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Rapid fault healing after seismic slip

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Abstract

Fault strength recovery (healing) following an earthquake is a key process in controlling the recurrence of future events; however, the rates and mechanisms of fault healing are poorly constrained. Here, by performing high-velocity friction experiments, we show that granite and gabbro fault gouges recover their strength rapidly after experiencing dynamic weakening during seismic slip. The healing rates are two orders of magnitude faster than those observed in typical frictional healing experiments performed at slow slip velocities. Analysis of the sheared gouges using Raman spectroscopy suggests that enhanced healing after seismic slip is associated with thermally activated chemical bonding at frictional contacts in the gouge. Our results imply that seismogenic faults regain their strength early during interseismic periods, indicating that healing may not be the dominant control on earthquake recurrence, with other processes, such as far-field tectonic loading or frictional stability transitions, likely dictating the occurrence of future events.

Teaser

Faults regain their strength rapidly after experiencing dynamic weakening during earthquake slip events.

22 **1. Introduction**

23 Faults slip suddenly during earthquakes, accelerating to velocities on the order of a few meters per
24 second. At these seismic slip velocities a significant reduction in fault strength occurs(1) as a result of
25 various dynamic weakening mechanisms becoming activated by shear heating(2). Although our knowledge
26 of dynamic fault weakening processes has increased significantly over the last 25 years since the advent of
27 high-velocity friction experiments(3), our understanding of how faults regain their strength after dynamic
28 weakening, once seismic slip has ceased, is more limited. Fault restrengthening is a fundamental process in
29 the earthquake cycle that may control the recurrence time(4), the maximum strength that can be attained(5,
30 6), and the nature of radiated energy(7) in future events.

31 The rate of fault restrengthening can vary with both time and space along the fault during the earthquake
32 cycle(8, 9). Restrengthening may occur initially during coseismic slip itself, as sometimes observed during
33 the deceleration phase of high-velocity friction experiments(10–13). Coseismic restrengthening (Fig. 1) is
34 a potentially important process in the generation of pulse-like earthquake ruptures(14), which require that
35 faults rapidly regain their strength (self-heal) after the passage of the rupture front. However, the
36 mechanisms of coseismic restrengthening are poorly constrained and it is a phenomenon that is not always
37 observed in experiments, or it may only partially recover the strength lost during high-velocity fault slip(15–
38 19). In such cases, the majority of fault restrengthening must occur in the postseismic regime instead, when
39 the fault is held in quasi-stationary contact.

40 The process of strength recovery as a fault is held in quasi-stationary contact, known as fault healing,
41 has been extensively studied in experiments performed at slow sliding velocities (on the order of
42 micrometers per second)(20). The common procedure for studying fault healing in the laboratory is to
43 perform slide-hold-slide (SHS) experiments(21, 22), whereby the shearing of fault materials is paused for
44 predetermined durations and then shear strength is monitored as sliding is resumed after the hold period.
45 Previous low-velocity SHS experiments have shown that frictional strength increases linearly with the
46 logarithm of hold time, with healing rate being dependent on the composition of the fault materials(23).

47 The physical mechanisms responsible for fault healing are debated, with time-dependent growth of real
48 contact area due to asperity creep often invoked to explain healing behavior(24). However, more recent
49 work has suggested other processes such as chemical bond formation could be responsible for fault healing
50 observed in laboratory experiments(25, 26).

51 In some specific cases, the healing rates determined from low-velocity SHS experiments correlate well
52 with stress drops observed during sequences of small repeating earthquakes in nature (i.e., the magnitude
53 of the stress drop increases as the duration of the recurrence interval increases)(4, 27). However, following
54 large earthquakes, geophysical observations suggest rapid fault restrengthening can occur in comparison to
55 typical recurrence intervals, with the majority of the strength being recovered early during the interseismic
56 period. For example, shear-wave splitting measurements following the 1995 Kobe earthquake (moment
57 magnitude M_w 6.9) on the Nojima fault indicate that the majority of fault strength had recovered within 33
58 months of the main event (recurrence interval of approximately 2000 years)(28). Borehole permeability
59 measurements from the Longmenshan fault zone that hosted the 2008 Wenchuan earthquake (M_w 7.9),
60 suggest that the fault healed within 0.6 to 2.5 years after the earthquake(29). Seismic velocity measurements
61 made following the same event, and also the nearby 2013 Lushan earthquake (M_w 6.6), support the notion
62 of rapid healing on the fault(8), with similar enhanced strength recovery rates also observed after the 2004
63 Parkfield earthquake (M_w 6.0) on the San Andreas fault(9). Geophysical observations thus potentially
64 indicate that different postseismic healing processes are in operation immediately following large
65 earthquakes, leading to more rapid restrengthening, than the classic “Dieterich-type” healing
66 mechanisms(21, 24) responsible for fault strengthening in low-velocity SHS experiments (Fig. 1). It should
67 also be noted that over typical recurrence intervals of large earthquakes (up to several hundreds of years),
68 processes such as pressure solution will increase cohesion of fault materials, contributing to the long-term
69 strength evolution of the fault during interseismic periods(30–32).

