives the

78   rapid nucleation of spheroidal magnesian-kutnahorite as metastable precursor, which serves

79 as an epitaxial scaffold, lowering the kinetic barrier for the subsequent crystallization of

80 disordered dolomite.

81   This electrochemical mechanism provides a framework for explaining the paragenesis

82 of delicate, fabric-retentive dolostones observed throughout the rock record. We propose that

83 this coupling, linking manganese availability to precursor dolomite precipitation, was a

84 fundamental driver for the massive, basin-wide fabric retentive dolomicrite facies

85 characteristic of the Precambrian. By identifying this catalytic engine, we can also explain the

86 distribution of massive dolomite in the Phanerozoic sedimentary rock record as a

87 consequence of the spatial (de)coupling of manganese redox cycling from shallow carbonate

88 platforms.

89 **2. Methods**

90 **2.1 Precipitation setup**

91 The experimental design comprises a H-cell electrochemical system (150 mL borosilicate glass

92 reaction vessels) connected to an automatic titrator (SI Analytics Titroline 7000). The

93 electrochemical system was filled with 125 mL of 80 mM $NaHCO_3$ electrolyte solution and it

94 cells were separated by an anion exchange membrane (Fumasep® FKE-50, Fumatech GmbH);

95 and were continuously stirred at 250 rpm via synchronized MIXDrive stirrers (2Mag). One cell

96 was open through a 0.2 μm cellulose acetate filter to maintain the dissolved inorganic carbon

97 of our electrolyte in equilibrium with the atmosphere at 25°C. The headspace of the other cell

98 was kept anoxic by a constant flow of argon, initiated 5 min before experiments. In the anoxic

99 cell, a graphite rod working electrode (WE, CP-2200, CP-Graphitprodukte GmbH) and an

100 Ag/AgCl reference electrode (sat. KCl, +197 mV *vs.* standard hydrogen electrode (SHE); RE-

101 1B, ALS Japan) were inserted. In the oxic cell, another graphite rod was used as the counter

102 electrode (CE). A schematic representation of the electrochemical system is shown in Figure

103 1.

104   To investigate the influence of carboxyl functionalization, to a set of experiments we added

105 to the electrolyte Na-alginate (300 mg·L^{-1}, Sigma-Aldrich)—a commercially available

106 compound composed exclusively of carboxyl functional groups. Alginate, a carboxyl

107 biomolecule, is key component of EPS in microbial mats colonizing hypersaline, dolomite-

108 forming settings (Bontognali et al., 2010). At the environmentally relevant pH range (i.e.,

109 >pKa) alginate is crosslinked by metal ions, forming a hydrogel (Petrash et al., 2011). This

110 property can render the resulting hydrogels conductive (e.g., Ji et al., 2022).

111   To isolate the effects of organic functionalization and electrochemical modulation on

112 mineral formation, four experimental conditions were tested: **(E$_{00}$)** control, lacking both

113 functionalization and electrochemical modulation, served as the baseline for surface

114 precipitation; **(E$_{10}$)** carboxyl-functionalized without electrochemistry, this condition isolated

115 the effect of the carboxyl groups on mineral formation in the absence of an applied potential;

116 **(E$_{01}$)** a non-carboxyl-functionalized sample was subjected to electrochemical modulation to

117 assess the role of redox cycling in the absence of the biopolymer; **(E$_{11}$)** a carboxyl-

118 functionalized surface with an applied electrochemical treatment to evaluate potential

119 synergistic or antagonistic interactions. All conditions were conducted in duplicate.

120 **2.2 Electrochemical precipitation protocol**

121 During each experimental run, an initial pH value of 8.5 ± 0.1 in the anodic cell was raised to

122 9.1 units after additions of 100 to 200 μL of 0.5 M NaOH solution as pre-titrant. Following

123 pre-titration, a cation-rich feedstock (0.6 M (Ca,Mg)Cl$_2$; Mg:Ca = 6:1) was dosed into the

124 anodic chamber at a rate of 1 mL min^{-1} for 20 minutes (20 mL total); the final working volume

125 of the H-cell setup was 145 mL. The electrolyte pH was maintained at 9.2 ± 0.4 by

126 automatically dosing the titrant into the anodic cell (1.2 to 3.0 mL in total) using the automatic

127 titrator in pH-stat mode. when the value dropped below the experimental pH-envelope.

128   After the first 5 min of feedstock dosing (*ca*. 5 mL), a single addition of $MnCl_2$ stock

129 solution was added to the anodic cell to achieve a concentration of 10 mM Mn. At this

130 nucleation onset (t = 5 min), the system had reached estimated concentrations of Mg $\approx$ 21 mM

131 and Ca $\approx$ 3.6 mM. This established a far-from-equilibrium state—supersaturated with respect

132 to Mg-Ca phases but undersaturated for Mn-precursors (Table S1)—effectively isolating Mn-

133 nucleation to the electrode interface. As feedstock dosing continued to completion (t = 20 min),

134 the background concentrations rose to Mg $\approx$ 86 mM and Ca $\approx$ 14 mM, sustaining the high

135 thermodynamic driving force. Deionized water (18.2 MΩ·cm^{-1}) and analytical-grade reagents

136 were used to prepare all solutions. Concurrently with reactant dosing, oscillations in the redox

137 potential of the electrolyte were induced by using a Dropsens µStat 8000 multichannel

138 potentiostat (Metrohm, Spain) in cyclic voltammetry mode (CV) for 28 min. The electric

139 potential was continuously swept between -0.1 and 0.5 V at a scan rate of 2 mV s^{-1}. This

140 environmentally relevant Eh window (e.g., Jørgensen et al., 1979; Di Loreto et al., 2021) targets

141 the Mn(II)/Mn(III) couple, driving oxidative cycling at the periphery of the working electrode

142 (WE) without inducing water hydrolysis or insoluble Mn(IV) oxide precipitation. It aligns with

143 the nitrogenous and manganous zone where incomplete manganese cycling is

144 thermodynamically dominant (Petrash et al., 2025). After titration, the electrolyte solution in

145 the anodic cell was left undisturbed for 24 h to permit stabilization of the surface precipitates.

146 Following this step, the WEs were retrieved, disassembled, and stored under anoxic conditions

147 until solid-phase characterization.

148 **2.3 Solution cation concentration determinations**

149 Samples were collected from the anodic chamber at three distinct experimental stages: **(S_i)**

150 immediately before the addition of 10 mM $MnCl_2$—when 5 mL of the $(CaMg)Cl_2$ reactant

151 (further referred here to as cation feedstock) had been added (i.e., $[Ca]_{added}$ = 3.6mM, $[Mg]_{added}$

152 = 21mM; Mn = 10 mM); **(S$_{ii}$)** following the complete dosing of 20 mL of the cation feedstock

153 (i.e., $[Ca]_{added}$ = 14mM, $[Mg]_{added}$ = 86 mM, $[Mn]_{added}$ = 10mM); and **(S$_{iii}$)** after the 24 h

154 stabilization period. The working solutions were acidified with trace grade HNO_3 (67% v/v,

155 Sigma-Aldrich). The cations crosslinking alginate are recovered after acidification, and the

156 concentrations of Ca^{2+}, Mg^{2+}, and Mn^{2+} relative to the total analyte added to the system are

157 determined using a triple quadrupole inductively coupled plasma mass spectrometer (ICP-MS;

158 Agilent 8800). Precision was verified against certified standards, and the relative standard

159 deviation was <10% of the reported value for all measurements.

160 **2.4 Multiscale microscopy and nanoscale characterization**

161 Electron-transparent lamellae of surface precipitates were prepared for (scanning) transmission

162 electron microscopy ((S)TEM) using a Helios Nanolab G3 (Thermo Fisher Scientific) focused

163 ion beam scanning electron microscope (FIB-SEM) at the University of Utrecht. This

164 instrument was also employed to assess the micrometre-scale morphological features of the

165 precipitates. Sub-micron scale imaging of lamellae, with thicknesses ranging from 250 to 300

166 nm, was conducted in bright-field (BF) TEM, dark-field (DF) TEM-EDS, and high-angle

167 annular dark field (HAADF) STEM modes. These were performed using either a Thermo

168 Fisher Talos F200X coupled to a Super-X detection system, or a Spectra 300 instrument,

169 operated at accelerating voltages of 200 kV and 300 kV, respectively. Selected area electron

170 diffraction (SAED) and electron energy loss spectroscopy (EELS) were further utilized in

171 multiple regions of interest to resolve crystallographic and compositional features of the

172 precipitates at the nanoscale. Dual-EELS datasets (core- and low-loss regimes) were acquired

173 using a CCD camera and DigiScan within the Gatan microscopy suite, with a dispersion of

174 0.15 eV/pixel and a convergence angle of 0 mrad. The low loss spectrum was used to align the

175 core region and evaluate the lamella thickness. To estimate the manganese valence state, we

176 employed a combined approach: semi-quantitative background-subtracted white line intensity

177 ratios (L_3/L_2) and the deep-learning spectral decomposition framework MnEdgeNet (Ji et al.,

178 2023). For the latter, spectra were analyzed using the noise-free model to strictly constrain

179 oxidation states based on peak position and spectral shape. Additionally, energy-dispersive X-

180 ray spectroscopy (EDS) was employed to map the spatial distribution of Mn, Ca, and Mg within

181 sacrificial areas of the precipitates.

3. Results

3.1 Response to titration and electrochemical treatments

184 Titration data (Fig. 2) show that control samples (E_{00}) required maximal external neutralization

185 due to unbuffered proton release during precipitation. Conversely, potential sweeping alone

186 (E_{01}) reduced titrant demand via cathodic proton consumption (Eq. 1). Conversely, potential

187 sweeping alone (E_{01}) reduced titrant demand via cathodic proton consumption (Eq. 1). In E_{01},

188 proton-coupled electron transfer partially offsets the release associated with carbonate

189 precipitation, explaining the reduced external titrant demand observed in the electrochemical

190 condition.

$$Mn^{III}(OH)_{org} + e^- + H+ \rightarrow Mn^{II}_{org} + H_2O \tag{1}$$
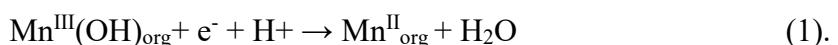
192 The carboxyl functionalization alone (E_{10}) provided intrinsic buffering capacity,

193 attributed to carboxyl groups (pKa ~3.5–4.5; e.g., Petrash et al., 2011). At the experimental

194 pH, the carboxyl-rich biopolymer forms a hydrogel upon deprotonation and in the presence of

195 cations. As carbonate mineral nucleates, it pulls CO_3^{2-} from the equilibrium, forcing the

196 deprotonation of HCO_3^-. This locally weakens the bond between metals, M, cross-linking the

197 hydrogel as the crystal growth front, releasing these metals back in solution right at the interface

198 of the growing mineral phase (2).

$$R\text{-}COO\text{-}M\text{-}OOC\text{-}R + H^+ \rightarrow R\text{-}COOH + M^{2+} + {}^-OOC\text{-}R \tag{2}$$

200   The combined system (E_{11}) achieved maximal pH buffering capacity by synergizing

201   active cathodic reduction with passive hydrogel buffering. This dual mechanism effectively

202   mitigated the acidification front driven by rapid carbonate nucleation, resulting in minimal

203   titrant consumption during cation feedstock addition (Fig. 2)

204   **3.2 Mn redox features developed during electrochemical carbonate synthesis**

205   CV was used to investigate the electrochemical behavior of Mn in the system. In experiments

206   lacking Na-alginate, the graphite electrode showed negligible faradaic activity within the

207   potential window (-0.1 V to +0.5 V vs. SHE), displaying a non-faradaic capacitive current

208   profile (Fig. 3A). Upon the introduction of Na-alginate, distinct quasi-reversible redox

209   features emerged (Fig. 3B). Accordingly, an anodic peak attributed to the oxidation of Mn(II)

210   → Mn(III) was observed at approximately +0.35 to +0.45 V, with a corresponding cathodic

211   peak for the reduction reaction appearing at ca. +0.30 V during the reverse scan (Fig. 3B).

212   This quasi-reversibility indicates that carboxyl functionalization does not merely lower the

213   activation energy but enables the electrochemical reaction. By acting as inner-sphere ligands,

214   the carboxyl groups near the surface of WEs could facilitate electron transfer that is otherwise

215   kinetically inhibited at the bare graphite interface. Therefore, the presence of the return

216   (cathodic) peak confirms that the surface ligand field stabilizes the highly reactive Mn(III)

217   intermediate.

218   To also monitor the progression of the surface-mediated reaction in functionalized

219   experiments, we recorded the temporal evolution of the current response. The system exhibited

220   reproducible periodic current oscillations, where each 4'-cycle (corresponding to a full

221   potential sweep) featuring sharp anodic and cathodic spikes (Fig. 3C). The pattern confirms

222   the sustained cyclic oxidation and reduction of manganese bound to the carboxylated hydrogel

223   during the active synthesis stage. The signal intensity displayed a progressive attenuation over

224   time (Fig. 3C), indicating that the electroactive surface area was being systematically

225   passivated by deposition of carbonate on the surface of the electrode, leading to fouling and

226   decreased anodic and cathodic capacity.

3.3. Aqueous composition of the reactive solutions

3.3.1. Magnesium removal: kinetic enhancement

229   The bulk of Mg removal initiated after 5 doses of the cation feedstock and the single Mn

230   addition. Therefore, S_i recorded [Mg] below certainty levels. By the end of the reactant

231   dosing and titration stage (S_{ii}), between $8.7 \pm 1.1\%$ to $28.0 \pm 1.0\%$ Mg^{2+} has been removed

232   from solution; enhanced by the electrochemical treatment (Fig 4A). Removal, however,

233   remained incomplete even after the stabilization time lag (S_{iii}), with 66% to 87% of the total

234   Mg^{2+} added to the system persisting in solution across the disparate conditions tested.

235   Electrochemistry alone (E_{01}) achieved the highest removal rates, surpassing the control (E_{00})

236   by approximately 20%. Electrochemistry with functionalization (E_{11}) was also effective.

