

1 Pre-print Statement for

2 **“Fe<sub>5</sub>S<sub>2</sub> identified as a host for sulfur in Earth’s core”**

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15  
16 Dear Earth Arxiv,

17 This manuscript is a non-peer reviewed preprint submitted to *EarthArXiv*. It has been  
18 submitted for peer review at *Nature Geoscience*.

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21 Sincerely,  
22 Claire Zurkowski

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# 35 **Fe<sub>5</sub>S<sub>2</sub> identified as a host for sulfur in Earth's core**

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## 48 **Abstract**

49  
50 Planetary habitability, as we experience on Earth, is linked to a functioning geodynamo  
51 which is in part driven by the crystallization of the liquid iron-nickel-alloy core as a planet cools  
52 over time. Cosmochemical considerations suggest that sulfur is a candidate light alloying  
53 element in rocky planetary cores of varying sizes and oxidation states; such that, iron sulfide  
54 phase relations at extreme conditions contribute to outer core thermochemical convection and  
55 inner core crystallization in a wide range of planetary bodies. Here we experimentally investigate  
56 the structural properties of the Fe-S system and report the discovery of the sulfide, Fe<sub>5</sub>S<sub>2</sub>,  
57 crystallizing in equilibrium with iron at Earth's outer core pressures and high temperatures.  
58 Using single-crystal X-ray diffraction techniques, Fe<sub>5</sub>S<sub>2</sub> was determined to adopt the complex  
59 Ni<sub>5</sub>As<sub>2</sub>-type structure (*P*6<sub>3</sub>*cm*, *Z* = 6). These results conclude that Fe<sub>5</sub>S<sub>2</sub> is likely to crystallize at  
60 the interface of Earth's core and mantle and will begin to crystallize during the freezing out of  
61 Earth and Venus' core overtime. The increased metal-metal bonding measured in Fe<sub>5</sub>S<sub>2</sub>  
62 compared to the other high *P-T* iron sulfides may contribute to signatures of higher conductivity  
63 from regions of Fe<sub>5</sub>S<sub>2</sub> crystallization. Fe<sub>5</sub>S<sub>2</sub> could serve as a host for Ni and Si as has been

64 observed in the related meteoritic phase, perryite,  $(\text{Fe, Ni})_8(\text{P, Si})_3$ , adding intricacies to  
65 elemental partitioning during inner core crystallization. The stability of  $\text{Fe}_5\text{S}_2$  presented here is  
66 key to understanding the role of sulfur in the multicomponent crystallization sequences that drive  
67 the geodynamics and dictate the structures of Earth and rocky planetary cores.

68

## 69 **Introduction**

70 Earth and the other terrestrial planets along with rocky exoplanets that continue to be  
71 discovered are composed of silicate mantles and iron-rich metallic cores (e.g. Birch 1952; Scott  
72 and Wasson 1975; Jephcoat and Olsen 1987; McDonough and Sun 1995; Rubie et al. 2011). The  
73 likelihood of planetary habitability as we know it demands the presence of a geomagnetic field,  
74 sustained by a convecting metallic liquid core, that protects the planet from harmful cosmic rays  
75 (e.g. Buffet, 2000). In Earth's liquid outer core, the presence of light alloying elements  
76 introduces complex pressure-temperature-dependent melting and crystallization thermodynamics  
77 that drives core convection (Fearn and Loper, 1981; Stevenson 1981; 1988; Nimmo 2015).

78 While there remains no method to directly sample Earth's or any terrestrial planetary core, iron  
79 meteorites are recognized as relics of the cores of planetesimal building blocks of terrestrial  
80 planets, and have long provided insight into the cosmochemically abundant light elements  
81 contributing to planetary core dynamics (e.g. Scott and Wasson 1975; Malvin et al. 1984). Of the  
82 candidate core-alloying elements such as Si, O, S, C, and H, sulfur is present in nearly all iron  
83 meteorites, suggesting that sulfur is a core alloying element in rocky planets with varying sizes,  
84 oxidation states, and formation histories (Scott and Wasson 1975; Jones and Drake, 1983;  
85 Kruijer et al. 2014). Examination of the structural properties of iron sulfides at high pressures

86 and temperatures ( $P$ - $T$ ) is therefore critical to ascertaining the chemistry and thermodynamics of  
87 Earth, terrestrial and exoplanetary cores.

88 At present, the crystallographic assessment of iron sulfides at Earth and planetary core  
89 conditions remains limited, even though properties such as density and elasticity are directly  
90 related to the atomic arrangement of core-crystallizing phases and are critical for assessing the  
91 seismic and dynamic observations of planetary cores. Previous work on Fe-rich systems indicate  
92 that tetragonal  $\text{Fe}_3\text{S}$  ( $\text{Fe}_3\text{P}$ -type structure) is stable from Martian core conditions to Earth's outer  
93 core conditions (Fei et al. 2000; Kamada et al. 2010; Ozawa et al. 2013; Mori et al. 2017), and an  
94 orthorhombic  $\text{Fe}_2\text{S}$  ( $\text{Co}_2\text{Si}$ -type structure) phase becomes stable near Earth's inner-core boundary  
95 (ICB) conditions (Tateno et al. 2019). The presence of Si has also been observed to stabilize the  
96  $\text{Fe}_2(\text{S}, \text{Si})$  chemistry over  $\text{Fe}_3\text{S}$  at Mercurian core-mantle-boundary (CMB) conditions (Tao and  
97 Fei 2021). These reports were predominantly based powder X-ray diffraction studies at high  
98 pressures and temperatures and chemical analyses of recovered samples. However, recent work  
99 has demonstrated that the iron-sulfur phase diagram is more complex at high  $P$ - $T$ , but that this  
100 complexity requires a superior method for accurately characterizing crystal structures at extreme  
101 conditions. Using single-crystal X-ray diffraction techniques on multigrain samples,  $\text{Co}_2\text{P}$ -type  
102  $\text{Fe}_2\text{S}$  ( $Pnma$ ,  $Z = 4$ ) was recently identified and characterized at 90 GPa (Zurkowski et al. *in*  
103 *press*). It remains, however, that these methods have not been applied to iron-sulfide phases  
104 synthesized at high temperatures above 90 GPa, leaving the role of sulfur in the crystallization  
105 sequences occurring in Venus, Earth and larger rocky exoplanets open to further discovery.

106

107

## 108 **Results and Discussion**

109           In the current study, Fe-rich Fe-S compositions were probed using both single-crystal and  
110 powder X-ray diffraction techniques in a laser-heated diamond anvil cell to 200 GPa and 3300 K.  
111 At high temperatures between 120 and 200 GPa, spanning Earth's CMB to mid-outer-core  
112 conditions, crystallization of hcp-Fe (Figure 1a) is observed in the diffraction patterns along with  
113 diffuse streaks and sets of closely spaced reciprocal nodes (Figure 1a, Figure S1), suggesting a  
114 complex atomic arrangement of the coexisting sulfide. Upon temperature quenching in this  
115 pressure range, diffraction images were collected while rotating the DAC across a +/- 17–30°  
116 range (depending on the DAC configuration). Grains of a hexagonal lattice were identified in the  
117 reciprocal space with indexed parameters:  $a = 5.979(3) \text{ \AA}$ ,  $11.088(6) \text{ \AA}$  at 140(2) GPa and  
118 3070(180) K (Table S1). Assessment of the systematic absences for the structure factors reduced  
119 for this hexagonal sulfide suggests a  $P6_3cm$  space group, and structural solution and positional  
120 and displacement parameter refinement converged to the  $\text{Ni}_5\text{As}_2$ -type  $\text{Fe}_5\text{S}_2$  ( $Z=6$ ) (Table S2, S3;  
121 Figure 2) (Oryshchyn et al. 2011). The  $\text{Ni}_5\text{As}_2$ -type  $\text{Fe}_5\text{S}_2$  is derived from the  $\text{Pb}_5\text{Sb}_2$  structure  
122 (El-Boragy et al. 1970) where the Fe6 site (Table S3) is split about its position and given half  
123 occupancy, changing its Wyckoff site from  $6c$  to  $12d$  (Oryshchyn et al. 2011) (Figure 2, Table  
124 S3, Appendix A1). Within error, all sites are fully occupied, except the Fe6 site that is ascribed  
125 half occupancy (Oryshchyn et al. 2011).

126           Additionally, two polytypes associated with stacking variations along the  $c$  direction were  
127 measured at lower temperatures and in stiff pressure media (Table S1). The  $\text{Fe}_5\text{S}_2$  polytypism is  
128 accompanied by observations of diffuse scattering between reciprocal nodes along the  $c$  direction  
129 (Figure S1), indicating positional disorder as the atoms configure at high temperatures.  
130 Analogous polytypes have been reported in related phases such as  $\text{Pb}_5\text{Sb}_2$  (Saini et al. 1964).

131 X-ray diffraction from seven heating cycles performed between 100 and 200 GPa and to  
132 3300 K provide insight into the Fe-rich Fe-S phase relations at outer core pressures and to high  
133 temperatures. In each heating experiment, temperatures near melting were attained, and lattices  
134 of Fe<sub>5</sub>S<sub>2</sub> were indexed in the reciprocal space upon quenching. By further probing locations  
135 across the thermal gradient of the laser heated spots, additional Fe-sulfides were characterized,  
136 offering information on the lower temperature Fe-saturated sulfide crystal chemistries (Table S1,  
137 S2). These include the Fe<sub>3</sub>P-type Fe<sub>3</sub>S (*I*-4, *Z* = 8) phase (Fei et al. 2000; Seagle et al. 2006;  
138 Morard et al. 2007; Kamada et al. 2010; Kamada et al. 2012; Thompson et al. 2020), the *C*23  
139 Fe<sub>2</sub>S (*Co*<sub>2</sub>P-type, *Pnma*, *Z* = 4) phase, and the *C*37 Fe<sub>2</sub>S (*Co*<sub>2</sub>Si-type, *Pnma*, *Z*=4) phase, in lower  
140 temperature regions at 119 GPa, 131 GPa, and 140 GPa, respectively (Figure 3; Table S1, S2).  
141 Notably, identification and crystal-structure analysis of Fe<sub>5</sub>S<sub>2</sub> and Fe<sub>3</sub>S in these Fe-saturated  
142 experiments clarifies that Fe<sub>5</sub>S<sub>2</sub>, not Fe<sub>3</sub>S is stable on the solidus above 120 GPa, despite reports  
143 from previous powder diffraction studies (e.g. Kamada et al. 2010; Kamada et al. 2012; Ozawa  
144 et al. 2013; Mori et al. 2017).

