

1 Pre-print Statement for

2 **“Fe<sub>5</sub>S<sub>2</sub> identified as a host for sulfur in Earth and**  
3 **planetary cores”**

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16  
17 Dear Earth Arxiv,

18 This manuscript is a non-peer reviewed preprint submitted to *EarthArXiv*. It has been  
19 submitted for peer review at *Earth and Planetary Science Letters*.

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22 Sincerely,  
23 Claire Zurkowski

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# Fe<sub>5</sub>S<sub>2</sub> identified as a host of sulfur in Earth and planetary cores

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## ABSTRACT

Cosmochemical considerations suggest that sulfur is a light alloying element in rocky planetary cores of varying sizes and oxidation states. High pressure-temperature (*P-T*) iron-sulfide phase relations therefore play a role in inner core crystallization and outer core thermochemical convection in a wide range of planetary bodies. The iron-saturated Fe-S phase relations were investigated to 200 GPa and 3250 K using single-crystal and powder X-ray diffraction techniques in a laser-heated diamond anvil cell. At high temperatures between 120 and 200 GPa, a hexagonal sulfide was synthesized with recrystallized hcp-Fe. The unit-cell parameters and diffraction intensities reveal an Fe<sub>5</sub>S<sub>2</sub> stoichiometry adopting the Ni<sub>5</sub>As<sub>2</sub> structure (*P6<sub>3</sub>cm*, *Z* = 6). Fe<sub>5</sub>S<sub>2</sub> is characterized by significant Fe-Fe bonding, complex coordination environments, and positional disorder and polytypism along the *c*-axis. Upon heating at 120 GPa, *I*-4 Fe<sub>3</sub>S is observed to break down into Fe<sub>5</sub>S<sub>2</sub> + Fe, whereas with heating above 120 GPa, Fe<sub>2</sub>S + Fe reacts to form Fe<sub>5</sub>S<sub>2</sub> + Fe at high temperatures. A *C23–C37* Fe<sub>2</sub>S transition is observed above 130 GPa. These results establish that Fe<sub>5</sub>S<sub>2</sub> is the stable Fe-rich sulfide across much of Earth's outer core and will crystallize over an extensive depth up to the core-mantle boundary as the core

63 cools overtime. The increased metal-metal bonding observed in  $\text{Fe}_5\text{S}_2$  compared to the other high  
64 *P-T* iron sulfides would likely contribute to signatures of higher conductivity from regions of  
65  $\text{Fe}_5\text{S}_2$  crystallization. As Earth's core is multicomponent,  $\text{Fe}_5\text{S}_2$  could further serve as a host for  
66 Ni and Si as has been observed in the related meteoritic phase perryite,  $(\text{Fe}, \text{Ni})_8(\text{P}, \text{Si})_3$ , adding  
67 intricacies to elemental partitioning during inner core crystallization. The stability of  $\text{Fe}_5\text{S}_2$   
68 presented here is key to understanding the role of sulfur in the crystallization sequences that  
69 drive the geodynamics and dictate the structures of Earth and rocky planetary cores.

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## 1. INTRODUCTION

72 Earth and the terrestrial planets are composed of silicate mantles and iron-rich metallic  
73 cores (e.g. Birch 1952; McDonough and Sun 1995). Earth's core is comprised of a convecting  
74 liquid metal outer core and an actively crystallizing, denser inner core (Birch 1952; Dziewonski  
75 and Anderson 1981; Fearn and Loper, 1981). The seismically determined 3-8% density deficit in  
76 Earth's core compared to pure iron is thought to be accounted for by cosmochemically abundant  
77 light alloying elements, such as Si, O, S, C, and H (Masters and Gubbins, 2003; Irving et al.  
78 2018; Kuwayami et al. 2020; McDonough 2003), that introduce complex *P-T*-dependent melting  
79 and crystallization thermodynamics and help drive core convection as planets cool over time  
80 (Fearn and Loper, 1981; Stevenson 1981; 1988; Nimmo 2015).

81 While there remains no method to directly sample Earth's or any terrestrial planetary  
82 core, iron meteorites are recognized as core relics of disrupted planetesimals, and have long  
83 provided insight into the light elements that sequester into the metallic component of planetary  
84 interiors (e.g. Scott and Wasson 1975; Malvin et al. 1984). Of the candidate core-alloying  
85 elements, sulfur is present in nearly all iron meteorites, suggesting that sulfur is a core alloying

86 element in rocky planets of varying sizes, oxidation states, and formation histories (Scott and  
87 Wasson 1975; Jones and Drake, 1983; Kruijer et al. 2014). Sulfur easily alloys with iron to high  
88 pressures and temperatures, it is depleted in the silicate mantle compared to chondrites, and it  
89 lowers the melting temperature of pure iron (e.g. Evans 1970; Fei 1995; McDonough and Sun  
90 1995; Fei et al. 1997; Ozawa et al. 2013; Tateno et al. 2019). It follows that during the high  
91 energy impact conditions of late stage proto-Earth formation, sulfur would have facilitated metal  
92 melt formation and density driven core segregation (Ringwood 1966; Murthy and Hall 1970;  
93 Shannon and Agee, 1996; Stevenson, 1988; Yoshino et al., 2003). Examination of the structural  
94 properties of iron sulfides at high  $P$ - $T$  is therefore fundamental to characterizing the chemistry  
95 and thermodynamics of Earth and planetary cores.

96         The Fe-S phase relations are sensitive to  $P$ - $T$ -composition conditions, and numerous Fe-  
97 rich sulfide compounds have been observed: FeS, Fe<sub>3</sub>S<sub>2</sub>, Fe<sub>2</sub>S, and Fe<sub>3</sub>S (Evans 1970; King and  
98 Prewitt 1982; Fei et al. 1995; Fei et al. 1997; Fei et al. 2000; Koch-Müller et al. 2002; Kamada et  
99 al. 2010; Ozawa et al. 2013; Mori et al. 2017; Tateno et al. 2019). Fe<sub>3</sub>S is reported to be the  
100 stable Fe-rich sulfide between 21 and 250 GPa (Fei et al. 2000; Kamada et al. 2010; Ozawa et al.  
101 2013), making it relevant at Earth's outer core conditions. Single crystal analysis of recovered  
102 Fe<sub>3</sub>S from 21 GPa establishes that it adopts a Fe<sub>3</sub>P-type structure ( $I$ -4,  $Z$ =8) (Fei et al. 2000), and  
103 no structural transitions have been reported in Fe<sub>3</sub>S to 250 GPa (Ozawa et al. 2013; Kamada et  
104 al. 2010; Thompson et al. 2020; Seagle et al. 2006; Morard et al. 2008; Mori et al. 2017). Above  
105 ~250 GPa, Fe<sub>3</sub>S breaks down at low temperatures into a  $B2$  FeS + Fe phase assemblage, and  
106 FeS+Fe reacts at high temperatures to form  $C37$  Fe<sub>2</sub>S + Fe at 306 GPa (Ozawa et al. 2013; Mori  
107 et al. 2017; Tateno et al. 2019).

108           Although tetragonal Fe<sub>3</sub>S has been reported to be stable to pressures encompassing  
109 Earth's upper-to-mid-outer core, crystal structure analysis of Fe<sub>3</sub>S has not been conducted at  
110 Earth's core pressures, and unit-cell parameters for Fe<sub>3</sub>S have only been reported up to 120 GPa  
111 at high temperatures (Seagle et al. 2006; Morard et al. 2008; Thompson et al. 2020). In the  
112 current study, Fe-rich Fe-S compositions were probed using single crystal and powder X-ray  
113 diffraction techniques which revealed that at high-temperatures between 120–200 GPa, hcp-Fe  
114 coexists with a hexagonal Fe<sub>5</sub>S<sub>2</sub> sulfide (Ni<sub>5</sub>As<sub>2</sub>-type, *P6<sub>3</sub>cm*, *Z* = 6). This complex crystal  
115 structure exhibits pervasive positional disorder and various stacking variations, such that single-  
116 crystal diffraction techniques were critical in identifying its crystal-chemistry. Increasing across  
117 the pressure range of 120–200 GPa, we find that Fe<sub>3</sub>P-type Fe<sub>3</sub>S + Fe, *C23* Fe<sub>2</sub>S + Fe, and *C37*  
118 Fe<sub>2</sub>S + Fe all react at high temperatures to form Fe<sub>5</sub>S<sub>2</sub> + hcp-Fe. The single-crystal derived  
119 structures for *C23* Fe<sub>2</sub>S, *I-4* Fe<sub>3</sub>S, and *C37* Fe<sub>2</sub>S are also presented. The stability of Fe<sub>5</sub>S<sub>2</sub> with  
120 recrystallized iron to 200 GPa at high temperatures reveals that Fe<sub>5</sub>S<sub>2</sub> will eventually crystallize  
121 over a significant depth range of Earth's outer core.

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## 2. MATERIALS AND METHODS

124           Experiments targeting multigrain synthesis in the Fe-S system consisted of mixtures of Fe  
125 (99.9+%, <10µm, Alfa Aesar) and iron sulfide (FeS, 99.99%, Alfa Aesar) in Fe+12.5 wt% S  
126 (Fe<sub>80</sub>S<sub>20</sub>), Fe + 23 wt% S (Fe<sub>67</sub>S<sub>33</sub>) compositions. The Fe-FeS compositions were ground in  
127 ethanol in a pestle and mortar for 1 hour then mixed dry for a short interval to homogenize any  
128 density settling during alcohol evaporation.