237   However, the total amount of Mg co-precipitated in E_{11} was lower than in the functionalized-

238   only condition (E_{10}). Subsequent removal during stabilization was sluggish, increasing by

239   only 3% to 6% relative to the amount removed by the end of titration (Fig. 4A; Table 1).

3.3.2. Calcium removal: Calcium availability as a rate-limiting factor

241   Prior to the addition of Mn (S_i), approximately 40% of the total Ca had already been removed

242   from the solution (Fig. 4B). This initial depletion is attributed to a quasi-instantaneous

243   homogeneous nucleation of fine-grained suspended precipitate formed upon dosing the cation-

244   rich feedstock into the electrolyte. Similar rapidly formed metastable phases have been

245   characterized as amorphous (e.g., Purgstaller et al., 2016). Considering our Mg^{2+} removal

246   trends via co-precipitation (Fig. 4A), and the extremely low Mg-distribution coefficient

247   between amorphous calcium carbonates (ACC) and the mother solution (Purgstaller et al.,

248   2019), we deduce that the bulk of this solid residue may be initially an ACC precursor that

249   actively rejected Mg^{2+} during exsolution. Consistent with this assertion, XRD analysis of the

250   residual solids from experiment E_{11} identifies them as aragonite associated with secondary

251   hydrated Mg-carbonate, dypingite (Fig. S1). This assemblage directly reflects the bulk

252   saturation state of the residual solution, S_{iii} (SI_{Arag} = 2.41 and SI_{Hmag} = 5.01; Table S1). The

253   result is consistent with solid-state transformation of the initial ACC that segregated into

254   distinct Ca and hydrated-Mg phases upon aging and dehydration of S_{iii} at ambient temperature.

255   During the active synthesis stage (S_{ii}), Ca removal continued, reaching between 65–

256   84% across all experiments. Following stabilization (S_{iii}), most conditions exceeded 95%

257   removal, except for the functionalized, and electrochemically active experiment (E_{11}, ~75%),

258   where ligand complexation and interfacial electrochemical dynamics rendered it unavailable

259   for co-precipitation (Fig. 4B; Table 1).

260   *3.3.3. Manganese removal: Redox-driven sequestration*

261   Mn removal exhibited a distinct, redox-driven pattern compared to Mg. After 20 doses of the

262   Mg-Ca reactant (S_{ii}), removal of Mn varied across experimental conditions, and showed the

263   strongest kinetic response to electrochemical treatment. Accordingly, Mn removal ranged from

264     $10.2 \pm 7.2\%$ to $61.4 \pm 17.3\%$, with electrochemical sweeping without functionalization (E_{01})

265   yielding the highest scavenging efficiency, far exceeding both the only functionalized

266   experiment (E_{10}) and the control (E_{00}). Remarkably, after the 24 h equilibration (S_{iii}), Mn

267   depletion approached >99% in most conditions, indicating near-complete incorporation into

268   the solid phases. The significant exception occurred in E_{11} (electrochemical + functionalized),

269   where 25–38% of the total Mn remained in solution. This indicates that while the potential

270   sweep drives oxidation-reduction, some of the carboxyl ligands in E_{11} can stabilize a significant

271   fraction of the Mn ions available in the system (see Tordi et al., 2025), thus preventing the

272   almost complete sequestration observed in non-functionalized controls (Fig. 4C; Table 1).

3.4 Mineralogical and textural characterization of precipitates

273

274 Neither electrochemical cycling nor functionalization alone produced dolomite-
275 composition-like phases. Controls (E_{00}) yielded only minimal, sparse Mn-rich precipitates on
276 the WE (Fig. 5A). Functionalization without electrochemistry (E_{10}) increased precipitate
277 density, forming manganous Mg-Ca carbonate spheroids (Fig. 5B–C). TEM in various modes
278 (Fig 5D-F) showed that the crystallites comprising these fabrics grow outward from central
279 nucleation points, forming spherulitic structures with radially arranged crystallites.

280 Electrochemical cycling without functionalization (E_{01}) yielded disperse aggregates
281 resulting from the rapid coalescence (self-assembly) of multiple nanocrystalline domains. In
282 these experiments, a Mg-carbonate phase forming rosettes was observed (Fig. 6A). The delicate
283 texture—likely a hydrous phase—was unstable under the ion beam and could not be recovered
284 during FIB-lamella preparation (Fig. 6B). STEM-HAADF imaging reveals that the recovered
285 aggregates exhibited significant internal porosity ("hollow"), consistent with self-assembly
286 during growth (Fig. 6C). Chemically, these aggregates display a magnesian-calcian
287 rhodochrosite composition. It displays distinct zoning indicated by Z-contrast gradients
288 portraying denser Mn-enriched domains transitioning into Ca-Mg enriched inner domains, and
289 with Mg homogenously distributed along the aggregate rims (Fig. 6C–E). SAED ring
290 assignments generated a d_{104} spacing of ~2.84 Å, alongside d_{113} (2.39 Å) and d_{110} (2.16 Å)
291 reflections (Fig. 6F).

292 Condition E_{11} yielded dense carbonate coatings consisting of 1–3 μm polycrystalline
293 spheroidal aggregates (Fig. 7A–C). Internally, these aggregates are defined by the coalescence
294 of smaller (200–600 nm) spheroids with Mn-enriched, thread-like cores and variably Mg-
295 enriched cortices (Fig. 7C–D). TEM-EDS confirms preferential Mn partitioning in the cores,
296 with oscillatory zoning likely induced by potential swapping (Fig. 7E). To resolve the potential
297 for solid solution extending to the dolomite-kutnahorite series (*cf.* Peacor et al., 1987), we

298   plotted the EDS data on a ternary Ca-Mg-Mn diagram (Fig. 7H). The data reveal a continuous

299   kinetic trajectory where metastable magnesian-calcian rhodochrosite, formed in the cores,

300   transitioned into manganoan protodolomite cortices. Notably, these intermediate compositions

301   plot directly within the thermodynamic miscibility gap defined by Peacor et al. (1987),

302   indicating that rapid kinetic growth stabilized a continuous solid solution that reflects

303   disequilibrium crystallization. In contrast, the E_{01} as magnesian- rhodochrosite products cluster

304   distinctly at the magnesite- calcian rhodochrosite solvus (Fig. 7H); i.e., the E_{01} phase is

305   presumably stable.

306   Structurally, HR-TEM reveals that our two-phase manganoan carbonate spheroids

307   posses a mosaic texture composed of nanodomains separated by broad, diffuse low-angle

308   boundaries (Fig. 7F). FFT analysis (Fig. S3) confirms crystallographic misorientations (3–11°)

309   between these regions. Diffraction rings in the SAED patterns (Fig. 7G) confirm that E_{11}

310   yielded a polycrystalline, randomly oriented precipitate. Overall, the measured interplanar

311   distances of the cores are consistent with disordered kutnahorite, while the cortices exhibit

312   reflections consistent with a disordered dolomite with near-stoichiometric cation ratio (i.e., Ca

313   : (Mg + Mn) = 1) as per d_{104} $ca.$ 2.88 Å. Other reflections that can confidently assigned are d_{101}

314   (4.029 Å), d_{110} (2.404 Å), d_{113} (2.191 Å), and d_{116} (1.786 Å). The characteristic b-type

315   superstructure reflections (e.g., d_{015}, d_{021}, etc), which signify cation ordering, were unresolved.

316   **3.5 EELS of the cortical precipitates**

317   The results from EELS targeting the synthetic products from E_{11} are shown in Figure 8A-D.

318   The carbon K-edge spectra are characterized by sharp pre-edge peaks at 290.4 eV and broader

319   features at 300.9 eV, corresponding to the π* and σ* resonances of the C=O bond, respectively.

320   These transitions combined with corresponding O K-edges at ~531 eV, uniquely identify the

321   matrix as carbonate (Fig. 8A-B). The Ca $L_{3,2}$-edge (Fig. 8C) shows well-resolved spin-orbit

322   splitting (L_3 at 349 eV), confirming the integration of Ca into a crystal lattice. Most

323 significantly, the spectra capture characteristic Mn $L_{3,2}$-edge features in the 640 to 650 eV range

324 (Fig. 8D). Mn $L_{2,3}$ edge spectra ($N = 4$) from the core of a representative spheroid were analyzed

325 using the deep-learning framework MnEdgeNet (Ji et al., 2023). The Supplementary Data 1

326 (Fig. S4) contains the MnEdgeNet modelling results. The combined data exhibit L_3/L_2 white

327 line intensity ratio ≈ 2.88 with an L_3 absorption maximum located at 639.5–640.0 eV. The

328 applied MnPredictor decomposition was the 'noise-free' model, which prioritizes peak position

329 fidelity and spectral shape. It returned to an average oxidation state (AOS) of +2.03, indicating

330 ≥98.6% Mn(II) (Fig. 8D). This predominantly divalent state is physically confirmed by the

331 deep spectral valley between the L_3 and L_2 edges (arrow in Fig. 8D), which reaches baseline

332 intensity.

333 **4. Discussion**

334 We presented an electrochemical synthetic pathway that circumvents the kinetic barriers

335 inhibiting the incorporation of the Mg^{2+} ion in dolomite by leveraging the redox chemistry of

336 Mn on reactive functionalized surfaces. This result establishes a mechanistic link between

337 dolomite and manganese redox-cycling (Petrash et al, 2015), organic functionalization and

338 ionic lattice distortion (Han et al., 2024), and links the increasingly recognized role of electron

339 transfer mechanisms in marine sediments (Nielsen et al., 2010).

340 Cyclic voltammetry demonstrates that carboxyl functionalization modulates the

341 electrochemical behavior of manganese, facilitating quasi-reversible Mn(II) ↔Mn(III) valence

342 state transitions that are otherwise kinetically inhibited at the WE interface. Our central finding

343 is the formation of compositionally zoned spheroids, which featured Ca-Mn-enriched cores

344 and Mg-enriched cortices. The discussion below expands on observations, their interpretations

345 and implications.

346 **4.1 The Mn-templating mechanism**

347 Morphological and nanostructural characterization reveals that low-temperature formation

348 proceeds via a non-classical, multi-stage pathway involving metastable precursors. In the

349 solid-phase products of E_{11}, STEM-HAADF imaging, SAED, and EDS mapping identify Mn-

350 rich cores with minor Mg incorporation. These cores serve as epitaxial templates for the

351 subsequent overgrowth of a Mg-enriched cortex—a compositional evolution significantly

352 promoted by carboxyl functionalization and potential modulation. The composite E_{11}

353 spheroids have cores analogous to "pseudo-kutnahorite" (Mucci, 1988), and cortices that are

354 consistent with a manganoan (proto)dolomite precursor—given the absece of ordering

355 reflections.

356 It is worth noticing, however, that the absence of ordering reflections may be a

357 methodological artifact rather than intrinsic disorder of the products. Relative thickness

358 mapping $(t(\lambda)^{-1})$ derived from the low-loss EELS spectrum reveals that the dolomitic cortices

359 correspond to the thinner regions of the lamellae yet these exhibit local $t(\lambda)^{-1}$ values ranging

360 from 0.86 to 0.88 (Fig. S5). In this regime, approximately 59-64% of the incident electrons

361 undergo inelastic scattering (Egerton, 2011). This substantial background contribution,

362 combined with the polycrystalline nature of our synthetic products, likely lowers the signal-

363 to-noise ratio sufficiently to obscure weak superlattice reflections possibly associated with

364 incipient cation ordering. Consequently, while the current SAED data are consistent with a

365 disordered precursor, the sample thickness was not optimal for definitively resolving any

366 faint b-type reflections characteristic of substantial Ca occupancy in the B site. Attenuation of

367 b-type reflections may also be from manganese occupying a substantial portion of the

368 magnesium sites within this disordered lattice structure (Peacor et al, 1987).

369 Although kutnahorite is metastable at near-surface temperatures (Peacor et al., 1987;

370 Mucci, 2004; Böttcher and Dietzel, 2010), our results show that formation of this phase is

371 kinetically favored over disordered dolomite under functionalized conditions. While

372 thermodynamics alone dictates that dolomite is the more stable phase ($\Delta G^o_{f\_dol} < \Delta^o_{f\_Kutn}$;

373 Rosenberg and Foit, 1979), kinetic factors—specifically those governing the descent of the

374 free energy landscape, ultimately influence the reaction pathway. In this regard, despite the

375 bulk electrolyte being undersaturated with respect to kutnahorite (SI = -2.57; Table S1), the

376 out-of-equilibrium experiments implemented here indicate that this precursor presents a

377 significantly lower activation energy barrier to nucleation than direct precipitation of

378 oversaturated dolomite (SI = 5.54). This advantage is probably rooted in coordination

379 chemistry as Mn^{2+} has a lower magnitude of dehydration enthalpy compared to Mg^{2+} (-1851

380 vs. -1926 kJ mol^{-1}; Marcus, 1987), and its water exchange rate is nearly two orders of

381 magnitude faster (2.1×10^7 s^{-1} vs. 6.7×10^5 s^{-1}; Helm and Merbach, 2005). This allows Mn^{2+} ions

382 to desolvate and occupy the carbonate lattice rapidly, effectively bypassing the barrier that

383 stalls Mg incorporation.

384 Other factors may also be at play. The high lattice strain inherent to dolomite

385 (determined by its cationic mismatch ratio, $R_{Ca}/R_{Mg} \sim 1.38$) may contribute substantially to

386 the energetic barrier inhibiting its formation at low temperature. In contrast, within the

387 octahedral geometry, Mn^{2+} possesses an effective ionic radius ~15% larger than that of Mg^{2+}

388 (0.83 Å vs. 0.72 Å). This yields a significantly lower mismatch ratio for kutnahorite (R_{Ca}/R_{Mn}

389 ~ 1.20) and places the latter phase within a (meta)stability window where lattice strain is

390 sufficiently low to avoid the kinetic penalties that prevent dolomite nucleation/growth.

