New analogue materials for nonlinear lithosphere rheology, with an application to slab break-off

Taco Broerse*1, Ben Norder², Rob Govers¹, Dimitrios Sokoutis¹, Ernst Willingshofer¹, and Stephen Picken²

¹Department of Geosciences, Utrecht University, Utrecht, The Netherlands

²Department of Chemical Engineering, Delft University of Technology, Delft, The Netherlands

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Abstract

Stress-dependent nonlinear upper mantle rheology has a firm base in rock mechanical tests, where this nonlinearity results from dislocation creep of minerals. In the last few decades there has been some attention to nonlinear, power law, materials for application in scaled analogue experiments for tectonic problems. However, studies describing the rheology of analogue materials with the same nonlinear dependency on stress as observed for lithospheric mantle materials at relevant stress levels, are still lacking.

In this study we have developed and rheologically tested materials based on combinations of silicone polymers and plasticine, with the aim of obtaining a material that can serve as a laboratory analogue to the power-law rheology of olivine aggregates at lithospheric mantle conditions. From our steady-state creep tests we find that it is possible to obtain such a power-law material, with effective viscosities over relevant model stress ranges [5-4000 Pa] that allow for non-linear deformation at laboratory time scales.

We apply the developed material to settings where localized deformation of the lithosphere can be expected: slab break-off. We study this process using analogue models, where we apply the new nonlinear material to the lithospheric mantle domains, while we use Newtonian glucose to represent the low viscous asthenosphere. From these models we observe that lithospheric tearing occurs in a localized mode.

^{*}Address: Princetonlaan 8a, Utrecht, The Netherlands; Email: d.b.t.broerse@uu.nl; Corresponding author

1 Introduction

Permanent deformation of the Earth's lithosphere can occur in brittle or ductile (viscous) modes, where the latter prevails for increasing temperatures and depths (Goetze and Evans, 1979; Kohlstedt et al., 1995). Ductile deformation involves a permanent change of shape without fracturing, and can be caused by various microphysical creep mechanisms, each of which results in deformation rates with different dependencies on external conditions such as stress or temperature, and microstructure of the material (Ranalli, 1995). For ductile steady-state deformation we can discriminate between Newtonian rheology, where strain rate is proportional to stress and hence viscosity is constant, and nonlinear rheologies, where viscosity has a nonlinear dependency on stress (Ranalli, 1995). Nonlinear ductile deformation often comes in the form of power-law creep, where strain rate $\dot{\epsilon}$ has a power-law relation to stress σ , leading to material weakening at elevated stresses, a phenomenon often called pseudoplasticity:

$$\dot{\epsilon} = C\sigma^n \quad n > 1 \tag{1}$$

Many different materials, from tomato puree to metals exhibit both types of ductile deformation: linear Newtonian creep at low stresses and power-law creep at high stresses (Barnes, 1999). Also for mantle materials, such as olivine aggregates, laboratory tests at high temperatures show this transition from Newtonian to power-law creep with increasing differential stress (Hirth and Kohlstedt, 2003).

The existence of plate boundaries implies localized deformation, not only in the brittle, but also ductile regime in relatively weak shear zones (Bercovici, 1993). Localization of ductile deformation is often attributed to strain softening as a result of previously accumulated deformation (e.g. Poirier (1980); Bercovici and Karato (2002)), by mechanisms such as recrystallization (Karato et al., 1980) or shear heating (Regenauer-Lieb and Yuen, 1998). The amount of weakening or strengthening due to strain softening or hardening, respectively, in shear zones is however debated (De Bresser et al., 2001; Drury, 2005; Platt and Behr, 2011; Tasaka et al., 2017). Alternatively, in settings with highly varying stresses, power-law creep may result in high rates of deformation, as the strain rate will localize stronger than the stress itself. Such a system weakening will be especially efficient when it results in geometrical changes of tectonic units such that stress becomes increasingly localized by a positive feedback, as is observed in slab necking or tearing processes in numerical and analytical models (Andrews and Billen, 2009; Schmalholz, 2011). Several numerical studies suggest the importance of power-law stress weakening of the lithosphere for transient localization phenomena; in subduction models power-law creep has been invoked as mechanism required for slab break-off (Davies and von Blanckenburg, 1995; Van Hunen and Allen, 2011; Duretz et al., 2011). Furthermore, power-law creep has been invoked to explain rapidly decaying postseismic deformation following several earthquakes (Freed et al., 2006, 2012; Masuti et al., 2016; Sobolev and Muldashev, 2017). In combination with strain softening a strongly stress-dependent rheology is a precondition for stable shear zones (Rutter, 1999).

Analogue models of tectonic problems aim at constructing a scaled representation of a natural tectonic setting, where application of geometrical, kinematic and dynamic scaling of the natural setting to the laboratory analogue is required (Hubbert, 1937; Weijermars and Schmeling, 1986). A precondition for correct dynamic similarity is that the rheological properties of the materials used are properly scaled with respect to the rheological properties of the respective natural rocks. Often in analogue models of tectonic problems, only the Newtonian behavior of rocks deforming in the ductile regime is modeled, by using materials that show near-Newtonian behavior, such as silicone polymers (Weijermars, 1986; Ten Grotenhuis et al., 2002; Rudolf et al., 2016). While for many tectonic settings the assumption of Newtonian behavior can be justified, problems that may involve a high variability of differential stresses call for materials that exhibit both Newtonian and power-law rheology.

Over the last few decades there have been a number of studies that tested power-law materials for the application in analogue models of tectonic problems (see an overview in section 4). However, rheological studies of analogue materials that are tailored to represent the rheology of the lithospheric mantle are currently missing. Especially, there is a lack of materials that show the same nonlinearity to stress, expressed by the stress exponent n, as the most abundant upper mantle material olivine ($n \approx 3.5$ Hirth and Kohlstedt (2015)). For existing analogue materials, the power-law nonlinearity is either too weak, or overly strong compared to olivine aggregates (see section 4.2). Furthermore, rheological data for power-law lithosphere analogue materials is lacking at the low stresses that are typical for analogue models at the upper mantle scale (< 100 Pa, Boutelier and Oncken (2011)). In our paper we develop materials based on silicones and plasticines, and study their behavior to determine how these can serve as mechanical analogs for nonlinear ductile creep in the lithospheric mantle. Furthermore, we are interested in the type of deformation behavior of the developed materials, distributed or localized, in tectonic settings where lithospheric tearing may occur.

This paper consists of two parts: in the first part we estimate the average rheology of lithospheric mantle and test the rheology of silicone-plasticine materials at stress levels typical for analogue models at the mantle-lithosphere scale. In the second part we present an application in a model of break-off of subducted lithospheric slabs, where localisation of strain is a prerequisite and a power-law material may potentially facilitate this mode of deformation.

2 Nonlinear rheology of the lithospheric mantle

2.1 Laboratory flow laws

We intent to design a material that can function as a rheological analogue for the average lithospheric mantle rheology. Olivine $((Mg, Fe)_2SiO_4)$ makes up roughly half of the upper mantle composition at depths down to 400 km (Putnis, 1992) and is the weakest upper mantle constituent (Karato and Wu, 1993). Therefore, oliving aggregate rheology is generally thought to represent the rheology of the upper mantle (e.g. Goetze and Evans (1979); Hirth and Kohlstedt (2003)). The steady-state rheology of olivine aggregates at high temperatures shows a transition from a viscosity that is stress-independent to a stress-dependent viscosity at increasing stress (Mei and Kohlstedt, 2000b). After application of a constant stress, olivine shows recoverable deformation, both immediate (elastic) and time-dependent (anelastic), next to permanent (viscous) deformation. After a transitional period where both anelastic and viscous deformation occur, strain rates decrease and converge to a steady-state regime (Chopra, 1997). Steady-state strain rates from rheological experiments on olivine can be explained from contributions from diffusion creep (Newtonian) and dislocation creep (power-law) (Hirth and Kohlstedt, 2003):

$$\dot{\epsilon} = \dot{\epsilon}_{\text{diff}} + \epsilon \dot{\alpha}_{\text{isl}} = C_{\text{diff}} \sigma + C_{\text{disl}} \sigma^n \tag{2}$$

where power-exponent n has been estimated to be in the range 3 - 4.9 (Mei and Kohlstedt, 2000a; Korenaga and Karato, 2008) and is likely close to 3.5 (Hirth and Kohlstedt, 2015). Parameters C are functions of grain size, temperature, pressure and water content (Hirth and Kohlstedt, 2003):

$$C = A d^{-p} C_{OH}^r exp\left(-\frac{E + PV}{RT}\right)$$
 (3)

with material parameter A, grain size d, grain size exponent p, C_{OH} the water content with exponent r, activation energy E, activation volume V, pressure P, gas constant R and temperature T.

2.2 Ellis rheology

Comparison and scaling of rheologies is simpler when the strain-rate vs. stress equation (2) is rewritten to a viscosity-stress relation:

$$\eta_{\text{eff}} = \frac{\sigma}{2\dot{\epsilon}} = \frac{1}{2C_{\text{diff}} + 2C_{\text{disl}}\sigma^{n-1}} = \frac{\frac{1}{2C_{\text{diff}}}}{1 + \frac{C_{\text{disl}}}{C_{\text{disf}}}\sigma^{n-1}} \tag{4}$$

which we rewrite to the Ellis rheological model:

$$\eta_{\text{eff}} = \frac{\eta_0}{1 + (\frac{\sigma}{\sigma_t})^{n-1}} \tag{5}$$

with:

$$\eta_0 = \frac{1}{2C_{\text{diff}}} \tag{6}$$

$$\sigma_t = \left(\frac{C_{\text{diff}}}{C_{\text{disl}}}\right)^{\frac{1}{n-1}} \tag{7}$$

The Ellis model describes the combination of power-law and Newtonian rheological behavior in terms of the zero-stress Newtonian viscosity η_0 and transition stress σ_t (Barnes, 1999). At the transition stress, the strain rate contributions from the Newtonian and nonlinear mechanisms in equation (2) are equal, and σ_t thus marks the change of rheological behavior: Newtonian for stresses below σ_t and power-law above σ_t .

2.3 Average ductile lithosphere rheology

We are interested in an average ductile lithospheric mantle rheology, which we cast in terms of an Ellis rheology. For first order estimates of the variability of η_0 and σ_t , we use equations (3),(6) and (7), where we apply parameters for dry and wet (constant C_{OH}) olivine diffusion and dislocation creep from Hirth and Kohlstedt (2003), table 1, including the value for n = 3.5 for dislocation creep. We apply a correction to C for parameters based on uni-axial experiments: $C' = C \frac{2}{3(n+1)/2}$ (Ranalli, 1995). We assume a linear geotherm between the Moho and 1350°C at the base of the thermal boundary layer, where for continental lithosphere we place the Moho with a 400°C temperature at 40 km depth and the 1350°C isotherm at 200 km (e.g. Jaupart and Mareschal (2007)); and for an oceanic lithosphere we situate the Moho with a 50°C temperature at $5~\mathrm{km}$ depth, and the $1350^{\circ}\mathrm{C}$ isotherm at 100 km depth. We define the ductile lithosphere between a brittle-plastic transition at a strain rate of 10^{-14} s⁻¹ using the Goetze criterion for the brittle strength of the lithosphere $(\sigma_1 - \sigma_3 < P)$ (Kohlstedt et al., 1995) and a lithospheric-asthenospheric boundary (LAB) defined by $\eta_0 = 10^{20}$ Pa s.