129           Pressure was generated using BX-90-type (Kantor et al. 2012) diamond anvil cells (DAC)  
130 with type 1 Boehler-Almax conical diamonds and seats, and Mao symmetric-type DACs with

131 Type 1 standard cut diamonds mounted on tungsten carbide of cubic-boron nitride (cBN) seats.  
132 Diamond culets ranged from 150-50  $\mu\text{m}$  in diameter. Foils of  $\text{Fe}_{80}\text{S}_{20}$  or  $\text{Fe}_{67}\text{S}_{33}$  were produced  
133 by pressing the Fe–S starting powders between two ungasketed diamond anvils, then loaded  
134 between pellets of KCl or  $\text{SiO}_2$ . Sample chambers ranged from 180-25  $\mu\text{m}$  in diameter and were  
135 drilled from pre-indented rhenium gaskets. Samples were dried at 100° C for 30 minutes prior to  
136 pressurization.

137 Angle dispersive X-ray diffraction (XRD) experiments were conducted at Argonne  
138 National Laboratory, Sector 13 ID-D, of the Advanced Photon Source. At Sector 13 ID-D, a 2.5  
139  $\mu\text{m}$  x 3.54  $\mu\text{m}$  full width, half maximum (FWHM) monochromatic X-ray beam tuned to 37 or 42  
140 keV was utilized, and diffracted X-rays were collected with a CdTe 1M Pilatus detector. Sample-  
141 to-detector geometries were calibrated using a  $\text{LaB}_6$  NIST standard and a single crystal of  
142 enstatite was used for calibration of the rotational geometry of the X-ray beam and detector.

143 Double-sided laser heating and *in-situ* XRD collection took place at Sector 13 ID-D. Fiber  
144 lasers shaped with  $\sim 10$   $\mu\text{m}$  radius flat tops were aligned with the X-ray beam using the X-ray  
145 fluorescence of the sample pressure media or the gasket (Prakapenka et al., 2008). During  
146 heating, thermal emission from a 6  $\mu\text{m}$  central region of the laser-heated spot was measured  
147 spectroradiometrically and fit to a gray body approximation (Heinz and Jeanloz, 1987). A 3%  
148 temperature correction was then applied to correct for axial gradients through the sample  
149 (Campbell et al., 2007; 2009). The laser power on each side of the sample was adjusted to  
150 maintain uniform double-sided heating and X-ray diffraction and temperature measurements  
151 were collected every  $\sim 200$  K with 1s exposure times. Samples were typically quenched at high  
152 temperatures within 30-45 minutes of heating or after a phase transition and suitable grain  
153 growth was observed. Powder diffraction patterns were processed using Dioptas (Prescher and

154 Prakapenka, 2015) and CrysaliPro (Rigaku OD, 2018). Pressure was determined using the  
155 equation of state of hcp-Fe (Dewaele et al. 2006).

156       Upon quenching, X-ray diffraction maps of the heated spot were collected across a 100  
157  $\mu\text{m}^2$  square region in 3  $\mu\text{m}$  steps. Map locations showing high intensity, spotty diffraction  
158 patterns were chosen for multigrain single-crystal type X-ray diffraction collection approach, as  
159 these features indicate high-temperature induced crystallite growth. At chosen map locations, X-  
160 ray diffraction images were collected across  $\pm 17$  to  $\pm 30^\circ$  rotational scans in  $0.25^\circ$ – $0.5^\circ$  steps with  
161 1–4 s exposure times. The diffraction reflections were then mapped in the reciprocal space and  
162 target grains were identified and separated from reflections associated with the pressure media,  
163 iron, and diamond (Rigaku OD, 2018).

164       Target Fe-S lattices were then indexed, and the peak intensities were integrated and  
165 reduced using CrysaliPro (Rigaku OD, 2018). Absorption corrections and scaling factors were  
166 applied to the structure factors in CrysaliPro using the multi-scan method via the Scale3  
167 Abspack program (Rigaku OD, 2018). The final structure factors and lattice geometries were  
168 then refined to known structure models (El-Boragy et al. 1970; Oryshchyn et al. 2011) using  
169 SHELXL2014/7 (Sheldrick 2015). Reflections showing anomalous calculated versus measured  
170 structure factors; likely due to overlap with diamond and other phases in the multigrain sample,  
171 resolution limitations, and volume of crystal illuminated by the X-ray beam, were omitted.  
172 Structure models were visualized using Vesta (Momma and Izumi 2011) during the refinement  
173 procedure.

### 3. RESULTS

#### 3.1 Synthesis and identification of $Fe_5S_2$ to 200 GPa

Upon compression between 119(2) and 193(4) GPa and with heating to temperatures between 2400–3300 K, the diffraction patterns are characterized by recrystallized of hcp-Fe and diffuse streaks and sets of closely spaced reciprocal nodes suggestive of a complex atomic arrangement of the coexisting sulfide (Figure 1, S1). With sustained heating, large sulfide grain growth (3 – 6  $\mu\text{m}$ ) is observed (Figure 1). The diffraction angles for this synthesized sulfide phase are not compatible with the  $Fe_3P$ -type  $Fe_3S$  at these conditions (Kamada et al., 2014). Recrystallization of hcp-Fe with the sulfide crystallites at high temperatures was observed in all experiments using the  $Fe_{80}S_{20}$  and  $Fe_{67}S_{33}$  starting powders as evidenced by the spotty (100) and (101) hcp-Fe rings in Figure 1a. Fe recrystallization with this high-temperature Fe-sulfide establishes that the probed sample locations were in a Fe-saturated phase field (Figure 1a) and this sulfide phase is important to consider further in the context of Fe-rich planetary cores.

Upon temperature quenching in this pressure range, diffraction images were collected while rotating the DAC across a +/- 17–30° range (depending on the DAC configuration). Grains of a hexagonal lattice were identified in the reciprocal space with indexed parameters:  $a = 5.979(3)$  Å,  $c = 11.088(6)$  Å at 140(2) GPa and 3070(180) K (Figure 1b, Table 1). However, across the  $P$ - $T$  explored, three polytypes were observed for this phase (Table 1). Each indexed grain exhibits an  $a$  axial length of  $\sim 6$  Å, while 3 differing  $c$  axial lengths are observed:  $\sim 11$ , 26, and 73 Å. Diffraction mappings show that the polytypism is accompanied by diffuse scattering along the  $c$  direction suggesting positional disorder along this axis (Figure S1). Decreased diffuse scattering is observed after continued heating at peak temperatures in the KCl pressure media,



196 and grains with  $a \sim 6 \text{ \AA}$ ,  $c \sim 11 \text{ \AA}$  were indexed (Figure 1b). This unit cell likely represents the  
197 most positionally ordered arrangement relevant at these high  $P$ - $T$  conditions.

198 The crystal structure of the Fe-sulfide synthesized at 140(2) GPa and 3070(180) K was  
199 determined based on 159 observed reflections at these extreme conditions (Table 2). Assessment  
200 of the systematic absences for the reduced structure factors suggests a  $P6_3cm$  space group, and  
201 structural solution and positional and displacement parameter refinement converged on an  $\text{Fe}_5\text{S}_2$   
202 compound adopting the  $\text{Ni}_5\text{As}_2$  structure type (Table 2, 3; Figure 2) (Oryshchyn et al. 2011). The  
203 measured unit cell parameters for this polytype are compatible with 6 formula units per cell  
204 volume of  $\text{Fe}_5\text{S}_2$ . Furthermore, analogous polytypism due to stacking variations along the  $c$   
205 direction have been observed in related transition metal binary phases such as  $\text{Pb}_5\text{As}_2$  (Saini et al.  
206 1964). The diffraction angles and intensities measured in this study cannot be modeled based on  
207 the tetragonal  $\text{Fe}_3\text{S}$  structure previously reported at these conditions (Kamada et al., 2012;  
208 Ozawa et al., 2013; Mori et al., 2017).

209 The  $\text{Ni}_5\text{As}_2$  structure that  $\text{Fe}_5\text{S}_2$  adopts is a slight modification of the  $\text{Pb}_5\text{Sb}_2$  structure (El-  
210 Boragy et al. 1970), where the  $M6$  site ( $M = \text{metal}$ ) (Table 3) is split about its position given half  
211 occupancy, changing its Wyckoff site from 6c to 12d (Figure 2) (Oryshchyn et al. 2011). A  
212 significant improvement to the  $\text{Fe}_5\text{S}_2$  refinement statistics was observed when incorporating the  
213 disordered Fe6 site, suggesting that the  $\text{Ni}_5\text{As}_2$  structure model better describes the  $\text{Fe}_5\text{S}_2$   
214 structure factors (Table 2). Displacement parameters for the  $\text{Fe}_5\text{S}_2$  structure model were refined  
215 as isotropic. Displacement parameters showing errors  $> 3\sigma$  were fixed to a value equal to the  
216 average displacement parameter value for Fe or S sites (Table 3). The isotropic displacement  
217 parameter for the disordered Fe site was also fixed at an average  $\text{Fe}_{\text{Uiso}}$  value (Table 3). The  
218 number of reflections collected at these extreme pressures limits the number of statistically

219 meaningful parameters to refine, and fixing displacement parameters to reasonable values  
220 precludes overinterpretation of the current dataset. The high isotropic displacement parameter  
221 observed on the Fe4 site may represent some vacancies on this site, but the dataset is again  
222 deficient to refine site occupancies (Table 3).

223 A CIF file for the final Fe<sub>5</sub>S<sub>2</sub> structure model is provided in Appendix A1. Following  
224 previous descriptions of related *M*<sub>5</sub>*X*<sub>2</sub> phases (e.g. Kjekshus et al. 1973; Oryshchyn et al. 2011),  
225 Fe<sub>5</sub>S<sub>2</sub> can be viewed as an arrangement of 6 Fe sites and 3 S sites with the Fe1, Fe2, Fe4, and  
226 Fe6 sites in 13-fold coordination, the Fe3 and Fe5 sites in 12-fold coordination, and the S sites in  
227 10-fold coordination (Figure 2c, Table 4). All sites are coordinated by both Fe and S sites.