70 In order to investigate rapid postseismic healing processes in the laboratory we need to simulate
71 earthquake slip velocities, something that is not done in typical low-velocity SHS experiments. By shearing

72 at seismic slip velocities, the fault materials will also experience dynamic weakening(I), which more
73 closely mimics what happens during natural earthquakes. Here, we perform high-velocity (0.57 m/s) SHS
74 experiments on gabbro and granite gouges under room humidity conditions at a constant normal stress of
75 1.5 MPa in all experiments (see Methods), to investigate how the gouges regain their strength during quasi-
76 stationary hold periods after experiencing dynamic weakening. We varied the length of the static hold
77 period in order to determine whether the postseismic restrengthening behavior exhibits either, (i)
78 “Dieterich-type” healing as observed in low-velocity SHS experiments, (ii) a form of more rapid healing,
79 or (iii) a combination of rapid and slow healing; as shown schematically in Figure 1. We then analyze the
80 microstructures of the sheared gouges and perform Raman spectroscopy in an attempt elucidate the
81 underlying healing mechanisms in operation after seismic slip events.

82

83 **2. Results**

84 *Friction data*

85 The frictional strength evolution of the granite and gabbro gouge samples is shown in Figure 2 for
86 both sliding events in the SHS experiments. During the first high-velocity sliding event (slide 1) the gouge
87 layers experience dynamic weakening with the friction coefficient (μ) decreasing by ~ 0.25 , from a peak
88 value between 0.7-0.8, to a final value of ~ 0.5 after 15 m of slip (Fig. 2a and c). During the static hold
89 period between sliding events the gouge undergoes healing, with the peak friction of the second sliding
90 event (slide 2) being dependent on the duration of the hold period (Fig. 2b and d) – i.e., longer hold periods
91 lead to higher peak friction values. During slide 2, after reaching their respective peak friction values, the
92 gouge layers again experience dynamic weakening, returning to a final μ of ~ 0.5 after another 15 m of
93 high-velocity slip.

94 The gouge samples recover their strength rapidly during the static hold period, as shown in Figure
95 3 where $\Delta\mu$ (the difference between the peak friction of slide 2 (μ_{p2}) and the final friction of slide 1 (μ_{f1}),

96 $\Delta\mu = \mu_{p2} - \mu_{f1}$; see also Fig. S1) is plotted against hold time. After around 20 s of static hold, the granite
97 gouge had recovered the majority of the strength it lost during slide 1, with the gabbro gouge healing even
98 more rapidly (<10 s of static hold). For comparison, healing data from low-velocity SHS experiments
99 performed on intact samples of granite and gabbro at slip rates of 2.6 $\mu\text{m/s}$ has been included in Figure 3
100 (see Methods for more details). The healing rate ($\beta = \Delta\mu/\Delta\log(t_h)$, where t_h is the hold time) is around
101 two orders of magnitude greater for the experiments performed at seismic slip velocities than those
102 performed at micrometer per second slip velocities (Fig. 3). After the initial rapid strength recovery in the
103 high-velocity tests, the healing rate decreased to a rate that is comparable to those observed in the low-
104 velocity SHS experiments.

105 A major difference between high-velocity and low-velocity SHS experiments is that during high-
106 velocity slip there is a large temperature increase caused by shear heating, which is much less significant
107 during sliding at low-velocity. In order to measure the temperature evolution in our high-velocity SHS
108 experiments, we placed thermocouples next to the upper surface of the gouge layer on the stationary side
109 of the fault in some experiments (see Methods). We recorded peak temperatures of around 350-400 $^{\circ}\text{C}$
110 during the high-velocity sliding events, with the temperature decaying as the samples cooled during the
111 hold period, returning to the ambient temperature in the laboratory after several minutes of static hold (Fig.
112 3). However, we find that the rapid frictional healing, which begins immediately after the initiation of the
113 hold period, occurred when the gouge layer was still relatively hot, at temperatures $>200^{\circ}\text{C}$ (Fig. 3).

114

115 *Microstructural analysis and Raman spectroscopy*

116 We analyzed the microstructures of the sheared gouges by collecting backscatter electron (BSE)
117 and secondary electron (SE) images using a JEOL JSM-6500F field emission scanning electron microscope
118 (FE-SEM). Fig. 4a-b shows BSE images of granite and gabbro gouge samples after the SHS experiments,
119 where the sheared layers have been cut perpendicular to the shear plane and parallel to the shearing direction

120 at a distance equal to $2/3$ of the sample radius. (Note that the gouge layers were vacuum impregnated with
121 a low-viscosity epoxy resin before being cut and polished ready for BSE imaging). The sheared gouges
122 display a texture of well-rounded larger relict grains surrounded by fine-grained highly comminuted
123 material, indicating that they have undergone a significant grain size reduction and particle rounding
124 when compared to the starting gouge material (see Fig. S2), with this likely occurring via mechanical
125 grinding(33). In the granite gouge the deformation appears to be homogeneously distributed across the layer
126 (Fig. 4b), whereas the gabbro gouge displays evidence of a highly comminuted localized zone at the
127 boundary of the layer (Fig. 4a and c). Despite the apparent difference in localization behavior between the
128 different materials, their mechanical behavior is remarkably similar (Fig. 2), suggesting that shear
129 localization does not have a strong control on frictional strength evolution under these experimental
130 conditions. Our experiments were run under relatively low normal stress, previous studies suggest that
131 localization would become more prominent if the gouge layers were sheared under higher normal stress(34,
132 35), or if they were taken to greater shear strains(36).