Using the described wet and dry olivine rheologies, the continental and oceanic geotherms and a grain size range between 0.1 and 10 mm, we estimate η_0 and σ_t as a function of depth. The first two panels of figure 1 show results for an oceanic lithosphere with a dry rheology, where we normalise η_0 by $\eta_{0,\text{LAB}}$, the diffusion creep viscosity that defines our LAB. Similar figures for η_0 and σ_t for a wet olivine rheology, respectively continental lithosphere, show similar results as shown in the left and middle panel of figure 1 (see supplemental material B). The zero-stress viscosity η_0 of the lithospheric mantle, for a constant grain size throughout the lithosphere, is highly variable, mainly due to the strong temperature dependency of diffusion creep. For lithospheric temperatures, the transition stress σ_t is dominantly dependent on the grain size due to the grain size dependency of diffusion creep: σ_t increases for decreasing grain size. For the last panel of figure 1 we take the average of η_0 , and use σ_t from the uppermost part of the ductile lithosphere (as the most viscous part should be representative

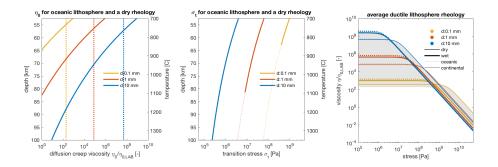


Figure 1: Average lithosphere rheology using olivine rheological parameters from Hirth and Kohlstedt (2003), left panel: η_0 for an oceanic lithosphere and a dry rheology for different values of grain size d, dotted lines denote average η_0 ; middle panel: σ_t for an oceanic lithosphere and a dry rheology, dotted lines denote values below the lithospheric-asthenospheric boundary (LAB) defined by $\eta_0 = 10^{20}$ Pa s. The top depth represents the brittle-plastic transition defined by the intersection of the Goetze criterion and the ductile stress at a strain rate of 10^{-14} s⁻¹. Right panel: combined results for oceanic and continental lithospheres with wet or dry olivine rheologies and 0.1 and 10 mm grain sizes, using the average η_0 and maximum σ_t and n = 3.5.

for the overall strength) to arrive at an average rheology. We repeat the same analysis for wet as well as dry rheologies and continental as well as oceanic lithospheres. While for a given pressure and temperature the olivine creep law consistently results in a lower viscosity for a wet rheology compared to a dry rheology (figure 22), this lower viscosity also results in a shallower brittle-plastic transition, resulting in a somewhat higher average η_0 .

Figure 1 shows that the average η_0 is not well constrained, as values are very sensitive the choice of grain size and to a lesser extent to the presence of water and type of lithosphere. Average ratio $\eta_0/\eta_{0,\text{LAB}}$ ranges between $2 \cdot 10^2$ and $4 \cdot 10^8$, with the lower value corresponding to the smallest considered grain size (0.1 mm). Similar large increases in viscosity throughout the lithosphere can be found in figure 2 of Karato and Wu (1993) or figure 2.1 of Freeburn (2016); Billen and Hirth (2007) consider a $10^6 - 10^7$ viscosity contrast for a subducting slab with respect to the asthenosphere, based on the same rheological parameters by Hirth and Kohlstedt (2003). The values for σ_t span a wide range: $2 \cdot 10^6 - 3 \cdot 10^9$ Pa, comparable to the transition stresses in figure 6 from Drury (2005). Nevertheless, once the power-law behavior is dominant at higher stresses, the right panel of figure 1 shows that the rheology follows a narrow band for all tested parameters.

Regarding parameters that describe recoverable deformation, the elastic shear modulus μ_e for the mantle lithosphere is on the order of 67 GPa (Dziewonski and Anderson, 1981), and anelastic olivine shear moduli μ_a have been estimated to be in the range 2.3 - 37 GPa (Chopra, 1997). We have thus obtained ranges

for the three parameters of the Ellis model $(n, \sigma_t \text{ and } \eta_0)$ that can describe the average lithospheric mantle rheology, next to shear moduli that constrain the amount of recoverable deformation. We will use these values in the next sections for developing an analogue lithospheric material.

3 Dynamic scaling of non-Newtonian materials

Weijermars and Schmeling (1986) developed rules for dynamic similarity of power-law materials in the case of negligible inertia. We propose a simpler approach using the Ellis rheological model (equation (5)) for experiments performed under standard gravity. The Ellis model equation contains three variables for which we can make use of existing scaling laws. Namely, the stress exponent n should be equal for model analogue (m) and natural (n) materials (Weijermars and Schmeling, 1986):

$$n_{\rm m} = n_{\rm n} \tag{8}$$

Furthermore, similarity of viscous forces requires constant viscosity ratios between model and nature values (Hubbert, 1937), which implies that the zero-stress viscosity η_0 scales with the same ratio as all other viscosities:

$$\frac{\eta_{0,n}}{\eta_{0,m}} = \frac{\eta_n}{\eta_m} \tag{9}$$

Finally, for σ_t we can simply use the existing scaling relations for stresses (e.g. Weijermars and Schmeling (1986)):

$$\frac{\sigma_{\rm t,n}}{\sigma_{\rm t,m}} = \frac{\rho_{\rm n} \ l_{\rm n}}{\rho_{\rm m} \ l_{\rm m}} \tag{10}$$

where ρ is density and l is length. Alternatively, if dynamic scaling is performed by using density differences between the sinking material and the surrounding fluid $\Delta\rho$ (e.g. Ribe and Davaille (2013); Schellart and Strak (2016)), the expression becomes:

$$\frac{\sigma_{\rm t,n}}{\sigma_{\rm t,m}} = \frac{\Delta \rho_{\rm n} \ l_{\rm n}}{\Delta \rho_{\rm m} \ l_{\rm m}} \tag{11}$$

The expression for transition stress σ_t implies that model dimension scale is important in the transition from linear to nonlinear behavior for a given material rheology. Time furthermore scales as:

$$\frac{t_{\rm n}}{t_{\rm m}} = \frac{\eta_{\rm n} \Delta \rho_{\rm m} l_{\rm m}}{\eta_{\rm m} \Delta \rho_{\rm n} l_{\rm n}} \tag{12}$$

4 Nonlinear analogue materials

4.1 Required analogue material rheology

Analogous to olivine aggregate rheology (section 2), the required lithospheric mantle material should behave Newtonian at low stresses, with a transition to

variable	natural values	scale factor $\frac{n}{m}$	target analogue values
$\overline{}$	3.5	1	3.5
$\bar{\eta}_{0,\mathrm{lith}}/\eta_{0,\mathrm{asth}}$	$2 \cdot 10^2 - 4 \cdot 10^8$	1	$2 \cdot 10^2 - 4 \cdot 10^8$
$\bar{\eta}_{0,\mathrm{lith}}$ [Pa s]	$2 \cdot 10^{22} - 4 \cdot 10^{28}$	$\approx 10^{18}$	$2 \cdot 10^4 - 4 \cdot 10^{10}$
σ_t [Pa]	$2 \cdot 10^6 - 3 \cdot 10^9$	$10^{6(I)} - 10^{7(II)}$	$2 - 300^{(1)}/0.2 - 30^{(11)}$
$\mu_{\rm e}~{ m [Pa]}$	$7 \cdot 10^{10}$	$10^{6(I)} - 10^{7(II)}$	$7 \cdot 10^{4(I)/3(II)}$
$\mu_{\rm a} \ [{\rm Pa}]$	$[0.2-4] \cdot 10^{10}$	$10^{6(I)} - 10^{7(II)}$	$[0.2 - 4] \cdot 10^{4(I)/3(II)}$

Table 1: Summary of average lithospheric mantle rheological values, scale values, and approximate target analogue rheological properties. Natural values have been discussed in section 2. Equation 8 prescribes that stress exponent n should be equal for nature and model. The scaling of stresses and moduli is dependent on the length scale of the model, as well as density or density difference scaling, which means that there is a possible range for the scaling of stress. We present two end-member scaling options; scaling option I uses a density difference scaling of $\frac{\Delta \rho_n}{\Delta \rho_m} = 0.5$ and uses a model height of 20 cm to represent an upper mantle depth of 660 km. This gives a stress scale $\sigma_n/\sigma_m \approx 10^6$ (equation (11)). This option resembles the setup of our models in the second part of this paper. For scaling option II we assume a 10 cm high model and a density ratio $\frac{\rho_n}{\rho_m} = 2$ (in line with densities of commonly used analogue materials), resulting in $\sigma_n/\sigma_m \approx 10^7$ (equation (10). From equation 9 follows that the ratio $\bar{\eta}_{0,\text{lith}}/\eta_{0,\text{asth}}$ should be equal for nature and model. The value for the average analogue lithospheric viscosity $\bar{\eta}_{0,\text{lith}}$ depends on the model asthenospheric viscosity, in our case $2 \cdot 10^4 < \bar{\eta}_{0,\text{lith}} < 4 \cdot 10^{10}$ Pa s when $\eta_{\text{asth}} \approx 10^2$ Pa s (i.e. glucose, Schellart (2011)).

power-law creep at increasing stress, with a stress exponent around 3.5. This parameter and other relevant rheological parameters for the analogue material are summarized in table 1. The table shows that the zero-stress viscosity η_0 for the lithospheric mantle should be in the range $2 \cdot 10^4$ - $4 \cdot 10^{10}$ Pa s (when the asthenospheric mantle is modeled using glucose), and that the transition stress σ_t should fall somewhere between 0.2-300 Pa, depending on the size of the model and the scaling of densities, as explained by the caption of table 1. Values for η_0 and σ_t are highly correlated: large σ_t apply to low values of η_0 . Elastic and anelastic strains in settings with large permanent deformation should be small, a constraint which is satisfied when the shear moduli (approximately) scale to the natural values. The shear modulus is scaled in the same manner as stress (Hubbert, 1937) (using equation (11) or (10)), which implies that for the analogue material the elastic shear modulus should be on the order of 10^4 Pa, and the anelastic shear modulus in the range 10^3 - 10^4 Pa.

4.2 Previous studies on power-law analogue materials

Over the last few decades there have been a number of studies that apply power-law (pseudoplastic) materials in analogue models of tectonic problems. Figure 2 summarizes the rheology of existing materials with stress dependent ductile rheologies. We also include commonly used Newtonian silicone polymers for reference such as PDMS (Rudolf et al., 2016) and rhodorsil gum (Treagus and Sokoutis, 1992). The required power-law behavior is indicated in figure 2 by the diagonal lines that represent a stress-exponent n of 3.5. The grey curved band represent the appropriate viscosities for analogue lithospheric mantle materials as a function of applied stress, using the requirements from section 4.1 (figure 1) in combination with an analogue asthenospheric viscosity of $\approx 10^2$ Pa s (i.e. glucose).