228 Final R<sub>1</sub> values  $\approx$  10% attest to the less-than ideal quality of the multigrain dataset as a  
229 possible result of the 1-2 megabar synthesis conditions and observed *c* axial disorder and  
230 stacking complexities in Fe<sub>5</sub>S<sub>2</sub> (Table 2). Previous studies of isomorphic Ni<sub>5</sub>As<sub>2</sub> and related  
231 Pb<sub>5</sub>As<sub>2</sub> at ambient conditions have also reported similar quality of refinements despite obtaining  
232 significantly more reflections in the absence of a DAC (e.g. Saini et al. 1964; El-Boragy et al.  
233 1970; Kjekshus et al. 1973). Notable challenges regarding the refinement of the Fe<sub>5</sub>S<sub>2</sub> structure  
234 model at these extreme conditions are discussed here and compared with similar difficulties  
235 reported in previous characterizations of this structure type.

236 Thirteen violations of the *P*6<sub>3</sub>*cm* systematic absence condition:  $l = 2n + 1$  for (*okl*), were  
237 flagged during the refinement of Fe<sub>5</sub>S<sub>2</sub>. The reflections associated with these systematic absence  
238 violations were examined in the raw diffraction images and show low, diffuse intensity.  
239 Discrepancy over the presence or absence of weak reflections with (*okl*),  $l=2n$  has been reported  
240 in previous investigations of Ni<sub>5</sub>As<sub>2</sub> and Pb<sub>5</sub>As<sub>2</sub> (e.g. Saini et al. 1964; El-Boragy et al. 1970;  
241 Kjekshus et al. 1973), suggesting that these studies also faced difficulties with space-group

242 determination. Observations of these low-intensity reflections could be a result of residual  
243 disorder along the  $c$  direction, and longer heating cycles may be required for the atoms to arrange  
244 into equilibrium positions. It is likewise possible that varying synthesis methods for  $\text{Ni}_5\text{As}_2$  and  
245  $\text{Pb}_5\text{As}_2$  in previous ambient condition studies affected the  $c$  axial atomic arrangement.  $\text{Ni}_5\text{As}_2$   
246 and isomorphic  $\text{Ni}_5\text{P}_2$  also exhibits a homogeneity range of  $\sim 71.25 - 72.7$  atomic % As  
247 (Kjekshus et al. 1973; Litasov et al., 2019), and slight modifications of this structure based  
248 stoichiometric and stacking variations result in trigonal structures such as  $\text{Ni}_{31}\text{Si}_{12}$  (Frank and  
249 Schubert 1971) and  $(\text{Fe, Ni})_8(\text{Si, P})_3$  (perryite) (Okada et al. 1991). Attempts to refine the current  
250 Fe-sulfide phase with the  $\text{Ni}_{31}\text{Si}_{12}$  or perryite structure models did not significantly improve the  
251 refinement statistics as the current high  $P$ - $T$  dataset is too limited to resolve the intricacies that  
252 differentiate these structures. Based on the presence of disorder, polytypism, anisotropic  
253 vibrational motion, and nonstoichiometry in the related  $M_5X_2$  phases, additional nuances to the  
254  $\text{Fe}_5\text{S}_2$  structure model may be developed in future studies; however, the identification and  
255 characterization of the  $\text{Fe}_5\text{S}_2$  crystal structure determined here to 200 GPa is novel, and the  
256 observations and challenges reported in this study align with that of previous analyses of  $\text{Ni}_5\text{As}_2$   
257 and  $\text{Pb}_5\text{Sb}_2$  that were not affected by the limitations associated with performing microdiffraction  
258 in a DAC at extreme conditions.

259

### 260 *3.2 Other sulfides observed during heating to 200 GPa and comparison of their structures with* 261 *$\text{Fe}_5\text{S}_2$*

262 X-ray diffraction from seven heating cycles performed between 100 and 200 GPa and to  
263 3300 K provide insight into the Fe-rich Fe-S phase relations at outer core pressures and to high  
264 temperatures. In each heating experiment, temperatures near melting were attained, and lattices

265 of Fe<sub>5</sub>S<sub>2</sub> were indexed in the reciprocal space upon quenching. By further probing locations  
266 across the thermal gradient of the laser heated spots, additional Fe-sulfides were characterized,  
267 offering information on the lower temperature Fe-saturated sulfide crystal chemistries (Table 1,  
268 2). These include Fe<sub>3</sub>P-type Fe<sub>3</sub>S (*I*-4, *Z* = 8), *C*23 Fe<sub>2</sub>S (Co<sub>2</sub>P-type, *Pnma*, *Z* = 4), and *C*37 Fe<sub>2</sub>S  
269 (Co<sub>2</sub>Si-type, *Pnma*, *Z*=4), in lower temperature regions at 119 GPa, 131 GPa, and 140 GPa,  
270 respectively (Figure 3; Table 1, 2).

271 Grains of tetragonal Fe<sub>3</sub>S (*I*-4, *Z* = 8) (referred to herein as *I*-4 Fe<sub>3</sub>S) were indexed with  
272 parameters:  $a = 8.094(3)$  Å,  $c = 3.990(2)$  Å at 119(2) GPa, and refined to the Fe<sub>3</sub>P-type structure  
273 model (*I*-4, *Z* = 8) (referred to herein as *I*-4 Fe<sub>3</sub>S), in agreement with previous studies (Fei et al.  
274 2000; Seagle et al. 2006; Morard et al. 2007; Kamada et al. 2010; Kamada et al. 2012;  
275 Thompson et al. 2020) (Table 2, Figure 3a, Appendix A2). The structure can be viewed as  
276 containing 3 tetrahedrally coordinated Fe sites (Blanchard et al. 2008): one Fe-site is coordinated  
277 only by S atoms with an average bond length of 2.083(9) Å, another Fe site is coordinated by 3 S  
278 atoms and 1 Fe atom with an average bond length of 2.146(8) Å, and the third Fe site is  
279 coordinated by 2 S and 2 Fe atoms with an average bond length of 2.137(8) Å (Figure 3). The  
280 measured interatomic distances are comparable to reports on other transition metal *M*<sub>3</sub>*X*  
281 structures (e.g. Aronsson 1955; Rundqvist 1979).

282 After heating at 131(2) GPa, grains of Fe<sub>3</sub>S were not observed across the heated spot.  
283 Instead, orthorhombic lattices were also identified with parameters  $a = 4.869(2)$  Å,  $b = 3.256(2)$   
284 Å,  $c = 6.139(2)$  Å, compatible with 4 formula units of Fe<sub>2</sub>S. Structural solution and refinement  
285 indicate that the Fe<sub>2</sub>S grains adopt the *C*23 structure (Co<sub>2</sub>P type, *Pnma*, *Z* = 4) in agreement with  
286 previous structural analyses at lower pressures (Zurkowski et al. *in press*) (Table 2, Figure 3b,  
287 Appendix A3). The structure is composed of columns of FeS<sub>4</sub> tetrahedra and columns of FeS<sub>5</sub>

288 square pyramids linked along edges in the *b* direction (Figure 3b). The average Fe–S bond  
289 lengths are 2.213(8) Å and 2.011(2) Å in the square pyramids and tetrahedra, respectively. These  
290 values are comparable with those observed in *C23* Fe<sub>2</sub>S at 90 GPa (Zurkowski et al. *in press*).

291       Upon temperature quenching at 140(2) GPa, orthorhombic grains were identified in the  
292 sample chamber with *a* = 4.667(2) Å, 3.289(1) Å, 6.186(4) Å. This unit cell is similar to the *C23*  
293 Fe<sub>2</sub>S cell measured at 130 GPa, but it exhibits a 4 % contraction of the *a* axial length, a 1%  
294 extension of the *b* and *c* axial lengths, and a 2% volume decrease. Structure solution and  
295 refinement establishes that Fe<sub>2</sub>S adopts the *C37* structure (Co<sub>2</sub>Si-type, *Pnma*, *Z*=4) (Table 2,  
296 Figure 3c, Appendix A4). This result confirms the previously proposed *C23*–*C37* transition in  
297 Fe<sub>2</sub>S at these conditions and presents a comparable volume change (Zurkowski et al., *in press*).  
298 Inherent to the *C23*–*C37* transition is coordination change from the 4-fold Fe1 coordination  
299 polyhedra in the *C23* structure to the 5-fold dipyramid polyhedra in the *C37* structure (Figure 3b,  
300 c). Comparing the *C23* and *C37* structure models determined at 130 and 140 GPa, respectively, a  
301 10% contraction of the interatomic distance involved in the coordination change is observed  
302 (Figure S2). The average Fe–S bond lengths measured in *C37* Fe<sub>2</sub>S are 2.196(2) Å and 2.165(3)  
303 Å in the square pyramids and dipyramids, respectively. These values are comparable with those  
304 observed in *C23* Fe<sub>2</sub>S in this study.

305       Comparison of the interatomic distances and coordination environments in the Fe-  
306 sulfides observed in this study distinguishes Fe<sub>5</sub>S<sub>2</sub> as particularly unique among them. Within an  
307 interatomic distance of 2.7 Å (Table 4, Appendix A1-A4), *C23* and *C37* Fe<sub>2</sub>S consist of Fe sites  
308 coordinated only by S, while Fe<sub>3</sub>S and Fe<sub>5</sub>S<sub>2</sub> mark a drastic increase in Fe-Fe bonding with Fe  
309 surrounded by up to 9 Fe sites in both structures. However, Fe<sub>5</sub>S<sub>2</sub> exhibits Fe-Fe bond lengths up  
310 to 25% shorter than that observed in Fe<sub>3</sub>S at these pressures, further distinguishing the increased

311 iron-iron interactions in the novel Fe<sub>5</sub>S<sub>2</sub> at Earth and planetary core pressures (Figure 2c, Table  
312 4).