391 However, lattice-ordering in multicomponent phases is governed by factors beyond simple

392 cation radii (Jakob et al., 2025). In our system, cation-partitioning can also be critical. Mucci

393 (1988) and Böttcher (1998) demonstrated that the Mg distribution coefficient (D_{Mg} is notably

394 higher in Mn-bearing environments. As the [Mg]/([Mn]+[Ca]) ratio progressively rises

395 during precipitation, D_{Mg} increases, facilitating the incorporation of Mg^{2+} into the Mn-

396 expanded lattice of the kutnahoritic precursor. Being manganoan, this phase—exhibiting a

397   small lattice mismatch (<1.5%; a_{Kut} = 4.8732(8) Å, a_{Dol} = 4.8012(1) Å), still has some degree

398   of structural distortion (Rosenberg and Foit, 1979).

399   HRTEM imaging of E_{11} synthetic products reveals that they consist of aggregated

400   nanocrystals with incoherent orientations, possibly implying formation through an assembly

401   of nanoparticle subunits sharing a common crystallographic motif (Cölfen and Mann, 2003).

402   Such variations can force the coexistence of regions with disparate unit cell parameters that

403   may locally exhibit higher-order superstructures (Fang and Xu, 2019; Meister et al., 2023).

404   Forming as the crystal progresses towards a Mg-Ca stoichiometry that is thermodynamically

405   closer to equilibrium, the nanoscale misorientations accommodate lattice misfit (Van

406   Tendeloo et al., 1985). These circumvent the „step-pinning barrier" by stabilizing the strained

407   surface, thereby preventing rapid monolayer dissolution inherent to slow Mg-dehydration

408   (Higgins and Hu, 2005). In the absence of such a favourable isostructural template, the slow

409   dehydration of magnesium typically leads to strained initial surface layers that undergo rapid

410   monolayer dissolution before stable critical nuclei can grow. Our results demonstrate that the

411   isostructural Mn-rich core stabilizes these initial layers, preventing dissolution and enabling

412   the heteroepitaxial growth of the disordered dolomite cortices on kutnahoritic cores.

413   **4.2 Carboxyl biomolecules and redox potential oscillations**

414   The nucleation mechanism is governed by the synergy between organic functionalization and

415   the applied potential sweep. This mimics the interfacial processes active at the reactive

416   surface of sedimentary Mn-bearing phases that can be present in chemically stratified

417   dolomite-precipitating depositional systems (e.g., Petrash et al., 2015). The carboxyl acts not

418   merely as a passive template, but as an active agent that modulates, in its interaction with

419   reactive Mn, the free energy landscape, driving the distinct morphological and mineralogical

420   outcomes (Mercedes-Martín et al. 2016).

421   The interplay between the polymeric matrix and the electric field dictates the growth

422   habit. In the passive functionalized condition (E_{10}), precipitation follows a radial, isotropic

423   pathway. The presence of the biopolymer drives supersaturation and nucleation near the WE

424   surface occur stochastically from the initial seed. As described by Gránásy et al. (2005), such

425   'non-crystallographic branching disrupts the crystalline anisotropy early in the growth

426   process, leading to isotropic spherulitic growth.

427   In contrast, the synergy of carboxyl functionalization and electrochemistry in E_{11}

428   shifts the system toward the Category 2 growth mode of Gránásy et al. (2005), shifting the

429   reaction profile from themodynamic to kinetic driving forces (Cölfen and Mann, 2003). In

430   this condition, the polarized alginate chains likely establish a directional electric field along

431   the hydrogel, initially inducing a fibrillar habit by stabilizing metal ions via oriented

432   attachement (De Yoreo et al., 2015). However, the defining driver is the evolution of

433   supersaturation near the electrode interface. As the growing crystal entered a compositional

434   zone where lattice strain blocked classical step-flow mechanisms, the interfacial fluid built up

435   extreme supersaturation. To bypass this kinetic barrier, the crystal is forced to grow by

436   rapidly incorporating Mg-rich nanoparticle arrays that may result in oscillatory zoning. This

437   mechanism explains the continuous trajectory observed in the ternary plot (Fig. 7H), as rapid

438   kinetic attachment in an increasingly Mg-rich solution increasingly captures metastable

439   phases, effectively bridging the miscibility gap as the composition converges upon the

440   double-carbonate stability field.

441   *4.2.1 The electrochemical "engine"*

442   The reoxidation of aqueous Mn(II) requires an inner-sphere mechanism where water is

443   displaced by anionic ligands (Luther, 2005). Alginate carboxyl groups serve this function,

444   substituting for water molecules to enable electron transfer to the electroactive manganese

445   reactant (Fig. 9A). As the local activity of Mn^{2+} is progressively decreased, these same

446   ligands transition to binding the increasingly abundant Mg^{2+} ions (Fig. 9B). This evolution in

447   the mother solution is chemically coupled to the solid product via the pre-existing Mn-rich

448   isostructural surface. Driven by ionic radius compatibility and the distribution coefficient

449   (D_{Mg}) covariation (Mucci, 1988; Böttcher, 1998), the initial kutnohoritic core can serve as a

450   lattice bridge that facilitates heteroepitaxy.

451   This mechanism is fueled by localized interfacial acidification arising from two

452   synergistic sources: (i) carbonate precipitation, and (ii) the stoichiometric proton release

453   coupled to the incomplete anodic oxidation of Mn(II) (Fig. 3B). Estimates using a steady-

454   state Nernst approximation suggest this anodic flux induces a local acidification ($\Delta pH \sim$ -0.5

455   to -1.2 units) within the diffusion boundary layer at the WE surface (see Note S1 in the

456   Supplementary Data 1). This transiently protonates the carboxyl ligands ($pK_a \sim$4), triggering

457   the "depinning" of cross-linked cations, Mg^{2+}, Mn^{2+}, and Ca^{2+}, that become available for

458   incorporation into growing cortices (Fig. 9B). Transient accumulation of H^+ near the reactive

459   (WE) surface create oscillating supersaturation regimes required to stabilize pre-nucleation

460   clusters (De Yoreo et al., 2015) and generate oscillatory zoning inthe overgrowths (e.g.,

461   Katsikopoulos et al., 2009).

462   Finally, carboxyl functionalization alters the thermodynamic fate of the intermediate

463   Mn species. As delineated by the Eh-pH stability fields (Fig. S6), our experimental conditions

464   (-0.1 to +0.4 V. $pH_{surf} \pm$1.2 units) oscillate across the critical solubility threshold separating

465   hausmannite (Mn_3O_4) and manganite (MnOOH). The absence of organic ligands, crossing

466   this boundary, results in rapid Mn(III) disproportionation and irreversible precipitation

467   (Luther, 2005). However, as evidenced by cyclic voltammetry (E_{01}), the organic ligand-free

468   interface is kinetically inhibited, showing negligible faradaic activity. Substantial Mn

469   removal observed in this experiment (Table 1) thus acts as a control for readily occurring,

470   non-electrochemical Mn-carbonate precipitation. In contrast, functionalization in E_{11} enables

471 the quasi-reversible regeneration of Mn(III) in the system (Fig. 3), isolating a fraction of it

472 from precipitation. This paradoxically results in lower net Mn removal (Table 1), confirming

473 that the carboxyl ligand field induces kinetic stabilization of Mn(III) in the hydrogel and

474 progressive Mn(II) incorporation in carbonate. The ligands allow the electrode to reductively

475 cycle Mn(III), sustaining the redox engine (Fig. 9A-B), while avoiding the irreversible

476 elctoractive reactant loss observed in E_{01}. Thus, the organic matrix (alginate) transforms the

477 system from a passive chemical sink into an active electrochemical engine (Eq. 3):

478 $$[R\text{-}COO\text{-}Mn^{III}(OH)] + e^- + H^+ \rightarrow [R\text{-}COO\text{-}Mn^{II}] + H_2O \qquad (3)$$
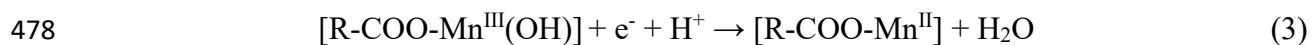
479 **4.3. Broader geo(bio)logical implications**

480 *4.3.1. Resolving the synsedimentary deep-time dolomite paradox*

481 In this section, we interpret the manganese-rich bands frequently observed in ancient

482 dolomite crystals not merely as diagenetic artifacts, but as a persistent chemo-textural

483 fingerprint of Mn-catalysis (Petrash et al., 2021). We propose that this catalytic mechanism,

484 by lowering the kinetic barrier for nucleation, acted as the fundamental driver that scaled up

485 to produce the massive, basin-wide, fabric-retentive dolostone facies characteristic of the

486 Precambrian. It likewise accounts for the cryptomicrobial dolomicrite facies that exhibit

487 punctuated abundance in the Phanerozoic, such as those deposited in the Western Tethyan

488 realm during the Triassic (e.g., Aljinović et al., 2025). With this crystal-to-facies link

489 established, the lithostratigraphic and sedimentological metadata for the dolostone deposits

490 considered in this section (listed in Note S2, Supplementary Data 1) can be understood as the

491 macroscopic record of this chemical coupling. For this purpose, we examine the temporal

492 correlation between massive dolostone intervals (Li et al., 2021, Shang, 2023) and pulses of

493 sedimentary manganese deposition (Spinks et al., 2022; Robbins et al., 2023; Fig. 10A).

494   The Mn-shuttle driven mechanism explored here would have operated efficiently

495   during the Paleoproterozoic (~2.3–2.05 Ga), establishing a link between primary kutnahorite

496   formation and massive fabric-retentive dolomite precipitation (e.g.,Préat et al., 2011; Nzamba

497   et al., 2026). This coupling persisted into the Mesoproterozoic, where dolomite abundance

498   exhibits a relative peak. Notably, during the Calymmian-Ectasian (1.6–1.2 Ga), a sedimentary

499   successions in the McArthur Basin host massive, fabric-retentive dolostones that preserve

500   early eukaryotic cells and their Mn-mineralized sheaths (Muir, 1983). This specific

501   association supports the hypothesis of Mn-catalysis as a critical microbial cell-fossilization

502   agent in ancient carbonate platforms (Bontognali, 2019). Although the Mesoproterozoic was

503   characterized by tectonic stability and expanding euxinia—conditions that possibly disrupted

504   the long-range transport of hydrothermal Mn(II) required for massive ore formation (Poulton

505   et al., 2010; Planavsky et al., 2014; Robbins et al., 2023)—newly quantified sedimentary Mn

506   data challenge the concept of a complete mechanistic hiatus in shallow marine Mn-cycling,

507   required for Mn-carbonate deposition (Spinks et al., 2022; Xu et al., 2025). Instead, the data

508   suggest that the requisite redox-stratified water columns were maintained locally, preventing

509   a total collapse of the Mn-shuttle dolomite coupling.

510   The dynamic re-emerged in the Neoproterozoic. High terrigenous Mn and silica

511   fluxes and benthic microbial primary productivity during the Tonian (Liang et al., 2025) and

512   Cryogenian–Ediacaran (Fang and Xu 2022; Cai et al., 2023) re-established conditions for

513   syndiagenetic, fabric retentive dolomite formation (e.g., Liang et al., 2025; Hood et al., 2011;

514   Stacey et al., 2023). Statistical analysis reveals positive correlation ($\rho = 0.50$) between

515   dolomite abundance and manganese ores in the Precambrian (Supplementary Data 2). While

516   statistical significance is inevitably constrained by the stochastic preservation of massive Mn

517   ores and the inherent discontinuity of ancient dolostone abundance determinations (Petrash et

518   al., 2017), this strong monotonic relationship supports that widespread redox stratification not

519   only favored Mn accumulation but simultaneously sustained the elusive seawater chemistry,

520   pervasive in the Proterozoic and recurrent in the Phanerozoic, that seems to be requisite for

521   the widespread deposition of syndiagenetic dolomicrite (Chang et al., 2020).

522   *4.3.2. Phanerozoic decoupling and restriction to continental basins*

523   The Phanerozoic record displays a contrasting negative correlation. ρ = -0.80,

524   marking a quantitative reversal in the relationship (Supplementary Data 2). This change

525   reflects the fundamental decoupling of these systems in oxygenated oceans and underscore

526   the shift in paragenesis of Mn deposits, as the drivers of massive manganese precipitation

527   (deep-ocean redox boundaries) became spatially and chemically separated from the loci of

528   shallow-marine dolomitization. In other words, as the Earth system transitioned toward a

529   fully oxygenated state (Fig. 10B), the ventilation of the deep ocean fundamentally disrupted

530   the shallow-water Mn-shuttle (Spinks et al., 2022; Chen et al., 2023; Robbins et al., 2023)

531   required for the ancient dolomite-forming engine proposed here.

532   As Mn-carbonates progressively gave way to Phanerozoic oxidized phases (Johnson

533   et al., 2016), dolomite formation became increasingly restricted. However, this long-term

534   decoupling was punctuated during intervals of Paleozoic and Mesozoic oceanic anoxia (Fig.

535   10B; Zhang et al., 2020; Li et al., 2021), when oceanographic conditions needed for shallow-

536   marine Mn-cycling were transiently re-established (Robbins et al., 2023). By the Neogene,

537   the Mn-templating mechanism for dolomite did not vanish but retreated, persisting in non-

538   marine settings such as rift lakes, where conditions analogue to the Mn-redox dynamics of

539   the Precambrian ocean existed (e.g., McCormack et al., 2024; Petrash et al., 2025).

540   Observations in post-Pliocene sabkha deposits (Chafetz et al., 1999; Bontognali et al., 2010;

541   Di Loreto et al., 2021) suggest that analogue (bio)geochemical conditions may also be locally

542   maintained in modern costal settings.