Figure 2 shows that none of the available materials can be used as a lithospheric mantle analogue without any modifications or additional rheometric measurements. Plasticines, either pure (McClay, 1976; Treagus and Sokoutis, 1992; Barnes, 1999) or mixed with an oil (Zulauf and Zulauf, 2004) have too high viscosities, while paraffin has a similar disadvantage Rossetti et al. (1999). Furthermore plasticines weaken too drastically for elevated stress, indicated by slopes that are steeper than the diagonals that represent n = 3.5; for plasticine stress exponents n have been estimated to be in the range 5-10 at room temperature (Weijermars, 1986; Schöpfer and Zulauf, 2002; Zulauf and Zulauf, 2004). However, an advantage of plasticine is its Newtonian behavior at low stresses, roughly below a transition stress σ_t of 10⁴ Pa (Barnes, 1999). The rheology of the silicone putty Dow Corning 3179 falls within the indicated area, but the nonlinearity of the material is too high, expressed by a stress exponent of ≈ 6 (Treagus and Sokoutis, 1992). The petrolatum + paraffin oil mixtures (designed to represent a subduction interface (Duarte et al., 2013)) display a reasonable Newtonian platform at low stresses as well, however these materials suffer from

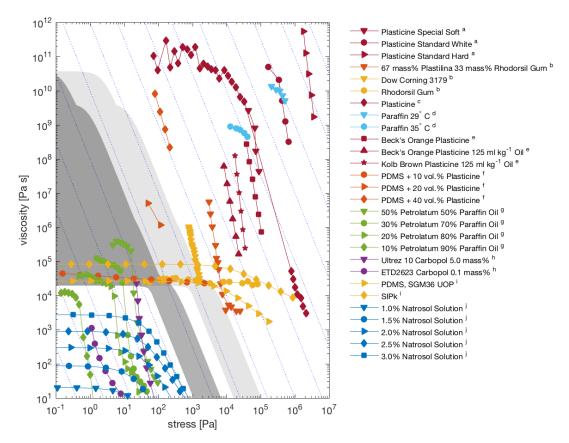


Figure 2: Overview of published rheological data on nonlinear materials used in analogue models: viscosity vs. stress, at approximate room temperature. References to tables and figures refer to the cited source. Diagonals represent a stress exponent of 3.5. The grey areas show appropriate viscosity vs. stress ranges for lithospheric mantle analogs in case analogue asthenosphere materials with an $\approx 10^2$ Pa s viscosity (e.g. glucose) are used. The light grey represents stress scaling option I, the darker grey option II, see section 4. a McClay (1976) (table 1): three different plasticines at a temperature $T=25^{\circ}\text{C}$. Treagus and Sokoutis (1992) (figure 2): plasticine-silcione polymer mix, a power-law and a Newtonian silicone polymer, at $T=24^{\circ}$, 23° , 24° C, respectively. ^c Barnes (1999) (figure 19): plasticine at a wide range of stresses showing the transition from stress-independent to stress-dependent viscosity (unknown T). Rossetti et al. (1999) (figure 2): paraffin viscosities at $T=29^{\circ}$ and 35° C, higher temperature measurements not included as these show near-Newtonian behavior. ^e Zulauf and Zulauf (2004) (using fitted parameters from table 1): plasticine and plasticine-oil mixtures at $T=25^{\circ}$ C, we included materials that show relatively strain-insensitive viscosities (see their figures 2 and 12). f Boutelier et al. (2008) (using fitted parameters from table 1): plasticine-silicone polymer mixtures at $T=25^{\circ}\mathrm{C.}^{g}$ Duarte et al. (2013) (figure 4): various petrolatum and paraffin oil mixtures, we included those with the lowest reported yield stresses. ^h Di Giuseppe et al. (2015) (using parameters from table 3): Carbopol at $T=23^{\circ}$ C, we include two-end members: one sample with the highest power-law exponent (Ultrez 10 5.0 mass%, pH 2.5, handmade, strain rate exponent m = 0.29, which translates to a stress exponent n = 3.5), and secondly, the sample with the lowest reported yield stress (ETD2623 0.1 mass\% pH 5.7, machine stirred). Rudolf et al. (2016) (using fitted parameters from table 3): a commonly used silicone-polymer and a silicone-polymer with fillers (other tested materials have very similar properties at the viscosity and stress ranges shown here), both at $T=23^{\circ}\mathrm{C}$. Boutelier et al. (2016) (using fitted parameters of table 1): various concentrations of Natrosol in water at T = 20 °C.

the same extreme weakening in the nonlinear regime as plasticines, which seems approximately constant for different petrolatum to paraffin oil ratios.

Carbopol has overall lower viscosities, but has been described to exhibit a yield stress (i.e. a stress below which no flow occurs) instead of a low-stress Newtonian plateau (Di Giuseppe et al., 2015), and viscosity decreases excessively at the stress range of interest. Natrosol solutions such as described by (Boutelier et al., 2016) nicely show a constant viscosity for low stresses, a slightly too low stress exponent ($n \approx 2$), while the viscosities are less than needed for lithospheric application.

The silicone polymer (PDMS) - plasticine mixtures as tested by Boutelier et al. (2008) approach the target rheology, but have either a slightly too low stress exponent (n=2.8) for 20 volume % of plasticine or too high overall viscosity (n=4.6) for 40 volume % of plasticine. Next, the stress levels over which the rheological parameters have been obtained are rather limited, and do not cover the expected stress range for upper mantle laboratory models. Silicone polymer - plasticine mixtures seem the most promising materials, which however need further adjustments to conform to the target rheology over wider stress ranges.

4.3 Material use in this study

Our goal is to develop an analogous material for olivine aggregate rheology at lithospheric conditions, combining both linear viscous (diffusion creep) behavior as well as power law behavior, with a stress exponent n in the range 3-4, a zero-stress viscosity η_0 in the range $2 \cdot 10^4$ - $4 \cdot 10^{10}$ Pa s and a transition stress σ_t in the range 2-300 Pa (corresponding to the light grey area in figure 2). We choose to work with plasticine-PDMS mixtures, as Boutelier et al. (2008) show that the stress exponent can be adjusted by the ratio of PDMS to plasticine. Plasticine-silicone polymer materials have been used or tested in previous studies (Weijermars, 1986; Sokoutis, 1987; Treagus and Sokoutis, 1992; Schrank et al., 2008), and we aim to overcome the problem of the high viscosity of these materials by adding a low viscosity silicone oil, which is likely to have only limited effects on the stress exponent but can significantly reduce the overall viscosity (Zulauf and Zulauf, 2004). Both plasticine and PDMS exhibit Newtonian behavior at low stresses in accordance with the Ellis rheology. We use the following material components to develop a lithospheric mantle analogue material:

Plasticine

Plasticine, and plasticine-like materials made by various manufacturers, are nondrying materials, and sold commercially as modeling clays. The components of plasticine or its varieties that go by names such as plastilin or plastilina are usually confidential, but generally these materials consist of an organic matrix (wax, vaseline, oils) with inorganic fillers (McClay, 1976). Creep tests have shown the limited applicability of many plasticines, which often do not reach steady-state creep but exhibit increasing viscosities with increasing strain (Zulauf and Zulauf, 2004; Boutelier et al., 2008). Zulauf and Zulauf (2004) describe a plasticine-like material, that they label Beck's orange, containing organic fillers (potato starch) and that shows only limited strain dependency. However, this material is no longer in production. For our experiments we use a similar material that is currently produced by the manufacturer Becks Plastilin and sold under the name Natur-Kreativknete, and that we refer to as organic plasticine. Relevant properties are listed in table 2. According to the manufacturer the organic plasticine is composed of beeswax, natural oils and plant-based filler (potato starch). We observe that plasticine is not very self-adhesive, which means that seams originating from the assemblage of different tectonic units may act as weak zones.

PDMS

The silicone polymer polydimethylsiloxane, PDMS, is a commonly used material in analogue models, especially for its Newtonian rheological properties for a wide range of strain rates (Johnson, 1961; Weijermars, 1986). PDMS is produced in a continuous range of viscosities, where longer polymer lengths result in higher viscosities (Weijermars, 1986). The high-viscosity PDMS that we use in our study, SGM36 produced by Dow Corning, has a viscosity of about $2 \cdot 10^4$ Pa s for stresses below 4 kPa, or alternatively, below strain rates below $2 \cdot 10^{-1}$ s⁻¹ (Rudolf et al., 2016). For higher stresses, or strain rates, the material increasingly starts to behave like a power-law material. One of the advantages of PDMS is the self-adhesive character, allowing the construction of multi-layered tectonic models without weak zones stemming from the assembly of the model.

Low-viscosity silicone oil

Zulauf and Zulauf (2004) have shown that addition of oil to plasticines reduces the effective viscosity while affecting the stress exponent only in a minor way. We add various amounts of silicone oil (50 cs Xiameter PMX-200 fluid), which is chemically similar to the SGM36 PDMS, but due its shorter polymer chains has a viscosity six orders of magnitude lower than PDMS (see table 2).

Iron powder

PDMS, low viscosity silicone oil and organic plasticine have similar densities, all about 450 kg m⁻³ lower than the density of glucose (table 2), the material that we use to represent the low-viscosity sub-lithospheric mantle (e.g. Davy and Cobbold (1988); Faccenna et al. (1999); Guillaume et al. (2009); Duarte et al. (2013)). By adding fine iron powder (Funiciello et al., 2004; Schellart, 2008; Guillaume et al., 2013) we adjust the mixture density to be able to create both buoyant and negatively buoyant materials. We mix and homogenise materials by hand. Table 3 lists the composite materials tested in this study.

Material	organic	PDMS	low viscosity	iron	glucose
name	plasticine		silicon oil	powder	
commercial name	Natur-Kreativknete	SGM36	50 cs Xiameter	unknown	Belgogluc
			PMX-200 fluid		HDE
producer	Becks Plastilin	Dow Corning	Dow Corning	unknown	Belgosuc
date purchased	06.2017	09.2015	07.2017	07.2017	12.2016
zero-stress viscosity [Pa s]	n.a.	$2 \cdot 10^4 \star$	$5 \cdot 10^{-2} \dagger$	-	$1.6 \cdot 10^2 \ddagger$
density [kg $m^{-}3$]	1143 •	965 •	957∙	7874	1405 •

Table 2: Overview of properties of materials used in this paper. All parameters determined at 20°C. \star Rudolf et al. (2016). † ‡ viscosity values according to manufacturer specifications. • Densities determined using a pycnometer, uncertainties are on the order \pm 10 kg m⁻³.