313

### 314 *3.3 Fe-rich sulfide phase relations at core-mantle boundary pressures*

315 The presented single-crystal derived structures of Fe<sub>5</sub>S<sub>2</sub>, Fe<sub>3</sub>S, C23 Fe<sub>2</sub>S and C37 Fe<sub>2</sub>S  
316 were then used to inform the changes in diffraction patterns collected during heating between  
317 110 and 200 GPa and to ~3250 K (Figure 4). Beginning at 112 GPa, diffraction signal from C23  
318 Fe<sub>2</sub>S coexisting with Fe was first identified upon heating of the Fe<sub>80</sub>S<sub>20</sub> starting material to  
319 ~2000 K (Figure 4). With continued heating, peaks associated with tetragonal Fe<sub>3</sub>S were  
320 observed over a limited temperature range ( $\lesssim$  2400 K) until diffuse scattering signal and Bragg  
321 reflections from Fe<sub>5</sub>S<sub>2</sub> first appeared. Crystallization of the Fe<sub>5</sub>S<sub>2</sub> grains with hcp-Fe occurred  
322 with continued heating to 119(2) GPa and 2840(180) K (Figure 4).

323 Upon heating beginning at 120 GPa, C23 Fe<sub>2</sub>S was observed coexisting with Fe to 122(1)  
324 GPa and 2290(120) K, above which the onset of diffraction from the Fe<sub>5</sub>S<sub>2</sub> phase was identified  
325 and Fe<sub>5</sub>S<sub>2</sub> crystallites formed coexisting with hcp-Fe to 131(2) GPa and 3050(140) K (Figure 4).  
326 I-4 Fe<sub>3</sub>S was not observed. With continued heating cycles between 133(1) GPa and 194(2) GPa  
327 and up to 3250 K, C37 Fe<sub>2</sub>S coexists with Fe at moderate temperatures and a reaction to form  
328 Fe<sub>5</sub>S<sub>2</sub> occurs at high temperatures (Figure 4). A pressure-induced C23–C37 Fe<sub>2</sub>S transition is  
329 therefore constrained between 125 and 135 GPa, in agreement with previous investigations of  
330 Fe<sub>2</sub>S (Zurkowski et al. *in press*).

## 4. DISCUSSION

### 4.1 Discrepancies among the current and previous studies

Transitional metal binary compounds with metal-to-nonmetal ratios ranging from 2.33–2.66 (~70–73 atm% metal) predominantly adopt complex trigonal or hexagonal structures with considerable metal-metal bonding (Chen and Whitmire 2018 and references therein). Our observations of the disorder, polytypism, and complex coordination environments inherent to  $\text{Fe}_5\text{S}_2$  demonstrate that it aligns with this systematic characterization. Interestingly, this work elucidates that the  $\text{Fe}_5\text{S}_2$  atomic arrangement is thermodynamically favored and expresses greater Fe-Fe interactions at the extreme conditions of Earth's outer core compared to that of  $I\text{-}4\text{ Fe}_3\text{S}$ .

These results are contrary to previous reports of  $\text{Fe}_3\text{S}$  stability on to high temperatures in Fe-rich systems to 250 GPa (Kamada et al. 2010; Kamada et al. 2012; Ozawa et al. 2013; Mori et al. 2017), but our observations may account for these discrepancies. First, interpretation of  $\text{Fe}_5\text{S}_2$  in the integrated powder diffraction patterns is difficult due to the variation in diffraction signal obtained from the  $\text{Fe}_5\text{S}_2$  polytypes and the low intensity scattering from the disordered sites during its formation (Figure S1, S3).  $\text{Fe}_5\text{S}_2$  also forms large crystallites, limiting the orientations of the phase and rendering indexation of all diffraction angles for  $\text{Fe}_5\text{S}_2$  quite challenging without rotating the sample (Figure 1a). The implementation of single-crystal X-ray diffraction techniques at high pressures was critical in the current study to accurately characterize the hexagonal unit cell geometry and complex structure of the  $\text{Ni}_5\text{As}_2$ -like  $\text{Fe}_5\text{S}_2$  and to constrain the stability field of  $I\text{-}4\text{ Fe}_3\text{S}$ .

Several studies have also reported chemically analyzed  $\text{Fe}_3\text{S}$  grains in samples recovered from high temperatures in the 200–250 GPa range (Mori et al. 2017; Yokoo et al. 2019; Ozawa et al. 2013), but the difference in Fe content between  $\text{Fe}_3\text{S}$  and  $\text{Fe}_5\text{S}_2$  may be as few as 3%. This

354 value is generally within  $3\sigma$  uncertainty reported for chemical analyses of samples recovered  
355 from these extreme conditions, posing another challenge for distinguishing between the synthesis  
356 of  $\text{Fe}_3\text{S}$  and  $\text{Fe}_5\text{S}_2$ . Furthermore, based on characterizations of the isomorphic  $\text{Ni}_5\text{As}_2$  and  $\text{Ni}_5\text{P}_2$ ,  
357 these structures exhibit a homogeneity range with up to  $\sim 73$  atm% metal (Kjekshus et al. 1973;  
358 Litasov et al., 2019). TEM analysis of a Fe–S sample recovered from 236 GPa and 2980 K  
359 reveals sulfide grains with on average  $\sim 73$  atm% Fe (Ozawa et al. 2013); this value is within  
360 0.3–1.25 atom % of the possible range of  $\text{Fe}_5\text{S}_2$  stoichiometries and 2 atomic percent less than an  
361  $\text{Fe}_3\text{S}$  composition. The results from previous chemical analyses of samples recovered from outer  
362 core pressures and high temperatures (Mori et al. 2017; Yokoo et al. 2019; Ozawa et al. 2013)  
363 therefore do not contradict the current results, and simply indicate previous misinterpretation of  
364  $\text{Fe}_5\text{S}_2$  as  $\text{Fe}_3\text{S}$ , based on EDS measurements alone, in the absence of single-crystal diffraction  
365 analysis.

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#### 367 *4.2 Sulfur in Earth and Planetary cores*

368 The stability of  $\text{Fe}_5\text{S}_2$  and the phase relations observed in this study present novel  
369 constraints on the material properties of Fe-sulfides at conditions relevant to Earth's outer core,  
370 Venus' core, and exoplanetary cores of similar size and core-mass fraction (CMF). Namely, at  
371 pressures  $\geq 120$  GPa,  $\text{Ni}_5\text{As}_2$ -type  $\text{Fe}_5\text{S}_2$ , not tetragonal  $\text{Fe}_3\text{S}$  as previously believed, is the  
372 relevant Fe-rich sulfide. As Earth's outer core crystallizes over time,  $\text{Fe}_5\text{S}_2$  will eventually  
373 crystallize over a large depth range up to the CMB. Tetragonal  $\text{Fe}_3\text{S}$  has been confirmed to 120  
374 GPa (Figure 4) (Seagle et al. 2006; Morard et al. 2008; Thompson et al., 2020), constraining the  
375  $\text{Fe}_3\text{S} + \text{Fe} \rightarrow \text{Fe}_5\text{S}_2 + \text{Fe}$  transition to  $\sim 120$  GPa. Rocky planetary bodies with sizes and CMFs  
376 like that of Venus (CMB = 114 GPa) (Aitta 2012) would likely crystallize  $\text{Fe}_3\text{S}$  at the CMB, but



377  $\text{Fe}_5\text{S}_2$  will be crystallizing over much of the core's depth.  $C37 \text{Fe}_2\text{S}$  is reported to be stable on the  
378 solidus above 306 GPa in Fe-rich systems (Tateno et al. 2019), suggesting a  $\text{Fe}_5\text{S}_2 + \text{Fe} \rightarrow \text{Fe}_2\text{S}$   
379 + Fe transition in the 200–300 GPa range.  $C37 \text{Fe}_2\text{S}$  remains the candidate Fe-rich sulfide at  
380 Earth's ICB, but for planets like Venus with a central pressure of 274 GPa (Aitta 2012),  $\text{Fe}_5\text{S}_2$   
381 may instead be the relevant Fe-rich sulfide, necessitating experimental determination of the  
382  $\text{Fe}_5\text{S}_2 \rightarrow \text{Fe}_2\text{S} + \text{Fe}$  decomposition  $P$ - $T$  boundary. Furthermore, as  $\text{Fe}_5\text{S}_2$  exhibits a complex  
383 crystalline arrangement with increased metal-metal bonding compared to  $\text{Fe}_2\text{S}$  and  $\text{Fe}_3\text{S}$  at core  
384 pressures, the significant depth over which  $\text{Fe}_5\text{S}_2$  crystallizes in and Venus- to Earth-sized  
385 planetary cores may also contribute to higher conductivity signatures in these regions.

386         The stability of  $\text{Fe}_5\text{S}_2$  coexisting with Fe between 120 and 200 GPa to to 3250 K also  
387 requires a change in melting behavior associated with the change in solidus phase from  $I-4 \text{Fe}_3\text{S}$   
388 to  $\text{Fe}_5\text{S}_2$  above 120 GPa. As  $\text{Fe}_5\text{S}_2$  is observed to temperatures of  $\sim 3250$  K, a kink in the Fe-  
389 sulfide solidus curve (Mori et al. 2017) is proposed to accommodate this (Figure 7). Further  
390 work examining melt signal will be helpful for accurately determining the shape of the solidus  
391 curve above 120 GPa, but the disorder observed in  $\text{Fe}_5\text{S}_2$  may challenge the determination of  
392 melting that is commonly based on diffuse scattering appearing in the diffraction background.  
393 Further changes in melting relations occur as  $C37 \text{Fe}_2\text{S}$  becomes stable to melting temperatures  
394 above 306 GPa, and future experiments must investigate the change in melting temperatures  
395 associated with this transition at lower outer core conditions.