133 The rapid healing observed after sliding at seismic slip rates in our experiments (Fig. 3) must be
134 caused by a strengthening of the frictional contacts in the gouge layer, possibly as a result of enhanced
135 interfacial chemical bonding. To investigate further the possible causes of the rapid restrengthening, we
136 analyzed the sheared gouges using Raman spectroscopy (see Methods), as this provides information about
137 the chemical structure of the gouge surface. We found that the gouges sheared at high-velocity all showed
138 the appearance of a small broad peak in the Raman spectra at a wavenumber of $\sim 1600\text{ cm}^{-1}$ (Fig. 4e and f),
139 which corresponds to the bending vibrational mode of water(37) adsorbed on the surface of the gouge. This
140 peak was not observed for the starting material or for samples sheared at low sliding velocities, only for
141 samples that had been subjected to sliding at seismic slip rates.

142 We hypothesize that the switch in vibration mode of adsorbed water is caused by a change in
143 chemical bonding on the gouge surface, potentially induced by elevated temperatures during high-velocity
144 shearing, which could be responsible for the rapid healing observed in the SHS experiments (Fig. 3). To

145 investigate this further, we heated undeformed samples of granite and gabbro in an oven to different
146 temperatures (leaving them for ~20 minutes at the target temperature), the samples were then removed from
147 the oven and left to cool at room atmosphere conditions (i.e., the same cooling conditions that the gouge
148 layers experienced during the hold period of the SHS experiments). We analyzed the oven-heated samples
149 using Raman spectroscopy and found the appearance of a small broad peak at $\sim 1600\text{ cm}^{-1}$ for samples that
150 had been heated to temperatures $\geq 250\text{ }^{\circ}\text{C}$ (Fig. 4g and h), which is similar to the temperatures that the gouge
151 layers experienced during high-velocity shearing where a similar Raman peak was observed (Fig. 4e and f)
152 and also the temperature conditions where rapid healing occurred (Fig. 3). We note that the size of the
153 adsorbed water peak in the oven-heated samples is often less than observed for the sheared gouge samples
154 (particularly for granite), which may be a result the sheared gouges having a much greater surface area due
155 to the presence of nanoparticles (Fig. 4d), producing a stronger Raman signal.

156

157 **Discussion**

158 The frictional strength data from our high-velocity SHS experiments show that the fault gouges
159 heal rapidly during static hold periods after shearing at seismic slip rates, in comparison to typical healing
160 rates observed in low-velocity SHS experiments performed at micrometer-per-second slip rates(20–23)
161 (Fig. 3). The rapid healing rates we observe for the granite and gabbro gouges in our study are a similar
162 order to those observed in previous high-velocity SHS experiments on clay-carbonate-bearing gouges from
163 the Longmenshan fault system (sheared at 0.8 MPa normal stress and a slip rate of 1.4 m/s)(19), suggesting
164 that rapid healing after high-velocity slip may be a universal phenomenon that is largely insensitive to the
165 lithology of the fault materials. Elevated healing rates have also been observed in experiments performed
166 at subseismic slip rates (on the order of a few millimeters-per-second)(38, 39), however the healing rates
167 in these experiments are an order of magnitude lower than we observe in our experiments at seismic slip
168 velocities.

169 Dynamic weakening during the high-velocity shearing events in our experiments (Fig. 2) is likely
170 caused by a combination of flash heating at asperity contacts(40) and the formation of amorphous wear
171 materials in the gouge(41). X-ray diffraction analysis of the sheared gouges confirms the presence of
172 amorphous material that was not present in the starting materials (Fig. S3). The microstructures of the
173 sheared gouges (Fig. 4a-d) show no evidence of other weakening mechanisms that have been reported in
174 previous studies such as frictional melting(42), silica-gel formation(38) or grain-size sensitive flow(43, 44).
175 Fault restrengthening during the hold periods is likely caused by the reformation of bonds at asperity
176 contacts in the gouge material. There are two prevailing hypotheses for the time-dependent strengthening
177 of frictional contacts during fault healing: (i) an increase in real contact area by asperity creep(24) (often
178 referred to as the contact ‘quantity’ hypothesis), or (ii) the formation of chemical bonds across the asperity
179 interface(25, 26) (often referred to as the contact ‘quality’ hypothesis).

180 If we first consider asperity creep, it is plausible that this process would be more active after seismic
181 slip, as creep is temperature-sensitive and the rapid healing we observe occurs immediately after high-
182 velocity slip while the gouge is still relatively hot (>200 °C, Fig. 3). The likely mechanisms that could
183 facilitate asperity creep are either solution-transfer processes(45) or indentation creep(46). Solution-transfer
184 is unlikely to be a dominant mechanism in our experiments as they were run without a pore-fluid (i.e., room
185 atmosphere conditions), therefore there is no solute to transfer chemical species. Furthermore, previous
186 fault healing experiments under hydrothermal conditions, where solution-transfer processes are operative,
187 show complex healing behavior(31, 47–49) that is quite different to the healing trends we observe in our
188 data (Fig. 3). Indentation creep can operate under atmospheric conditions in the absence of a pore-fluid(50),
189 however, although previous low-velocity fault healing experiments at elevated temperatures (up to 550°C)
190 under room humidity conditions indicate some temperature-dependence on healing rate(51, 52), the effect
191 is relatively minor and insufficient to explain the rapid healing in our experiments. It is therefore unlikely
192 that an increase in the real contact area via asperity creep is the cause of the rapid restrengthening we
193 observe during the static hold periods.