145 Comparison of the interatomic distances and coordination environments in the Fe-  
146 sulfides observed in this study separates Fe<sub>5</sub>S<sub>2</sub> as particularly unique among them. Within an  
147 interatomic distance of 2.7 Å (Table S4, Appendix A1-A4), *C*23 and *C*37 Fe<sub>2</sub>S consist of Fe sites  
148 coordinated only by S, Fe<sub>3</sub>S marks an increase in Fe-Fe bonding with Fe surrounded by up to 9  
149 Fe sites, and Fe<sub>5</sub>S<sub>2</sub> exhibits Fe surrounded by up to 9 Fe sites with Fe-Fe bond lengths up to 25%  
150 shorter in Fe<sub>5</sub>S<sub>2</sub> compared to Fe<sub>3</sub>S (Table S4). Following previous descriptions of related *M*<sub>5</sub>*X*<sub>2</sub>  
151 phases (e.g. (Kjekshus et al. 1973; Oryshchyn et al. 2011), Fe<sub>5</sub>S<sub>2</sub> can be viewed as an  
152 arrangement of 6 Fe sites and 3 S sites with the Fe1, Fe2, Fe4, and Fe6 sites in 13-fold

153 coordination, the Fe3 and Fe5 sites in 12-fold coordination, and the S sites in 10-fold  
154 coordination (Figure 2c, Table S4). All sites are coordinated by both Fe and S sites.

155 The presented single-crystal derived structures of Fe<sub>5</sub>S<sub>2</sub>, Fe<sub>3</sub>S, C23 Fe<sub>2</sub>S, and C37 Fe<sub>2</sub>S  
156 were then used to interpret the changes in powder X-ray diffraction patterns collected during  
157 heating between 110 and 200 GPa and to ~3300 K (Figure 4, Figure S4). C23 Fe<sub>2</sub>S is observed  
158 coexisting with Fe to ~125 GPa and up to ~2300 K. In combination with previous studies, C23  
159 Fe<sub>2</sub>S is stable at moderate temperatures between 25 and 125 GPa (Zurkowski et al. *in press*,  
160 Zurkowski et al. submitted). A pressure-induced C23–C37 Fe<sub>2</sub>S transition is constrained between  
161 125 and 135 GPa, in agreement with previous investigations of Fe<sub>2</sub>S (Zurkowski et al.  
162 submitted). C37 Fe<sub>2</sub>S is stable between 130–200 GPa and to ~3000 K in this study, and previous  
163 work suggests that C37 Fe<sub>2</sub>S may remain stable to 306 GPa at high temperatures (Tateno et al.  
164 2019). Tetragonal Fe<sub>3</sub>S forms from the reaction of C23 Fe<sub>2</sub>S + Fe between 2000–2400 K at ~115  
165 GPa. The stability of tetragonal Fe<sub>3</sub>S is constrained to pressures below ~120 GPa at high  
166 temperatures in Fe-rich systems (Figure 4) (Seagle et al. 2006; Morard et al. 2008; Kamada et al.  
167 2010; Kamada et al. 2012; Thompson et al, 2020). Fe<sub>5</sub>S<sub>2</sub> is observed coexisting with iron  
168 between ~120–200 GPa and in the 2400–3300 K range, making it the relevant sulfide in the  
169 melting and crystallization properties of Fe-rich metallic cores at these pressures (Figure 4).

## 170 **Sulfur in Earth and planetary cores**

171 The stability of Fe<sub>5</sub>S<sub>2</sub> and the phase relations observed in this study present novel  
172 constraints on the material properties of Fe-sulfides at conditions relevant to Earth's outer core,  
173 Venus' core, and exoplanetary cores of similar size and core-mass fraction (CMF). Namely, at  
174 pressures  $\geq 120$  GPa, Ni<sub>5</sub>As<sub>2</sub>-type Fe<sub>5</sub>S<sub>2</sub>, not tetragonal Fe<sub>3</sub>S as previously believed, is the

175 relevant Fe-rich sulfide. As Earth's outer core crystallizes over time,  $\text{Fe}_5\text{S}_2$  will eventually  
176 crystallize over a large depth range up to the CMB. Tetragonal  $\text{Fe}_3\text{S}$  has been confirmed up to  
177 120 GPa (Figure 4) (Seagle et al. 2006; Morard et al. 2008; Thompson et al., 2020), constraining  
178 the  $\text{Fe}_3\text{S} + \text{Fe} \rightarrow \text{Fe}_5\text{S}_2 + \text{Fe}$  transition to  $\sim 120$  GPa. Rocky planetary bodies with sizes and CMFs  
179 like that of Venus (CMB = 114 GPa) (Aitta 2012) would likely crystallize  $\text{Fe}_3\text{S}$  at the CMB,  
180 while  $\text{Fe}_5\text{S}_2$  will be crystallizing over much of the core's depth.  $C37$   $\text{Fe}_2\text{S}$  is reported to be stable  
181 on the solidus above 306 GPa in Fe-rich systems (Tateno et al. 2019), suggesting a  $\text{Fe}_5\text{S}_2 + \text{Fe} \rightarrow$   
182  $\text{Fe}_2\text{S} + \text{Fe}$  transition in the 200–300 GPa range.  $C37$   $\text{Fe}_2\text{S}$  remains the candidate Fe-rich sulfide  
183 at Earth's ICB, but for planets like Venus with a central pressure of 274 GPa (Aitta 2012),  $\text{Fe}_5\text{S}_2$   
184 may instead be the relevant Fe-rich sulfide, necessitating experimental determination of the  
185  $\text{Fe}_5\text{S}_2 \rightarrow \text{Fe}_2\text{S} + \text{Fe}$  decomposition  $P$ - $T$  boundary. Additional assessment of the volume and  
186 melting-temperature changes across these newly determined sulfide transitions are critical for  
187 matching the density and seismic profiles of Earth and rocky planetary cores. Furthermore, as  
188  $\text{Fe}_5\text{S}_2$  exhibits a complex crystalline arrangement with significant metal-metal bonding compared  
189 to  $\text{Fe}_2\text{S}$  and  $\text{Fe}_3\text{S}$  at core pressures, the significant depth over which  $\text{Fe}_5\text{S}_2$  crystallizes in and  
190 Venus- to Earth-sized planetary cores is likely to contribute to higher conductivity signatures in  
191 these regions.

192 Terrestrial core chemistries are multicomponent; Si and Ni are also important core-  
193 alloying elements in planets such as Earth, drawing attention to the potential stability of the  
194 perryite  $(\text{Fe}, \text{Ni})_8(\text{Si}, \text{P})_3$  structure in  $(\text{Fe}, \text{Ni}) - (\text{Si}, \text{S})$ -rich cores. Perryite, is a mineral observed  
195 in enstatite chondrites and aubrites that adopts a structure that is trigonal stacking variant of the  
196  $\text{Ni}_5\text{As}_2$ -type structure (Wasson and Wai, 1970; Okada et al. 1991) and iron phosphides tend to  
197 adopt analogous structures to that observed in iron sulfides (i.e., Fei et al. 2000; Dera et al. 2008;



198 Gu et al. 2014; Gu et al. 2016; Zurkowski et al., in press).  $\text{Fe}_5\text{S}_2$  could potentially serve as a host  
199 for nickel and silicon in the perryite structure at Earth's outer core conditions. Further, Si has  
200 recently been shown to dissolve into iron-sulfide phases and expand their stability fields with  
201 pressure (Tao and Fei, 2021). The dissolution of silicon into  $\text{Fe}_5\text{S}_2$  may expand its stability to  
202 higher pressures, making it potentially relevant at Earth's ICB conditions. If this is the case, the  
203 partitioning of Si and Ni between  $(\text{Fe}, \text{Ni})_5(\text{S}, \text{Si})_2$  and  $(\text{Fe}, \text{Ni}, \text{Si})$  and the density difference  
204 between the inner-core-crystallizing phase and remaining light-element-rich liquid will be crucial  
205 to investigations of the inner-core-density deficit and inner core seismic morphology.

206

## 207 **Materials and Methods**

208 Experiments targeting multigrain synthesis in the Fe-S system consisted of mixtures of Fe  
209 (99.9+%, <10 $\mu\text{m}$ , Alfa Aesar) and iron sulfide ( $\text{FeS}$ , 99.99%, Alfa Aesar) in Fe+12.5 wt% S  
210 ( $\text{Fe}_{80}\text{S}_{20}$ ), Fe + 23 wt% S ( $\text{Fe}_{67}\text{S}_{33}$ ) compositions. The Fe-FeS compositions were ground in  
211 ethanol in a pestle and mortar for 1 hour then mixed dry for a short interval to homogenize any  
212 density settling during alcohol evaporation.

213 Pressure was generated using BX-90-type (Kantor et al. 2012) diamond anvil cells (DAC)  
214 with type 1 Boehler-Almax conical diamonds and seats, and Mao symmetric-type DACs with  
215 Type 1 standard cut diamonds mounted on tungsten carbide or cubic-boron nitride (cBN) seats.  
216 Diamond culets ranged from 150-50  $\mu\text{m}$  in diameter. Foils of  $\text{Fe}_{80}\text{S}_{20}$  or  $\text{Fe}_{67}\text{S}_{33}$  were produced  
217 by pressing the Fe-S starting powders between two ungasketed diamond anvils, then loaded  
218 between pellets of KCl or  $\text{SiO}_2$ . Sample chambers ranged from 180-25  $\mu\text{m}$  in diameter and were  
219 drilled from pre-indented rhenium gaskets. Samples were dried at 100°C for 30 minutes prior to  
220 pressurization.

221 Angle dispersive X-ray diffraction (XRD) experiments were conducted at Argonne  
222 National Laboratory, sector 13 ID-D, of the Advanced Photon Source. At Sector 13 ID-D, a 2.5  
223  $\mu\text{m} \times 3.54 \mu\text{m}$  full width, half maximum (FWHM) monochromatic X-ray beam tuned to 37 or 42  
224 keV was utilized, and diffracted X-rays were collected with a CdTe 1M Pilatus detector. Sample-  
225 to-detector geometries were calibrated using a LaB<sub>6</sub> NIST standard and a single crystal of  
226 enstatite was used for calibration of the rotational geometry of the X-ray beam and detector.

227 Double-sided laser heating and *in-situ* XRD collection took place at Sector 13 ID-D. Fibers  
228 lasers shaped with  $\sim 10 \mu\text{m}$  radius flat tops were aligned with the X-ray beam using the X-ray  
229 fluorescence of the sample pressure media or the gasket (Prakapenka et al., 2008). During  
230 heating, thermal emission from a 6  $\mu\text{m}$  central region of the laser-heated spot was measured  
231 spectroradiometrically and fit to a gray body approximation (Heinz and Jeanloz, 1987). A 3%  
232 temperature correction was then applied to correct for axial gradients through the sample  
233 (Campbell et al., 2007; 2009). The laser power on each side of the sample was adjusted to  
234 maintain uniform double-sided heating and X-Ray diffraction and temperature measurements  
235 were collected every  $\sim 200 \text{ K}$  with 1s exposure times. Samples were typically quenched at high  
236 temperatures within 30-45 minutes of heating or after a phase transition and suitable grain  
237 growth was observed. Powder diffraction patterns were processed using Dioptas (Prescher and  
238 Prakapenka, 2015) and CrysAlisPro (Rigaku OD, 2018). Pressure was determined using the  
239 equation of state of hcp-Fe (Dewaele et al. 2006).