5. Conclusions and future work

We demonstrate that coupling manganese redox cycling with carboxyl functionalization unlocks a kinetically facile, non-classical pathway for the low-temperature formation of dolomite-group minerals. Our results show that electrochemical modulation accelerates Mn sequestration by over 60% compared to passive controls, stabilizing a metastable kutnahoritic precursor. This intermediate circumvents the high activation energy of Mg^{2+} dehydration, acting as a structural template that lowers the barrier for the heteroepitaxial nucleation of disordered dolomite in <24 h. The identification of mesocrystalline architectures, likely formed via crystallization by particle attachment, provides a physical explanation for the textural preservation of delicate features exceptionally observed in certain dolostones in the rock record. Unlike disruptive dissolution-reprecipitation models, this mechanism allows for compositional evolution while retaining primary aggregate morphology. Finally, our model offers a unifying explanation for the stratigraphic distribution of dolomite. We attribute the pervasive nature of Precambrian dolostones to the widespread availability of dissolved Mn(II) in redox-stratified ancient basins. Furthermore, we explain the distribution of massive dolomite in the Phanerozoic as a consequence of the spatial decoupling of manganese redox cycling from shallow carbonate platforms.

5.1. Future work

Moving forward, a significant technical hurdle remains the preparation of optimal FIB lamellae from these inherently friable, polycrystalline aggregates. While single-crystal diffraction would theoretically provide the definitive resolution for cation ordering, the pervasive polycrystalline nature of the spheroid precluded the isolation of discrete, defect-free domains large enough for standard tilt-series analysis. Consequently, future work must focus on optimizing ion polishing protocols—potentially identifying rare single-crystal domains—to achieve the ideal electron transparency ($(\lambda)^{-1} < 0.5$) required to resolve whether

568 faint superlattice reflections are present. Simultaneously, future research should prioritize

569 constraining specific isotopic fractionation factors (e.g., δ^{26}Mg, δ^{13}C-δ^{18}O, Δ_{47}-Δ_{48} REE

570 systematics) associated with this pathway, and with extended synthesis until electrode

571 passivation. To achieve this, experimental redesign can transition from closed-system batch

572 reactors to continuous-flow chemostat setups under controlled pCO$_2$ atmospheres.

573 Reengineering of the precipitation apparatus may allocate future work to systematically target

574 the saturation states characteristic of ancient 'greenhouse' intervals. By clamping reactant

575 activities (e.g., via external DIC buffering), it could be possible to effectively mitigate the

576 cation depletion artifacts (Rayleigh fractionation) driven by rapid precursor precipitation.

577 This steady-state approach would enable the precise determination of equilibrium distribution

578 coefficients (K_D), which otherwise remain elusive in the compositionally drifting regimes

579 typical of low-temperature dolomite synthesis.

580 **Data availability**

581 Data, including Python codes, are available through Zenodo at 10.5281/zenodo.18499776

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FIGURES
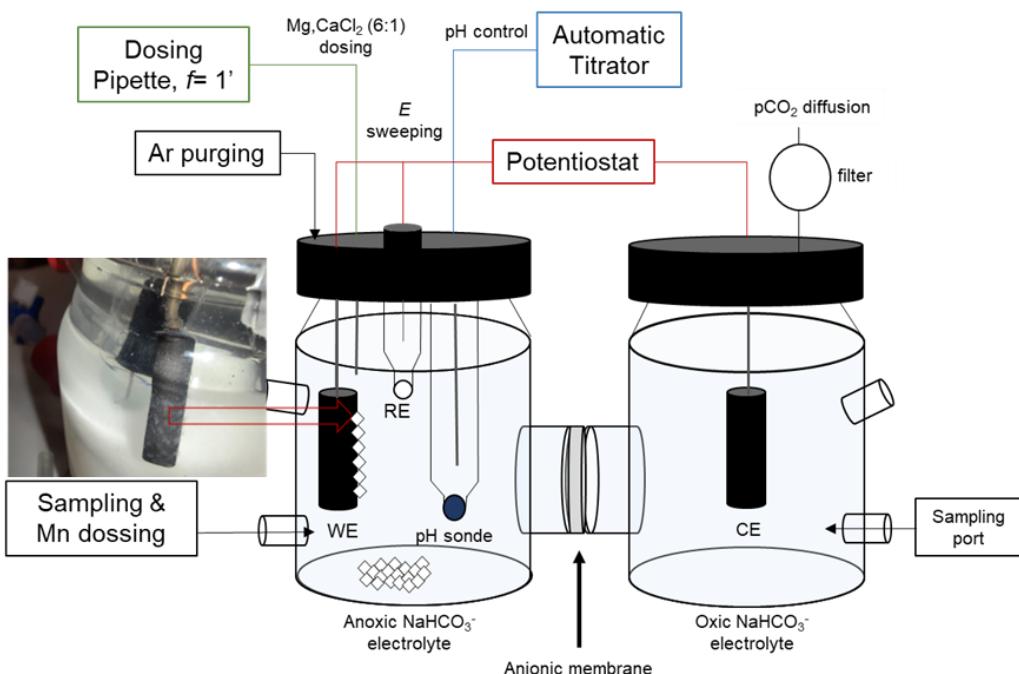
Figure 1. Schematic of the two-cell electrochemical precipitation apparatus. Two 150 mL borosilicate glass reaction vessels were connected via an anodic membrane. System hydrochemistry was controlled using an automatic pH-stat titrator and a potentiostat applying continuous potential sweeps. One cell, containing the working electrode (WE) and reference electrode (RE), was maintained under an Ar atmosphere, while the other, housing the counter electrode (CE), remained in equilibrium with air. Both WE and CE had surface areas of 4.2 cm^2. The electrolyte consisted of $NaHCO_3$ solution. Following pre-titration (see test for details), the cation feedstock was introduced at 1' intervals using a dosing pipette (n=20); a single $MnCl_2$ dose was at t ~ 5' . The inset shows WE and surface precipitates, with residual precipitates formed quasi-instantaneously in solution upon reactant additions. Attention of this study is placed on the surface precipitates.

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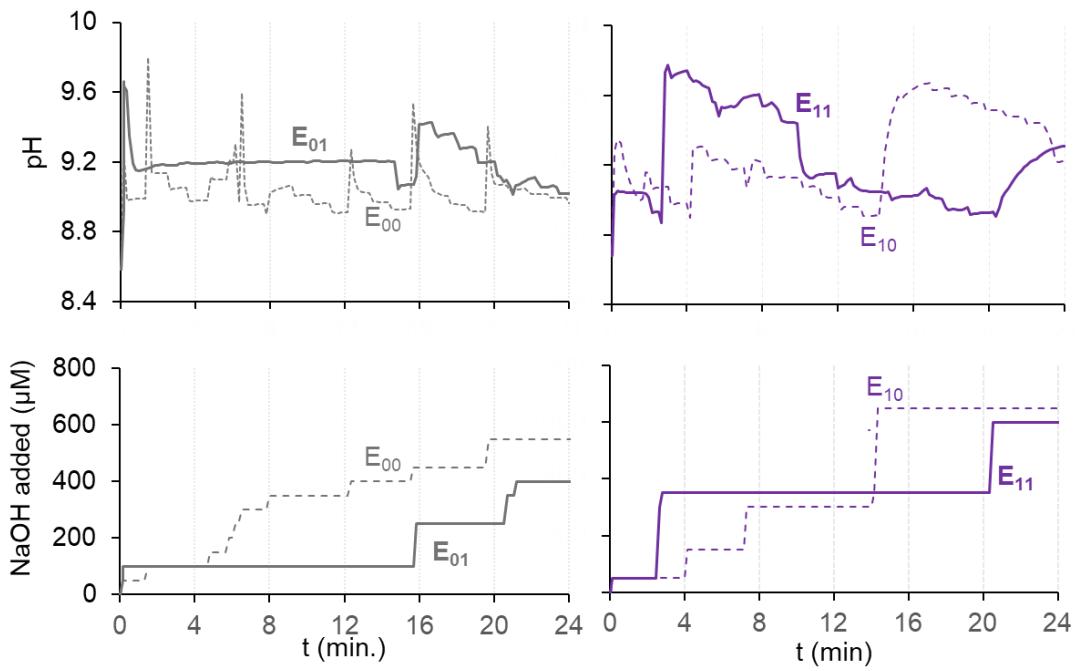
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792 **Figure 2**. Experimental pH and NaOH additions during carbonate synthesis under four
793 conditions: control (E_{00}), electrochemistry only (E_{01}), functionalized only (E_{10}), and
794 combined (E_{11}). Electrochemistry and functionalization reduced titrant demand, enhancing
795 buffering.