396         Terrestrial core chemistries are multicomponent; Si and Ni are also important core-  
397 alloying elements in planets such as Earth, drawing attention to the potential stability of the  
398 perryite  $(\text{Fe}, \text{Ni})_8(\text{Si}, \text{P})_3$  structure in (Fe, Ni) – (Si, S)-rich cores. Perryite, observed in enstatite  
399 chondrites and aubrites, adopts a trigonal stacking variant of the  $\text{Ni}_5\text{As}_2$ -type structure (Wasson

400 and Wai, 1970; Okada et al. 1991), and iron phosphides tend to adopt analog structures to those  
401 observed in iron sulfides (i.e., Fei et al. 2000; Dera et al. 2008; Gu et al. 2014; Gu et al. 2016;  
402 Zurkowski et al., in press). Hence,  $\text{Fe}_5\text{S}_2$  could potentially serve as a host for nickel and silicon  
403 in its perryite-like structure at Earth's outer core conditions. Furthermore, Si has recently been  
404 shown to dissolve into iron-sulfide phases and expand their stability field to higher pressure (Tao  
405 and Fei, 2021). The dissolution of silicon into  $\text{Fe}_5\text{S}_2$  may expand its stability to higher pressures,  
406 making it potentially relevant at Earth's ICB conditions. If this is the case, the partitioning of Si  
407 and Ni between  $(\text{Fe, Ni})_5(\text{S, Si})_2$  and  $(\text{Fe, Ni, Si})$  and the density difference between the inner-  
408 core-crystallizing phase and remaining light-element-rich liquid will be crucial to investigations  
409 of the inner-core-density deficit and inner core seismic morphology.

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## 5. CONCLUSIONS

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To assess the phase stability of Fe-rich sulfides at Earth's outer core conditions, Fe-rich sulfide compositions were examined to 200 GPa and 3250 K using single-crystal and powder X-ray diffraction techniques in a laser-heated diamond anvil cell. At high temperatures between 120 and 200 GPa  $\text{Fe}_5\text{S}_2$  is synthesized in the  $\text{Ni}_5\text{As}_2$ -type structure. The atomic arrangement of this hexagonal structure is characterized by positional disorder, complex coordination environments and significant Fe-Fe interactions. Additional polytypism and disorder associated with various stacking arrangements along the  $c$  direction are also observed. Along with  $\text{Fe}_5\text{S}_2$ , grains of  $I-4 \text{Fe}_3\text{S}$ ,  $C23 \text{Fe}_2\text{S}$ , and  $C37 \text{Fe}_2\text{S}$  were also characterized. The stability of tetragonal  $\text{Fe}_3\text{S}$  in Fe-rich systems is constrained to below 120 GPa.  $C23 \text{Fe}_2\text{S} + \text{Fe}$  is observed at moderate temperatures to 130 GPa, and  $C37 \text{Fe}_2\text{S} + \text{Fe}$  is observed at moderate temperatures above 140 GPa. Between 120 and 200 GPa at the highest temperatures probed,  $\text{Fe}_5\text{S}_2$  is stable coexisting

423 with Fe, therefore establishing that Fe<sub>5</sub>S<sub>2</sub>, not Fe<sub>3</sub>S, is stable in Fe-rich systems at outer core  
424 pressures associated with Earth- and Venus-sized bodies. Fe<sub>5</sub>S<sub>2</sub> crystallization can be expected  
425 over a extensive depth range in the cores of these planetary bodies.

426

## 427 **Acknowledgments**

428 Portions of this work were performed at GeoSoilEnviroCARS (The University of Chicago,  
429 Sector 13), Advanced Photon Source (APS), Argonne National Laboratory.

430 GeoSoilEnviroCARS is supported by the National Science Foundation - Earth Sciences (EAR -  
431 1634415). This research used resources of the Advanced Photon Source, a U.S. Department of

432 Energy (DOE) Office of Science User Facility operated for the DOE Office of Science by

433 Argonne National Laboratory under Contract No. DE-AC02-06CH11357. This material is based

434 upon work supported by a National Science Foundation Graduate Research Fellowship to C.C.Z.

435 This work was also supported by the National Science Foundation by grant EAR-1651017 to

436 A.J.C.

## 6. TABLES

438 **Table 1.** Unit cell parameters of Fe<sub>5</sub>S<sub>2</sub>, Fe<sub>3</sub>S, and Fe<sub>2</sub>S measured upon quenching from high *P-T*  
 439 synthesis in this study. Each cell was indexed in the reciprocal space, and the DAC opening, and  
 440 number of reflections obtained for each lattice are listed. For select experiments, high-  
 441 temperature synthesis was conducted without collecting X-ray diffraction data, and the synthesis  
 442 conditions for these experiments are listed as “high T, not measured.”

Sample name	Starting		Room temperature collection								Synthesis conditions	
	material	medium	Phase	P GPa	<i>a</i> Å	<i>b</i> Å	<i>c</i> Å	V Å <sup>3</sup>	DAC opening (°)	no. of reflections	<i>P</i> GPa	<i>T</i> K
C140_P2_map25	Fe80S20	KCl	Fe5S2	105(2)	6.020(4)		11.204(7)	351.6(5)	60	290	119(2)	2840(140)
C140_P2_map31	Fe80S20	KCl	Fe5S2	105(2)	6.050(2)		73.0(2)	2313(1)	60	134	119(2)	2840(140)
C140_P4_map10	Fe80S20	KCl	Fe5S2	118(1)	5.983(3)		11.078(6)	343.4(4)	60	489	140(2)	3070(180)
C140_P4_map14	Fe80S20	KCl	Fe5S2	118(1)	5.979(2)		11.084(4)	343.2(3)	60	379	140(2)	3070(180)
C140_P4_map14	Fe80S20	KCl	Fe5S2	118(1)	5.973(1)		11.094(2)	342.8(1)	60	276	140(2)	3070(180)
C129_P140_map71	Fe67S33	SiO2	Fe5S2	136.4(5)	5.957(4)		71.89(4)	2209(3)	34	84	159(2)	3230(150)
C129_P150_map9	Fe67S33	SiO2	Fe5S2	144.2(8)	5.939(3)		25.99(6)	794(2)	34	76	high T, not measured	
C129_P150_map9	Fe67S33	SiO2	Fe5S2	144.2(8)	5.923(1)		26.004(6)	789.9(3)	34	146	high T, not measured	
C137_P1_map25	Fe80S20	SiO2	Fe5S2	159(1)	5.890(3)		25.837(7)	776.2(5)	40	108	184(3)	3250(260)
C137_P1_map25	Fe80S20	SiO2	Fe5S2	159(1)	5.880(1)		25.789(3)	772.0(2)	40	108	184(3)	3250(260)
C137_P1_map25	Fe80S20	SiO2	Fe5S2	159(1)	5.876(2)		25.884(8)	774.0(4)	40	116	184(3)	3250(260)
C137_P1_map25	Fe80S20	SiO2	Fe5S2	159(1)	5.883(3)		25.85(1)	774.6(7)	40	126	184(3)	3250(260)
C137_P1	Fe80S20	SiO2	Fe5S2	159(1)	5.897(3)		25.90(1)	775.1(6)	40	116	184(3)	3250(260)
C137_P1	Fe80S20	SiO2	Fe5S2	159(1)	5.878(1)		25.784(3)	771.6(2)	40	149	184(3)	3250(260)
C137_P2_34	Fe80S20	SiO2	Fe5S2	179(1)	5.8400(7)		25.583(2)	755.7(1)	40	189	193(4)	3010(160)
C140_P1_map	Fe80S20	KCl	Fe3S	100(1)	8.156(3)		4.025(3)	267.8(3)	60	416	high T, not measured	
C140_P2_map25	Fe80S20	KCl	Fe3S	105(2)	8.094(3)		3.990(2)	261.4(2)	60	296	119(2)	2840(140)
C140_P3_map6	Fe80S20	KCl	C23Fe2S	111(1)	4.869(2)	3.256(2)	6.139(2)	97.3(1)	60	227	131(2)	3050(140)
C140_P4_map10	Fe80S20	KCl	C37 Fe2S	118(1)	4.677(2)	3.289(1)	6.186(4)	95.18(9)	60	188	140(2)	3070(180)

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458 **Table 2.** Select experimental details for crystal structure synthesis and analysis of Fe<sub>5</sub>S<sub>2</sub>, Fe<sub>3</sub>S,  
 459 and the Fe<sub>2</sub>S polymorphs.

Phase	Fe <sub>5</sub> S <sub>2</sub>	Fe <sub>3</sub> S	C23 Fe <sub>2</sub> S	C37 Fe <sub>2</sub> S
Sample Name	C140_P4_map10	C140_P2_map25	C140_P3_map6	C140_P4_map10
<b>Synthesis</b>				
Pressure (GPa)	140(2)	119(2)	131(2)	140(2)
Synthesis Temperature (K)	3070(180)	2840(140)	2050(140)	3070(180)
Symmetry	Hexagonal, <i>P6(3)cm</i> , Z=6	Tetragonal, <i>I-4</i> , Z=8	Orthorhombic, <i>Pnma</i> , Z = 4	Orthorhombic, <i>Pnma</i> , Z = 4
<i>a</i> (Å)	5.979(2)	8.094(3)	4.869(3)	4.677(2)
<i>b</i> (Å)			3.256(4)	3.289(1)
<i>c</i> (Å)	11.087(6)	3.990(2)	6.139(2)	6.186(4)
<i>V</i> (Å <sup>3</sup> )	343.2(3)	226.4(2)	97.3(1)	95.18(9)
<b>Reduction</b>				
No. of measured, independent and observed [ <i>I</i> > 2σ( <i>I</i> )] reflections	396, 309, 157	332, 314, 219	227, 208, 112	188, 158, 105
<i>R</i> <sub>int</sub> , <i>R</i> <sub>sigma</sub>	0.092, 0.062	0.053, 0.072	0.012, 0.016	0.009, 0.014
<b>Refinement</b>				
<i>R</i> [ <i>F</i> <sup>2</sup> > 2σ( <i>F</i> <sup>2</sup> )], <i>wR</i> ( <i>F</i> <sup>2</sup> ), <i>S</i>	0.098, 0.255, 1.12	0.067, 0.164, 1.05	0.053, 0.149, 1.21	0.047, 0.139, 1.22
No. of reflections	156	219	112	105
No. of parameters	21	17	10	10
Δρ <sub>max</sub> , Δρ <sub>min</sub> (e Å <sup>-3</sup> )	2.14, -2.18	1.92, -1.68	1.86, -2.02	1.91, -2.21

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465 **Table 3.** Atomic coordinates of the Fe<sub>5</sub>S<sub>2</sub> refinement model for the data collected at 140(2) GPa  
 466 and quenched from 3070(180) K.