194 Alternatively, rapid healing may be caused by enhanced chemical bonding across contacting
195 asperity interfaces. Our Raman data reveal a change in chemical bonding on the surface of the gouges
196 sheared at high-velocity, with a switch in the vibrational mode of adsorbed water to the H-O-H bending
197 mode, which only occurs after sample has been heated to temperatures ≥ 250 °C (Fig. 4g-h). Although we
198 observe a change in adsorbed water properties, we do not expect the adsorbed water itself to be responsible
199 for the rapid healing, as rapid healing occurs at temperatures > 200 °C (Fig. 3) where water would be in the
200 vapor state and desorbed from the gouge surface(53). Instead, we hypothesize that the rapid healing is a
201 result of hydrogen bonding on the surface of the sheared gouge materials, which subsequently causes water
202 to re-adsorb in the bending vibrational mode once the gouge has cooled to sufficiently low temperatures
203 (< 140 °C)(53) during the hold period. Hydrogen bonding can arise between hydroxylated silanol (Si-OH)
204 surfaces(54), which are readily formed on freshly cleaved surfaces of silicate materials during frictional
205 slip(37, 41, 55) (Fig. 5a). Once slip has stopped, the formation of hydrogen bonds between silanol surfaces
206 can take place on very short timescales ($< 10^{-2}$ s)(56). Therefore, if hydrogen bonding occurs during the first
207 few seconds of static hold in our experiments, it could be responsible for the rapid increase in friction we
208 observe. Furthermore, at elevated temperatures, like those produced by shear heating in our experiments,
209 silanol groups on opposite sides of an asperity interface can react to form strong covalent siloxane (Si-O-
210 Si) bonds(57, 58) (Fig. 5b). Previous molecular dynamics simulations of silica-silica interfaces have shown
211 that siloxane bond formation provides a plausible explanation for frictional healing, with frictional strength
212 being approximately proportional to the number of siloxane bonds(59) and the kinetics of interfacial bond
213 formation leading to a logarithmic time-dependent increase in strength(56), as observed in SHS experiments
214 (Fig. 3). Therefore, we postulate that rapid healing after high-velocity slip is caused by either hydrogen or
215 siloxane bond formation (or a combination of both) at asperity contacts in the sheared gouges. Once the
216 gouge has cooled to sufficiently low temperature water will re-adsorb(53) (Fig. 5c). The vibrational motions
217 of water molecules are sensitive to local hydrogen bonding on the adsorbent surface(37, 58), thus the switch
218 to the H-O-H bending mode we observe on the sheared gouges likely results from changes in the hydrogen
219 bonding on the gouge surface that occur during/after high-velocity slip while the gouge is still hot, hence

220 why the change in adsorbed water properties is only observed in samples that have been heated to
221 temperatures >250 °C (Fig. 4g-h) and not in the samples sheared at low velocity where the temperature
222 increase was low.

223 Regardless of the underlying restrengthening mechanism, our data clearly show that fault materials
224 heal rapidly after seismic slip, which has important implications for our understanding of the earthquake
225 cycle. Rapid healing may explain why geophysical observations suggest faults can regain their strength
226 early during interseismic periods after large earthquakes(28, 29). Fast-acting healing mechanisms, like
227 those in operation during our experiments, potentially also operate during coseismic slip on natural faults,
228 particularly when slip occurs heterogeneously along the fault such as during the propagation of pulse-like
229 ruptures(14, 60). The passage of a rupture pulse requires rapid healing in the just-slipped portions of the
230 fault(61), in order for them to stay locked and prevent further slip as they are reloaded by waves from the
231 actively slipping regions elsewhere along the fault. Results from recent dynamic rupture experiments
232 further highlight the complex interplay between rapid weakening and healing processes that occur in gouge
233 samples during dynamic rupture propagation(62).

234 Rapid fault strength recovery immediately following a seismic event suggests that earthquake
235 recurrence is not necessarily controlled by continuous restrengthening over time during interseismic
236 periods. Instead, if the majority of strength is recovered early during the interseismic period, as implied by
237 our results, then earthquake recurrence on natural faults may be more strongly controlled by far-field
238 tectonic loading (i.e., when the stress applied to the fault exceeds the strength an earthquake may occur).
239 Alternatively, other time-dependent processes in operation during interseismic periods may influence
240 earthquake recurrence. For example, over typical recurrence intervals of hundreds of years, fault cohesion
241 will increase by processes such as pressure solution(30–32). Increased cohesion will not only contribute to
242 the fault strength evolution, but will also influence the frictional stability of the gouge materials, with more
243 cohesive materials often displaying rate-weakening behavior required for earthquake nucleation(63, 64). It
244 is plausible that transitions from rate-strengthening to rate-weakening behavior may occur as the gouge

245 materials become more lithified during interseismic periods, potentially leading to earthquake recurrence
246 once the frictional properties have evolved to state that promotes earthquake nucleation and unstable slip.

247 In summary, we find that faults regain their strength rapidly after experiencing dynamic weakening
248 during seismic slip. After the initial rapid increase in strength, the healing rate decreases to a rate that is
249 comparable to those observed in low-velocity friction experiments. Rapid healing occurs while the gouge
250 is still hot from shear heating, and is likely promoted by enhanced chemical bonding across contacting
251 asperity interfaces. Further experimental and theoretical studies are needed to investigate the kinetics of
252 interfacial reactions over the range of stress, temperature and pore fluid conditions that faults experience
253 during and after earthquake slip, to understand better strength recovery at seismogenic depths. Our findings
254 motivate further study aimed at the quantification of rapid healing mechanisms and incorporation into
255 larger-scale constitutive laws for modelling dynamic fault processes, to provide insight into the driving
256 mechanisms of earthquake rupture and arrest, and hence seismic hazard.