A

B

$[Mn^{II}] + H_2O \rightarrow [Mn^{III}(OH)] + H^+ + e^-$

$[Mn^{II}] + H_2O \leftarrow [Mn^{III}(OH)] + e^- + H^+$

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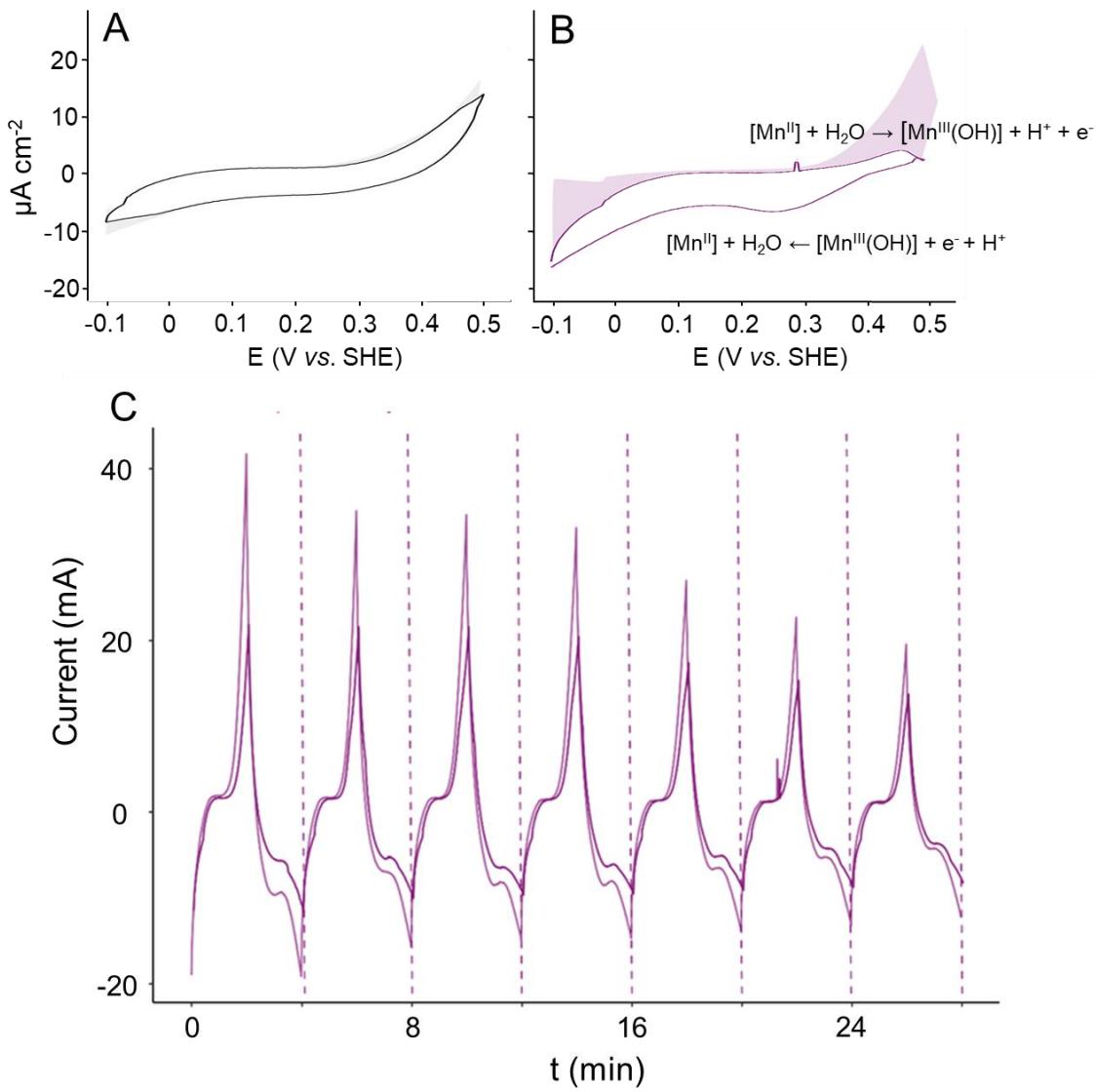
Figure 3. Electrochemical characterization of the Mn-functionalized system. Cyclic voltammograms (CV) recorded in carbonate-buffered electrolyte (E: [-0.1 to +0.5] V vs. SHE). (A) Control without functionalization exhibits a featureless profile characteristic of non-faradaic capacitive charging. (B) Carboxyl-functionalized hydrogel displays distinct quasi-reversible peaks, confirming active Mn redox cycling. Annotated equations detail the proposed proton-coupled electron transfer (PCET) mechanism: anodic oxidation coupled to hydrolysis (proton release) and cathodic reduction coupled to proton consumption. Solid lines denote average current density (j); shaded areas indicate standard deviation. (C) Temporal evolution of the current response (I) over 28 minutes in the functionalized system (E_{11}). Sharp peaks align with the 4-minute potential sweep periodicity (dashed lines). The progressive attenuation of peak intensity indicates gradual electrode passivation, consistent with the formation of non-conductive carbonate precipitate.
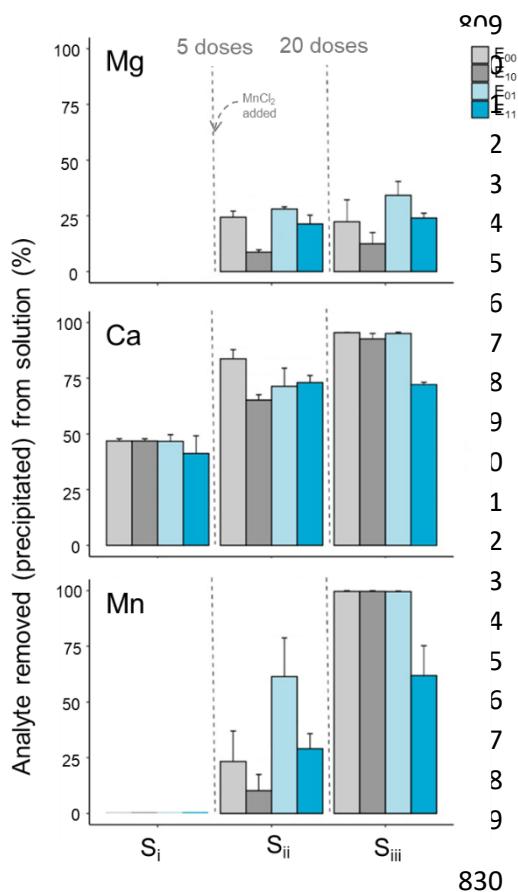
Figure 4. Percentage of Mg, Ca, and Mn removed (precipitated) from solution under four experimental conditions: without carboxyl groups (light grey), with carboxyl groups (dark grey), and their respective counterparts with electrochemical potential sweeping (light and dark blue). Samples were collected at three time points. These are S_i: 5' after start, with 5 doses of 1 mL $(Ca,Mg)Cl_2$ solution added to the system, and before the 10 mM $MnCl_2$ addition; S_{ii}: after 20', i.e., 20 doses of 1 mL of $(Ca,Mg)Cl_2$ solution; and S_{iii}: after the 24 h stabilization time lag. Consumption was calculated as the percentage difference between measured concentrations and maximum expected concentrations, i.e. @5' $\{[Mg]_{max}= 21$ mM, $[Ca]_{max}= 3.6$ mM, $[Mn]_{max}= 0.0$ mM$\}$; @20' and 24 h:$\{[Mg]_{max}= 86$ mM, $[Ca]_{max}= 14$ mM, $[Mn]_{max}= 10$ mM$\}$. Bars represent mean duplicate values, and whiskers indicate standard deviation based on duplicate experiments.
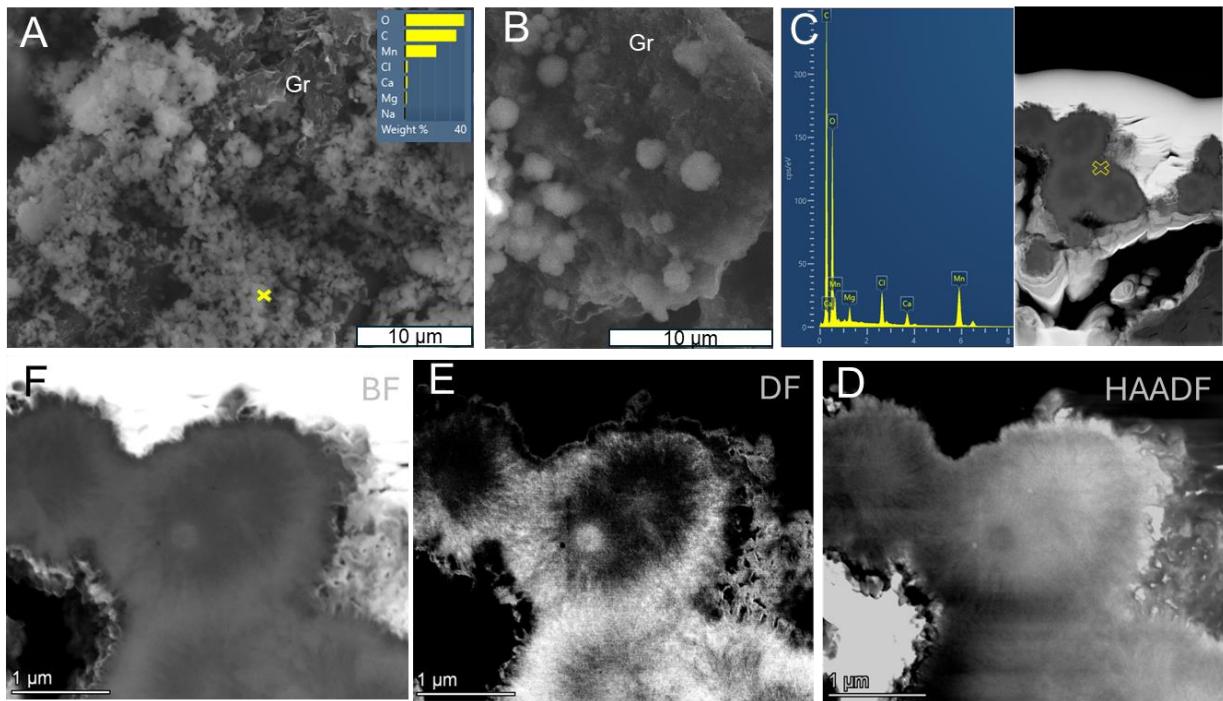
Figure 5. Solid phase characterization of carbonate precipitates in Control (E_{00}) vs. Functionalized (E_{10}) conditions. (A) SEM-EDS characterization of the control sample (E_{00}), where the absence of functionalization and cycling resulted in sparse, Mn-dominated precipitates (Mn >> Ca + Mg) with minimal surface accumulation. (B) SEM image of sample E_{10} (functionalization without electrochemical cycling), showing increased precipitate density and the formation of coherent carbonate spheroids on the graphite (Gr) substrate. (C) EDS spectrum of an E_{10} spheroid, confirming the phase as a manganous Mg-Ca carbonate. (D–F) Cross-sectional TEM analysis (HAADF, Dark Field, and Bright Field) of the E_{10} spheroids, revealing a spherulitic internal fabric composed of crystallites radiating outward from central nucleation centers.

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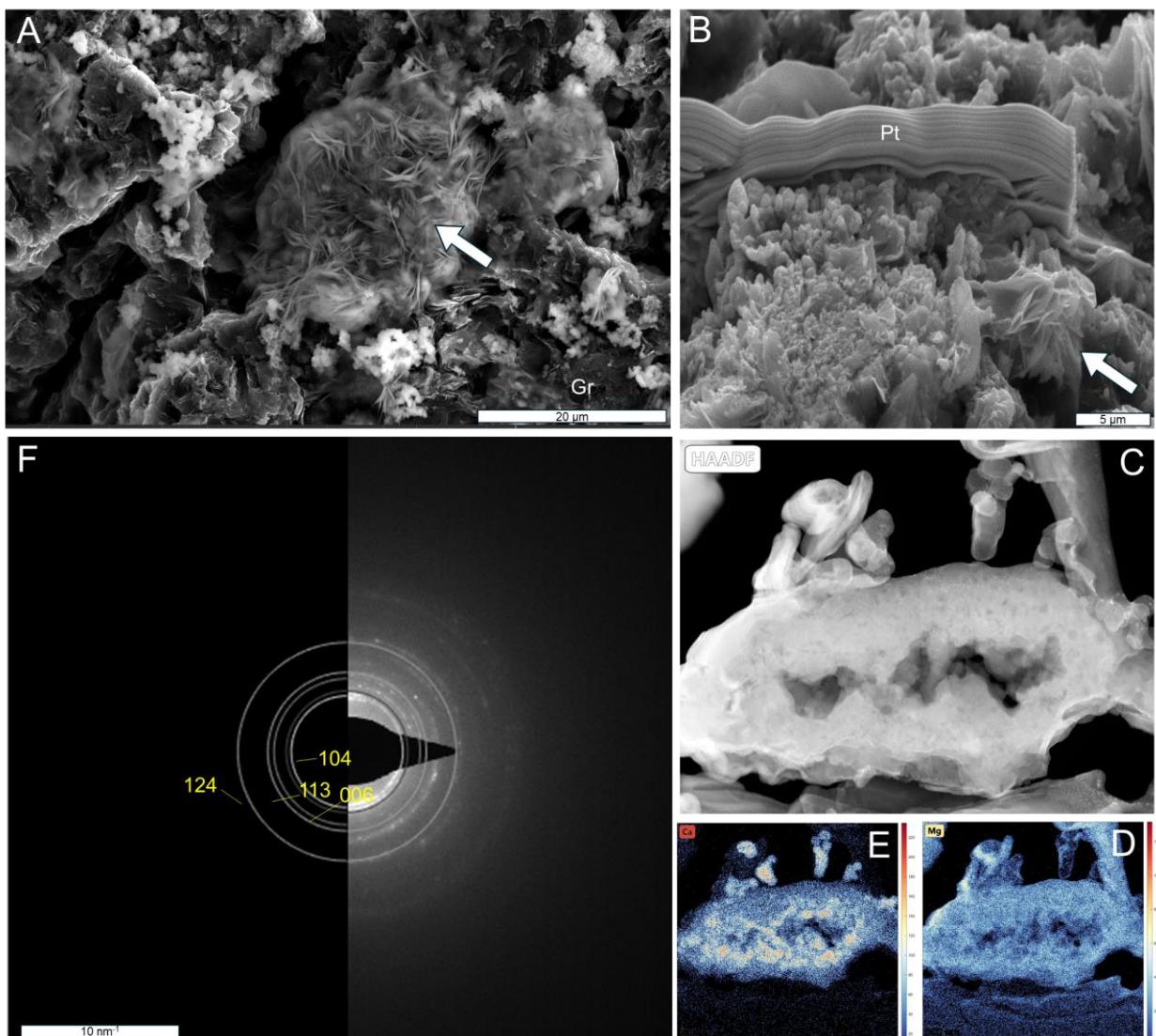
Figure 6. Morphology and structure of precipitates formed under electrochemical cycling without functionalization (E_{01}). (A) SEM of the electrode surface showing prominent rosette-like structures (arrow) intermixed with disperse, granular polycrystalline aggregates (B) FIB-milling site; the delicate rosettes (arrow) were unstable under the ion beam, leaving only the granular aggregates for analysis (C) STEM-HAADF cross-section revealing a porous, "vuggy" texture formed by coalesced nanocrystalline domains. Contrast variations highlight chemical zonation (inset mean EDS spectrum). (D–E) EDS maps contrasting the internal zonation of Ca with the homogeneously distributed Mg. (F) SAED pattern showing diffraction rings consistent with cation-substituted rhodochrosite lattice spacings.
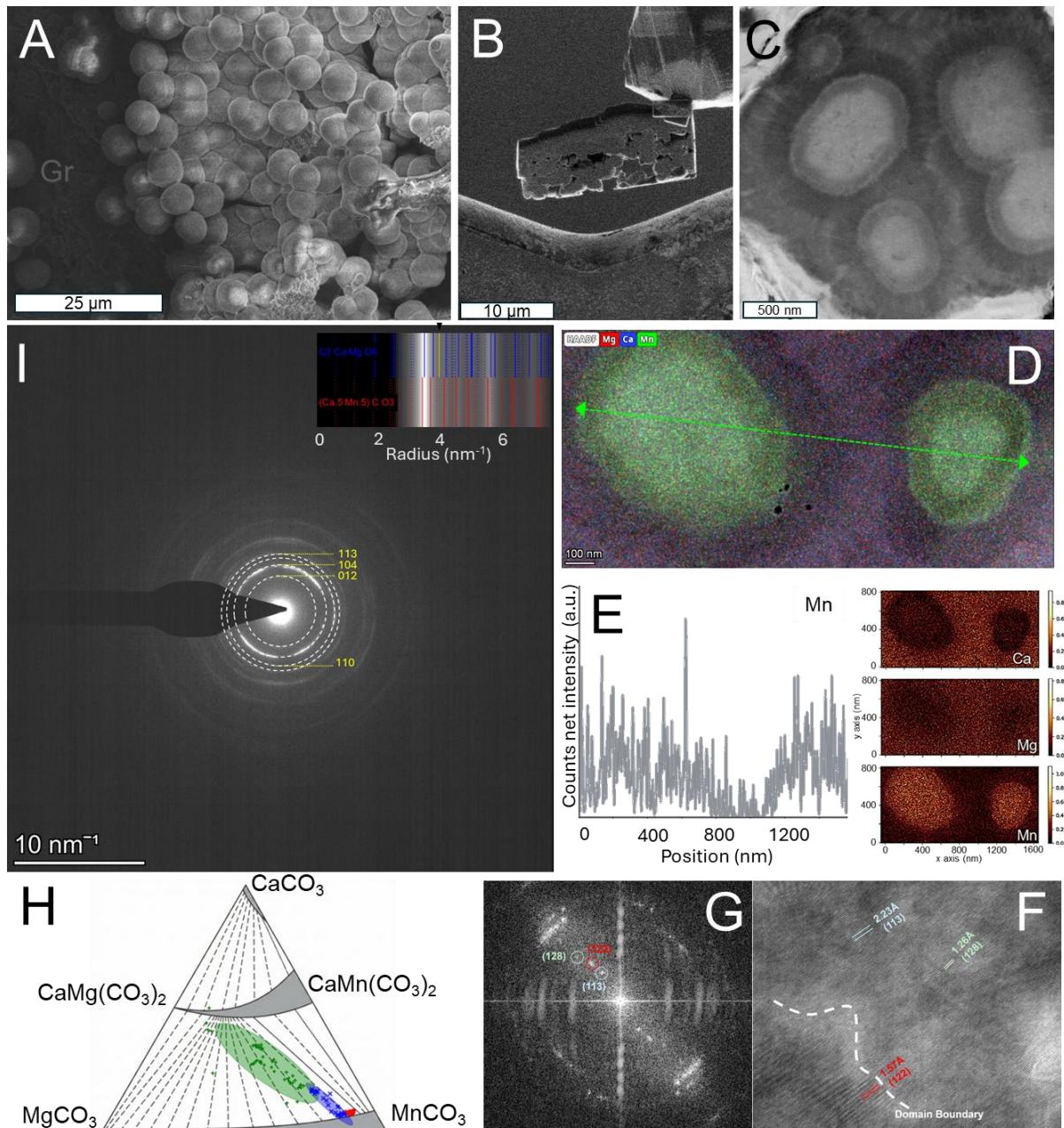
Figure 7. Multiscale characterization of Mn-templated protodolomite formed under electrochemical and carboxyl modulation (E_{11}). **(A–C) Morphology and texture: (A)** SEM micrograph showing the dense accumulation of spheroidal aggregates on the graphite electrode (Gr). **(B)** SEM view of a FIB-milled cross-section (lamella) extracted from the aggregate cluster. **(C)** STEM-HAADF image of the lamella, revealing the characteristic "eye-like" core-shell zonation and diverging sheaf texture. **(D–I) Nanostructure and chemistry**: **(D)** STEM-EDS composite map confirming cation partitioning: Mn-enriched cores (green) surrounded by Mg/Ca-rich cortices (red/blue). **(E)** EDS line-scan intensity profile (1.6 µm) for Mn across the aggregate (transect in D), showing the preferential sequestration of Mn in the core regions; the segregated distributions of Ca, Mg, and Fe are also shown. **(F)** High-resolution TEM (HRTEM) micrograph of a crystalline domain. The lattice fringes and the diffuse domain boundary (white dashed line) highlight the mosaic nanostructure, showing nanodomains (<30 nm) intergrowth together with different crystallographic orientations. **(G)**

867   Fast Fourier-Transform (FFT) of the HRTEM image **(H)** Compositional TEM-EDS data

868   projected onto the $CaCO_3$-$MgCO_3$-$MnCO_3$ ternary plot, the low temperature two-phase

869   manganoan carbonate stability fields/miscibility gap are shown (after Peacor et al.,1987).