240 Upon quenching, X-ray diffraction maps of the heated spot were collected across a 100  
241  $\mu\text{m}^2$  square region in 3  $\mu\text{m}$  steps. Map locations showing high intensity, spotty diffraction  
242 patterns were chosen for multigrain single-crystal type X-ray diffraction collection approach, as  
243 these features indicate high-temperature induced crystallite growth. At chosen map locations, X-

244 ray diffraction images were collected across  $\pm 17$  to  $\pm 30^\circ$  rotational scans in  $0.25^\circ$ – $0.5^\circ$  steps with  
245 1–4 s exposure times. The diffraction reflections were then mapped in the reciprocal space and  
246 target grains were identified and separated from reflections associated with the pressure media,  
247 iron, and diamond (Rigaku OD, 2018).

248 Target Fe-S lattices were then indexed and the peak intensities were integrated and reduced  
249 using CrysAlisPro (Rigaku OD, 2018). Absorption corrections and scaling factors were applied to  
250 the structure factors in CrysAlisPro using the multi-scan method via the Scale3 Abspack program  
251 (Rigaku OD, 2018). The final structure factors and lattice geometries were then refined to known  
252 structure models (El-Boragy et al. 1970; Oryshchyn et al. 2011) using SHELXL2014/7  
253 (Sheldrick 2015). Reflections showing anomalous calculated versus measured structure factors;  
254 likely due to overlap with diamond and other phases in the multigrain sample, resolution  
255 limitations, and volume of crystal illuminated by the X-ray beam, were omitted. Structure  
256 models were visualized using Vesta during the refinement procedure (Momma and Izumi 2011).

257

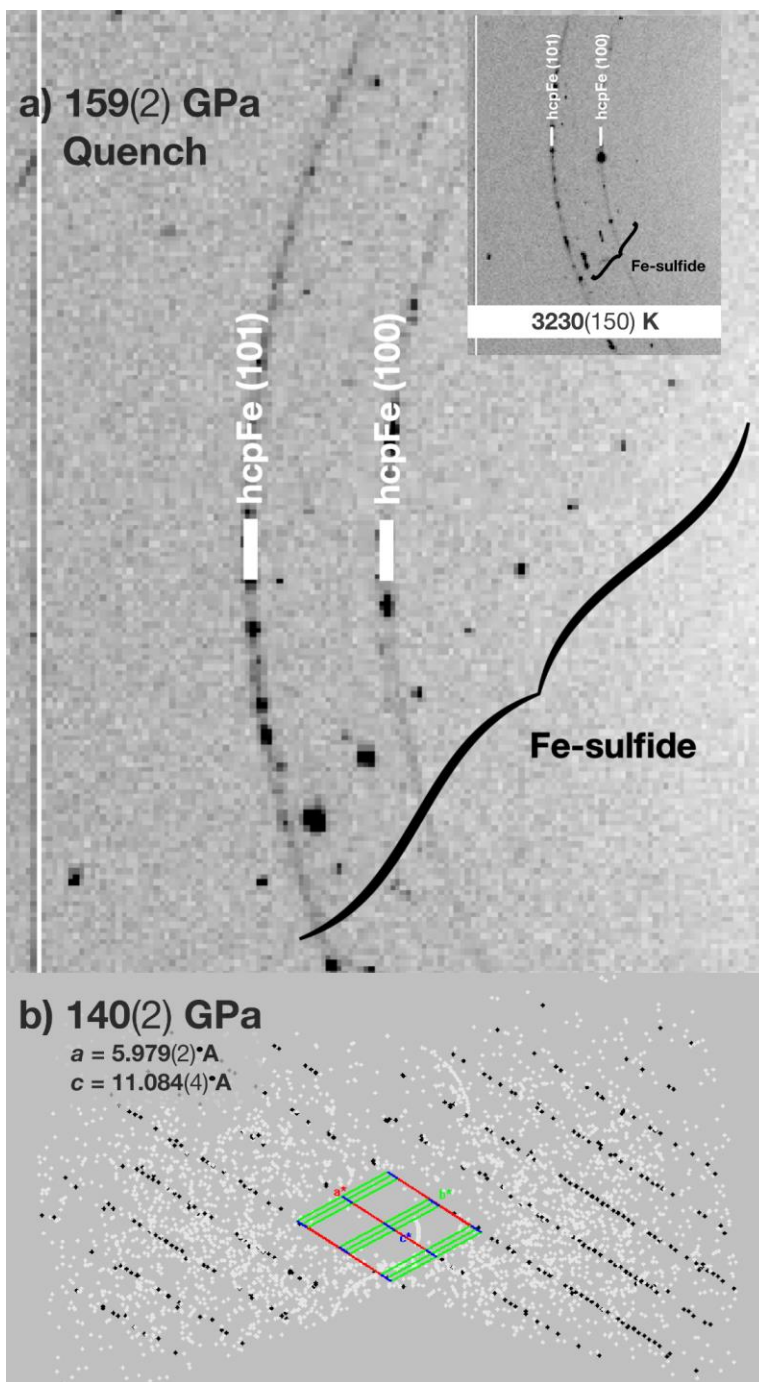
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273

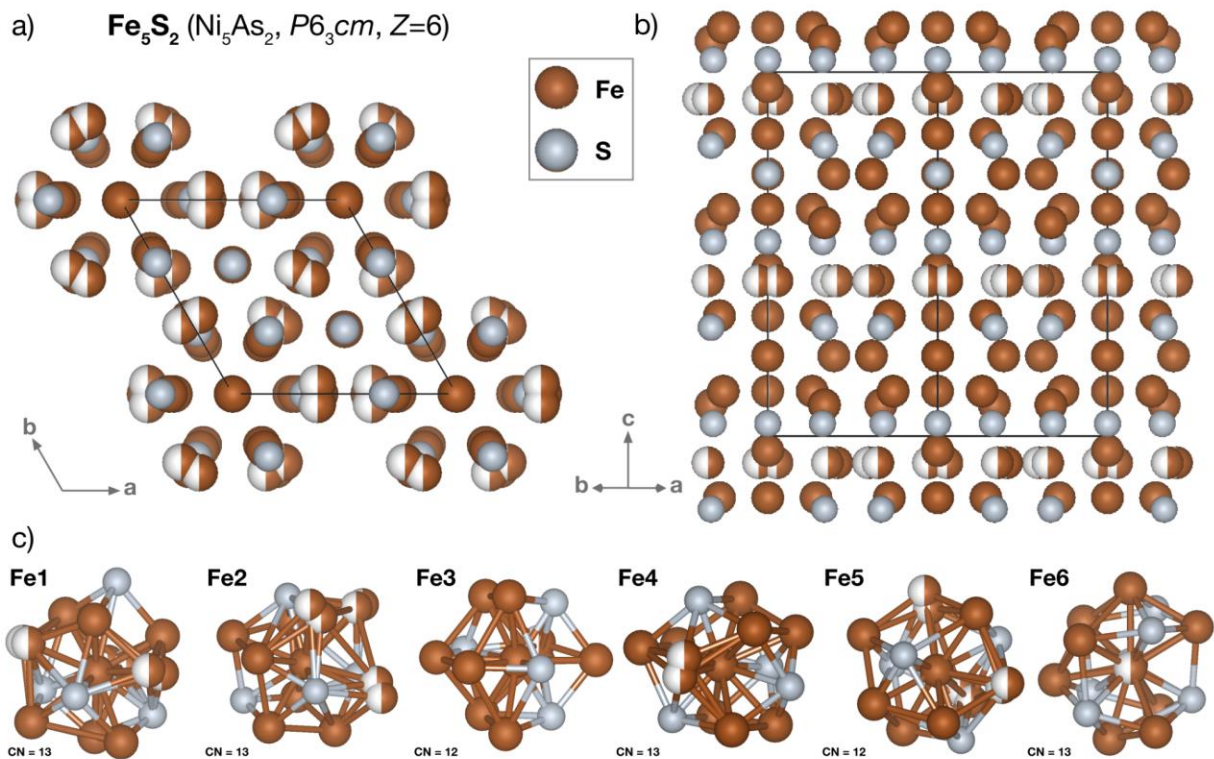


275

276 **Figure 1. Fe<sub>5</sub>S<sub>2</sub> diffraction in equilibrium with iron**

277 a) View of a crystallite of the high-temperature Fe<sub>5</sub>S<sub>2</sub> coexisting with recrystallized hcp-Fe after  
 278 synthesis at 159(2) GPa and 3230(130) K (inset). The spotty rings corresponding to the hcp-Fe  
 279 (100) and (101) planes demonstrate that iron equilibrated at 159(2) GPa and 3230(130) K. b)

280 View of the reciprocal space for reflections detected in an experiment on the Fe<sub>80</sub>S<sub>20</sub> composition  
 281 quenched from 140(2) GPa and 3070(140) K. The reciprocal lattice, colored in black, is  
 282 associated with a Fe<sub>5</sub>S<sub>2</sub> grain whose lattice parameters are provided in the top left.