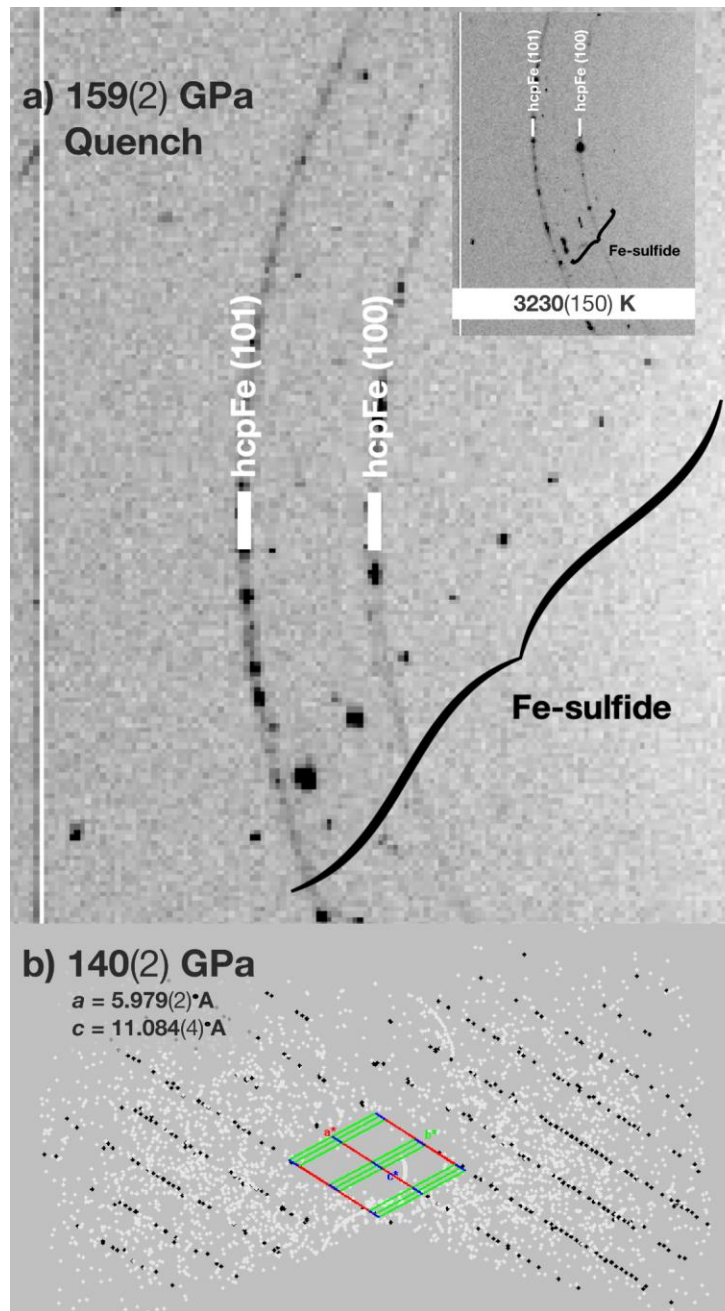
Wycoff site	ATOM	x	y	z	Uiso
<b>2a</b>	<b>Fe1</b>	0	0	0.963	0.021
	error			0.003	0.005
<b>4b</b>	<b>Fe2</b>	1/3	2/3	0.089	0.046
	error			0.003	0.005
<b>6c</b>	<b>Fe3</b>	0.259	0	0.126	0.021
	error	0.002		0.001	
<b>6c</b>	<b>Fe4</b>	0.613	0	0.222	0.021
	error	0.002		0.001	0.003
<b>6c</b>	<b>Fe5</b>	0.284	0	0.330	0.027
	error	0.002		0.001	0.003
<b>12d</b>	<b>Fe6*</b>	0.651	0.056	0.426	0.021
	error	0.003	0.003	0.002	
<b>2a</b>	<b>S1</b>	0	0	0.220	0.026
	error			0.004	
<b>4b</b>	<b>S2</b>	1/3	2/3	0.299	0.026
	error			0.002	
<b>6c</b>	<b>S3</b>	0.673	0	0.030	0.026
	error	0.005		0.003	

\*indicates half occupancy

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480 **Table 4.** Selected interatomic distances for the Fe sites measured in Fe<sub>5</sub>S<sub>2</sub> at 140 GPa. The “#”  
 481 column indicates that number of the “bonded to” sites around the central atom.

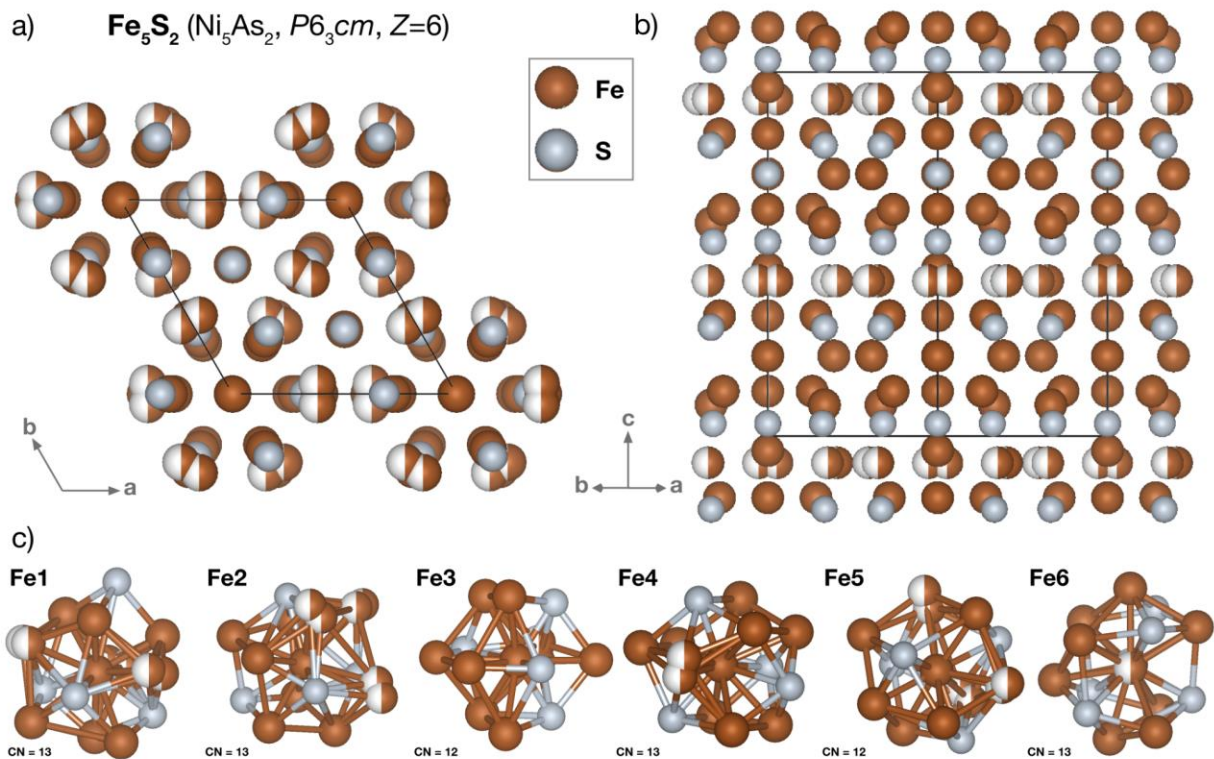
Atom	bonded to	#	distance (Å)
Fe1	-Fe3	3	2.39(2)
	-Fe5	3	2.22(3)
	-Fe6	3	2.313(17)
	-S1	1	2.66(8)
	-S4	3	2.09(3)
	Fe2	-Fe3	3
-Fe4		3	2.375(16)
-Fe6		3	2.40(3)
-S3		1	2.36(4)
-S4		3	2.108(13)
Fe3		-Fe1	1
	-Fe2	2	2.275(8)
	-Fe4	3	2.350(15)
	-Fe5	1	2.30(3)
	-Fe6	1	2.307(17)
	-S1	1	1.90(4)
	-S4	3	2.68(3)
Fe4	-Fe2	2	2.375(16)
	-Fe3	3	2.320(11)
	-Fe5	3	2.416(14)
	-Fe6	1	2.303(17)
	-S1	1	2.326(11)
	-S3	2	2.044(16)
	-S4	1	2.10(3)
Fe5	-Fe1	1	2.22(3)
	-Fe3	1	2.30(3)
	-Fe4	3	2.416(14)
	-Fe6	3	2.135(15)
	-S1	1	2.09(5)
	-S3	2	2.184(10)
	-S4	1	2.27(3)
Fe6	-Fe1	1	2.313(17)
	-Fe2	2	2.40(3)
	-Fe3	1	2.307(17)
	-Fe4	1	2.303(17)
	-Fe5	3	2.135(15)
	-S3	2	2.15(3)
	-S4	3	2.17(3)



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486 **Figure 1.** a) View of a crystallite of the high-temperature  $\text{Fe}_5\text{S}_2$  coexisting with recrystallized  
 487 hcp-Fe after synthesis at 159(2) GPa and 3230(130) K (inset). The spotty rings corresponding to  
 488 the hcp-Fe (100) and (101) planes demonstrate that iron equilibrated at 159(2) GPa and  
 489 3230(130) K. b) View of the reciprocal space for reflections detected in an experiment on the  
 490  $\text{Fe}_{80}\text{S}_{20}$  composition quenched from 140(2) GPa and 3070(140) K. The reciprocal lattice, colored  
 491 in black, is associated with a  $\text{Fe}_5\text{S}_2$  grain whose lattice parameters are provided in the top left.