257

258 **Methods**

259 *Experimental procedure*

260 The experimental samples were produced by crushing and sieving intact samples of Inada granite
261 and Belfast gabbro to form simulated fault gouges (powders) with grain sizes between 63-125 μm . A layer
262 of simulated gouge (measured by weight to produce a layer with an initial thickness of 1.5 mm) was then
263 sandwiched between two cylindrical stainless steel experimental forcing blocks (diameter = 25 mm). The
264 surface of the blocks contains radial grooves (0.5 mm deep) to minimize boundary shear between the gouge
265 layer and the forcing blocks during the experiments. To limit gouge loss during shearing, the gouge layer
266 was contained laterally by a 5 mm thick polytetrafluoroethylene (PTFE) sleeve (Fig. S4). The low-friction
267 PTFE sleeve was cut and tightened onto the forcing blocks using a hose clip (Fig. S4), following the
268 procedure outlined in the supplementary material of De Paola et al.,(43). We used a torque-screwdriver to

269 ensure the hose clip was tightened by the same amount for each experiment. Once the gouge sample was
270 constructed in between the forcing blocks, it was sheared using the PHV rotary shear apparatus(65) in the
271 Rock Mechanics Laboratory at the Kochi Institute for Core Sample Research (Japan).

272 Before the main SHS experiment, the gouge samples were pre-sheared for four complete
273 revolutions (equivalent to 0.2 m of slip) under a normal stress of 0.75 MPa at a rate of 1.7 mm/s, to ensure
274 the gouge layer thickness was even across the sample. The normal stress was then increased to the
275 experimental target value of 1.5 MPa. As all experiments were run under the same normal stress we did not
276 correct for the shear stress contribution from the PTFE sleeve, with previous work showing that the
277 mechanical contribution from the PTFE is negligible(15). All tests were conducted under room temperature
278 (22-25 °C) and humidity (30-50%) conditions. During the main SHS experiments the gouge layers were
279 sheared at 650 rpm for 285 revolutions (slide 1), they were then held in quasi-stationary contact for a
280 predetermined amount of time, before being sheared again for another 285 revolutions at 650 rpm (slide 2).
281 As the slip velocity varies with radial position, we use an “equivalent slip velocity” (v_e) which corresponds
282 to the velocity at 2/3 of the radius of the cylindrical specimens(43), given by:

$$283 \quad v_e = \frac{4\pi Rr}{3}$$

284 where R is the revolution rate of the motor and r is the sample radius. In our experiments, a revolution rate
285 of 650 rpm corresponds to an equivalent slip velocity of 0.57 m/s. During each sliding event the gouge layer
286 was sheared for 285 revolutions which corresponds to an equivalent slip displacement (d_e) of ~15 m ($d_e =$
287 $v_e t$ where t is time).

288 In some of the high-velocity experiments temperature measurements were made by placing
289 thermocouples next to the upper surface of the gouge layer. Two holes were drilled into the upper
290 experimental forcing block (on the stationary side of the fault) and thermocouples were inserted into the
291 holes and sealed into place using a ceramic bond. The thermocouples were positioned at 2/3 of the radius
292 so that the temperature measurements were consistent with the calculated v_e and d_e .

293 As well as the high-velocity SHS experiments, some additional tests were performed at
294 micrometer-per-second slip velocities to compare healing rates after low-velocity slip with the rates
295 determined in our high-velocity experiments (Fig. 3). In the low-velocity SHS experiments (performed at
296 an equivalent slip velocity of $2.6 \mu\text{m}\cdot\text{s}^{-1}$) we used intact cylindrical rock-to-rock samples of Inada granite
297 and Belfast gabbro, instead of gouge. Initially we tried performing the low-velocity SHS experiments using
298 gouge samples, however, we found negligible healing even after hold periods >1000 s (healing rate, $\beta \approx$
299 0). As the purpose of our low-velocity SHS experiments is just to provide an approximate representation
300 of typical healing rates at slow sliding velocities, we chose to instead include data from rock-to-rock
301 samples in Fig. 3, as the healing rates we determined from the rock-to-rock samples are close to typical
302 healing rates observed in many low-velocity healing studies(22, 23). Prior to the low-velocity SHS
303 experiments, the cylindrical rock samples were rotated for more than 1000 rotations at a constant speed of
304 4 rpm ($v_e = 3.5 \text{ mm}\cdot\text{s}^{-1}$) over a range of incrementally increasing normal stresses from 0.3 to 1.4 MPa. The
305 purpose of this procedure was to remove any heterogeneities and ensure the surfaces on opposites side of
306 the sliding interface were parallel. The wear materials produced on the sliding surface during this pre-
307 sliding were not removed before the SHS experiments, thus the rock samples were separated by a thin
308 gouge layer during the experiments. The wear materials produced during the experiments were allowed to
309 extrude from the slip zone (we did not use a PTFE containing ring for these tests). Once the sliding surface
310 was prepared, the normal stress was increased to 1.5 MPa and the samples were sheared for 0.26 mm during
311 each sliding event in the SHS experiment at a velocity of $2.6 \mu\text{m}\cdot\text{s}^{-1}$; the length of the hold time between
312 the sliding events was varied to determine the healing rate.