283

284

**Figure 2. Crystal structure of  $\text{Ni}_5\text{As}_2$ -type  $\text{Fe}_5\text{S}_2$**

285

Crystal structure of  $\text{Fe}_5\text{S}_2$  viewed along the a)  $c$  axis and the b) (100) plane. c) The Fe-

286

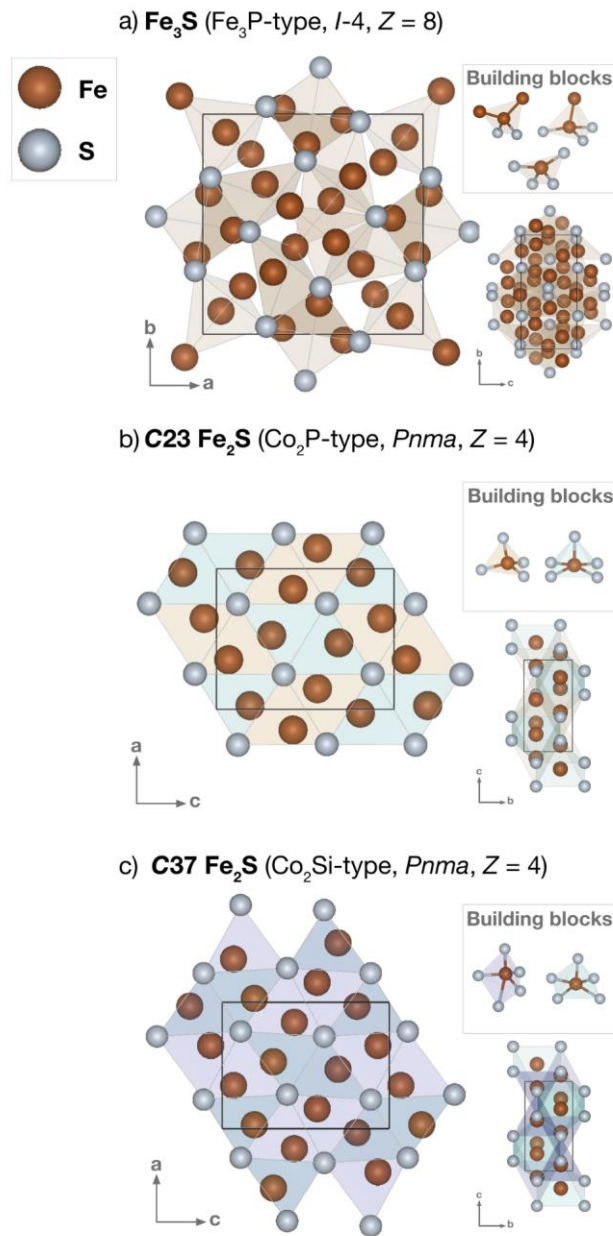
coordination polyhedra observed in this atomic arrangement are provided. Fe1, Fe2, Fe4, and

287

Fe6 are coordinated by 13 sites and Fe3 and Fe5 are coordinated by 12 sites. Each coordination

288

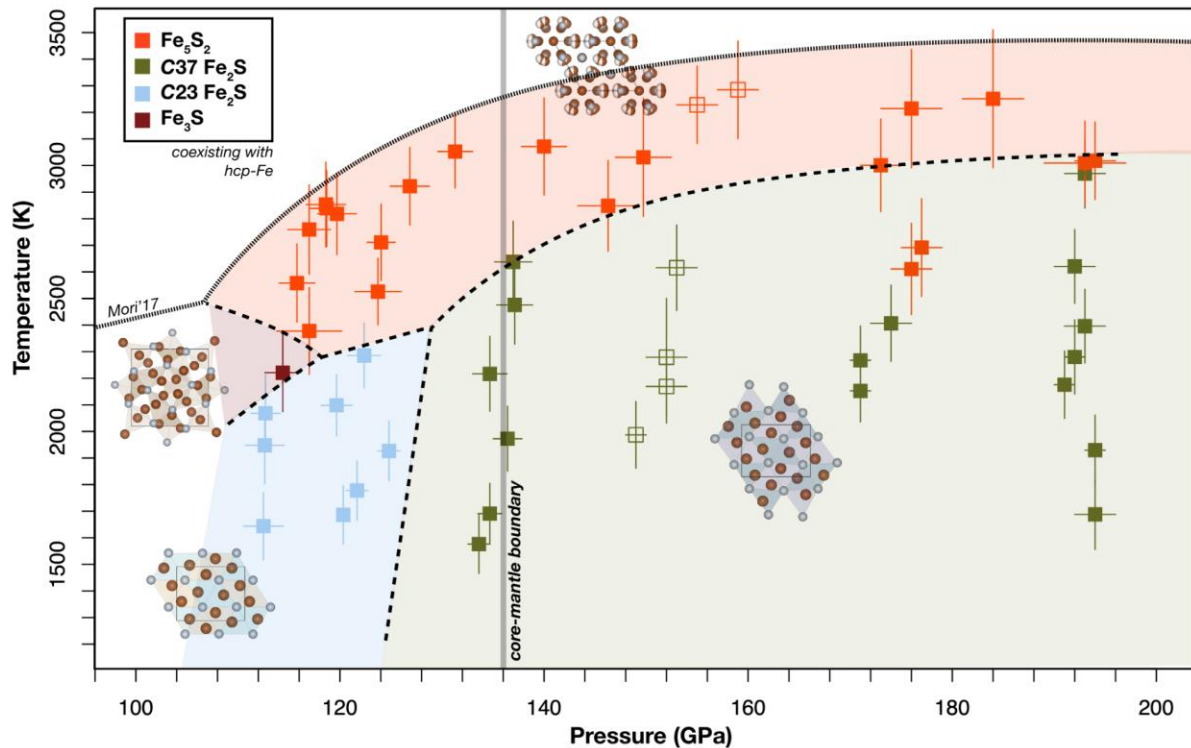
polyhedron consists of Fe and S sites.



289

290 **Figure 3. Crystal structures of  $\text{Fe}_3\text{S}$  and  $\text{Fe}_2\text{S}$  observed in this study**

291 a)  $\text{Fe}_3\text{S}$  adopts the  $\text{Fe}_3\text{P}$ -type structure ( $I-4$ ,  $Z = 8$ ) that is composed of three tetrahedrally  
 292 coordinated Fe-sites, each with increasing Fe–Fe bonding. b) The  $\text{C}23 \text{Fe}_2\text{S}$  structure ( $\text{Co}_2\text{P}$ -type,  
 293  $Pnma$ ,  $Z = 4$ ) is made up of columns of  $\text{FeS}_4$  tetrahedra and columns of  $\text{FeS}_5$  square pyramids  
 294 linked along edges in the  $b$  direction. c) The  $\text{C}37 \text{Fe}_2\text{S}$  structure ( $\text{Co}_2\text{Si}$ -type,  $Pnma$ ,  $Z = 4$ ) has  
 295 the same site symmetries as the  $\text{C}23$  structure, but is marked by a shortened  $a$  axis and elongated  
 296  $b$  and  $c$  axes accompanied by the formation of a 5-fold dipyramid.



297

298 **Figure 4. Iron-rich sulfide phase diagram to Earth's outer core pressures**

299 At moderate temperatures up to 130 GPa, C23 Fe<sub>2</sub>S is observed. At moderate temperatures,  
 300 tetragonal Fe<sub>3</sub>S (Fe<sub>3</sub>P-type) is observed below 120 GPa. At moderate temperatures between 130  
 301 and 200 GPa, C37 Fe<sub>2</sub>S is observed, and on the liquidus between ~115–200 GPa, Fe<sub>5</sub>S<sub>2</sub> is the  
 302 liquidus phase. A kink the Fe-S solidus curve (Mori et al. 2017), is presented to account for the  
 303 change in phase relations and the high temperatures at which Fe<sub>5</sub>S<sub>2</sub> was observed. The closed  
 304 squares represent experiments conducted on the Fe<sub>80</sub>S<sub>20</sub> starting material, and the open squares  
 305 represent experiments conducted on the Fe<sub>67</sub>S<sub>33</sub> starting material. Both show compatible results,  
 306 likely indicating that Fe-rich regions of the Fe<sub>67</sub>S<sub>33</sub> foils were probed.

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473 and high temperatures. <https://doi.org/10.2138/am-2022-8187>

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476 **Fe<sub>5</sub>S<sub>2</sub> identified as a host for sulfur in Earth's core**

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489 **Supplementary Information**

490 The Supplementary Information includes:

- 491 ➤ Supplementary Information Text
- 492 ➤ Table S1
- 493 ➤ Table S2
- 494 ➤ Table S3
- 495 ➤ Table S4
- 496 ➤ Figure S1
- 497 ➤ Figure S2
- 498 ➤ Figure S3
- 499 ➤ Figure S4
- 500 ➤ Supplementary Information References

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## 510 **Supplementary Information Text**

### 511 *Synthesis of Fe<sub>5</sub>S<sub>2</sub> to 200 GPa*

512 Fe<sub>5</sub>S<sub>2</sub> was synthesized between 119(2) and 193(4) GPa with heating to temperatures  
513 between 2400–3300 K. The formation of Fe<sub>5</sub>S<sub>2</sub> phase was first recognized in the diffraction  
514 patterns by the onset of diffuse streaks and sets of closely spaced reciprocal nodes in the powder  
515 diffraction images, suggesting a complex atomic arrangement (Figure 1, S1). With sustained  
516 high temperatures in the samples loaded in more hydrostatic pressure media (e.g. KCl vs. SiO<sub>2</sub>),  
517 large grain growth (3 – 6 μm) was observed and the diffuse streaks became less apparent. Further  
518 experiments are required to assess the role of hydrostaticity, kinetics, pressure, and temperature  
519 on the disorder of this phase.

520 Recrystallization of hcp-Fe with the Fe<sub>5</sub>S<sub>2</sub> crystallites at high temperatures was observed  
521 in all experiments using the Fe<sub>80</sub>S<sub>20</sub> and Fe<sub>67</sub>S<sub>33</sub> starting powders as evidenced by the spotty  
522 (100) and (101) hcp-Fe rings in Figure 1a. Fe-recrystallization with this high-temperature Fe-  
523 sulfide establishes that the probed sample locations were in a Fe-saturated phase-field (Figure  
524 1a) and this phase is important to consider further in the context of Fe-rich planetary cores.

525 Upon temperature quenching, rotational single-crystal X-ray diffraction scans were  
526 performed across a 37–60° omega range, depending on the DAC configuration, and grains were  
527 indexed in the reciprocal space to a hexagonal unit cell compatible with an Fe<sub>5</sub>S<sub>2</sub> volume (Table  
528 S1, Figure 1b). Three unit-cell polytypes were observed and indexed in our experiments and are  
529 listed in Table S1. All indexed grains exhibit an *a* axial length of ~ 6 Å, while 3 differing *c* axial  
530 lengths are observed: ~ 11, 26, and 73 Å. Diffraction mappings show that the polytypism is  
531 accompanied by diffuse scattering suggesting positional disorder as the atomic arrangement  
532 along the *c* axis configures at high temperatures (Figure S1). Analogous polytypism has also

533 been observed in the transition metal binary phase  $\text{Pb}_5\text{As}_2$  (Saini et al. 1964). Decreased diffuse  
534 scattering from the  $\text{Fe}_5\text{S}_2$  grains was observed after continued heating in the KCl pressure media,  
535 and grains with  $a \sim 6 \text{ \AA}$ ,  $c \sim 11 \text{ \AA}$  were indexed, indicating that this configuration is the most  
536 positionally ordered arrangement relevant to these high  $P$ - $T$  conditions. These measured unit cell  
537 parameters are compatible with 6 formula units per cell volume of  $\text{Fe}_5\text{S}_2$ .