	Sample	organic	PDMS	low viscosity	iron
	name	plasticine		silicon oil	powder
1	40OPL 60PDMS 38.8Fe	40	60	-	38.8
2	$60\mathrm{OPL}\ 40\mathrm{PDMS}\ 33.2\mathrm{Fe}$	60	40	-	33.2
3	$60\mathrm{OPL}\ 40\mathrm{PDMS}\ 37.3\mathrm{Fe}\ 8\mathrm{oil}$	60	40	8	37.3
4	60OPL 40PDMS 41.4Fe 16oil	60	40	16	41.4
5	$60\mathrm{OPL}\ 40\mathrm{PDMS}\ 45.6\mathrm{Fe}\ 24\mathrm{oil}$	60	40	24	45.6
6	$80\mathrm{OPL}$ 20PDMS 13.7Fe	80	20	-	13.7

Table 3: Overview of tested materials. Material name and mass fractions of the constituents. Mass percentages are chosen such that the high viscosity materials (organic plasticine and PDMS) represent 100%. All densities have been adjusted by addition of iron power to obtain a uniform density of 1365 kg/m 3 .

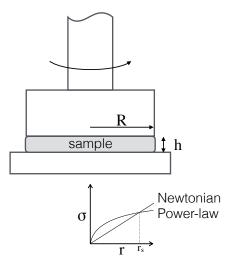


Figure 3: Schematic rheometer setup, with gap height h and disk radius R, including shear stress σ variation as function of radial distance r for Newtonian and power-law materials.

5 Measurement setup

We use a AR-G2 rheometer (by TA Instruments) that we operate in a parallel plate setup, as shown in figure 3. In all measurements the radius R of the upper plate is 12.425 mm and a gap height of 25 mm is applied. The rheometer applies a torque M and measures the angular deflection θ . Strain γ (engineering strain $\gamma = 2\epsilon$) is a simple linear function of radius r:

$$\gamma(r) = \frac{r}{h}\theta\tag{13}$$

Whereas the applied torque M is the integral of the product of shear stress and arm over the area:

$$M = \int_0^R \sigma(r) \ r \ 2\pi r \ dr \tag{14}$$

As Macosko (1994) shows, the shear stress at the rim (r = R) can be written as a function of the torque and the derivative of torque to strain:

$$\sigma(R) = \frac{M}{2\pi R^3} \left(3 + \frac{d \ln M}{d \ln \dot{\gamma}} \right) \tag{15}$$

The derivative is unknown, except for Newtonian materials, where it is 1. For that reason the rheometer reports apparent shear stress σ_a values at the rim that are equal to the true shear stress σ if the material would behave as a Newtonian material:

 $\sigma_a(R) = \frac{2M}{\pi R^3} \tag{16}$

To calculate the actual stress (for a non-Newtonian material) either the derivative of the torque to strain rate should be measured, or one can make use of the fact that at one radial distance (r_s) the curves for apparent Newtonian σ_a and true power-law stresses σ intersect (see figure 3):

$$\sigma(r_s) = \sigma_a(r_s) \tag{17}$$

As shown by Carvalho et al. (1994) at radial distance $r_s = 0.755 * R$, the apparent and true shear stresses are equal with an error less than 1% for power-law materials with stress exponents n larger than 0.8. As both strain and apparent shear stress vary linearly with radial distance, we can apply this so-called one point correction by taking shear stress and strain at r_s :

$$\sigma(r_s) = \frac{r_s}{R}\sigma_a(R) = 0.755\sigma_a(R) \tag{18}$$

$$\gamma(r_s) = \frac{r_s}{R}\gamma(R) = 0.755\gamma(R) \tag{19}$$

We test the materials under constant stress, with apparent stress levels σ_a in the range [5;4000 Pa], in a creep and recovery setup. Before the first application of stress we let the sample rest in the rheometer for at least 3 hours, to relax residual stresses. Stresses are applied in ascending order. As figure 4 shows, we let the sample relax in between the application of different shear stresses; the duration of each stress application and subsequent relaxation differs for each stress level, where we aim at creating permanent deformation during stress application and (nearly) full relaxation during the rest period.

6 Data analysis

The creep and recovery tests show that the recorded strain contains a considerable elastic and anelastic component; from figure 4 we can observe immediate elastic (recoverable) deformation, followed by a superposition of a transient anelastic (recoverable) and steady-state (permanent) deformation. Especially for low stresses, where long periods of loading (several hours) are needed to produce significant permanent deformation, it is difficult to discriminate between the early transient deformation and the permanent deformation, which deviates slightly from a constant strain rate due to strain hardening or strain softening. While the strain curve resembles the theoretical curve for a Burgers body at constant stress (see Findley et al. (1976)), observed strain hardening (at low

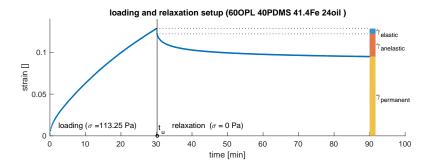


Figure 4: Loading and relaxation in the creep and recovery setup for each separate stress level, in this case 60OPL 40 PDMS 45.6Fe 24 oil at the stress level 113.25 Pa. Loading of the sample starts at t=0 and ends at $t=t_{\rm u}$. After relaxation the permanent strain $\gamma_{\rm p}$ can be determined from subtracting the recoverable strain from the total strain. The recoverable strains include the instantaneous, elastic strain $\gamma_{\rm e}$ at the time of unloading $t_{\rm u}$ and the subsequent time-dependent anelastic strain $\gamma_{\rm a}$.

stress) or strain softening (at high stress) does not permit the application of the Burgers model.

In the absence of a mathematical model that would allow us to estimate a steady-state viscosity at each stress level, we therefore determine the mean steady-state viscosity per stress level. The total strain that has accumulated during the loading period up to the unloading time t_u comprises of elastic γ_e , anelastic γ_a and permanent γ_p strains, where γ_e and γ_a recover during the relaxation phase, such that the remaining strain equals to γ_p (figure 4). The average steady-state viscosity then becomes:

$$\bar{\eta}(\sigma) = \frac{\sigma t_{\rm u}}{\gamma_{\rm p}} \tag{20}$$

To describe the stress-dependent viscosity we fit the Ellis model of equation (5) to the average viscosity $\bar{\eta}$ as a function of stress σ , or solely a power-law equation (equation (1)) in case no transition from linear to nonlinear viscosity could be observed. We perform the fitting in a weighted nonlinear least squares sense using the Levenberg-Marquardt algorithm, where we minimize:

$$\left(W * \left(\frac{\eta_0}{1 + \left(\frac{\sigma}{\sigma_t}\right)^{n-1}} - \bar{\eta}(\sigma)\right)\right)^2 \tag{21}$$

using weights $W = 1/\bar{\eta}(\sigma)$.

For strain rates below 10^{-4} 1/s, the rheometer reports increasingly noisy strain rates. To determine instantaneous effective viscosities η_{eff} , we therefore first smooth the strains using smoothing splines, and afterwards differentiate the

smoothed strains $\hat{\gamma}$ with respect to time to create smoothed strain rates $\hat{\dot{\gamma}}$. Smoothed effective (time-dependent) viscosities then become:

$$\hat{\eta}_{\text{eff}} = \frac{\sigma}{\hat{\dot{\gamma}}} \tag{22}$$

At each stress level we derive the elastic μ_e and an elastic shear modulus μ_a from the elastic and an elastic strains (see figure 4) determined from the relaxation curves:

 $\mu_{\rm e}(\sigma) = \frac{\sigma}{\gamma_{\rm e}} \qquad \mu_{\rm a}(\sigma) = \frac{\sigma}{\gamma_{\rm a}}$ (23)

7 Rheometry results

7.1 Stress-dependent viscosity

Figure 5 shows how the smoothed effective viscosities $\hat{\eta}_{\rm eff}$ for the six tested materials develop in time. Effective viscosity shows a relatively large increase during the start of the loading, which is most pronounced at lower stresses. In this phase the strain rates are dominated by transient, anelastic deformation. For those stresses where the average steady-state viscosity $\bar{\eta}$ is close to the observed effective viscosity $\hat{\eta}_{\rm eff}$ (depicted by a dot) the transient phase is over. We can still observe small (on logarithmic scale) variations in viscosity in the phase when only permanent deformation is occurring; here strain softening (decreasing viscosity) is observed for high stresses, strain hardening (increasing viscosity) for low stresses. For curves shown in grey there was an unknown amount of permanent deformation, as during the end of the chosen period for relaxation of the sample the strain rate was still too high (more than 10% of the final loading strain rate). All loading and relaxation strain curves are included in appendix A.

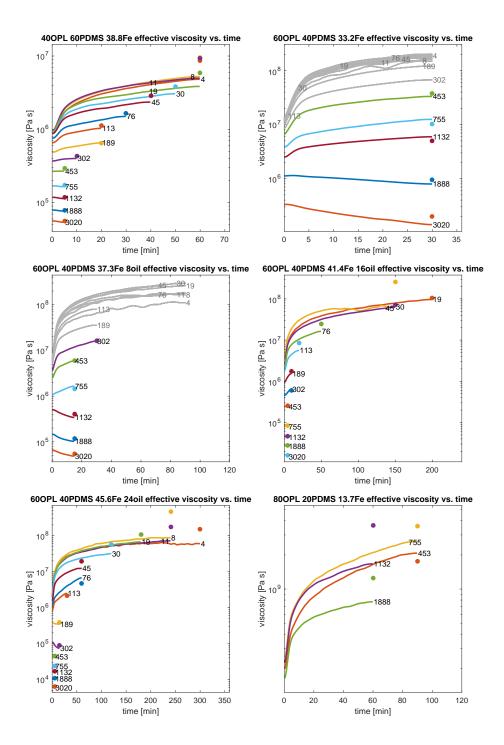


Figure 5: Effective smoothed viscosity $\hat{\eta}_{\text{eff}}$ versus time for the six tested materials. The coloured dot denotes the average viscosity based on the observed permanent strain $\bar{\eta}$. Grey lines stand for measurements for which we were not able to extract the permanent strain with sufficient confidence. Numbers stand for stress level in Pa.

As shown by figure 6, five of the six materials (all except the sample 80OPL 20PDMS) exhibit a clear nonlinear stress dependent viscosity in the applied stress ranges. The material with the lowest plasticine fraction (40OPL 60PDMS) shows the lowest zero-stress viscosity η_0 and the weakest stress nonlinearity, demonstrated by the power exponent n of 2. Increasing the plasticine content increases the power exponent, as expected: all materials having a plasticine to PDMS mass ratio of 60:40 have power exponents in the 3-4 range. Only for higher silicone oil contents (24% and 16% mass fractions) we are able to observe a transition from Newtonian to nonlinear steady state viscosity. The addition of the low-viscosity silicone oil lowers the viscosity over the whole stress range, but seems to have a minor influence on the power exponent n (comparable to the findings of Zulauf and Zulauf (2004)). For the 60% plasticine materials with no or little added silicone oil (8%) we have no good observations of permanent strain at the lower stress range (so we fit a power-law equation to the curves of these materials). At even higher fractions of plasticine (80%) we obtain only a weak stress-dependence of the viscosity. As the parallel-plate setup leads to slip between material and plates at stresses at the high end of our applied stress range, we are not able to determine a possible transition to nonlinear viscosities for this material at higher stresses. Table 4 summarizes the estimated parameters, next to the confidence intervals.