313

314 *Raman spectroscopy*

315 After the experiments the PTFE ring was removed and the sample holders were gently opened to
316 expose the sheared gouge sample. The surface of the gouge was then analyzed using Raman spectroscopy.
317 (Note that Raman spectra were acquired on the exposed gouge surface before it was impregnated with

318 epoxy resin and prepared for microstructural imaging). Raman spectra of the test samples were obtained
319 with a 514.5 nm Ar laser (Showa Optronics Co., Ltd.) and T64000 Raman system (Jobin Yvon Horiba).
320 The laser passed through a 40× objective and the laser power at the sample surface was set at 2–5 mW. The
321 scattered light was collected by backscattered geometry with a 25 μm pinhole and a holographic notch filter,
322 and finally dispersed using a 1800 grids/mm grating and analyzed by a Peltier cooled CCD detector
323 (SPECTRUM ONE, Jobin Yvon Horiba). Spatial resolution is about 1 μm, and wavenumber resolution is
324 about 1 cm⁻¹. Frequencies of the Raman bands were calibrated by measuring silicon standards.

325

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482

483 **Acknowledgments**

484 We are grateful to T. Suzuki and O. Tadai for technical support with the experiments and analyses.

485

486 **Funding**

487 This work was supported by a Japan Society for the Promotion of Science (JSPS) International Research

488 Fellowship and JSPS KAKENHI Grant Numbers JP20F20786 and JP19K21907.

489

490 **Author Contributions**

491 Conceptualization: JDB, TH, YH

492 Methodology: JDB, TH, YH

493 Investigation: JDB, YH

494 Visualization: JDB

495 Writing – original draft: JDB

496 Writing - review and editing: JDB, TH, YH

497

498 **Competing Interests**

499 The authors declare they have no competing interests.

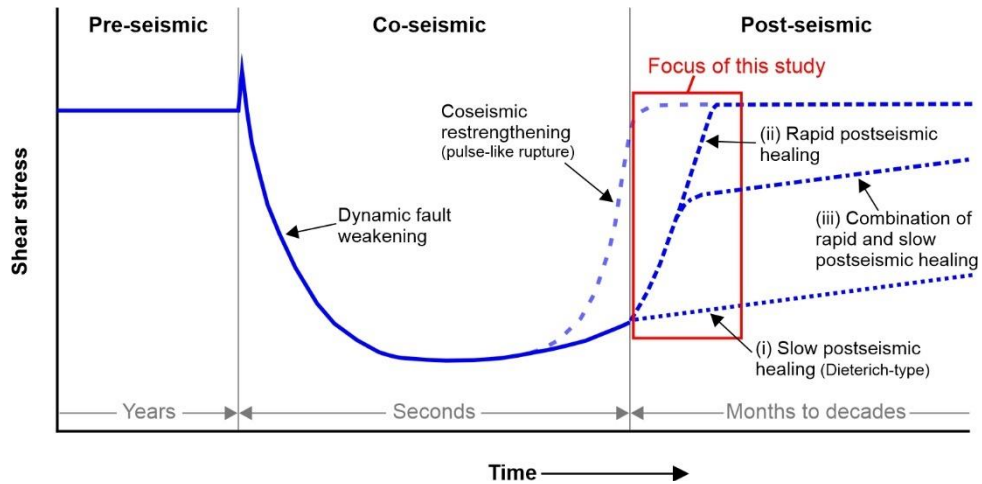
500

501 **Data and Materials Availability**

502 The associated experimental data files for this research can be accessed at:

503 <https://data.mendeley.com/datasets/rw3ndtwkgt/1>

504



505

506 **Fig. 1. Schematic diagram of fault strength evolution during the seismic cycle.** During coseismic slip,