870   EDS data renormalized to [Ca] + [Mg] + [Fe] = 1. The confidence ellipses are calculated

871   from the variance (scatter) of the data points. **(I)** SAED pattern acquired from the cortex with

872   identified diffraction rings. The inset/overlay displays the theoretical ring positions for

873   dolomite and kutnohorite, highlighting the expected location of the d_{015} superstructure

874   reflection. This reflection may be highly attenuated by broadening and superposition of d_{104}

875   peaks in polycrystalline matrices comprised of both minerals (see text for details).

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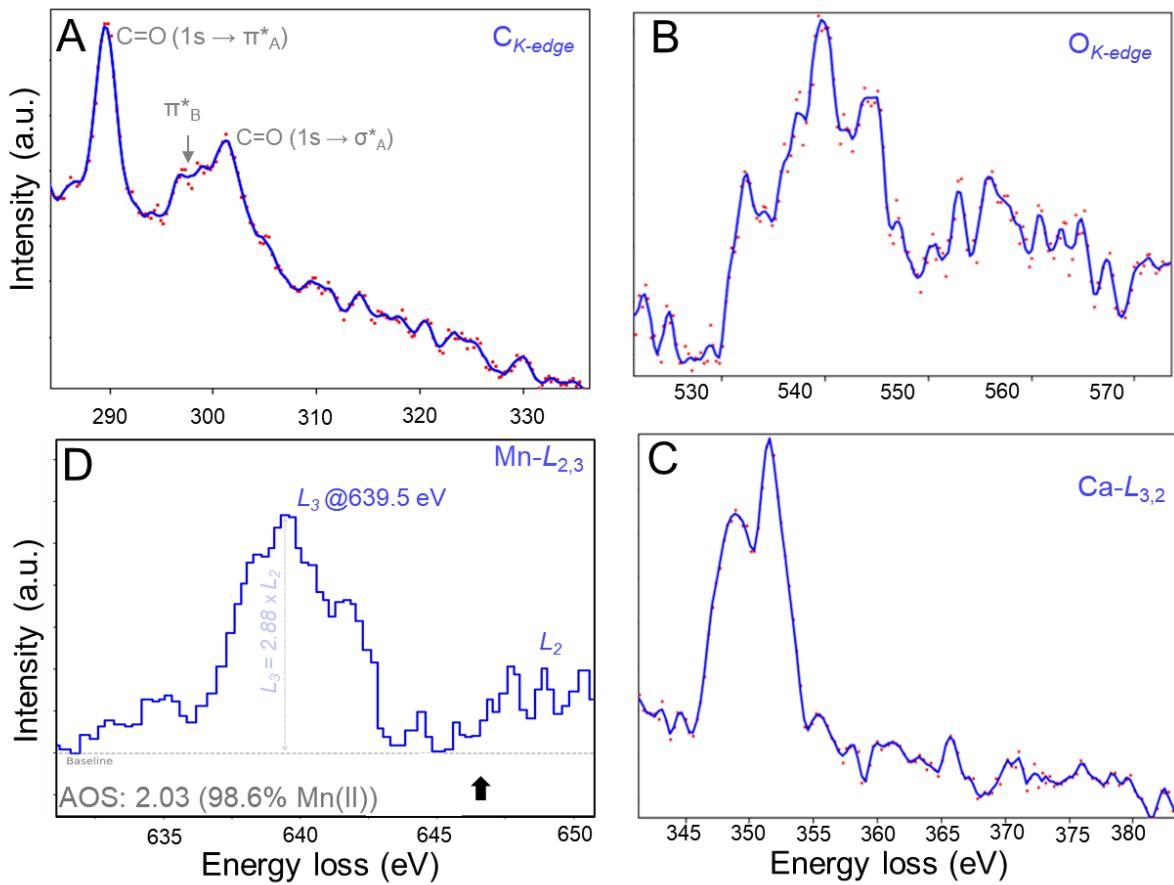
Figure 8. Representative core-loss EELS spectra acquired from the inner cortex region. (A) Carbon K-edge showing the characteristic π∗ and σ∗ transitions of the carbonate. (B) Oxygen K-edge exhibiting fine structure consistent with carbonate bonding. (C) Calcium $L_{3,2}$ -edge displaying sharp white lines indicative of a crystalline coordination environment. (D) Manganese $L_{2,3}$ edge spectrum. The data is presented as a raw step plot to preserve peak shape fidelity. The analysis reveals a sharp L_3 maximum at 639.5 eV and an intensity ratio of $L_3/L_2 \approx 2.88$, consistent with high-spin Mn(II). A key feature is the complete return to baseline intensity in the inter-peak valley (~645 eV, black arrow), effectively ruling out significant mixed-valence contributions.

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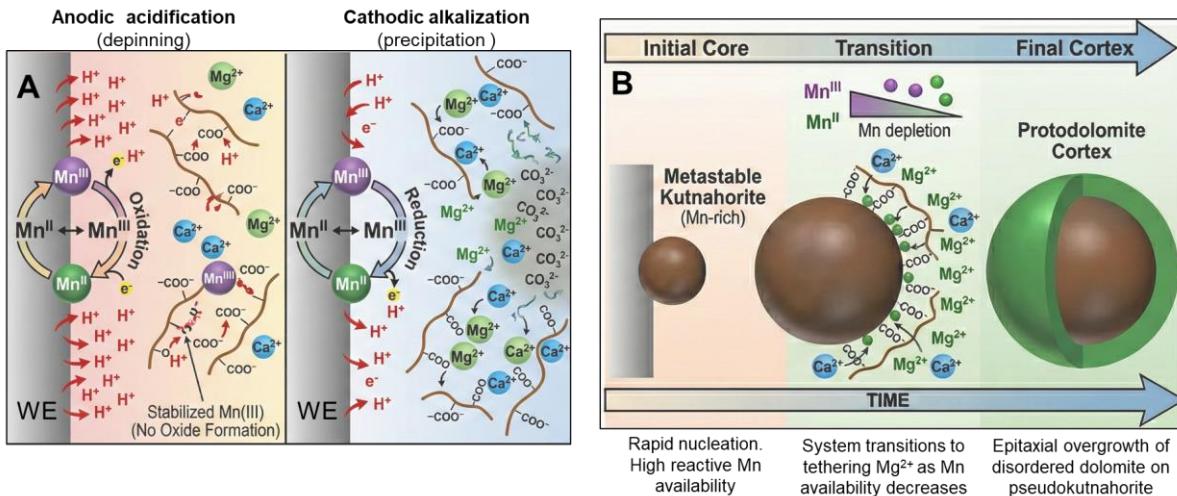
Fig. 9. Proposed mechanism of electrochemically induced dolomite growth. (A) The proton-coupled redox engine. At the working-electrode interface, anodic oxidation of Mn and concurrent water splitting generate Mn and protons (H^+). This drives anodic acidification, in the vicinity of the electrode which transiently protonates alginate ligands and causes the depinning of bound cations (Mg^{2+}, Ca^{2+}). The subsequent cathodic alkalization consumes protons, reactivating the ligands to bind cations and producing a local supersaturation spike. The carboxyl ligands form stabilized Mn complexes, acting as a kinetic brake that prevents oxide formation and sustains the quasi-reversible cycle. (B) Core-to-cortex evolution. Initially high concentrations of reactive Mn drive the rapid nucleation of a pseudo-kutnahorite template. As aqueous Mn availability progressively decreases (see transition wedge), the pumping mechanism shifts toward tethering the increasingly abundant Mg ions. These are delivered to the growth front, facilitating the epitaxial overgrowth of a protodolomite cortex.
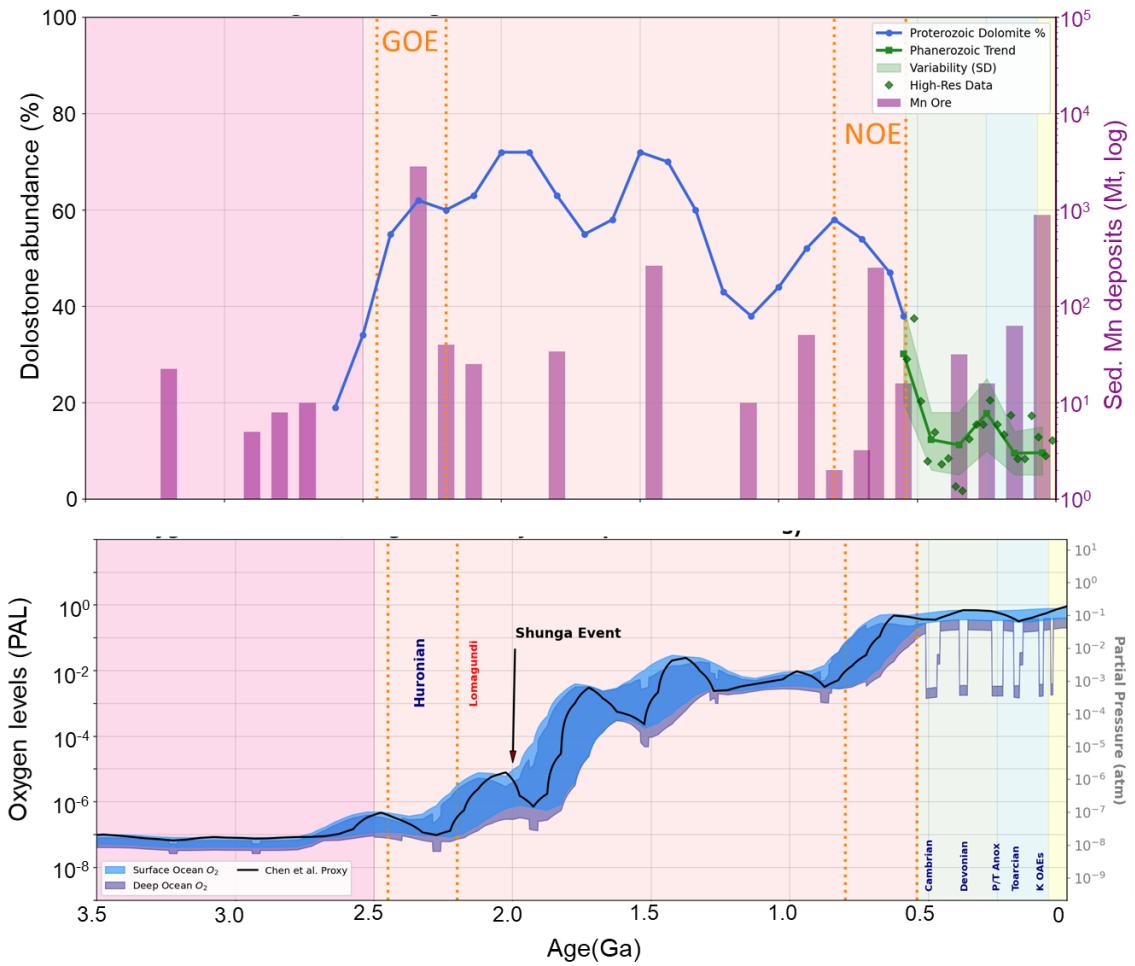
Figure. 10 Integrated evolution of global ocean redox conditions, carbonate mineralogy, and manganese deposition. (A) Secular trends in dolomite abundance (blue: Proterozoic; green: Phanerozoic mean ± SD; derived from Shang et al., 2023; Li et al., 2021) plotted against Mn ore deposits (purple bars; Robbins et al., 2023). Note the strong co-occurrence of dolomite peaks with Mn pulses in the Precambrian. (B) Evolution of oxygenation states (pO_2). Surface envelopes are calibrated to Chen et al. (2022). The deep ocean tracks surface conditions in the Mesoproterozoic but decouples during high-amplitude redox fluctuations: (i) the Great Oxidation Event (GOE) and Huronian glaciations; (ii) the Lomagundi-Jatuli overshoot; (iii) the Shunga Event oxygen crash; and (iv) the Neoproterozoic Oxygenation Event (NOE). In the Phanerozoic, the deep-water "Mn-pump" is largely suppressed, re-emerging only during transient Ocean Anoxic Events (e.g., P-Tr, Toarcian, K-OAEs) where deep waters dropped to <1% surface saturation.