538         The structure of a grain  $\text{Fe}_5\text{S}_2$  synthesized at 140(2) GPa and 3070(180) K was  
539 determined based on 159 observed reflections at these extreme conditions (Table S2).  
540 Assessment of the systematic absences for the structure factors reduced for  $\text{Fe}_5\text{S}_2$  suggests a  
541  $P6_3cm$  space group. Structural solution and positional and displacement parameter refinement  
542 converged on the  $\text{Ni}_5\text{As}_2$  structure type (Table S2, 3; Figure 2) (Oryshchyn et al. 2011). The  
543  $\text{Ni}_5\text{As}_2$  structure is a slight modification of the  $\text{Pb}_5\text{Sb}_2$  structure (El-Boragy et al. 1970) where the  
544 Fe6 site (Table S3) is split about its position and given half occupancy, changing its Wyckoff site  
545 from 6c to 12d (Figure 2) (Oryshchyn et al. 2011). This structural modification was initiated  
546 based on the large  $U_{22}$  parameter observed on the  $M6$  site ( $M = \text{metal}$ ) in  $\text{Ni}_5\text{As}_2$  when  
547 anisotropic displacement parameters were refined using a  $\text{Pb}_5\text{As}_2$  starting model (Figure S2). In  
548 the current dataset, a comparably large  $U_{22}$  value was observed on the Fe6 site resulting in an  
549 oblong displacement ellipse when refining the data against the initial  $\text{Pb}_5\text{As}_2$  structure model.  
550 The Fe6 site was positionally disordered following Oryshchyn et al. (2011), and the resultant  $R_1$   
551 value was improved by around  $\sim 5\%$ , rendering the  $\text{Ni}_5\text{As}_2$  structure model more compatible with  
552 the current  $\text{Fe}_5\text{S}_2$  data (Table S2). Isotropic displacement parameters showing errors  $> 3\sigma$  were  
553 fixed to a value equal to approximately the average displacement parameter value for Fe or S  
554 (Table S2). The isotropic displacement parameter for the disordered Fe site was also fixed at an  
555 average  $\text{Fe}_{\text{Uiso}}$  value (Table S2). The number of reflections collected at these extreme pressures

556 limits the number of statistically meaningful parameters to refine, and fixing displacement  
557 parameters to reasonable values precludes overinterpretation of the current dataset. The high  
558 isotropic displacement parameter observed on the Fe4 site may represent some vacancies on this  
559 site, but the dataset is again deficient to precisely describe the site occupancy.

560         The experimental details for the solution and refinement of Fe<sub>5</sub>S<sub>2</sub> to the Ni<sub>5</sub>As<sub>2</sub> solution  
561 model are provided in Table S2 and the atomic coordinates of Fe<sub>5</sub>S<sub>2</sub> synthesized at 140(2) GPa  
562 and 3070(180) K are provided in Table S3. In accordance with previous descriptions of related  
563 *M<sub>5</sub>X<sub>2</sub>* transitional-metal binary phases (e.g. (Kjekshus et al. 1973; Oryshchyn et al. 2011), Fe<sub>5</sub>S<sub>2</sub>  
564 can be viewed as an arrangement of 6 Fe sites and 3 S sites with the Fe1, Fe2, Fe4, and Fe6 sites  
565 are in 13-fold coordination, the Fe3 and Fe5 sites are in 12-fold coordination, and the S sites are  
566 in 10-fold coordination (Figure 2c). The measured bond lengths for the Fe-coordinated polyhedra  
567 are provided in Table S4. All sites are coordinated by both Fe and S sites. A CIF file for Fe<sub>5</sub>S<sub>2</sub>  
568 is provided in Appendix A1.

569         Final R<sub>1</sub> values  $\simeq$  10% attest to the less-than ideal quality of the multigrain dataset as a  
570 possible result of the 1-2 megabar synthesis conditions and observed *c* axial disorder and  
571 stacking complexities in Fe<sub>5</sub>S<sub>2</sub> (Table S2). Previous studies of isomorphic Ni<sub>5</sub>As<sub>2</sub> and related  
572 Pb<sub>5</sub>As<sub>2</sub> at ambient conditions have also reported similar quality of refinements despite obtaining  
573 significantly more reflections in the absence of a DAC (e.g. Saini et al. 1964; El-Boragy et al.  
574 1970; Kjekshus et al. 1973). Notable challenges regarding the refinement of the Fe<sub>5</sub>S<sub>2</sub> structure  
575 model at these extreme conditions are discussed here and compared with similar difficulties  
576 reported in previous characterizations of this structure.

577         Thirteen violations of the *P6<sub>3</sub>cm* systematic absence condition:  $l = 2n$  for (*0kl*), were  
578 flagged during the refinement of Fe<sub>5</sub>S<sub>2</sub>. The reflections associated with these systematic absence



579 violations were examined in the raw diffraction images and all show low, diffuse intensity.  
580 Discrepancy over the presence or absence of weak reflections with  $(0kl)$ ,  $l=2n+1$  has been  
581 reported in previous investigations of  $\text{Ni}_5\text{As}_2$  and  $\text{Pb}_5\text{As}_2$  (e.g. Saini et al. 1964; El-Boragy et al.  
582 1970; Kjekshus et al. 1973), suggesting that these studies may have also faced difficulties with  
583 space group determination. Observations of these low-intensity reflections may be a result of  
584 residual disorder in the stacking arrangement along the  $c$  direction, and longer heating cycles  
585 may be required for the atoms to arrange into equilibrium positions. It is likewise possible that  
586 varying synthesis methods for  $\text{Ni}_5\text{As}_2$  and  $\text{Pb}_5\text{As}_2$  in previous ambient condition studies affected  
587 the  $c$  axial atomic arrangement.  $\text{Ni}_5\text{As}_2$  and isomorphic  $\text{Ni}_5\text{P}_2$  also exhibits a homogeneity range  
588 of  $\sim 71.25 - 72.7$  atomic % As (Kjekshus et al. 1973; Litasov et al., 2019), and slight  
589 modifications of this structure based on varying stacking arrangements result in structures such  
590 as  $\text{Ni}_{31}\text{Si}_{12}$  (Frank and Schubert 1971) and  $(\text{Fe, Ni})_8(\text{Si, P})_3$  (perryite) (Okada et al. 1991).  
591 Attempts to refine the current Fe-sulfide phase with the  $\text{Ni}_{31}\text{Si}_{12}$  or perryite structure models did  
592 not improve the refinement results. Based on the presence of disorder, polytypism, anisotropic  
593 vibrational motion, and nonstoichiometry in the related  $M_5X_2$  phases, additional nuances to the  
594  $\text{Fe}_5\text{S}_2$  structure model may be developed in future studies; however, the identification and  
595 characterization of the  $\text{Fe}_5\text{S}_2$  crystal structure determined here to 200 GPa is novel, and the  
596 observations and challenges reported in this study align with that of previous analyses of  $\text{Ni}_5\text{As}_2$   
597 and  $\text{Pb}_5\text{Sb}_2$  that were not limited by microdiffraction in a DAC at extreme conditions.

598

#### 599 *Other sulfides observed during heating to 200 GPa*

600 Three experiments and 7 heating cycles were performed in this study between 100 and  
601 200 GPa and to 3400 K, and the X-ray diffraction patterns collected upon heating provide insight

602 into the Fe-rich Fe-S phase relations at outer core pressures and to high temperatures. In each  
603 heating experiment, near melting temperatures were targeted and grains of Fe<sub>5</sub>S<sub>2</sub> were identified  
604 upon quenching. Additional Fe-sulfides were also observed across the thermal gradient of the  
605 heated spot, providing information on the lower temperature Fe-saturated phase relations in each  
606 heating cycle. Upon quenching from high-temperature synthesis at 119(2) GPa, the other grains  
607 in the sample were identified and indexed to a tetragonal cell with  $a = 8.094(3)$  Å,  $c = 3.990(2)$   
608 Å, compatible with 8 formula units of Fe<sub>3</sub>S (Table S1). The structure was then solved and refined  
609 to the Fe<sub>3</sub>P-type structure ( $I-4$ ,  $Z = 8$ ) (referred to herein as  $I-4$  Fe<sub>3</sub>S) in agreement with previous  
610 studies (Fei et al. 2000; Seagle et al. 2006; Morard et al. 2007; Kamada et al. 2010; Kamada et  
611 al. 2012; Thompson et al. 2020). The resultant structure model for Fe<sub>3</sub>S at these conditions is  
612 shown in Figure (3a), a CIF file for Fe<sub>3</sub>S is provided in Appendix A2, and the refinement details  
613 are given in Table S2. The structure can be viewed as containing 3 tetrahedrally coordinated Fe  
614 sites (Blanchard et al. 2008): one Fe-site is coordinated only by S atoms with an average bond  
615 length of 2.083(9) Å, another Fe site is coordinated by 3 S atoms and 1 Fe atom with an average  
616 bond length of 2.146(8) Å, and the third Fe site is coordinated by 2 S and 2 Fe atoms with an  
617 average bond length of 2.137(8) Å (Figure 3). The measured interatomic distances are  
618 comparable to reports on other transition metal  $M_3X$  structures (e.g. Aronsson 1955; Rundqvist  
619 1979).

620 After temperature quenching from 131(2) GPa and 3050(140) K, grains of Fe<sub>3</sub>S were not  
621 observed across the heated spot. Instead, orthorhombic lattices were also identified with  
622 parameters  $a = 4.869(2)$  Å,  $b = 3.256(2)$  Å,  $c = 6.139(2)$  Å, compatible with 4 formula units of  
623 Fe<sub>2</sub>S. Structural solution and refinement indicate that the Fe<sub>2</sub>S grains adopt the  $C23$  structure  
624 (Co<sub>2</sub>P type,  $Pnma$ ,  $Z = 4$ ) in agreement with previous structural analyses at lower pressures

625 (Zurkowski et al. *in press*). Refinement details for the structure model are given in Table S2 and  
626 a CIF file for *C23* Fe<sub>2</sub>S is provided in Appendix A3. The structure is composed of columns of  
627 FeS<sub>4</sub> tetrahedra and columns of FeS<sub>5</sub> square pyramids linked along edges in the *b* direction  
628 (Figure 3b). The average Fe–S bond lengths are 2.213(8) Å and 2.011(2) Å in the square  
629 pyramids and tetrahedra, respectively. These values are comparable with that observed in *C23*  
630 Fe<sub>2</sub>S at 90 GPa (Zurkowski et al. *in press*).

631       Upon compression and heating at 140(2) GPa and 3070(180) K, orthorhombic grains  
632 were identified in the sample chamber with *a* = 4.667(2) Å, 3.289(1) Å, 6.186(4) Å. This unit  
633 cell is similar to the *C23* Fe<sub>2</sub>S observed at 130 GPa, but exhibits a 4 % contraction of the *a* axial  
634 length, a 1% extension of the *b* and *c* axial lengths, and a 2% volume decrease. Structural  
635 solution and refinement using the measured structure factors establish that Fe<sub>2</sub>S adopts the *C37*  
636 structure (Co<sub>2</sub>Si-type, *Pnma*, *Z*=4) at these conditions (Figure 3c). A CIF file for *C37* Fe<sub>2</sub>S is  
637 provided in Appendix A4 and the analysis details are given in Table S2. In agreement with the  
638 proposed *C23*–*C37* Fe<sub>2</sub>S phase transition around 140 GPa that is accompanied 1.6% volume  
639 decrease (Zurkowski et al., *in press*), this work confirms that the *C23*–*C37* transition in Fe<sub>2</sub>S  
640 occurs between 130 and 140 GPa with a similar volume change. Inherent to the *C23*–*C37*  
641 transition is coordination change from the 4-fold Fe1 coordination polyhedra in the *C23* structure  
642 to the 5-fold dipyramid polyhedra in the *C37* structure (Figure 3b, c). Comparing the *C23* and  
643 *C37* structure models determined at 130 and 140 GPa, respectively, a 10% contraction of the  
644 interatomic distance involved in the coordination change is observed (Figure S3). The average  
645 Fe–S bond lengths measured in *C37* Fe<sub>2</sub>S are 2.196(2) Å and 2.165(3) Å in the square pyramids  
646 and dipyramids, respectively. These values are comparable with *C23* Fe<sub>2</sub>S.