Material	n [-]	η_0 [Pa s]	σ_t [Pa]	$C \left[\mathrm{Pa}^{-n}/s \right]$	$\bar{\mu_{\rm e}}$ [Pa]	$\mu_{\rm a} \ [{\rm Pa}]$
40OPL 60PDMS 38.8Fe	2.0(1.9 - 2.1)	$1.10^7 (9.10^6 - 2.10^7)$	11.0(4.5 - 17.4)	$6.2 \cdot 10^{-9}$	$2.5 \cdot 10^4$	$1.2 \cdot 10^3 - 6.4 \cdot 10^4$
60OPL 40 PDMS 33.2 Fe	3.7(3.4 - 4.1)	-	-	$1.2 \cdot 10^{-15} (1.2 \cdot 10^{-16} - 1.3 \cdot 10^{-14})$	$3.9 \cdot 10^5$	$2.5 \cdot 10^4 - 5.1 \cdot 10^4$
$60\mathrm{OPL}\ 40\mathrm{PDMS}\ 37.3\mathrm{Fe}\ 8\mathrm{oil}$	3.6(3.3 - 3.8)	-	-	$3.0 \cdot 10^{-14} (7.0 \cdot 10^{-15} - 1.3 \cdot 10^{-13})$	$1.5 \cdot 10^{5}$	$1.4 \cdot 10^4 - 8.5 \cdot 10^4$
60OPL 40PDMS 41.4Fe 16oil	3.2(2.8 - 3.7)	$2 \cdot 10^8 (1 \cdot 10^1 - 5 \cdot 10^8)$	25.3(0.1 - 53.5)	$3.9 \cdot 10^{-12}$	$2.8 \cdot 10^4$	$4.6 \cdot 10^3 - 3.3 \cdot 10^5$
60OPL 40PDMS 45.6Fe 24oil	3.6(3.2 - 4.1)	$2 \cdot 10^8 (4 \cdot 10^7 - 4 \cdot 10^8)$	18.5(7.6 - 29.5)	$2.4 \cdot 10^{-12}$	$1.8 \cdot 10^4$	$2.8 \cdot 10^3 - 5.2 \cdot 10^5$
$80\mathrm{OPL}$ 20PDMS 13.7Fe	1	$2 \cdot 10^9 (3 \cdot 10^8 - 3 \cdot 10^9)$	-	-	$1.1 \cdot 10^6$	$1.2 \cdot 10^5 - 4.4 \cdot 10^5$

Table 4: Estimated parameters of steady-state viscosity as a function of stress (see figure 6): stress exponent n, zero-stress viscosity η_0 , transition stress σ_t , power-law material constant C, average elastic modulus μ_e and range of anelastic shear modulus μ_a . Values within brackets provide the 95% confidence intervals.

7.2 Elastic parameters

As figure 7 shows, the elastic shear modulus $\gamma_{\rm e}$ changes only little with stress, and decreases for larger fractions of PDMS and low viscosity silicone oil. Table 4 summarises the average shear modulus for each material. The anelastic shear modulus $\gamma_{\rm a}$ is stress dependent: while constant at those stress intervals where the steady-state viscosity is Newtonian, $\gamma_{\rm a}$ increases in the power-law viscosity stress domain, implying a decreasing amount of anelastic deformation at increasing stress.

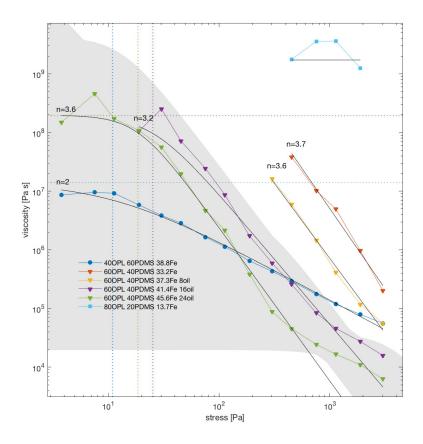


Figure 6: Steady state viscosities per stress level for the six tested materials (table 3) with a fit (equation (6)) to the theoretical models: power-law, Ellis and Newtonian rheologies. The grey area represents the desired range in viscosity vs. stress as described in section 4 for stress scale option I.

7.3 Temperature effects

To infer the viscosity sensitivity to temperature we test the material 60OPL 40PDMS 45.6Fe 24oil at 30°C. Figure 8 shows a non-uniform and modest reduction (on average 2) in viscosity for an increase in temperature from 20° to 30° C. The degree of nonlinearity, expressed by stress-exponent n is not significantly affected, see table 5.

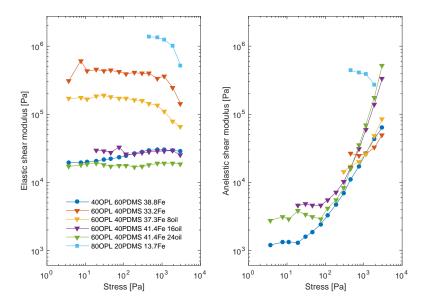


Figure 7: Shear moduli for all tested materials as function of applied stress. Left panel shows elastic shear modulus μ_e ; right panel shows anelastic shear modulus μ_a for the same stress levels as featured in figure 6.

Material	n [-]	η_0 [Pa s]	σ_t [Pa]	$\bar{\mu_{\rm e}}$ [Pa]	$\mu_{\rm a} \ [{\rm Pa}]$
60OPL 40PDMS 45.6Fe 24oil 20°C	3.6(3.2 - 4.1)	$2 \cdot 10^8 (4 \cdot 10^7 - 4 \cdot 10^8)$	18.5(7.6 - 29.5)	$1.8 \cdot 10^4$	$2.8 \cdot 10^3 - 5.2 \cdot 10^5$
$60\mathrm{OPL}$ 40PDMS $45.6\mathrm{Fe}$ 24 oil $30^{\circ}\mathrm{C}$	3.9(3.1 - 4.6)	$7 \cdot 10^7 (7 \cdot 10^6 - 1 \cdot 10^8)$			$1.3 \cdot 10^3 - 2.7 \cdot 10^5$
60OPL 40PDMS 37.0Fe 24oil	3.7(2.9 - 4.6)	$1.10^8(1.10^1 - 3.10^8)$	20.0(1.1 - 38.8)	$2.0 \cdot 10^4$	$2.4 \cdot 10^3 - 2.8 \cdot 10^5$
60OPL 40PDMS 45.6Fe 24oil	3.6(3.2 - 4.1)	$2 \cdot 10^8 (4 \cdot 10^7 - 4 \cdot 10^8)$	18.5(7.6 - 29.5)	$1.8 \cdot 10^4$	$2.8 \cdot 10^3 - 5.2 \cdot 10^5$
60OPL 40PDMS 92.0Fe 24oil	2.8(2.4 - 3.2)	$1 \cdot 10^8 (1 \cdot 10^1 - 3 \cdot 10^8)$	8.8(0.1 - 23.1)	$1.9 \cdot 10^4$	$2.2 \cdot 10^3 - 3.5 \cdot 10^5$

Table 5: Estimated parameters of steady-state viscosity as a function of stress for material 60OPL 40PDMS 45.6Fe 24oil at temperatures 20°C and 30°C (see figure 8) and for different iron filler contents (see figure 9): stress exponent n, zero-stress viscosity η_0 , transition stress σ_t , average elastic modulus μ_e and range of anelastic shear modulus μ_a . Values within brackets provide the 95% confidence intervals.

7.4 Iron filler effects

Up to this point we have adjusted the iron powder content to achieve a uniform density for all samples. For analogue models, variations in density are needed to accommodate naturally occurring density variations throughout the lithosphere. To that end we test the influence of significant iron filler content variations on the rheology. Figure 9 shows that large changes in iron powder contents, which we vary between 37 - 92% in mass, have relatively small effects on the rheology. A higher iron filler content seems to slightly lower the viscosity at stresses below 100 Pa, and slightly increase viscosity above. The power-law behavior is hardly

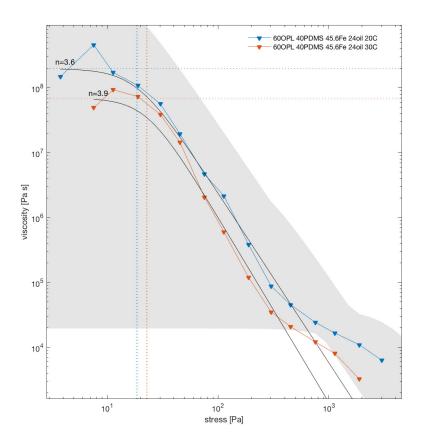


Figure 8: Steady state viscosities per stress level for 60OPL 40PDMS 45.6Fe 24oil at 20° C and 30° C with an Ellis fit (equation (6)). The grey area represents the desired range in viscosity vs. stress as described in section 4 for stress scale option I.

affected based on visual inspection of figure 9, but the estimated exponent n is lower for the 92.0 mass % iron sample. However, this can merely be attributed to a worse fit to the data, as the sharp kink in viscosity at around 20 Pa cannot be properly represented by the Ellis model.

8 Discussion analogue material rheology

In our search for analogs for the ductile deformation of olivine aggregates we find that the materials with a plasticine to PDMS mass ratio of 60:40 have the desired rheological characteristics. Firstly, the material shows a nonlinear stress dependence (expressed by a power exponent of around 3.5) that nicely

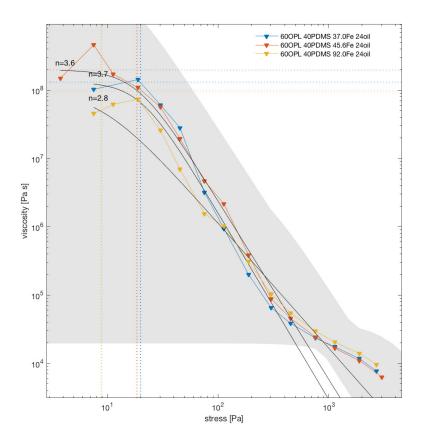


Figure 9: Steady state viscosities per stress level for 60OPL 40PDMS 24oil with three different iron filler ratio's, with an Ellis fit (equation (6)). The grey area represents the desired range in viscosity vs. stress as described in section 4 for stress scale option I

matches the stress dependence reported for olivine (in the range 3-4.9 e.g. (Mei and Kohlstedt, 2000a; Korenaga and Karato, 2008)). Secondly, the zero-stress viscosity of about 10^8 Pa s (for a 24% inclusion of silicone oil) allows a viscosity contrast with the asthenosphere in the intended range of $2 \cdot 10^4$ - $4 \cdot 10^{10}$ Pa s when glucose is used to represent the asthenosphere ($\eta_a \approx 10^2$ Pa s). The material can thus represent the average viscosity of that part of the lithosphere that deforms in a ductile regime. Furthermore, the transition stress σ_t (for 24% silicone oil added, see table 4) is in the appropriate range compared to scaled values for olivine: 2-300 Pa (see section 4). This indicates that the power-law behavior as witnessed in our materials can be effectuated in properly scaled laboratory models, as this transition occurs at realistic tectonic stresses. Finally,

we observe that elastic and anelastic strains are limited and that both elastic as well as anelastic shear modulus properly scale with natural values (target γ_e and γ_a shear moduli are approximately 10^4 and 10^3 - 10^4 Pa, respectively, section 4.1), which implies that the viscous behavior of the material is dominant for large deformation. All mixtures have self-adhesive properties, which prevents material boundaries to persist as weak zones in the final model assembly.