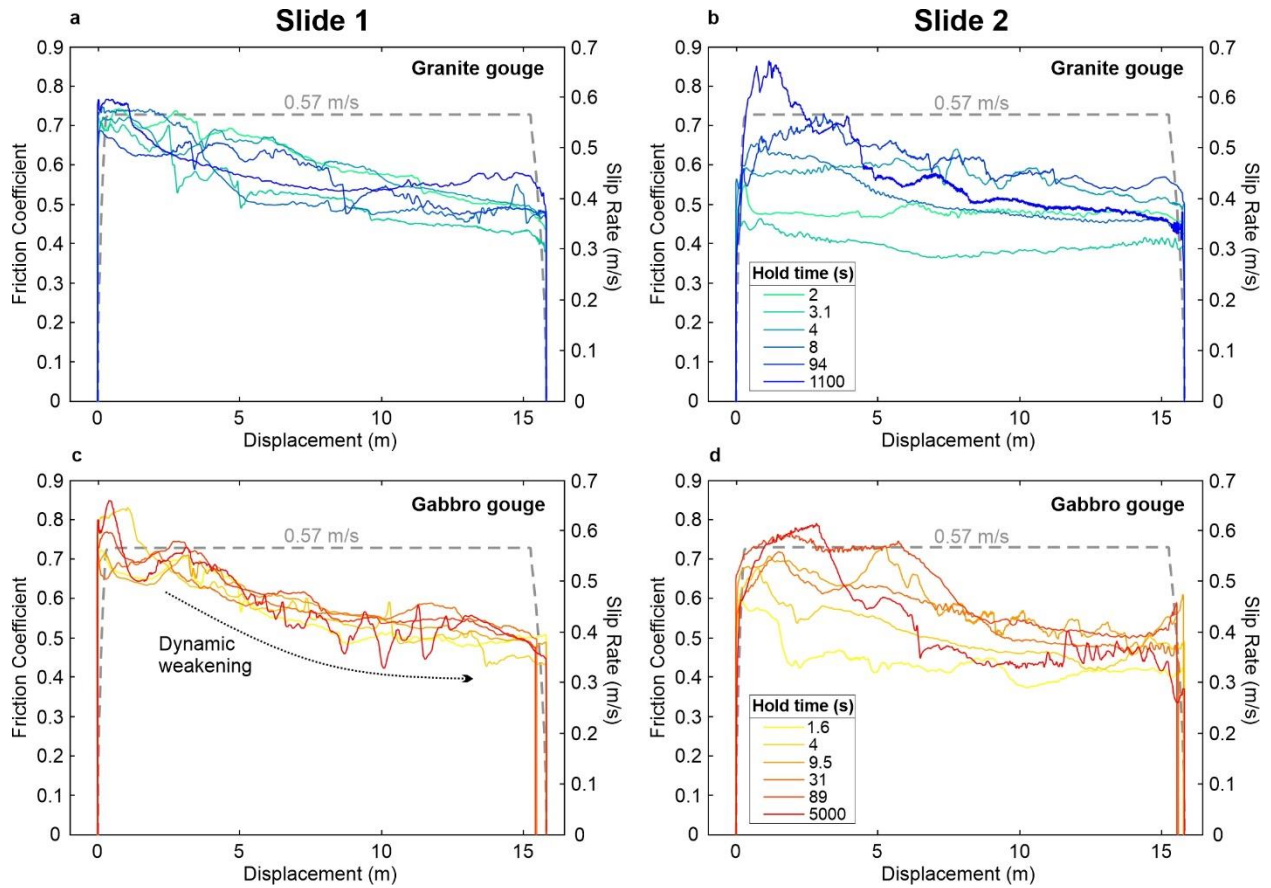
507 a significant reduction in shear stress occurs as a result of dynamic fault weakening. In the postseismic

508 regime the fault regains its strength as it is held in quasi-stationary contact. The aim of this study is to

509 determine whether fault strength recovery immediately following seismic slip occurs via (i) slow

510 “Dieterich-type” healing, (ii) rapid postseismic healing, or (iii) a combination of rapid and slow healing.

511



512

513 **Fig. 2. Example mechanical data from the two high-velocity sliding events in the slide-hold-slide**

514 **experiments.** The plots show the evolution of the friction coefficient with displacement for the granite

515 gouge during (a) the first sliding event (slide 1), and (b) the second sliding event (slide 2). The same data

516 are shown for the gabbro gouge in panels (c) and (d), respectively. The velocity-displacement history during

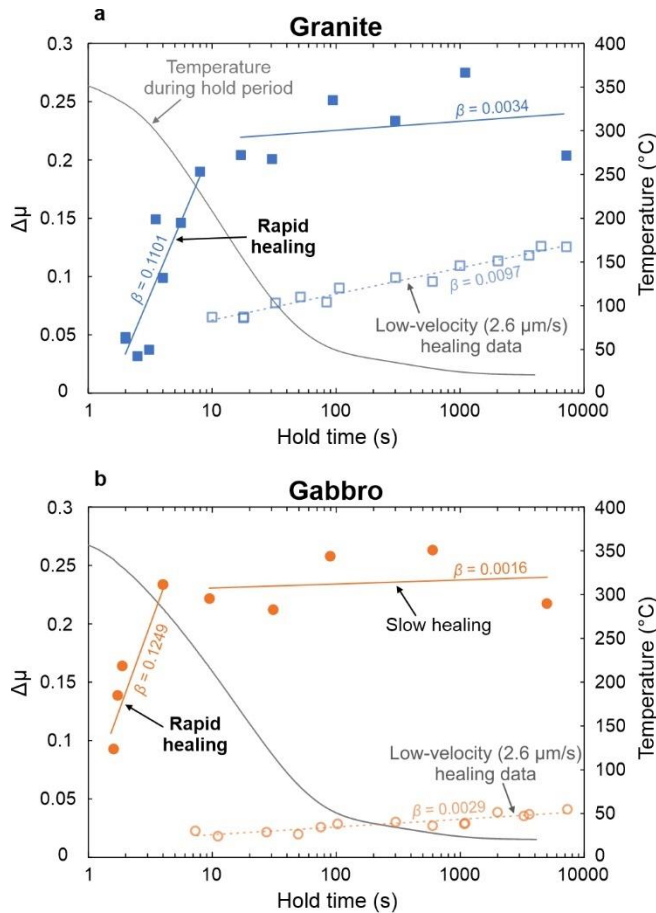
517 the experiments is shown by the grey dashed line. The gouge layers all show similar dynamic weakening

518 during slide 1, with the friction coefficient decreasing by ~ 0.25 after 15 m of displacement. The peak

519 friction during slide 2 is controlled by the duration of the static hold time between the sliding events, with

520 longer hold times leading to higher peak friction.