647

648 *Fe-rich sulfide phase relations at core-mantle boundary pressures*

649           The presented single-crystal derived structures of  $\text{Fe}_5\text{S}_2$ ,  $\text{Fe}_3\text{S}$ ,  $C23 \text{Fe}_2\text{S}$  and  $C37 \text{Fe}_2\text{S}$   
650 were then used to inform the changes in diffraction patterns collected during heating between  
651 110 and 200 GPa and to  $\sim 3300$  K. Beginning at 112 GPa, diffraction signal from  $C23 \text{Fe}_2\text{S}$   
652 coexisting with Fe was first identified upon heating of the  $\text{Fe}_{80}\text{S}_{20}$  starting material to  $\sim 2000$  K.  
653 With continued heating, peaks associated with tetragonal  $\text{Fe}_3\text{S}$  were observed over a limited  
654 temperature range ( $\lesssim 2400$  K) until diffuse scattering signal and Bragg reflections from  $\text{Fe}_5\text{S}_2$   
655 first appeared. Crystallization of the  $\text{Fe}_5\text{S}_2$  grains with iron occurred with continued heating to  
656 119(2) GPa and 2840(180) K. Upon heating beginning at 120 GPa,  $C23 \text{Fe}_2\text{S}$  was observed  
657 coexisting with Fe to 122(1) GPa and 2290(120) K, above which the onset of diffraction from  
658 the  $\text{Fe}_5\text{S}_2$  phase was identified and  $\text{Fe}_5\text{S}_2$  crystallites formed coexisting with hcp-Fe to 131(2)  
659 GPa and 3050(140) K.  $I-4 \text{Fe}_3\text{S}$  was not observed. With continued heating cycles between 133(1)  
660 GPa and 194(2) GPa and up to 3300 K,  $C37 \text{Fe}_2\text{S}$  coexists with Fe at moderate temperatures and  
661 a phase transition to  $\text{Fe}_5\text{S}_2$  occurs at high temperatures. A phase diagram satisfying these  
662 observations is proposed in Figure 4, and integrated XRD patterns for the heating cycles at  
663 119(2) and 184(3) GPa and to 3300 K are shown in Figure S4 to demonstrate the identification  
664 of  $C23 \text{Fe}_2\text{S}$ ,  $\text{Fe}_3\text{S}$ ,  $C37 \text{Fe}_2\text{S}$ , and  $\text{Fe}_5\text{S}_2$  coexisting with hcp-Fe. The peak intensities from the  
665  $\text{Fe}_5\text{S}_2$  lattices vary during heating up at high temperatures as a result of disorder, polytypism, and  
666 the formation of large crystallites at high temperatures (Figure 1, S1). These structural  
667 complexities render the powder diffraction patterns difficult to characterize without  
668 incorporating single-crystal XRD techniques.

669

670 *Discrepancies among the current and previous studies*

671           Transitional metal binary compounds with metal-to-nonmetal ratios ranging from 2.33–  
672 2.66 (~70–73 atm% metal) predominantly adopt complex trigonal or hexagonal structures with  
673 considerable metal-metal bonding (Chen and Whitmire 2018 and references therein). Our  
674 observations of the disorder, polytypism, and complex coordination environments inherent to  
675 Fe<sub>5</sub>S<sub>2</sub> demonstrate that it aligns with this systematic characterization. Interestingly, the Fe<sub>5</sub>S<sub>2</sub>  
676 atomic arrangement is thermodynamically favored at the extreme conditions of Earth's outer  
677 core over that of *I*-4 Fe<sub>3</sub>S. These results are contrary to previous reports of Fe<sub>3</sub>S stability on to  
678 high temperatures in Fe-rich systems to 250 GPa (Kamada et al. 2010; Kamada et al. 2012;  
679 Ozawa et al. 2013; Mori et al. 2017). Few lines of reasoning may account for these  
680 discrepancies. Interpretation of Fe<sub>5</sub>S<sub>2</sub> in the integrated powder diffraction patterns is difficult due  
681 to the variation in diffraction signal obtained from the Fe<sub>5</sub>S<sub>2</sub> polytypes and the low intensity  
682 scattering from the disordered sites during its formation. Fe<sub>5</sub>S<sub>2</sub> also forms large crystallites,  
683 limiting the orientations of the phase and rendering indexation of all diffraction angles for Fe<sub>5</sub>S<sub>2</sub>  
684 quite challenging without rotating the sample. The implementation of single-crystal X-ray  
685 diffraction techniques at high pressures was critical in the current study to accurately  
686 characterize the hexagonal unit cell geometry and complex structure of the Ni<sub>5</sub>As<sub>2</sub>-like Fe<sub>5</sub>S<sub>2</sub>.

687           Several studies have also reported chemically analyzed Fe<sub>3</sub>S grains in samples recovered  
688 from high temperatures in the 200–250 GPa range (Mori et al. 2017; Yokoo et al. 2019; Ozawa  
689 et al. 2013), but the difference in Fe content between Fe<sub>3</sub>S and Fe<sub>5</sub>S<sub>2</sub> may be as few as 3%. This  
690 value is generally within 3 sigma of the atomic percent error reported for chemical analyses of  
691 samples recovered from these extreme conditions, posing another challenge for distinguishing  
692 between the synthesis of Fe<sub>3</sub>S and Fe<sub>5</sub>S<sub>2</sub>. Furthermore, based on characterizations of the

693 isomorphous  $\text{Ni}_5\text{As}_2$  and  $\text{Ni}_5\text{P}_2$ , these structures exhibit a homogeneity range with up to ~73 atm%  
694 metal (Kjekshus et al. 1973; Litasov et al., 2019). For example, TEM analysis of a Fe–S sample  
695 recovered from 236 GPa and 2980 K reveals sulfide grains with on average ~73 atm% Fe  
696 (Ozawa et al. 2013). This value is within 0.3–1.25 atomic percent of the possible range of  $\text{Fe}_5\text{S}_2$   
697 stoichiometries and 2 atomic percent less than an  $\text{Fe}_3\text{S}$  composition. The results from this  
698 previous work as well as other chemical analyses of samples recovered from outer core pressures  
699 and high temperatures therefore do not contradict the current results (Mori et al. 2017; Yokoo et  
700 al. 2019; Ozawa et al. 2013) and may reveal a previous misinterpretation of  $\text{Fe}_5\text{S}_2$  as  $\text{Fe}_3\text{S}$  based  
701 on measured chemistries.

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704 **Table S1.** Unit cell parameters of Fe<sub>5</sub>S<sub>2</sub>, Fe<sub>3</sub>S, and Fe<sub>2</sub>S measured upon quenching from high *P*-  
705 *T* synthesis in this study. Each cell was indexed in the reciprocal space, and the DAC opening,  
706 and number of reflections obtained for each lattice are listed. For select experiments, high-  
707 temperature synthesis was conducted without collecting X-ray diffraction data, and the synthesis  
708 conditions for these experiments are listed as “high T, not measured.”

| Sample name     | Starting |        | Phase   | Room temperature collection |           |          |           |                |             |                    | Synthesis conditions |           |
|-----------------|----------|--------|---------|-----------------------------|-----------|----------|-----------|----------------|-------------|--------------------|----------------------|-----------|
|                 | material | medium |         | P                           | <i>a</i>  | <i>b</i> | <i>c</i>  | V              | DAC opening | no. of reflections | <i>P</i>             | <i>T</i>  |
|                 |          |        |         | GPa                         | Å         | Å        | Å         | Å <sup>3</sup> | (°)         |                    | GPa                  | K         |
| C140_P2_map25   | Fe80S20  | KCl    | Fe5S2   | 105(2)                      | 6.020(4)  |          | 11.204(7) | 351.6(5)       | 60          | 290                | 119(2)               | 2840(140) |
| C140_P2_map31   | Fe80S20  | KCl    | Fe5S2   | 105(2)                      | 6.050(2)  |          | 73.0(2)   | 2313(1)        | 60          | 134                | 119(2)               | 2840(140) |
| C140_P4_map10   | Fe80S20  | KCl    | Fe5S2   | 118(1)                      | 5.983(3)  |          | 11.078(6) | 343.4(4)       | 60          | 489                | 140(2)               | 3070(180) |
| C140_P4_map14   | Fe80S20  | KCl    | Fe5S2   | 118(1)                      | 5.979(2)  |          | 11.084(4) | 343.2(3)       | 60          | 379                | 140(2)               | 3070(180) |
| C140_P4_map14   | Fe80S20  | KCl    | Fe5S2   | 118(1)                      | 5.973(1)  |          | 11.094(2) | 342.8(1)       | 60          | 276                | 140(2)               | 3070(180) |
| C129_P140_map71 | Fe67S33  | SiO2   | Fe5S2   | 136.4(5)                    | 5.957(4)  |          | 71.89(4)  | 2209(3)        | 34          | 84                 | 159(2)               | 3230(150) |
| C129_P150_map9  | Fe67S33  | SiO2   | Fe5S2   | 144.2(8)                    | 5.939(3)  |          | 25.99(6)  | 794(2)         | 34          | 76                 | high T, not measured |           |
| C129_P150_map9  | Fe67S33  | SiO2   | Fe5S2   | 144.2(8)                    | 5.923(1)  |          | 26.004(6) | 789.9(3)       | 34          | 146                | high T, not measured |           |
| C137_P1_map25   | Fe80S20  | SiO2   | Fe5S2   | 159(1)                      | 5.890(3)  |          | 25.837(7) | 776.2(5)       | 40          | 108                | 184(3)               | 3250(260) |
| C137_P1_map25   | Fe80S20  | SiO2   | Fe5S2   | 159(1)                      | 5.880(1)  |          | 25.789(3) | 772.0(2)       | 40          | 108                | 184(3)               | 3250(260) |
| C137_P1_map25   | Fe80S20  | SiO2   | Fe5S2   | 159(1)                      | 5.876(2)  |          | 25.884(8) | 774.0(4)       | 40          | 116                | 184(3)               | 3250(260) |
| C137_P1_map25   | Fe80S20  | SiO2   | Fe5S2   | 159(1)                      | 5.883(3)  |          | 25.85(1)  | 774.6(7)       | 40          | 126                | 184(3)               | 3250(260) |
| C137_P1         | Fe80S20  | SiO2   | Fe5S2   | 159(1)                      | 5.897(3)  |          | 25.90(1)  | 775.1(6)       | 40          | 116                | 184(3)               | 3250(260) |
| C137_P1         | Fe80S20  | SiO2   | Fe5S2   | 159(1)                      | 5.878(1)  |          | 25.784(3) | 771.6(2)       | 40          | 149                | 184(3)               | 3250(260) |
| C137_P2_34      | Fe80S20  | SiO2   | Fe5S2   | 179(1)                      | 5.8400(7) |          | 25.583(2) | 755.7(1)       | 40          | 189                | 193(4)               | 3010(160) |
| C140_P1_map     | Fe80S20  | KCl    | Fe3S    | 100(1)                      | 8.156(3)  |          | 4.025(3)  | 267.8(3)       | 60          | 416                | high T, not measured |           |
| C140_P2_map25   | Fe80S20  | KCl    | Fe3S    | 105(2)                      | 8.094(3)  |          | 3.990(2)  | 261.4(2)       | 60          | 296                | 119(2)               | 2840(140) |
| C140_P3_map6    | Fe80S20  | KCl    | C23Fe2S | 111(1)                      | 4.869(2)  | 3.256(2) | 6.139(2)  | 97.3(1)        | 60          | 227                | 131(2)               | 3050(140) |
| C140_P4_map10   | Fe80S20  | KCl    | C37Fe2S | 118(1)                      | 4.677(2)  | 3.289(1) | 6.186(4)  | 95.18(9)       | 60          | 188                | 140(2)               | 3070(180) |