The occurrence of a transition from Newtonian behavior to power-law behavior, as described by Barnes (1999) has been subject of recent debate. The Newtonian plateau at low stress for Carbolpol solutions - a material that has also received attention to serve as an analogue model for tectonic studies, e.g. Schrank et al. (2008); Di Giuseppe et al. (2015) and figure 2 - observed by Barnes (1999) has been interpreted by Møller et al. (2009) as an experimental artefact (see also Denn and Bonn (2011); Dinkgreve et al. (2017)). These studies show that at low stresses the effective viscosity of a Carbopol solution increases with time, which these authors attribute to an elastic deformation rather than Newtonian steady-state viscosity. They claim that there exists a yield stress below which no flow occurs, and that marks the transition material transforms from a solid to a liquid state. For two of our tested materials (40OPL 60PDMS and 60OPL 40PDMS 24oil) we find permanent strain for the complete applied stress range, which means that flow occurs down to the lowest stress level. For these two materials we observe a Newtonian plateau at low stress, a similar behavior as Boutelier et al. (2016) find for the low-viscosity natrosol solutions ($\eta \approx 10^1 - 10^4$ Pas, see figure 2). The Ellis model fits well to the observed combined low-stress Newtonian and high-stress power-law viscosities, only at the highest stress levels the fit becomes less and the observations suggest a decrease in stress exponent.

9 Material application: slab break-off

To demonstrate the capability of our developed power-law material to create localized deformation in settings with significant variation of differential stress and a positive stress feedback, we apply the developed material in a model of continental collision and possible slab break-off (Davies and von Blanckenburg, 1995; Wong A Ton and Wortel, 1997). The subduction of oceanic lithosphere and the following closure of the ocean leads to the subsequent forced subduction of buoyant continental lithosphere. Whereas the negative buoyancy of oceanic lithosphere initially drives sinking of the oceanic slab into the asthenosphere, the positive buoyancy of continental will eventually halt subduction. Already in the early days of plate tectonic theory it has been proposed that when subduction has been brought to standstill, stresses in the subducted plate ultimately lead to break-off of the oceanic slab (McKenzie, 1969). Occurrence of slab break-off is supported by seismic tomographic images that show gaps at about 200 km depth in the downgoing slab, in several subduction zones of the Mediterranean (Spakman et al., 1988; Wortel and Spakman, 1992, 2000).

Numerical models suggest the detachment process takes place by necking of the slab due to tensile stresses, where localized thinning serves as separation mechanism between continental and oceanic lithospheres, with the subsequent sinking of the oceanic fragment (Gerya et al., 2004; Andrews and Billen, 2009; Van Hunen and Allen, 2011).

There are a few continental collision studies that make use of analogue models. We discriminate between models that describe the rheological properties of the applied materials as elasto-plastic, in the sense that these materials should not exhibit viscous behavior, and studies that use visco-elastic materials for modeling lithosphere deformation. Another distinction can be made between studies that use external forcing as driving forces and models that are driven solely by buoyancy forces (Schellart and Strak, 2016).

Analogue models using elasto-plastic materials and external forcing by a piston include studies by Chemenda et al. (1995); Boutelier et al. (2004) and Boutelier and Cruden (2017), where the latter two studies are set up as thermo-mechanical experiments and lead in some of the experiments to break-off of the oceanic slab. As far as we know, the model setup by Boutelier and Cruden (2017) (next to that of Boutelier and Oncken (2011)) is unique in the sense that the horizontal forces exerted by the piston are actually measured, and can thus be checked for proper dynamic scaling. The question however remains how the strain rate insensitive rheologies applied in these experiments can be related to viscous properties that are commonly attributed to the deeper lithosphere (e.g. Bürgmann and Dresen (2008)).

Regard et al. (2003, 2005, 2008) and Faccenna et al. (2006) use externally forced analogue models to study continental collision using Newtonian viscoelastic materials to represent the lower crust and upper mantle, next to an upper crust that deforms in a brittle manner. In the case the slab is allowed to penetrate the lower mantle, these models lead to detachment, in the form of distributed thinning. The distributed thinning, resulting in dripping of the oceanic lithosphere, may be resulting from both the Newtonian rheology (Schmalholz, 2011), as well as the relatively low viscosity contrast (two orders of magnitude) between the viscous lithosphere and asthenosphere (see for a comparable numerical result Pysklywec et al. (2000)). Guillaume et al. (2013) present a model that includes subduction of continental lithosphere, where the model is driven by buoyancy only and Newtonian viscous materials are applied. However, no break-off occurs after continental material has subducted. In the following we test whether it is possible to create localized slab break-off using the nonlinear viscoelastic material that we have developed.

10 Model setup

We set up a laboratory model of oceanic and continental subduction, that includes a nonlinear viscoelastic lithosphere and a Newtonian viscoelastic asthenosphere, see figure 10. We present two versions of the model: model I features a long oceanic plate that after subduction is supported by the bottom of the model tank (i.e. the upper mantle-lower mantle boundary) before the continent is forced to subduct; model II features a very short oceanic slab and a continental slab, designed such that after subduction the slab will hang freely in the surrounding asthenosphere (comparable to the numerical model by Tscharner et al. (2014)), while the bottom of the tank still coincides with the upper to lower mantle transition. We enforce a no-displacement boundary condition at the continental end of the plates for both models. Both models have been performed at a room temperature of 20° C.

We use the organic plasticine-PDMS-silicone oil-iron powder mixture 60OPL 40PDMS 24oil to create our oceanic and continental lithospheres, whereas we use glucose with a Newtonian rheology for the low viscous upper mantle. Densities of the oceanic and continental lithospheres are representing bulk lithospheric densities, where we use varying iron powder content to correct densities (see table 6). We base the density differences between each lithospheric unit and the asthenosphere on Cloos (1993), see equation (11). Table 6 provides all material properties and scale factors between model and nature, complemented by realistic natural ranges from relevant literature. The table shows that elastic shear moduli $\mu_{\rm e}$ for the lithosphere are of the right magnitude compared to natural values; anelastic shear moduli $\mu_{\rm a}$ (down to one order of magnitude lower than $\mu_{\rm e}$) are in the range of estimated natural values (Chopra, 1997). All model images have been corrected for lens distortion and perspective distortion, such that all views are perpendicular to the model axes, and straight lines in the model are indeed depicted as straight lines in the images.

11 Results slab break-off model

After we initiate subduction by pulling glucose over the first few cm of the oceanic plate, subduction progresses in both models by roll-back of the slab and retreat of the trench. In Model I the slab touches the bottom of the tank, representing the upper - lower mantle discontinuity, and all negatively buoyant oceanic lithosphere subducts, which finally forces subduction of a part of the buoyant continental lithosphere as well (figure 11). The subduction velocity slows down to zero, and while thinning around the ocean-continent margin takes place, this deformation stalls without causing separation between the oceanic and continental parts of the lithosphere. We left the model intact for another 8 days (representing $4 \cdot 10^9$ scaled years), however no significant deformation occurred within this time frame (see movie S1 in our online repository).

Table 6: Geometric and rheological parameters used in model I and model II, including scaled values and natural values. $Density \star Reference$ densities of the asthenosphere, and continental and oceanic lithosphere are based on Cloos (1993), where the lithospheric densities represent bulk density. The oceanic lithospheric density of 3310 kg m⁻³ ($\Delta \rho_{ol-a} = 80 \text{ kg m}^{-3}$) assumes an oceanic crust that has been transformed into eclogite during subduction. Values within brackets denote values likely for lithosphere around a continental margin. • To arrive at these densities, 37% and 92% mass fraction iron powder (3.6 and 8.7 volume % respectively) has been used for continental and oceanic lithospheric materials (with 60% organic plasticine, 40% PDMS and 24% silicone oil in terms of mass). As thenosphere viscosity \dagger Asthenosphere steady-state viscosities estimates for subduction zones are commonly on the order of 10¹⁹ Pa s, based on observations of postseismic deformation (generally estimates are in the range $[10^{18} - 10^{20}]$ see table 1 of Wang (2007)). This viscosity estimate is in line with results from recent studies that interpret postseismic deformation after recent megathrust earthquakes, such as the 2004 Sumatra-Andaman earthquake (Hoechner et al., 2011; Broerse et al., 2015), the 2010 Maule earthquake (Klein et al., 2016; Bedford et al., 2016) or 2011 Tohoku earthquake (Hu et al., 2016). However, the asthenosphere viscosities reported for subduction zones may be more applicable to the mantle wedge than the sub-oceanic asthenosphere Hirth and Kohlstedt (2003), for which we adopt the higher value of $\approx 10^{20}$ Pa s (Hu et al., 2004; Broerse et al., 2015). Analogue material rheology ‡ Viscosity at 20° C according to manufacturer specifications. Lithosphere rheology • The estimate for the mean zero-stress mean viscosity of the ductile lithosphere η_{0_1} is based on diffusion creep parameters from Hirth and Kohlstedt (2003), see explanation in section 2. * Estimates for power-law exponent n for olivine range from 3 to 4.9 (Karato and Wu, 1993; Korenaga and Karato, 2008).

The transition stress σ_t has a large range, but correlates strongly with $\bar{\eta_0}$, see figure 1. High values of σ_t apply to low values of $\bar{\eta_0}$. Elastic parameters \blacklozenge mantle lithosphere shear modulus from PREM (Dziewonski and Anderson, 1981). § Anelastic shear modulus values from Chopra (1997) (table 2).

	model values		scaled val	ues	range in nature
	model I	model II	model I	model II	
Geometry					
$l_{ m n}/l_{ m m}$	$6.60 \cdot 10^6$	$3.30 \cdot 10^6$			
$d_{\rm a}~[{ m m}]$	$100 \cdot 10^{-3}$	$200 - \cdot 10^{-3}$	$660 \cdot 10^3$	$660 \cdot 10^3$	$660 \cdot 10^3$
$d_{ m cl}$	16.10^{-3}	$32 \cdot 10^{-3}$	$105 \cdot 10^3$	$105 \cdot 10^3$	
$d_{ m ol}$	$12 \cdot 10^{-3}$	$20 \cdot 10^{-3}$	79.10^{3}	66.10^{3}	$0 - 110 \cdot 10^3$
w_1	$100 \cdot 10^{-3}$	$160 \cdot 10^{-3}$	$660 \cdot 10^3$	$528 \cdot 10^3$	
$l_{ m cl}$	$150 \cdot 10^{-3}$	$100 \cdot 10^{-3}$	$990 \cdot 10^{3}$	$330 \cdot 10^3$	
$l_{ m ol}$	$200 \cdot 10^{-3}$	$80 \cdot 10^{-3}$	$1320 \cdot 10^3$	$264 \cdot 10^3$	
Density					
$\rho_{\rm a} [{\rm kg \ m^{-3}}]$	1405	1405			3230 ★
$ ho_{ m cl}$	1241	1241 0			3140 ★
$ ho_{ m ol}$	1603	1603 ∘			3310 ★
$\Delta ho_{ m n}/\Delta ho_{ m m}$	0.48	0.48			
$\Delta ho_{ m cl-a}$	-164	-164	-79	-79	-9070 (-90)
$\Delta ho_{ ext{ol-a}}$	198	198	95	95	-330 - 170 (80) *