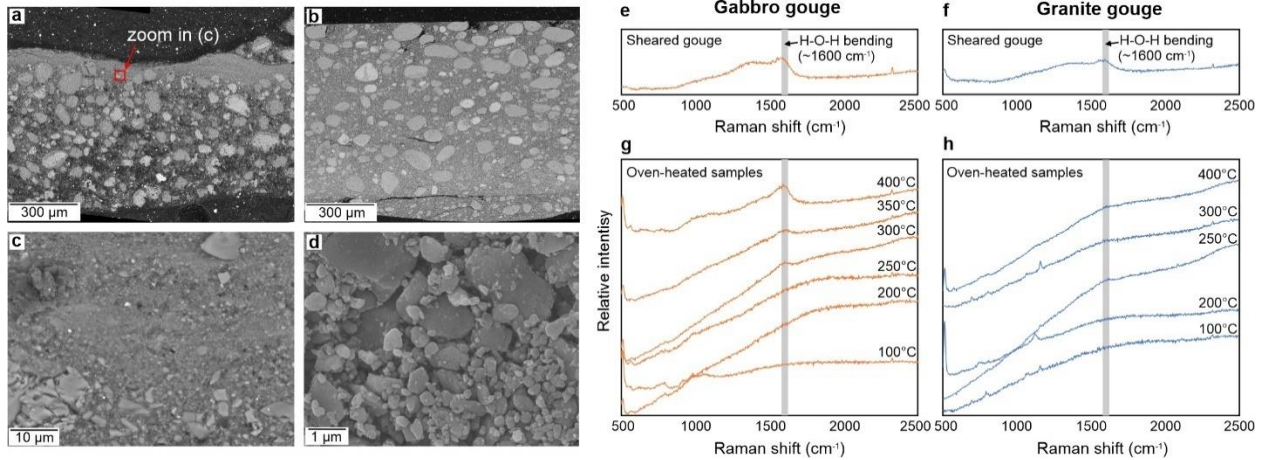
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522

523 **Fig. 3. Frictional healing data from the high-velocity SHS experiments.** The slide-hold-slide parameter
 524 $\Delta\mu$ is plotted against hold time for (a) granite gouge and (b) gabbro gouge. The gouges experience rapid
 525 healing immediately after the initiation of the hold period; the healing rate then decreases to a rate
 526 comparable to those observed in low-velocity SHS experiments. Healing data from experiments performed
 527 at 2.6 $\mu\text{m/s}$ has been included (hollow symbols) for comparison. The temperature evolution was monitored
 528 during the hold period (grey line); rapid healing occurs while the gouges are still relatively hot ($>200\text{ }^\circ\text{C}$)
 529 after the high-velocity first sliding event.

530



531

532 **Fig. 4. Microstructures of the sheared gouges and Raman spectra.** Backscatter electron images of (a)

533 gabbro and (b) granite gouge layers recovered at the end of the SHS experiments. (c) Zoom of the localized

534 zone within the gabbro gouge layer (from the red box in (a)). (d) Secondary electron image of the surface

535 of the gabbro gouge layer showing the presence of sub-micron particles. (e) and (f) Raman spectra of the

536 surface of the sheared gabbro and granite layers, respectively. Both show a broad peak at a wavenumber of

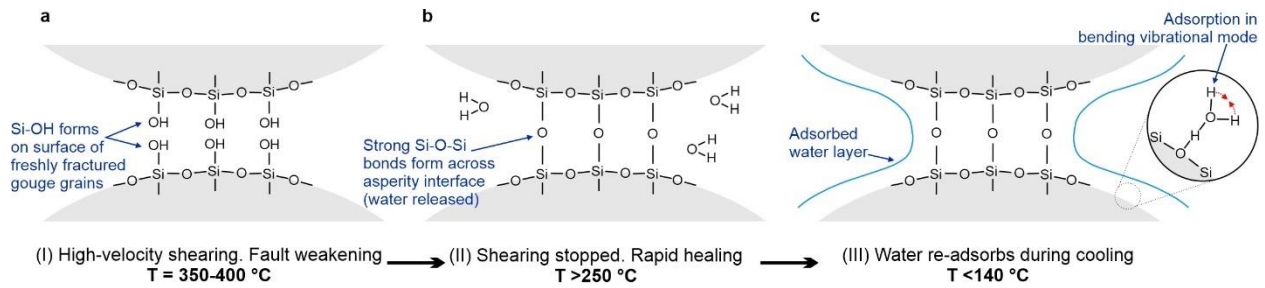
537 $\sim 1600 \text{ cm}^{-1}$, indicating the bending vibrational mode of H-O-H. Panels (g) and (h) show Raman spectra

538 for undeformed gabbro and granite samples heated to different temperatures in an oven and then left to cool

539 under atmospheric humidity conditions. The broad peak at 1600 cm^{-1} appears in samples that have been

540 heated to temperatures $\geq 250 \text{ }^\circ\text{C}$.

541



542

543 **Fig. 5. Schematic cartoon showing the evolution of chemical bonding during and after high-velocity**

544 **slip. (a)** Silanol bonds (Si-OH) form on freshly fractured gouge surfaces during high-velocity slip. During

545 the hold period, once fault slip has ceased, we hypothesize that rapid healing occurs as a result of either

546 hydrogen bonding between adjacent silanol surfaces, or **(b)** the formation of strong siloxane bonds across

547 the asperity interface. **(c)** Once the gouge has cooled to temperatures <140 °C during the hold period, water

548 re-adsorbs onto the surface in the bending vibrational mode.