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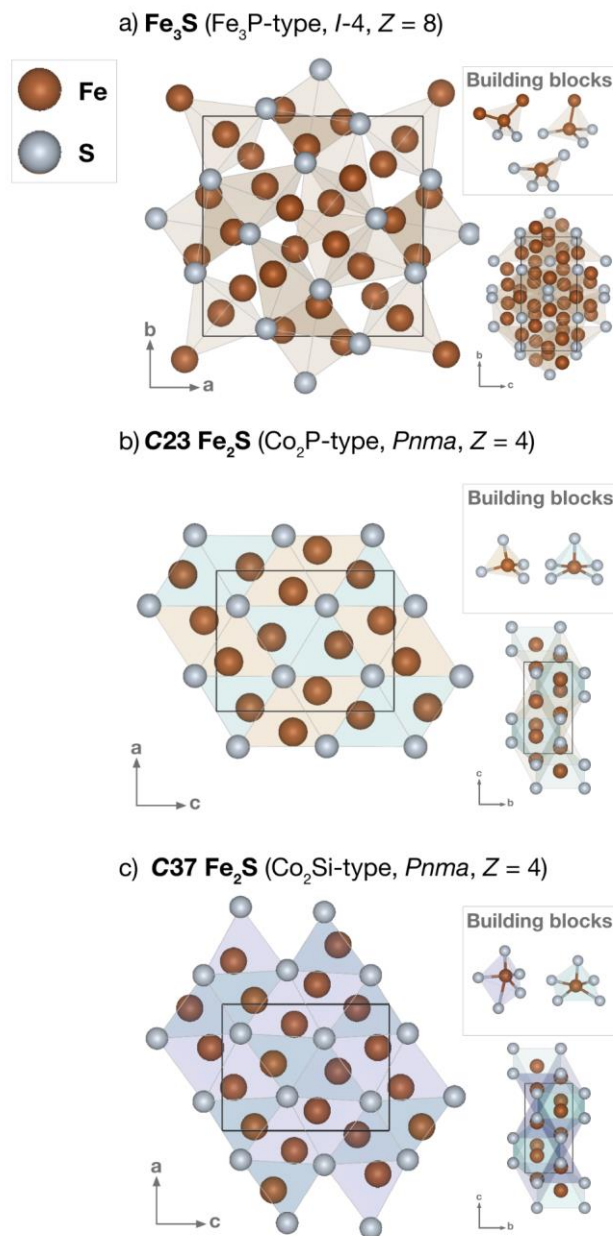
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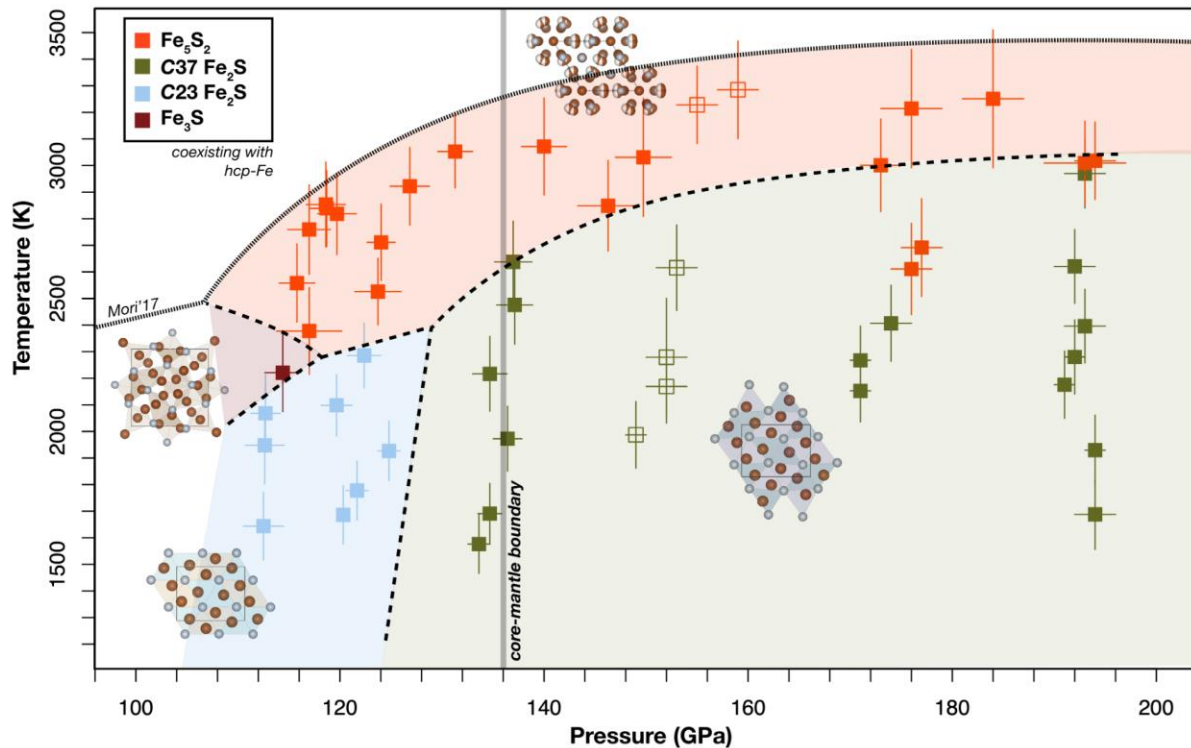
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**Figure 2.** Crystal structure of  $\text{Fe}_5\text{S}_2$  viewed along the a)  $c$  axis and the b)  $(100)$  plane. c) The Fe-  
 coordination polyhedra observed in this atomic arrangement are provided. Fe1, Fe2, Fe4, and  
 Fe6 are coordinated by 13 sites and Fe3 and Fe5 are coordinated by 12 sites. Each coordination  
 polyhedron consists of Fe and S sites.



497  
 498 **Figure 3.** Crystal structures of  $\text{Fe}_3\text{S}$  and  $\text{Fe}_2\text{S}$  observed in this study. a)  $\text{Fe}_3\text{S}$  adopts the  $\text{Fe}_3\text{P}$ -  
 499 type structure ( $I-4$ ,  $Z = 8$ ) that is composed of three tetrahedrally coordinated Fe-sites, each with  
 500 increasing Fe–Fe bonding. b) The  $\text{C}23 \text{Fe}_2\text{S}$  structure ( $\text{Co}_2\text{P}$ -type,  $Pnma$ ,  $Z = 4$ ) is made up of  
 501 columns of  $\text{FeS}_4$  tetrahedra and columns of  $\text{FeS}_5$  square pyramids linked along edges in the  $b$   
 502 direction. c) The  $\text{C}37 \text{Fe}_2\text{S}$  structure ( $\text{Co}_2\text{Si}$ -type,  $Pnma$ ,  $Z = 4$ ) has the same site symmetries as  
 503 the  $\text{C}23$  structure, but is marked by a shortened  $a$  axis and elongated  $b$  and  $c$  axes accompanied  
 504 by the formation of a 5-fold dipyrmaid.



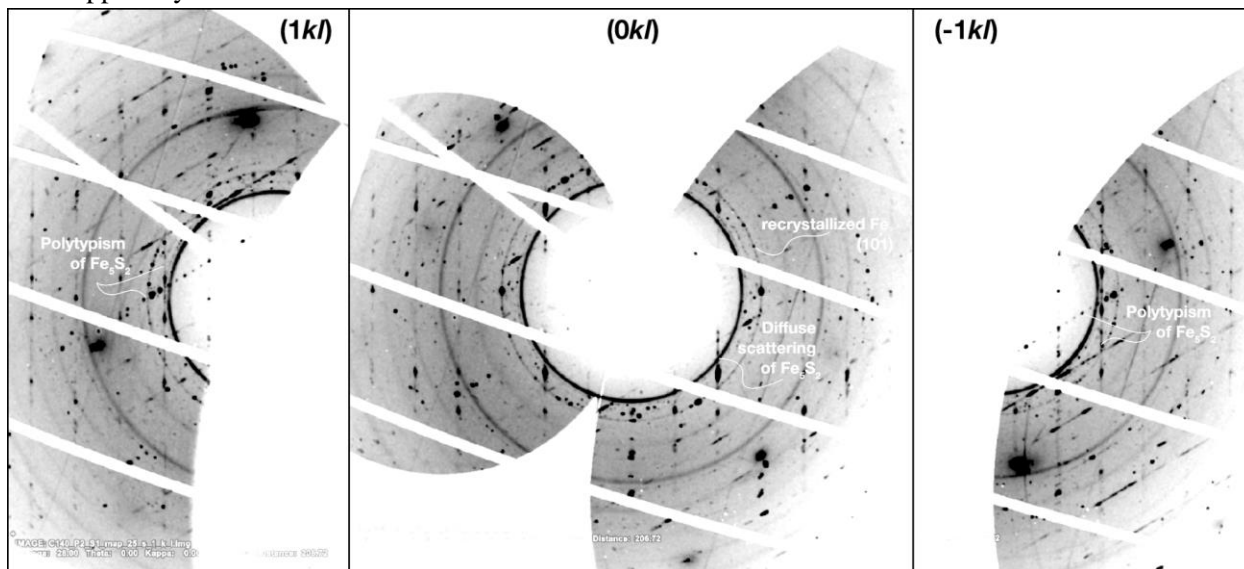
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 506 **Figure 4.** The single-crystal derived structures of  $\text{Fe}_5\text{S}_2$ ,  $\text{Fe}_3\text{S}$ , and  $\text{Fe}_2\text{S}$  were then used to  
 507 interpret the diffraction patterns collected during heating (Figure S3). At moderate temperatures,  
 508  $\text{Fe}_3\text{S}$  is observed below 120 GPa,  $\text{C}23 \text{Fe}_2\text{S}$  is observed below 130 GPa, and  $\text{C}37 \text{Fe}_2\text{S}$  is  
 509 observed above 130 GPa. At high temperatures to 200 GPa,  $\text{Fe}_5\text{S}_2$  is stable, and a kink in the Fe-  
 510 S solidus curve (Mori et al., 2017) is presented to account for the change from  $\text{Fe}_3\text{S}$  to  $\text{Fe}_5\text{S}_2$   
 511 melting in Fe-rich systems above  $\sim 120$  GPa. The closed squares represent experiments  
 512 conducted on the  $\text{Fe}_{80}\text{S}_{20}$  starting material, and the open squares represent experiments  
 513 conducted on the  $\text{Fe}_{67}\text{S}_{33}$  starting material. Both show compatible results and iron  
 514 recrystallization at high temperatures, indicating that Fe-rich regions of the  $\text{Fe}_{67}\text{S}_{33}$  foils were  
 515 probed.