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726 **Table S2.** Select experimental details for crystal structure synthesis and analysis of Fe<sub>5</sub>S<sub>2</sub>, Fe<sub>3</sub>S,  
 727 and the Fe<sub>2</sub>S polymorphs.

| Phase  | Fe <sub>5</sub> S <sub>2</sub>     | Fe <sub>3</sub> S               | C23 Fe <sub>2</sub> S                | C37 Fe <sub>2</sub> S                |
|--|------------------------------------|---------------------------------|--------------------------------------|--------------------------------------|
| Sample Name  | C140_P4_map10                      | C140_P2_map25                   | C140_P3_map6                         | C140_P4_map10                        |
| <b>Synthesis</b>   |                                    |                                 |                                      |                                      |
| Pressure (GPa)   | 140(2)                             | 119(2)                          | 131(2)                               | 140(2)                               |
| Synthesis Temperature (K)  | 3070(180)                          | 2840(140)                       | 2050(140)                            | 3070(180)                            |
| Symmetry   | Hexagonal, <i>P6(3)cm</i> ,<br>Z=6 | Tetragonal, <i>I-4</i> ,<br>Z=8 | Orthorhombic,<br><i>Pnma</i> , Z = 4 | Orthorhombic,<br><i>Pnma</i> , Z = 4 |
| <i>a</i> (Å)   | 5.979(2)                           | 8.094(3)                        | 4.869(3)                             | 4.677(2)                             |
| <i>b</i> (Å)   |                                    |                                 | 3.256(4)                             | 3.289(1)                             |
| <i>c</i> (Å)   | 11.087(6)                          | 3.990(2)                        | 6.139(2)                             | 6.186(4)                             |
| <i>V</i> (Å <sup>3</sup> )   | 343.2(3)                           | 226.4(2)                        | 97.3(1)                              | 95.18(9)                             |
| <b>Reduction</b>   |                                    |                                 |                                      |                                      |
| No. of measured,<br>independent and observed<br>[ <i>I</i> > 2σ( <i>I</i> )] reflections                       | 396, 309, 157                      | 332, 314, 219                   | 227, 208, 112                        | 188, 158, 105                        |
| <i>R</i> <sub>int</sub> , <i>R</i> <sub>sigma</sub>  | 0.092, 0.062                       | 0.053, 0.072                    | 0.012, 0.016                         | 0.009, 0.014                         |
| <b>Refinement</b>  |                                    |                                 |                                      |                                      |
| <i>R</i> [ <i>F</i> <sup>2</sup> > 2σ( <i>F</i> <sup>2</sup> )], <i>wR</i> ( <i>F</i> <sup>2</sup> ), <i>S</i> | 0.098, 0.255, 1.12                 | 0.067, 0.164, 1.05              | 0.053, 0.149, 1.21                   | 0.047, 0.139, 1.22                   |
| No. of reflections   | 156                                | 219                             | 112                                  | 105                                  |
| No. of parameters  | 21                                 | 17                              | 10                                   | 10                                   |
| Δρ <sub>max</sub> , Δρ <sub>min</sub> (e Å <sup>-3</sup> )   | 2.14, -2.18                        | 1.92, -1.68                     | 1.86, -2.02                          | 1.91, -2.21                          |

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734 **Table S3.** Atomic coordinates of the Fe<sub>5</sub>S<sub>2</sub> refinement model for the data collected at 140(2) GPa  
 735 and quenched from 3070(180) K.

| Wycoff site | ATOM        | x     | y     | z     | Uiso  |
|-------------|-------------|-------|-------|-------|-------|
| <b>2a</b>   | <b>Fe1</b>  | 0     | 0     | 0.963 | 0.021 |
|             | error       |       |       | 0.003 | 0.005 |
| <b>4b</b>   | <b>Fe2</b>  | 1/3   | 2/3   | 0.089 | 0.046 |
|             | error       |       |       | 0.003 | 0.005 |
| <b>6c</b>   | <b>Fe3</b>  | 0.259 | 0     | 0.126 | 0.021 |
|             | error       | 0.002 |       | 0.001 |       |
| <b>6c</b>   | <b>Fe4</b>  | 0.613 | 0     | 0.222 | 0.021 |
|             | error       | 0.002 |       | 0.001 | 0.003 |
| <b>6c</b>   | <b>Fe5</b>  | 0.284 | 0     | 0.330 | 0.027 |
|             | error       | 0.002 |       | 0.001 | 0.003 |
| <b>12d</b>  | <b>Fe6*</b> | 0.651 | 0.056 | 0.426 | 0.021 |
|             | error       | 0.003 | 0.003 | 0.002 |       |
| <b>2a</b>   | <b>S1</b>   | 0     | 0     | 0.220 | 0.026 |
|             | error       |       |       | 0.004 |       |
| <b>4b</b>   | <b>S2</b>   | 1/3   | 2/3   | 0.299 | 0.026 |
|             | error       |       |       | 0.002 |       |
| <b>6c</b>   | <b>S3</b>   | 0.673 | 0     | 0.030 | 0.026 |
|             | error       | 0.005 |       | 0.003 |       |

\*indicates half occupancy

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749 **Table S4.** Selected interatomic distances for the Fe sites measured in Fe<sub>5</sub>S<sub>2</sub> at 140 GPa.

| Atom | bonded to | # | distance (Å) |
|------|-----------|---|--------------|
| Fe1  | -Fe3      | 3 | 2.39(2)      |
|      | -Fe5      | 3 | 2.22(3)      |
|      | -Fe6      | 3 | 2.313(17)    |
|      | -S1       | 1 | 2.66(8)      |
|      | -S4       | 3 | 2.09(3)      |
|      |           |   |              |
| Fe2  | -Fe3      | 3 | 2.275(3)     |
|      | -Fe4      | 3 | 2.375(16)    |
|      | -Fe6      | 3 | 2.40(3)      |
|      | -S3       | 1 | 2.36(4)      |
|      | -S4       | 3 | 2.108(13)    |
|      |           |   |              |
| Fe3  | -Fe1      | 1 | 2.39(2)      |
|      | -Fe2      | 2 | 2.275(8)     |
|      | -Fe4      | 3 | 2.350(15)    |
|      | -Fe5      | 1 | 2.30(3)      |
|      | -Fe6      | 1 | 2.307(17)    |
|      | -S1       | 1 | 1.90(4)      |
|      | -S4       | 3 | 2.68(3)      |
|      |           |   |              |
| Fe4  | -Fe2      | 2 | 2.375(16)    |
|      | -Fe3      | 3 | 2.320(11)    |
|      | -Fe5      | 3 | 2.416(14)    |
|      | -Fe6      | 1 | 2.303(17)    |
|      | -S1       | 1 | 2.326(11)    |
|      | -S3       | 2 | 2.044(16)    |
|      | -S4       | 1 | 2.10(3)      |
|      |           |   |              |
| Fe5  | -Fe1      | 1 | 2.22(3)      |
|      | -Fe3      | 1 | 2.30(3)      |
|      | -Fe4      | 3 | 2.416(14)    |
|      | -Fe6      | 3 | 2.135(15)    |
|      | -S1       | 1 | 2.09(5)      |
|      | -S3       | 2 | 2.184(10)    |
|      | -S4       | 1 | 2.27(3)      |
|      |           |   |              |
| Fe6  | -Fe1      | 1 | 2.313(17)    |
|      | -Fe2      | 2 | 2.40(3)      |
|      | -Fe3      | 1 | 2.307(17)    |
|      | -Fe4      | 1 | 2.303(17)    |
|      | -Fe5      | 3 | 2.135(15)    |
|      | -S3       | 2 | 2.15(3)      |
|      | -S4       | 3 | 2.17(3)      |
|      |           |   |              |