TABLE

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919   **Table S1. Summary of aqueous geochemical evolution and cation removal efficiencies.**

Parameter	Initial Phase (Doses 1–5)	End of titration (20 doses)[†][‡]	Stabilization 24h[†]	Primary controlling factors
NaOH Titrant	**Delayed onset:** Lag in consumption (0–15 min) due to proton consumption from Mn(III) reduction	**Linear increase:** Steady demand to buffer proton release from precipitation	**Converged:** Total volume ~ 0.50–0.60 mL (0.25–0.30 mmol-eq OH^-) across all experiments.	Proton consumption by electrochemical Mn-reduction vs. alkalinity demand from carbonate equilibrium ($HCO_3^- \rightarrow CO_3^{2-} + H^+$)
Mg removal	**Negligible:** No significant uptake observed.	**Variable:** 8.7 ± 1.1 % (E_{10}) to $28.0 \pm 1.0\%$ (E_{01}). Highest in electrochemical setups, but reduced by carboxyl ($E_{11} < E_{01}$)	**Incomplete:** Slight increase (3-6%). Majority (66-87%) persists in solution.	Electrochemical treatment enhances Mg co-precipitation, but carboxyl functionalization partially mitigates this effect
Ca removal	**Rapid:** >40% removal prior to Mn addition (ACC formation)	**High:** $65.2 \pm 2.4\%$ to $83.7 \pm 4.1\%$ removal	**Near complete:** >95% removal, except E_{11} (~75%)	Initial homogeneous nucleation of ACC ; partially inhibited by ligand stabilization in E_{11}.
Mn removal	**N/A:** Mn added after Dose 5.	**Kinetic Response:** $E_{01} \gg E_{00}$, w./ E_{01} (61.4%) > E_{11} (29.0%) > E_{00} (23.3%) > E_{10} (10.2%)	**Complete vs. stabilized:** >99% removal in most units. Exception: E_{11} retains 25-38% in solution	Applied potential drives rapid sequestration; alginate in E_{11} stabilizes Mn(III) intermediates against precipitation

[†] Conditions denote E_{00} (control), E_{01} (electrochemistry only), E_{10} (functionalized only), and E_{11} (combined).

[‡] Removal (%)calculated relative to the total molar mass added to the H-cell reactor.

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924   **SUPPLEMENTARY MATERIAL**

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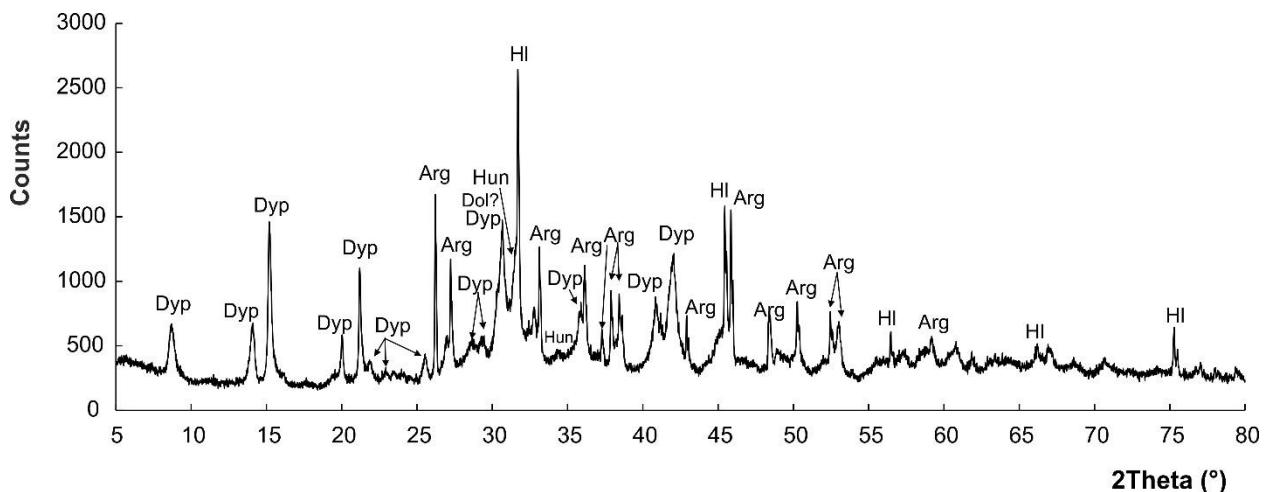
Fig. S1. 1-D X-ray diffraction pattern of the residual solid phase. The mineral
assemblage was analyzed after complete evaporation of the parental H-cell solution at 25 °C.
Identified phases include aragonite (Arg), dypingite, $Mg_5(CO_3)_4(OH)_2 \cdot 5H_2O$ (Dyp), and
halite. (Dyp) and halite.

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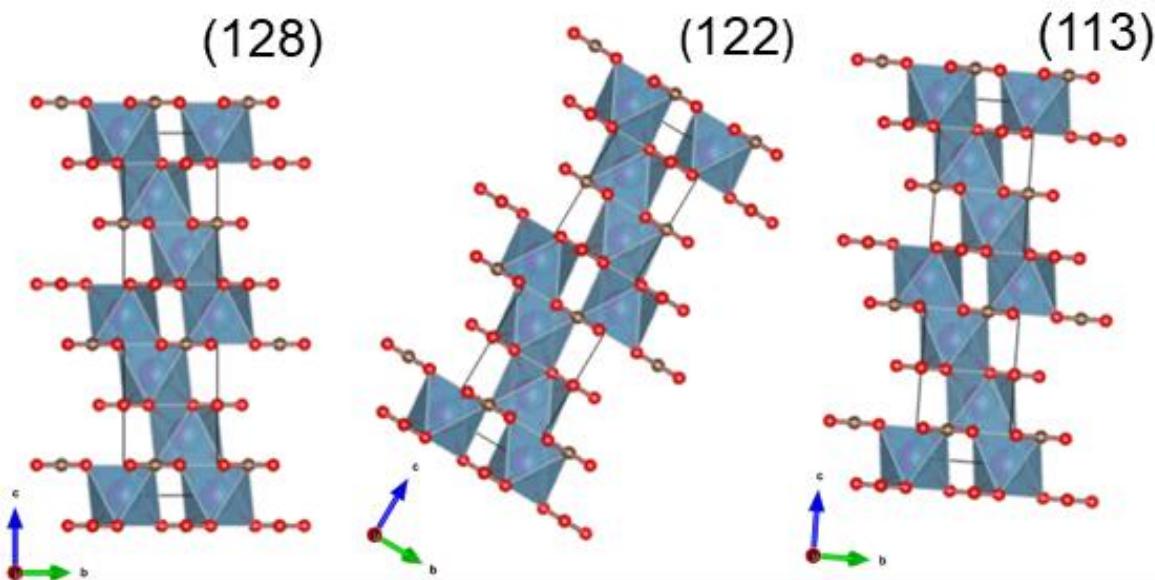
**Figure S2 | Fast Fourier Transform (FFT) generated from the HRTEM micrograph in
Figure 7F.** The pattern displays discrete diffraction spots corresponding to the (128), (122),
and (113) lattice planes. The characteristic arcing of these spots highlights the rotational
mismatch between adjacent nanodomains, quantifying the crystallographic misorientations
(~3 to 11°) associated with the mosaic fabric.

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Figure S3. Selected-area electron diffraction (SAED) pattern (inset) acquired from the core of one of the spheroidal features resulting from E_{11}. The diffraction geometry is consistent with kutnaharite, based on estimation analyses using AMCSD entry 0001070 (after Peacor et al. 1987).

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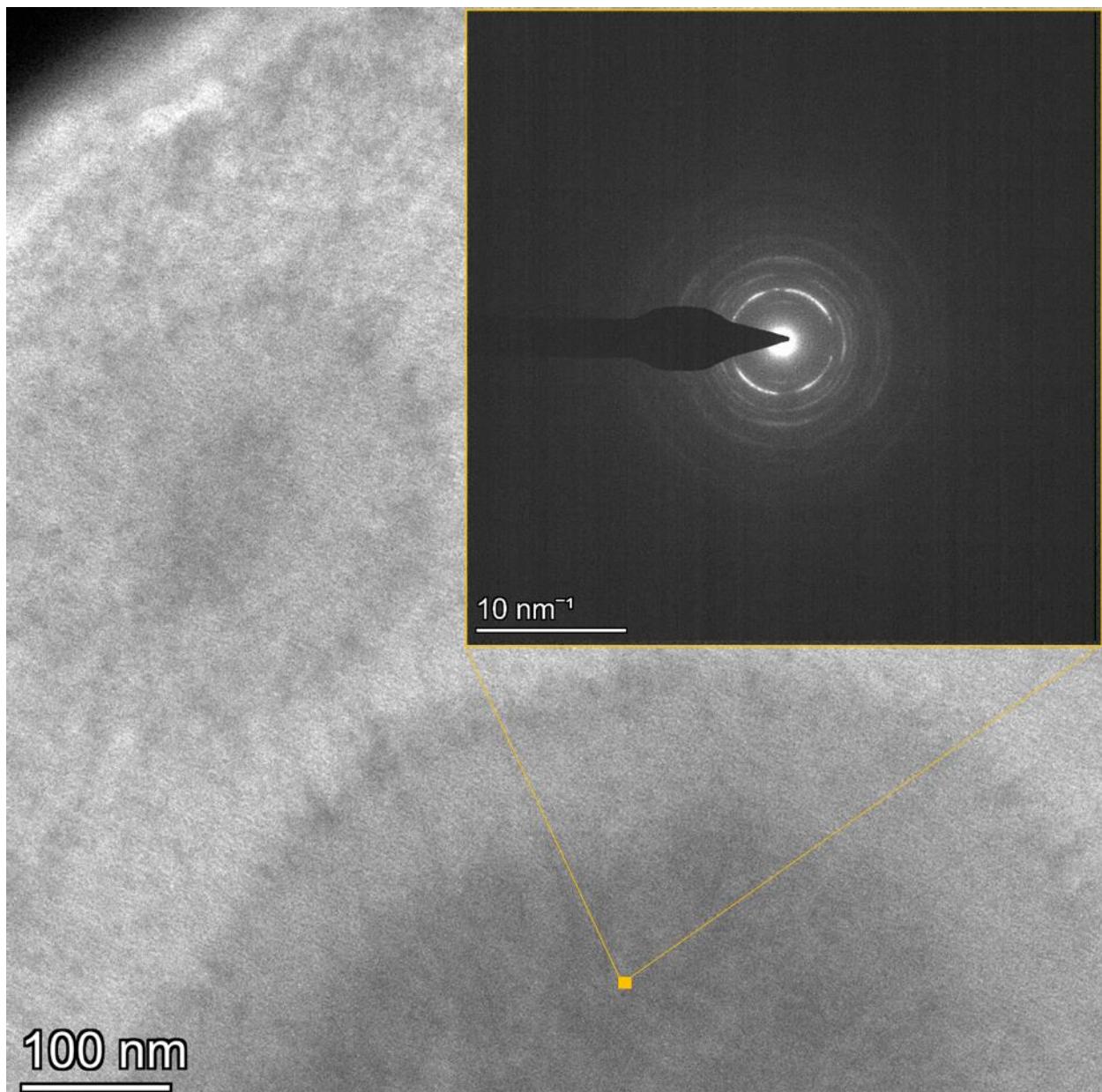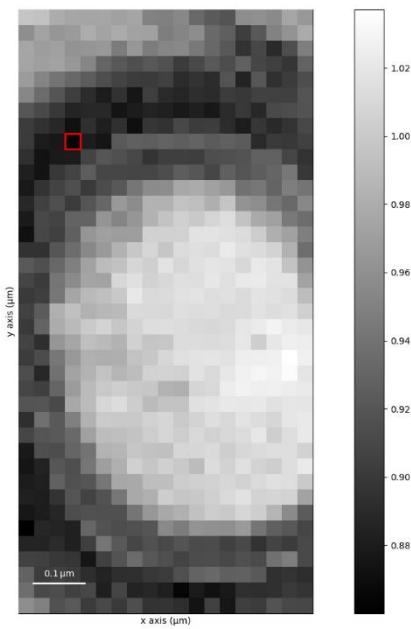
Figure S4. t(λ)⁻¹ map across a spheroidal ROI. Derived from the low-loss electron energy loss spectra (Malis et al., 1988), the map displays the spatial variation in sample thickness relative to the inelastic mean free path, $t(\lambda)^{-1}$. The darker grey regions correspond to the Mg-rich cortices (e.g., red square), while the bright central regions correspond to the thicker Mn-rich cores ($t(\lambda)^{-1} \approx 1.02$). In this regime (($\lambda)^{-1} > 0.5$), inelastic scattering significantly contributes to the background signal (unscattered fraction $= e^{\wedge}\text{-}t(\lambda)^{-1} \approx 0.41$, i.e., 41% of the beam is coherent (unscattered). The other 59% creates inelastic background noise, which hinders the detection of weak diffraction intensities such as the cation ordering (b-type) superlattice reflections. Calculated using HyperSpy (https://doi.org/10.5281/zenodo.14956374).

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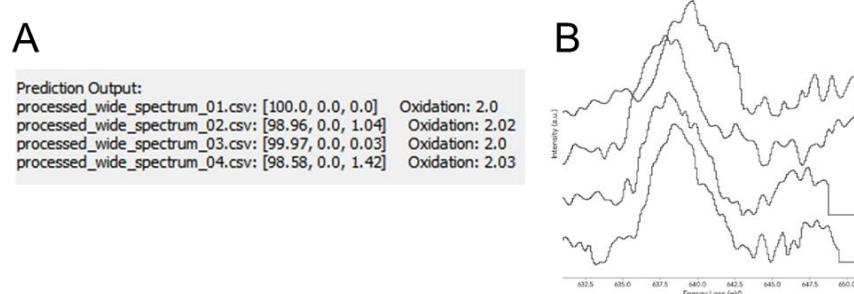
968 **Figure S5.** Results of the model MnEdgeNet (
969 https://github.com/xinhuolin/MnEdgeNet/tree/main 5). Stacked raw EELS spectra of the Mn
970 $L_{2,3}$-edge acquired from four representative locations within the spheroid core (range 631–
971 651 eV). Vertical offsets are applied for clarity. All scans consistently display a sharp L_3
972 absorption maximum at 693.5 to 640.0 eV and a deep inter-peak valley returning to baseline
973 intensity. These identical spectral features confirm the spatial homogeneity of the pure high-
974 spin Mn(II) oxidation state across the analyzed region.