Table 7: continuation of table 6

	model values		scaled values		range in nature
	model I	model II	model I	model II	
Rheology					
$\eta_{ m n}/\eta_{ m m}$	$6.25 \cdot 10^{17}$	$6.25 \cdot 10^{17}$			
$\eta_{\rm a} [{\rm Pa s}]$	$1.6 \cdot 10^{2}$	$1.6 \cdot 10^{2}$			$10^{18} - 10^{20} \dagger$
$\eta_{0_1}^-$	$2.0 \cdot 10^{8}$	$2.0 \cdot 10^{8}$	$1.2 \cdot 10^{26}$	$1.2 \cdot 10^{26}$	$2 \cdot 10^{22} - 4 \cdot 10^{28} \bullet$
$n_{ m n}/n_{ m m}$	1	1			
n_1 []	3.6	3.6	3.6	3.6	3 - 4.9 *
$\sigma_{ m n}/\sigma_{ m m}$	$3.17 \cdot 10^{6}$	$1.58 \cdot 10^{6}$			
σ_{t_1} [Pa]	20	20	$6.3 \cdot 10^{7}$	$3.1 \cdot 10^{7}$	$2 \cdot 10^6 - 3 \cdot 10^9 \diamond$
$ar{\mu_{ m e}}$	$1.8 \cdot 10^{4}$	$1.8 \cdot 10^{4}$	$5.7 \cdot 10^{10}$	$2.8 \cdot 10^{10}$	$6.7 \cdot 10^{10} \ lack$
$\min(\mu_{\mathrm{a}})$	$2.8 \cdot 10^{3}$	$2.8 \cdot 10^{3}$	$8.8 \cdot 10^{9}$	$4.4 \cdot 10^{9}$	$2 \cdot 10^9 - 37 \cdot 10^9 $ §
Time					
$t_{ m n}/t_{ m m}$	$1.97\cdot10^{11}$	$3.95 \cdot 10^{11}$			
model minutes:			$3.75 \cdot 10^5$	$7.50 \cdot 10^5$	
nature years					

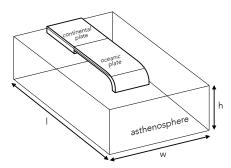


Figure 10: Schematic setup of the initial stage of the slab breakoff model. Model contains a lithosphere consisting of an oceanic part and a continental part overlying the asthenosphere that sits in a tank of with height h, width w and length l (for model dimensions see table 6).

In Model II (figure 12), the buoyant continental lithosphere is forced to subduct before the oceanic slab touches the bottom of the tank, as the oceanic slab is significantly shorter than the upper mantle depth. Subsequently, around the time that the trench retreat velocity approaches zero (at 20 minutes/15 \cdot 10⁶ scaled years), the subducted part of the lithosphere is fully suspended from the continental lithosphere at the surface of the model. At the same time, a tear initiates at the right slab edge, just above the oceanic - continental transition, at an equivalent depth of 280 km. While slab break-off commences with the edge tear, the break-off process is complemented by a slab window and wholesale thinning at the same depth, as visible in figure 12 from the frontal and side views. The total break-off period is 20 minutes, which represents 15 Myr in nature. As the two lithospheric domains have slightly different colors we can observe that the final tear occurred in the continental domain, just updip of the oceanic - continental transition. After break-off the continental lithosphere rebounds to the surface, while the oceanic lithosphere subducts further until it rests on the bottom of the model upper mantle. For both models holds that the Reynolds number is on the order of 10^{-4} , which justifies the assumption that inertia can be neglected.

12 Discussion slab break-off model

The absence of break-off in Model I and the occurrence of break-off in Model II suggests that the support of the slab by the bottom of the model tank highly reduces the stresses in the slab. The resulting increase in effective viscosity in the subducting slab subsequently prevents significant deformation. Van Hunen and Allen (2011) show results from a numerical study of continental collision and subsequent slab-breakoff, where the oceanic plate that is supported from below by the lower mantle needs considerable time to become thermally weakened before the viscosity becomes low enough to allow tearing. The location of tearing in Model II is where one would expect the largest tensile stresses for a stationary and fully hanging slab: at the transition between buoyant and negatively buoyant lithosphere.

The break-off in Model II is highly localized around the continental-oceanic transition, and the sinking oceanic plate behaves plate-like and does not drip-off, such as is observed in analogue models of continental collision where the ductile lithosphere has been modeled using Newtonian materials (Regard et al., 2003). We thus account the localization of deformation to the power-law rheology of the lithosphere materials. This interpretation is supported by the analytical study of slab detachment of Schmalholz (2011) that tests the influence of Newtonian or power-law slab rheologies on detachment modes. For Model II it is straightforward to approximate the tensile stresses at the continental-oceanic transition at the time before thinning commences, by calculating the buoyancy forces resulting from the hanging oceanic slab. As the rollback has ended we can assume frictional forces exerted by mantle flow can be neglected. The stress

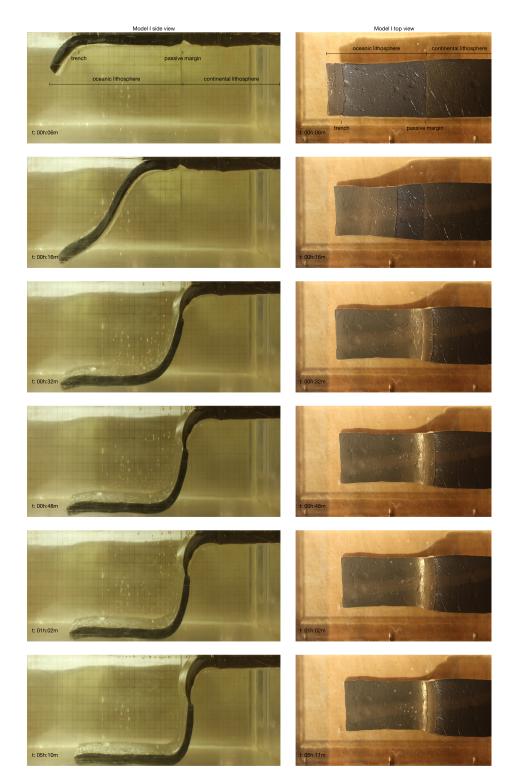


Figure 11: Snapshots of side and top views of Model I. Note the four hour step in time between the two last epochs. The location of the continental and oceanic lithospheric units, as well as the passive margin and the initial location of the trench are depicted in the upper panels.

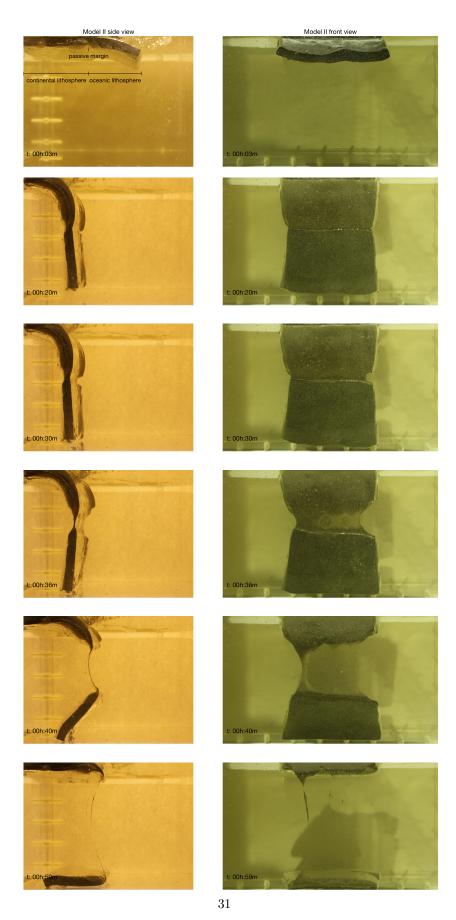


Figure 12: Snapshots of side and frontal views of Model II. The upper left panel describes the location of the continental and oceanic lithospheric units, as well as the location of the passive margin.

at this location and time can be approximated by: $\sigma = g l_{\rm ol} \Delta \rho_{ol-a}$ which is about 155 Pa. From figure 6 it shows that at this stress the material rheology is in the nonlinear regime, and resulting viscosity is about $7 \cdot 10^5$ Pa s.

Recently, Boutelier and Cruden (2017) published results on a thermal-mechanical laboratory model of slab break-off, but as they describe the model lithospheric materials as exhibiting a elasto-plastic rheology (i.e. yielding at a specific stress, without viscosity) the results are difficult to interpret knowing that the lithosphere likely has a viscosity. Our results however show that localization can occur in a properly dynamically scaled analogue model using a power-law material, in a tectonic setting without adverse boundary effects such as in previous studies of the influence of rheological nonlinearity on shear localization (e.g. Schrank et al. (2008)).

13 Conclusions

We have created new materials to represent the lithospheric mantle in analogue models, based on silicone polymers and organic plasticine using creep and recovery tests.

Combinations of organic plasticine and PDMS exhibit deformation rates with a nonlinear dependence on stress. If mixed in a mass ratio 60:40, the plasticine-PDMS mixture behaves as a power-law viscoelastic material with a stress exponent of around 3.5, which fits in the range of reported values for olivine aggregates. At low stresses we observe that the material shows Newtonian behavior, with a transition to power-law viscosity beyond a critical stress level, comparable to composite flow laws for olivine creep that include both Newtonian diffusion and nonlinear dislocation creep. Addition of low-viscosity silicone oil is effective in lowering the material viscosities to suitable values that allow the use of the material in scale models of the ductile lithosphere.

Application of the developed nonlinear material in a model of forced subduction of continental lithosphere, leads to highly localized deformation and ultimately resulting in slab break-off. We find that slab break-off is highly promoted for slabs that are not supported from below by the lower mantle.

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A Strain plots

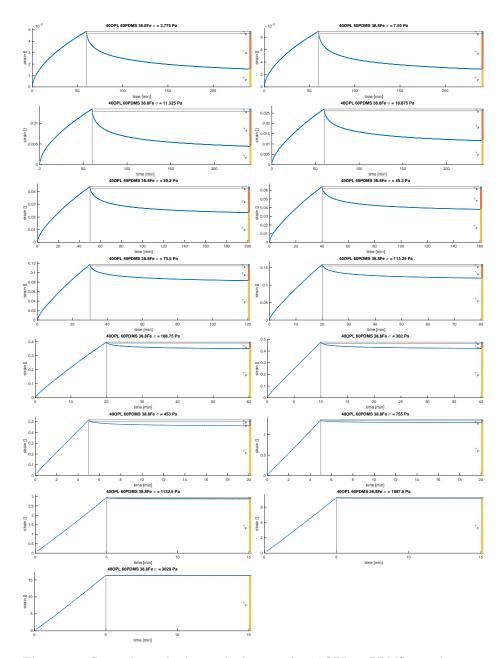


Figure 13: Strain during loading and relaxation for 40 OPL 60 PDMS at each stress level, with elastic strain γ_e , anelastic strain γ_a and permanent strain γ_p . Permanent strain is used to compute the mean strain rate and steady-state viscosity $\bar{\eta}$ (equation (20)).