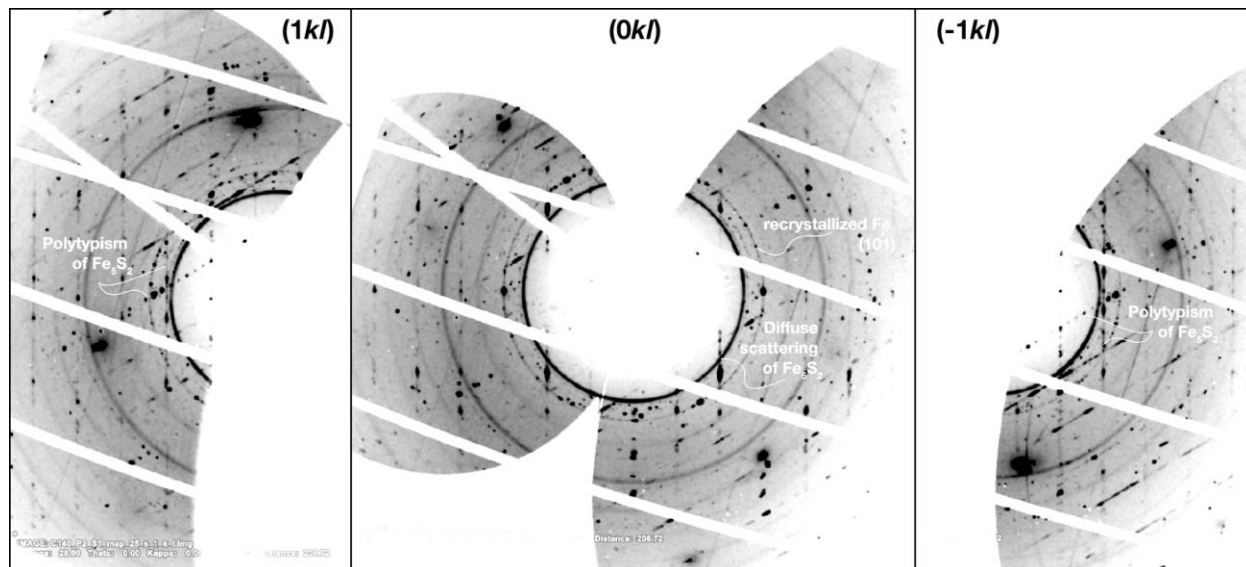
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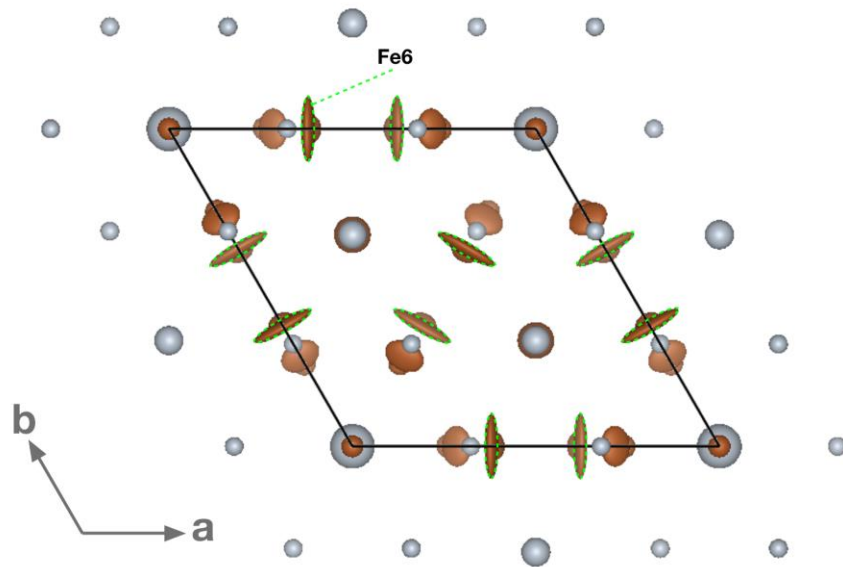
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754 **Figure S1.** Unwarped diffraction mappings of the  $(1kl)$ ,  $(0kl)$ , and  $(-1kl)$  directions for an  $\text{Fe}_5\text{S}_2$   
755 crystallite exemplifying the diffuse scattering signal collected at 119(2) and 2840(180) K. The  
756  $(1kl)$  and  $(-1kl)$  mappings also show the presence of a polytype with more closely spaced  
757 reflections oriented  $\sim 51^\circ$  from the mapped crystallite.



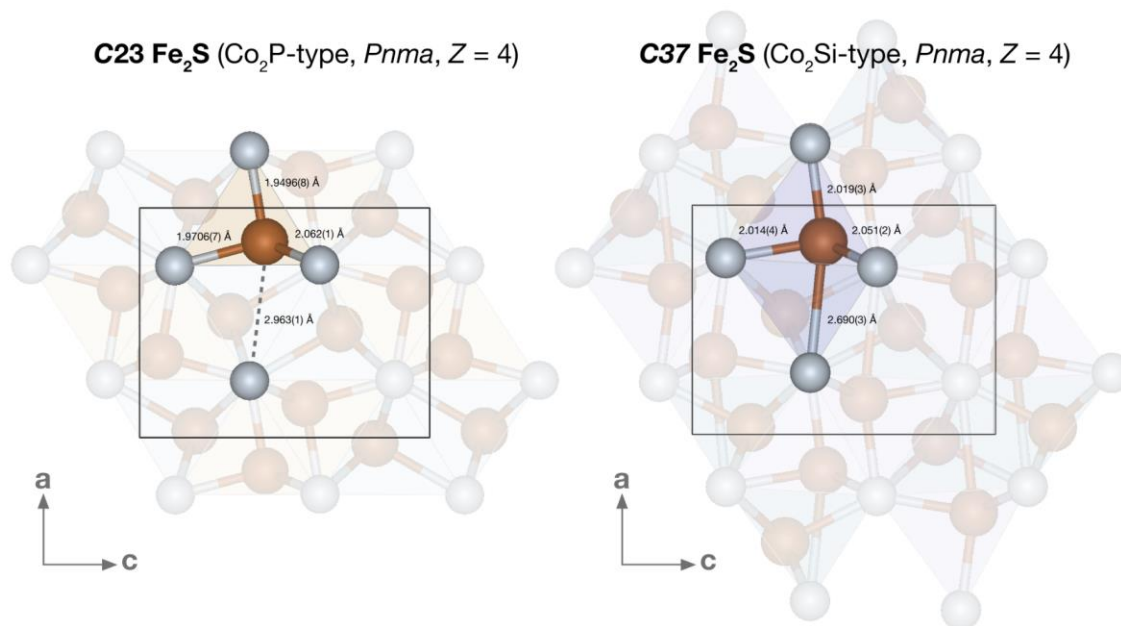
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765 **Figure S2.**  $\text{Fe}_5\text{As}_2$  structure model shown with anisotropic displacement ellipsoids using the  
766 starting  $\text{Pb}_5\text{As}_2$  structure model. In accordance with the  $\text{Ni}_5\text{As}_2$  structural characterization  
767 (Oryshchyn et al. 2011), these sites were disordered about their position and ascribed half  
768 occupancy.



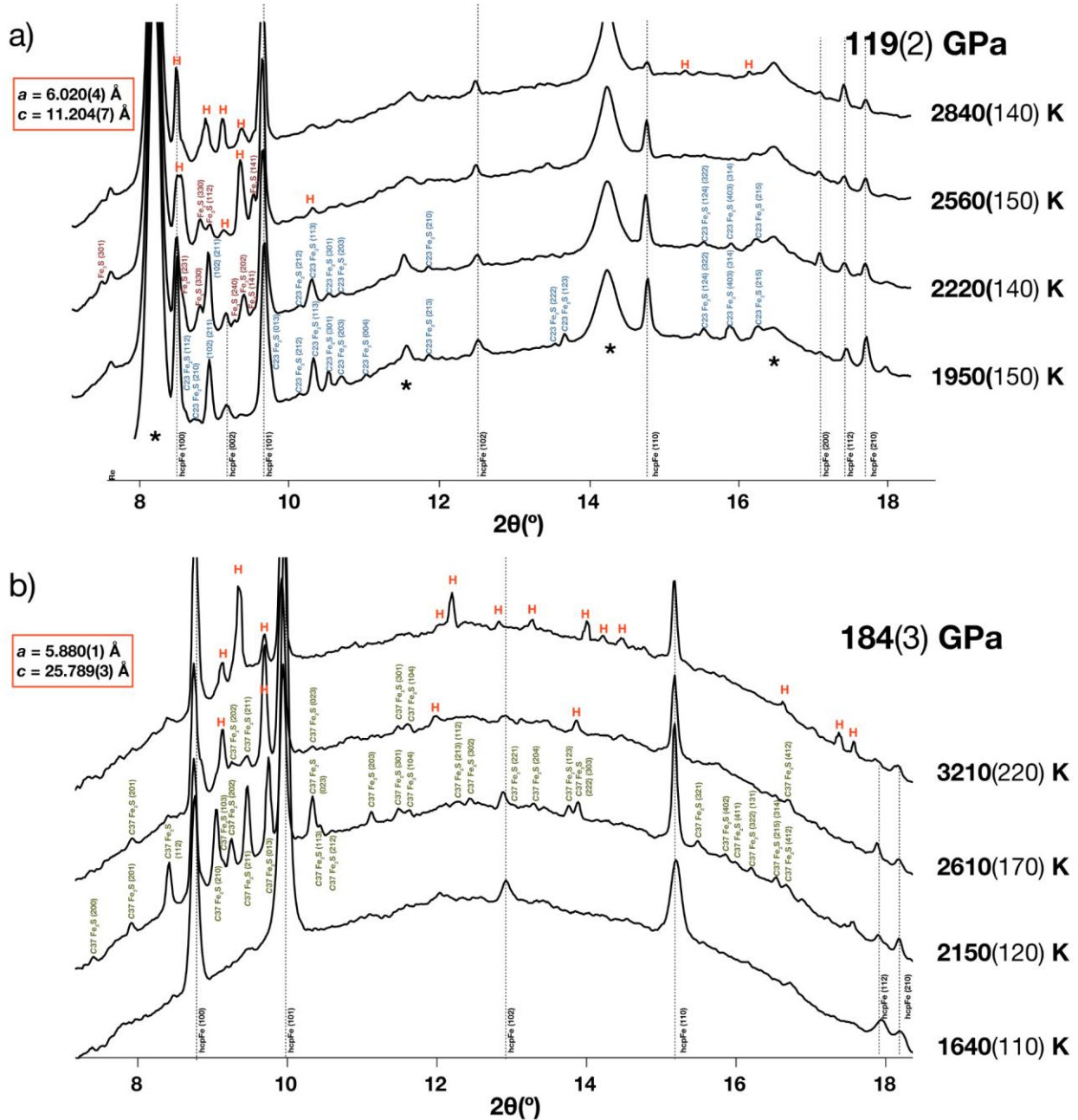
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788 **Figure S3.** Comparison of the *C23* and *C37* structure models refined at 130 and 140 GPa,  
789 respectively. In the *C23* Fe<sub>2</sub>S structure, the next nearest sulfur site to the Fe1 tetrahedral site is at  
790 a 2.963(1) Å distance at 130 GPa (dotted line). A 10% contraction of this interatomic distance  
791 and a coordination change is observed in the formation of *C37* Fe<sub>2</sub>S at 140 GPa (right).  
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812 **Figure S4.** X-ray diffraction patterns collected upon heating in an Fe<sub>80</sub>S<sub>20</sub> starting composition at  
 813 a) 119(2) GPa and b) 184(3) GPa. The miller indices for C23 Fe<sub>2</sub>S (blue), Fe<sub>3</sub>S (burgundy), and  
 814 C37 Fe<sub>2</sub>S (green) are provided, and the red “H” symbols represent the observations of the  
 815 formation of Fe<sub>5</sub>S<sub>2</sub>. The growth of large crystallites, along with the disorder and polytypism of  
 816 this phase make for challenging powder diffraction indexing, but the lattice parameters of Fe<sub>5</sub>S<sub>2</sub>  
 817 indexed in the reciprocal space after quenching from these high *P-T* conditions are provided.



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