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## 8. SUPPLEMENTAL

524 **Figure S1.** Unwarped diffraction mappings of the  $(1kl)$ ,  $(0kl)$ , and  $(-1kl)$  directions for an  $\text{Fe}_5\text{S}_2$  crystallite  
525 exemplifying the diffuse scattering signal collected at 119(2) and 2840(180) K. The  $(1kl)$  and  $(-1kl)$   
526 mappings also show the presence of a polytype with more closely spaced reflections oriented  $\sim 51^\circ$  from  
527 the mapped crystallite.



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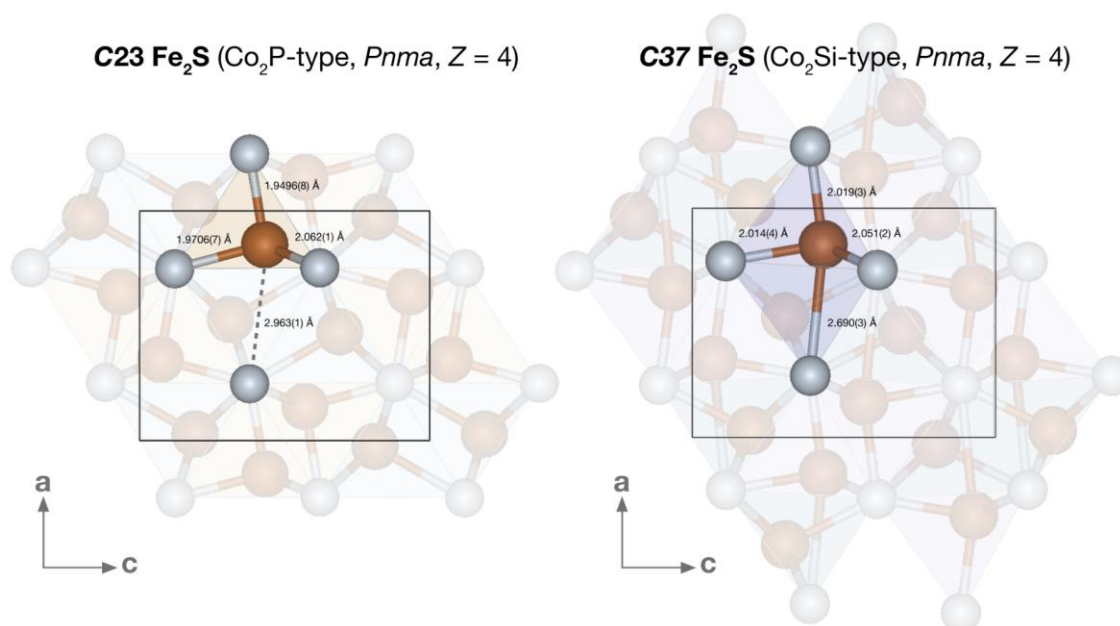
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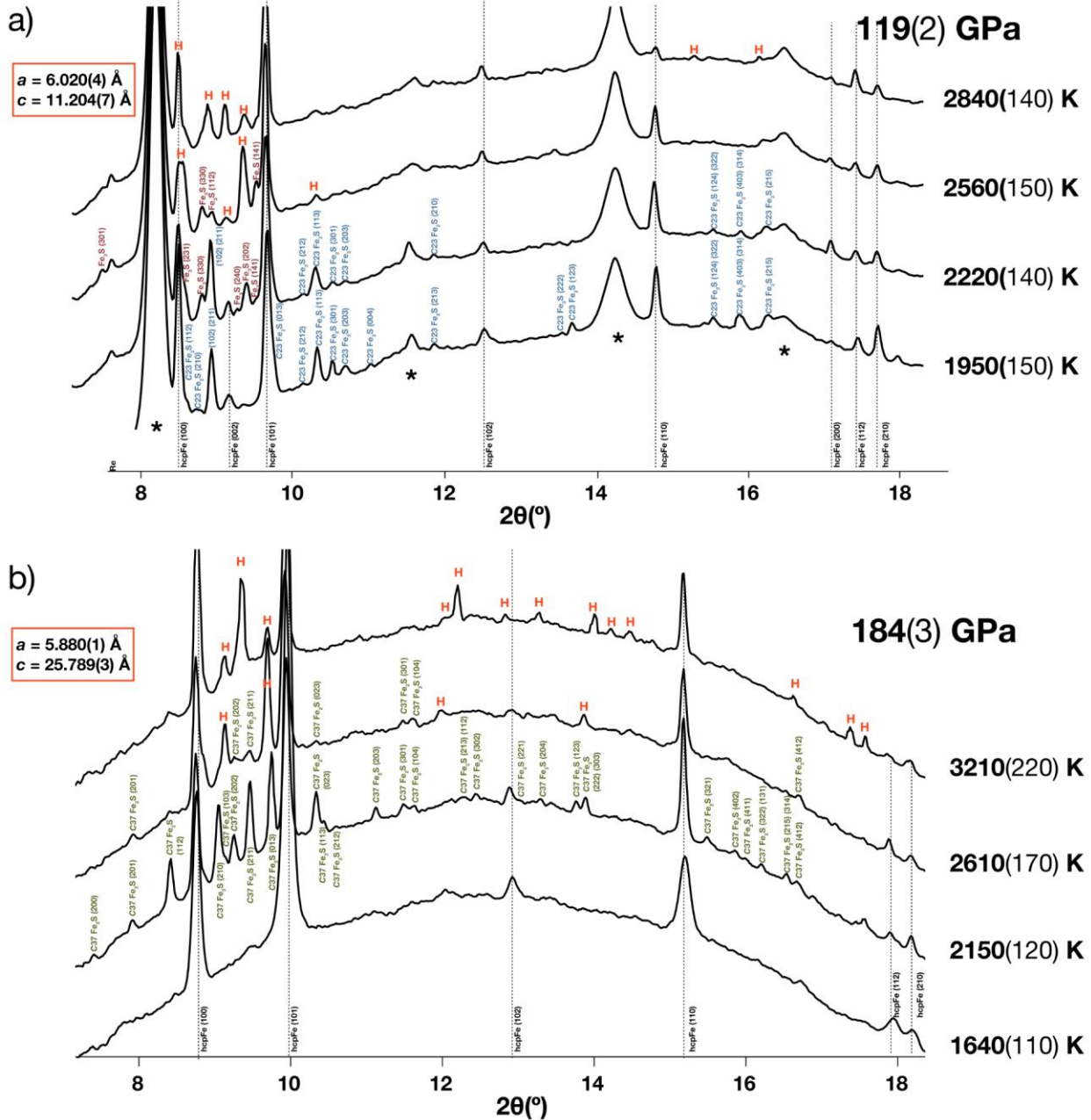
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552 **Figure S2.** Comparison of the *C23* and *C37* Fe<sub>2</sub>S structure models refined at 130 and 140 GPa,  
553 respectively. In the *C23* Fe<sub>2</sub>S structure, the next nearest sulfur site to the Fe1 tetrahedral site is at  
554 a 2.963(1) Å distance at 130 GPa (dotted line). A 10% contraction of this interatomic distance  
555 and a coordination change is observed in the formation of *C37* Fe<sub>2</sub>S at 140 GPa (right).  
556



579 **Figure S3.** X-ray diffraction patterns collected upon heating in an Fe<sub>80</sub>S<sub>20</sub> starting composition at  
 580 a) 119(2) GPa and b) 184(3) GPa. The miller indices for C23 Fe<sub>2</sub>S (blue), Fe<sub>3</sub>S (burgundy), and  
 581 C37 Fe<sub>2</sub>S (green) are provided, and the red “H” symbols represent the observations of the  
 582 formation of Fe<sub>5</sub>S<sub>2</sub>. The growth of large crystallites, along with the disorder and polytypism of  
 583 this phase make for challenging powder diffraction indexing, but the lattice parameters of Fe<sub>5</sub>S<sub>2</sub>  
 584 indexed in the reciprocal space after quenching from these high *P-T* conditions are provided.



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