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2	Low Environmental Impact Remediation of Microplastics: Visible-Light
3	Photocatalytic Degradation of PET Microplastics Using Bio-Inspired C,N-
4	TiO ₂ /SiO ₂ Photocatalysts
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27	

28 Abstract

29 Microplastics (MPs) are plastic particles of ≤ 5 mm with a ubiquitous presence in 30 aquatic ecosystems. MPs harm marine life and can cause severe health 31 problems for humans. Advanced oxidation processes (AOPs) that involve the 32 in-situ generation of highly oxidant hydroxyl radicals can be an alternative to 33 fight MPs pollution. Of all the AOPs, photocatalysis has been proven a clean 34 technology to overcome microplastic pollution. In this work, we propose novel 35 C,N-TiO₂/SiO₂ photocatalysts with proper visible-active properties to degrade 36 polyethylene terephthalate (PET) MPs. Photocatalysis was performed in an 37 aqueous medium and at room temperature, evaluating the influence of two pH 38 values (pH 6 and 8). The obtained results demonstrated the degradation of the 39 PET MPs by C,N-TiO₂/SiO₂ semiconductors and achieved mass loss between 40 9.35 and 16.22%, carbonyl indexes up to 28.7% lower, and a significant 41 decrease in crystallinity compared to the original PET MPs. 42 **Keywords:** PET; photocatalysis; microplastics; C,N-TiO₂/SiO₂; crystallinity; DSC. 43 44 45 1. Introduction

46 The presence of plastics in the ocean is one of the most pressing environmental issues of the current century. Plastic bags, six-pack rings, or ghost nets cause 47 48 entanglement and asphyxia of the marine biota. The harmful effects of those 49 and other big plastic items in the marine environment are readily visible. Hence 50 various efforts to combat marine pollution caused by such plastic items are now 51 being implemented worldwide. However, since 2010, the scientific community 52 has started to keep records of the smaller plastic debris, the so-called 53 microplastics (MPs). MPs are plastics with sizes between 1 µm and 5 mm that

54 were intentionally fabricated in those sizes or that, more often, come from the 55 breakdown of larger plastic items (Wang et al., 2021). Due to their high area-tovolume ratio and surface chemical properties, MPs adsorb persistent organic 56 57 pollutants (POPs) when they are present in contaminated aquatic environments 58 (Gao et al., 2019). Marine biota frequently consumes MP, which promotes 59 physical damage to their tissues by direct contact (Hale et al., 2020), as well as 60 other general health issues caused when MPs act as POPs vectors in the body 61 (Koelmans et al., 2016). This represents a serious ecological problem as it is 62 well-known that MPs are being transferred through the trophic web. Thus, their 63 presence has recently been confirmed in humans (Ragusa et al., 2021;

64 Schwabl et al., 2019).

65 Formerly, it was assumed that MPs remain inactive in oceans once released. 66 However, studies revealed that they could be transferred to the atmosphere by 67 sea spray (Allen et al., 2020). MPs are already so ubiquitous that, similar to 68 global biogeochemical cycles, MP now spiral around the globe with different 69 atmospheric, oceanic, cryospheric and terrestrial residence times (Brahney et 70 al., 2021). In terrestrial environments, MPs cause similar damages to those 71 observed in the marine environment. MPs are consumed by terrestrial biota 72 leading to impaired health and mortality, transfer up the trophic web and act as 73 vectors for pollutants (Brahney et al., 2021). Additionally, MPs interfere with soil 74 dynamics and plant production, altering the microbial community's composition. 75 Due to the previous arguments, MPs are considered an emergent pollutant of 76 global concern, and there is an urgent need for technological solutions to solve 77 MP pollution. The ocean now not only acts as a MP's sink but has already become a source; hence, a logical approach to fight MP pollution is to stop their 78

79 inputs into the ocean. A feasible and relatively straightforward approach to 80 accomplish this goal is to effectively eliminate the MP present in wastewater 81 effluents before their discharge into water bodies. This approach is based on 82 the fact that wastewater treatment plants (WWTPs) have been identified as a 83 source of MPs in aquatic and terrestrial environments (Barceló and Picó, 2019; 84 Sharma et al., 2020). Photocatalysis is a water treatment process that can be 85 adapted as tertiary treatment into WWTPs. By photocatalysis, MP can be 86 mineralized to CO₂ and H₂O or degraded into less toxic substances (Jiang et 87 al., 2021; Uheida et al., 2021). Furthermore, if performed in visible or solar light 88 and if the photocatalyst is prepared using renewable feedstocks, the overall process respects the 6th, 7th, and 9th principles of Green Chemistry (Jiménez-89 90 González and Constable, 2011).