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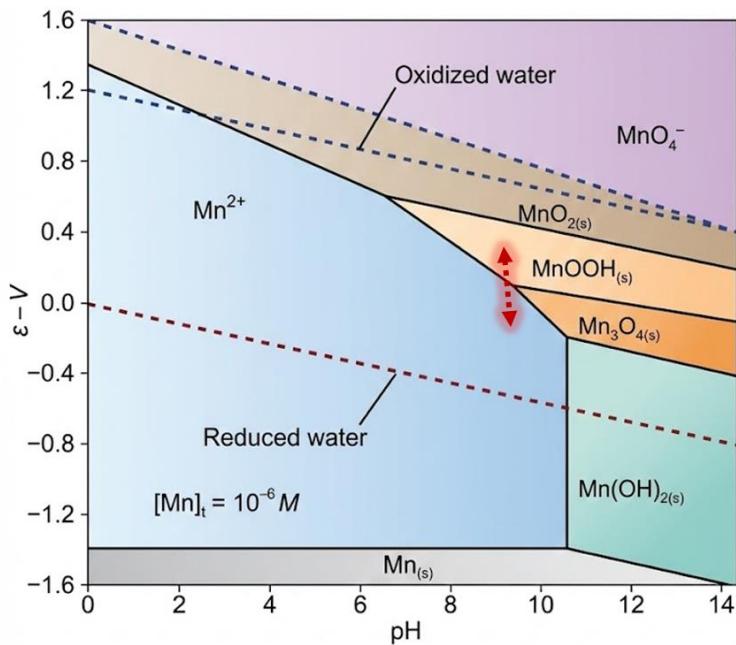
Figure S6. Eh-pH stability diagram for the Mn water system at 25°C. The diagram plots redox potential (ε–V) against acidity (pH), showing the thermodynamically stable fields for various manganese species at a total manganese molar concentration of $[Mn]_t = 10^{-6}$ M. Solid lines delimit the stability boundaries between species. Aqueous species shown are the manganous ion (Mn^{2+}) and permanganate (MnO_4^-). Solid phases, denoted by the subscript (s) and distinct color shading, include pyrochroite ($Mn(OH)_2(s)$), hausmannite ($Mn_3O_4(s)$), manganite ($MnOOH(s)$), and pyrolusite ($MnO_2(s)$). Adapted from Faust & Aly (1998) and Stumm & Morgan (1996). The approximate experimental trajectory is shown (arrow)

987 **Table S1. Saturation state of relevant mineral phases in the functionalized active**
988 **electrolyte (E_{11})**. Values represent the Saturation Index (SI = log(IAP) - log(K)) calculated
989 for the mixture composition S_{ii}. Positive SI values (>0) indicate supersaturation, while
990 negative values (<0) indicate undersaturation. Note that the S_{ii} is strongly supersaturated with
991 respect to aragonite, dolomite, and hydromagnesite, consistent with the observed mineral
992 assemblage.

Phase	SI	logIAP	logK	Formula
Huntite	8.81	-21.15	-29.97	$CaMg_3(CO_3)_4$
Dolomite	6.09	-11	-17.09	$CaMg(CO_3)_2$
Dolomite(d)	5.54	-11	-16.54	$CaMg(CO_3)_2$
Hydromagnesite	5.01	-3.75	-8.76	$Mg_5(CO_3)_4(OH)_2 \cdot 4H_2O$
Rhodochrosite	4.18	-6.95	-11.13	$MnCO_3$
Rhodochrosite(d)	3.44	-6.95	-10.39	$MnCO_3$
Magnesite	2.95	-5.08	-8.03	$MgCO_3$
Calcite	2.55	-5.93	-8.48	$CaCO_3$
Aragonite	2.41	-5.93	-8.34	$CaCO_3$
Artinite	1.88	11.48	9.6	$MgCO_3:Mg(OH)_2:3H_2O$
Nesquehonite	0.54	-5.09	-5.62	$MgCO_3:3H_2O$
Pseudo-kutnahorite†	-2.57	-12.87	-10.3	$MnCa(CO_3)_2$
Nahcolite	-3.12	-3.67	-0.55	$NaHCO_3$
Natron	-5.06	-6.37	-1.31	$Na_2CO_3:10H_2O$
Thermonatrite	-6.47	-6.35	0.12	$Na_2CO_3:H_2O$
Trona	-9.23	-10.02	-0.8	$NaHCO3:Na2CO3:2H_2O$
Pyrochroite	-0.5	14.7	15.2	$Mn(OH)_2$
Brucite	-0.27	16.57	16.84	$Mg(OH)_2$
Portlandite	-7.08	15.72	22.8	$Ca(OH)_2$
Manganite	-5.11	20.23	25.34	$MnOOH$
Hausmannite	-5.87	55.16	61.03	Mn_3O4
Bixbyite	-9.96	-10.57	-0.61	Mn_2O_3
$MnCl_2 \cdot 4H_2O$	-7.87	-5.16	2.71	$MnCl_2 \cdot 4H_2O$

993 **Notes:** Calculations were performed using the U.S. Geological Survey geochemical code PHREEQC (Parkhurst
994 and Appelo, 2013) employing the wateq4f thermodynamic database. (d) denotes disordered or metastable phases
995 included in the database. † Solubility product (log K_{sp}) for pseudokutnahorite is adapted from Mucci (2004).
996 Data Availability: The input code and full dataset for these calculations are available in the Zenodo repository
997 (see Data Availability Statement).
998

Supplementary Note S1: Estimation of local interfacial pH and saturation state of dolomite near the electrode-mineral interface

1. Model assumptions

To estimate the magnitude of pH oscillations at the working electrode (WE) surface, we apply a simplified steady-state Nernst-Planck approximation. The model assumes a stagnant diffusion boundary layer (Nernst diffusion layer, Δ) separating the electrode surface from the bulk electrolyte.

We adopt the following governing assumptions:

1. Transport control: mass transport of protons (H^+) and reactants is governed primarily by diffusion across the boundary layer; migration effects are minimized by the supporting electrolyte.

2. Stoichiometry: the anodic oxidation of Mn(II) to Mn(III) coupled with hydrolysis generates protons. We conservatively assume a localized proton flux (J_{H+}) proportional to the measured current density (j).

3. Layer thickness: For an unstirred vertical electrode under natural convection, Δ is estimated at *ca.* 200 μm (2×10^{-2} cm) (Bard & Faulkner, 2001).

4. Bulk conditions: Bulk solution is maintained at pH 9.1 (experimental baseline).

2. Flux Calculation

The J_{H+} generated at the anode is derived from Faraday's Law:

$J_{H+} = j(nF)^{-1}$

Where:

- $j \approx 100$ μA cm^{-2} (T_{peak} current density observed in CV scans).

- n = 1 (electrons transferred per proton equivalent).

- F = 96,485 C mol^{-1} (Faraday constant).

$J_{H+} = 10^{-4} (96485) = 1.04 \times 10^{-9}$ mol cm^{-2} s^{-1}

3. Interfacial pH shift

According to Fick's First Law, the concentration gradient (ΔC) required to sustain this flux across layer Δ is:

$$\Delta H^+ \approx J_{H+} \times \delta (D_H)^{-1},$$

where:

- $D_H \approx 9.3 \times 10^{-5}$ cm^2 s^{-1} (diffusion coefficient of H^+ in water at 25°C).

- $\delta \approx 0.02$ cm

$\Delta H^+ \approx 1.04 \times 10^{-9} (0.02)/(9.3 \times 10^{-5}) \approx 2.21 \; 10^{-7}$ M

Resulting surface pH: the bulk proton concentration at pH 9.1 is $[H^+]_{bulk} \approx 7.9 \times 10^{-10}$ M.

The surface concentration $[H^+]_{surf}$ is the sum of the bulk concentration and the anodic accumulation $\Delta[H^+]$):

$$[H^+]_{surf} \approx 7.9 \text{ x } 10^{-10} + 2.2 \text{ x } 10^{-7} \approx 2.21 \text{ x } 10^{-7} \text{ M}$$

$$\underline{pH_{surf} = -\log(2.21 \text{ x } 10^{-7}) \approx 6.65}$$

The unbuffered anodic shift is $\Delta pH \approx -2.45$ units. However, in the presence of the (bi)carbonate/alginate buffering system, this theoretical drop can be substantially attenuated (see Fig. 2). Conservatively estimating a 50–70% buffering efficiency, then the effective localized shift is $\Delta pH \approx \pm 0.5$ to 1.2 units.

4. Impact on supersaturation (SI$_{dol}$)

The saturation state for dolomite (SI$_{dol}$) is defined as:

$$SI_{dol} = \{Ca^{2+}\}\{Mg^{2+}\}\{CO_3^{2-}\}^2 K_{sp}^{-1}$$

Since CO_3^{2-} is strictly pH-dependent, the calculated anodic acidification near the WE surface (pH 9.1 to ~8.0 to 8.5) causes a logarithmic decrease in local carbonate ion activity.

- Anodic phase (oxidation): local pH drops, SI_{dol} decreases significantly (undersaturation or low supersaturation), promoting the dissolution of kinetic defects.
- Cathodic phase (reduction): local pH is restored or increased (via proton consumption), SI_{dol}, driving rapid nucleation.

This confirms that electrochemical cycling imposes a high amplitude oscillation of the saturation state (SI$_{dol}$), mimicking the Ostwald ripening mechanism required for ordering.

5. Scan rate justification: The electrochemical experiments utilized a fixed scan rate of 10 mV s^{-1}. This rate was selected to establish a stable diffusion profile (δ) that balances the anodic proton production with the buffering capacity of the hydrogel, creating a sustained, reproducible saturation gradient necessary for crystal growth.

References

Bard, A. J., & Faulkner, L. R. (2001). *Electrochemical Methods: Fundamentals and Applications*. Wiley.

Stumm, W., & Morgan, J. J. (1996). *Aquatic Chemistry*. Wiley-Interscience.

Supplementary Note S2. A working-table compiling sedimentary units of interest to test the mechanism proposed in the main text. Have fun.

Period	Formation / Location	Dominant Texture & Key Features	References*
Holocene	Sabkha Flats (Al Jubayl, Saudi Arabia)	Microbial mats; Quaternary ordered dolomite.	(C'hafetz et al., 1999)
Holocene	Sabkha Flats (Abu Dhabi/Qatar)	Microbial mats; protodolomite forming in EPS-rich layers; seasonal cycling.	(Bontognali et al., 2010; Di Loreto et al., 2021)
Holocene	Lake Van (Turkey)	Cyclic early diagenetic dolomite; triggered by hydrochemical mixing and redox perturbations.	(McCormack et al., 2018, 2024)
Miocene	Eger Rift Paleolake (Czech Rep.)	Episodic carbonate cementation driven by Mn–N redox cycling in a non-marine setting.	(Petrash et al., 2025)
Eocene	Green River Formation (USA)	Lacustrine dolomite; vast deposits formed in organic-rich, stratified (meromictic) alkaline paleolakes.	(Pommer et al., 2023)
Triassic (Ladinian)	Dolomite Mtns / Sella Group (Italy)	The "Dolomite" Archetype. synsedimentary	Meister et al., 2013), Sedimentology 60(1):270-291
Permian-Triassic	Inner Platform (Dinarides, Croatia)	Dolomitization linked to oceanic anoxia and chemocline shifts; inner platform facies.	(Aljinović et al., 2025)
Devonian	Ratner Formation (Canada, SK)	Partially recrystallized medium-sized dolomite with high. Mn contents (ME)	Fu & Qinq, 2010, *Carbonates and Evaporites* 26, 111-115
Devonian	Grosmont Formation (Canada, AB)	Sabkha dolomite forming within burrowed stromatolites; microbial influence.	(Baniak et al., 2014)
Upper Cambrian (Furongian)	Chaomidian Fm. (North China Platform)	Fabric-retentive microbial dolomite; thrombolite-stromatolite reefs	(Han et al., 2024; Chen et al., 2014)

Period	Formation / Location	Dominant Texture & Key Features	References*
Lower Cambrian	Xiaoerbulake Fm. (Tarim Basin, China)	Fabric-retentive dolomicrite; microbialites (stromatolites/thrombolites).	(Zhang et al., 2022; Zheng et al., 2022)
Ediacaran (Shuram Excursion)	Khufai, Buah & Birba Fms (Oman)	Primary Nanoscale Fabrics; fibrous dolomite ooids, spherulitic stromatolites, and fibrous cements.	(Wilcots et al., 2025)
Ediacaran	Doushantuo Formation (China)	Cap dolostone featuring interlinked cycles of methane, manganese, and sulfate.	(Cai et al., 2023)
Mesoproterozoic (~1.6 Ga)	McArthur Basin (Amelia/Mara Dol.), Australia	Massive fabric-retentive dolostone; Mn-mineralized sheaths; exceptional eukaryotic cell preservation.	(Muir, 1976, 1983)
Mesoproterozoic (~1.6 Ga)	Chuanlinggou Formation, N China	Fabric retentive. Mn-bearing dolomitic siltstone	(Xu et al., 2025)
Mesoproterozoic (~1.5 Ga)	Gaoyuzhuang Formation, N China	Syndiagenetic, consist of dolommicrite, along with kutnohorite and rhodochrosite	(Xu et al., 2025)
Mesoproterozoic (~1.4–1.1 Ga)	Bangemall Supergroup (W. Australia)	Sizable Mn deposits associated with surface oxygenation events.	(Spinks et al., 2023)
Paleoproterozoic (~2.1 Ga)	Francevillian Succession (Gabon)	Intimate association Mn-Ca carbonate	(Nzamba et al., 2026; Préat et al., 2011)

1067 * (Refer to the main text). Unless otherwise indicated.