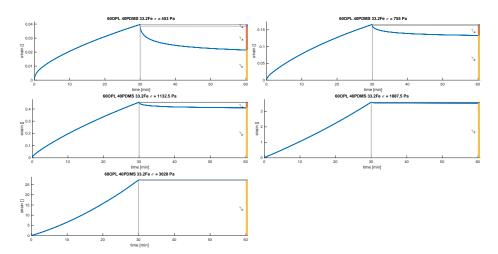


Figure 14: Strain during loading and relaxation for 60 OPL 40 PDMS at each stress level, with elastic strain γ_e , anelastic strain γ_a and permanent strain γ_p . Permanent strain is used to compute the mean strain rate and steady-state viscosity $\bar{\eta}$ (equation (20)).

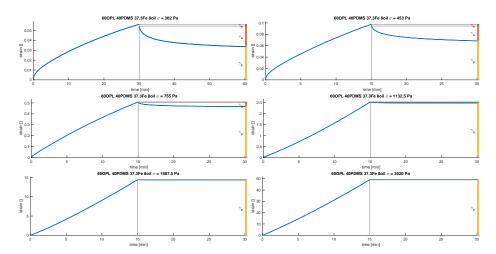


Figure 15: Strain during loading and relaxation for 60 OPL 40 PDMS 8 oil at each stress level, with elastic strain γ_e , anelastic strain γ_a and permanent strain γ_p . Permanent strain is used to compute the mean strain rate and steady-state viscosity $\bar{\eta}$ (equation (20)).

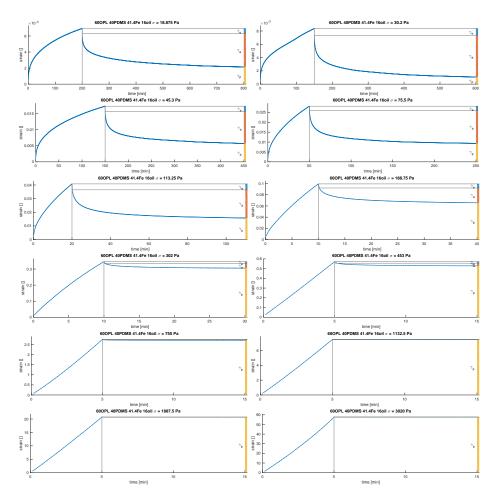


Figure 16: Strain during loading and relaxation for 60 OPL 40 PDMS 16 oil at each stress level, with elastic strain γ_e , anelastic strain γ_a and permanent strain γ_p . Permanent strain is used to compute the mean strain rate and steady-state viscosity $\bar{\eta}$ (equation (20)).

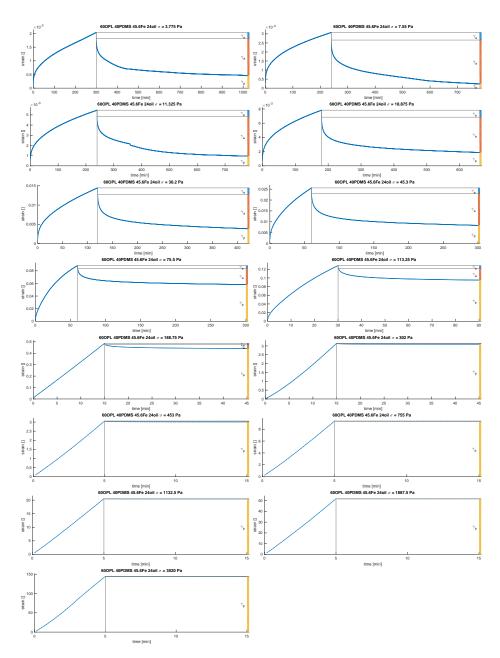


Figure 17: Strain during loading and relaxation for 60 OPL 40 PDMS 24 oil 45.6 Fe at each stress level, with elastic strain γ_e , anelastic strain γ_a and permanent strain γ_p . Permanent strain is used to compute the mean strain rate and steady-state viscosity $\bar{\eta}$ (equation (20)).

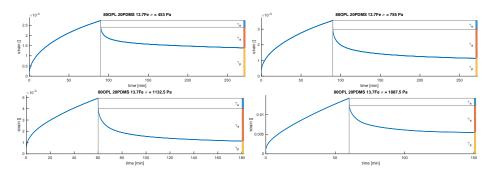


Figure 18: Strain during loading and relaxation for 80 OPL 20 PDMS at each stress level, with elastic strain γ_e , anelastic strain γ_a and permanent strain γ_p . Permanent strain is used to compute the mean strain rate and steady-state viscosity $\bar{\eta}$ (equation (20)).

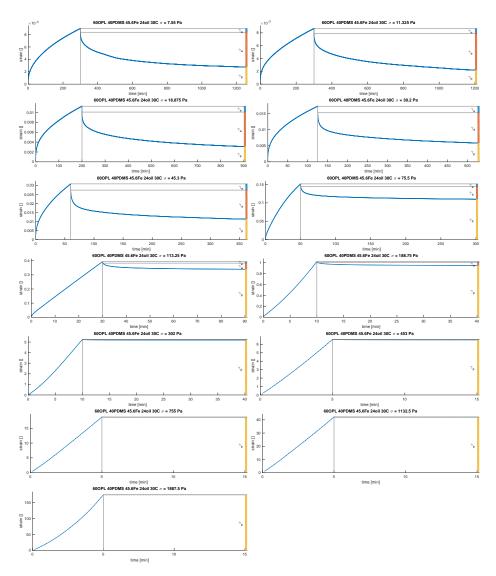


Figure 19: Strain during loading and relaxation for 60 OPL 40 PDMS 24 oil @ T = 30°C at each stress level, with elastic strain γ_e , anelastic strain γ_a and permanent strain γ_p . Permanent strain is used to compute the mean strain rate and steady-state viscosity $\bar{\eta}$ (equation (20)).

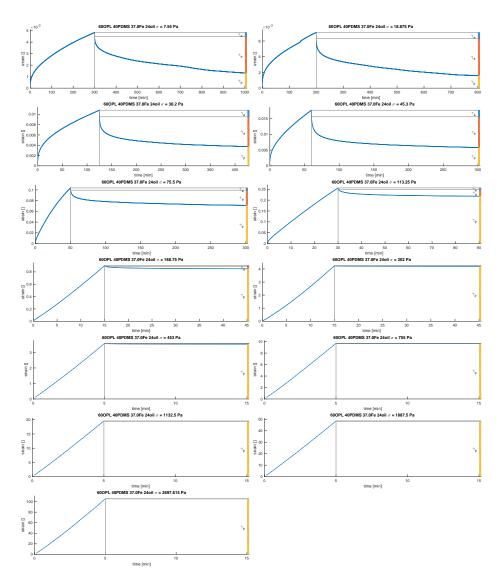


Figure 20: Strain during loading and relaxation for 60 OPL 40 PDMS 24 oil 37 Fe at each stress level, with elastic strain γ_e , anelastic strain γ_a and permanent strain γ_p . Permanent strain is used to compute the mean strain rate and steady-state viscosity $\bar{\eta}$ (equation (20)).

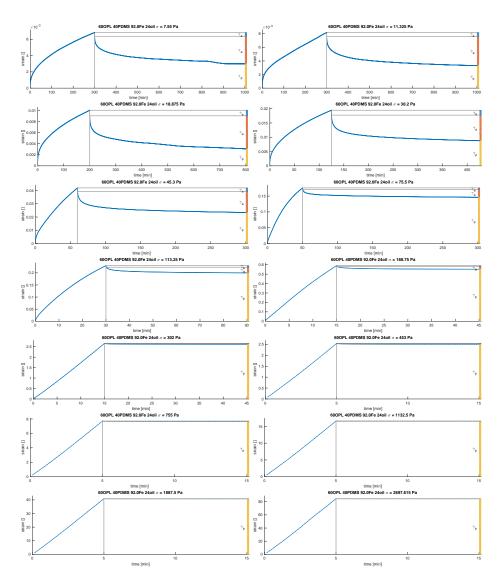


Figure 21: Strain during loading and relaxation for 60 OPL 40 PDMS 24 oil 92 Fe at each stress level, with elastic strain γ_e , anelastic strain γ_a and permanent strain γ_p . Permanent strain is used to compute the mean strain rate and steady-state viscosity $\bar{\eta}$ (equation (20)).

B Supplement: Average lithosphere rheology

	parameters	dry	wet	dry	wet
		diffusion		dislocation	
A	pre-exponential factor $[Pa^{-n}\mu m^p s^{-1}]$	1500	1	$1.1 \cdot 10^{-16}$	9.10^{-20}
p	grain size exponent [-]	3	3	0	0
C_{OH}	water content [ppm-H/Si]	0	1000	0	1000
r	$C_{\rm OH}$ exponent [-]	0	1	0	1.2
E	activation energy [J/mol]	$375 \cdot 10^{3}$	$335 \cdot 10^{3}$	$530 \cdot 10^3$	$480 \cdot 10^3$
V	activation volume [m ³ /mol]	5.10^{-6}	4.10^{-6}	$23 \cdot 10^{-6}$	11.10^{-6}
n	stress exponent [-]	1	1	3.5	3.5
C'/C	scale factor [-]	1.5	1.5	5.92	5.92
ρ_{crust}	density of the crust [kg/m ³]	2700			
$ ho_{ m mantle}$	density of the mantle $[kg/m^3]$	3300			
g	gravitational acceleration $[m/s^2]$	9.8			

Table 8: Rheological parameters used in the equation $C = A d^{-p} C_{\rm OH}^r \exp\left(-\frac{E+PV}{RT}\right)$ derived from table 1 of Hirth and Kohlstedt (2003). We apply a correction to C for parameters based on uni-axial experiments: $C' = C \frac{2}{3(n+1)/2}$ (Ranalli, 1995). Grain size d is in μ m, stress σ as well as pressure P are in Pa. Temperature T is in degrees Kelvin. Gas constant R is 8.314. We calculate lithostatic pressure P using the values listed for ρ and g.

Figure 22: Average lithosphere rheology using wet and dry olivine diffusion and dislocation creep rheological parameters from Hirth and Kohlstedt (2003) applied to continental and oceanic lithospheres (see section 2), left panels: η_0 for different values of grain size d, dotted lines denote average η_0 ; right panels: σ_t , dotted lines denote values below the LAB defined by $\eta_0 = 10^{20}$ Pa s. The top depth represents the brittle-plastic transition defined by the intersection of the Goetze criterion and the ductile stress at a strain rate of 10^{-14} s⁻¹.