91 Even though all types of plastics have been found in the marine environment, 92 polyethylene terephthalate (PET) is prevalent in many ecosystems due to its 93 extended use and particular properties. PET is a thermoplastic polyester 94 composed of glycolate and terephthalate subunits connected by ester bonds 95 (Zhang et al., 2021). Together with polypropylene, they represent 26.7% of 96 worldwide plastic demand (Magalhães et al., 2020), and their applications 97 include bottles for drinking and cleaning products and polyester fibers used in 98 clothing. PET is predominantly found in deep sea and sewage works because 99 of its high density compared to other plastics (1.36-1.37 g/cm³) (Bond et al., 100 2018) and its high resistance to biodegradation (Mueller, 2006). Hence, this 101 research attempts to degrade this pollutant through photochemical methods and 102 analyze the efficacy of the process through various characterization studies.

103 The photocatalytic degradation of PET in an aquatic environment involves two 104 main mechanisms: photo-oxidation and hydrolysis. The hydrolytic degradation 105 of PET strongly depends on the pH and temperature of the surrounding medium 106 (Nguyen-Tri et al., 2014). Even though PET has heteroatoms in the backbone, 107 which are highly susceptible to hydrolysis, the presence of aromatic groups 108 makes PET highly resistant to degradation under normal conditions (Niaounakis 109 et al., 2019). Because of this, PET hydrolysis is relatively slow, especially at a 110 pH of around 7 (Chamas et al., 2020; loakeimidis et al., 2016; Nguyen-Tri and 111 Prud'homme, 2019). Even though the mechanism for hydrolytic degradation is 112 not fully understood (Fotopoulou and Karapanagioti, 2017), it has been reported 113 that in an acidic or alkaline environment, the rate of hydrolysis has a substantial 114 increase (Allen et al., 1994; Edge et al., 1991). In an acidic environment, the 115 protonation of the oxygen atom of the ester group reacts with water to produce 116 hydroxyl and carboxyl end-groups (Allen et al., 1994). In an alkaline 117 environment, the hydroxide anion attacks the carboxyl oxygen atom, producing 118 hydroxyl and carboxyl end-groups (Edge et al., 1991). In this work, the 119 photocatalytic degradation of PET MPs was tested under ambient temperature 120 and two pH, one slightly acidic at pH 6 and the other slightly alkaline at pH 8, to 121 be closer to natural conditions while also promoting the hydrolytic degradation 122 at non-neutral pH values.

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- 124

1. Materials and methods

125 1.1. C,N-TiO₂/SiO₂ synthesis and characterization

126 Two C,N-TiO₂/SiO₂ photocatalysts were synthesized using the extrapallial fluid 127 (EPF) of two different mussel species. According to Zeng et al. (2015), the EPF 128 of mussels contains proteins that can be used as doping sources (C and N) and

129 pore-forming templates. One photocatalyst C,N-TiO₂/SiO₂, was synthesized 130 using Mytilus Edulis mussels fluid, referred to as TS-ME, and the other using 131 Mytilus Galloprovincialis mussels fluid, which will be referred to as TS-MG in 132 this manuscript. The synthesis consisted of two main procedures. The first one 133 is a modification from the procedure proposed by Zeng et al. (2015) (Zeng et 134 al., 2015), who intended to obtain a C,N-TiO₂ semiconductor. The methods for 135 extracting the extrapallial fluid and the synthesis process are already described 136 in detail in our previous work (Ariza-Tarazona et al., 2020). Briefly, 5 mL of 137 titanium butoxide and 25 mL of extrapallial fluid of *Mytilus edulis* or *Mytilus* 138 Galloprovincialis mussels were mixed in a beaker and kept under stirring for 2 139 h. Then, the mixture was left to mineralize for 4 h without stirring. The second 140 procedure intends to couple the C,N-TiO₂ photocatalyst with SiO₂. In another 141 beaker, 2 mL of tetraethyl orthosilicate and 3.2 mL of isopropyl alcohol were 142 mixed and kept under stirring for 15 minutes. After that, 0.8 mL of distilled water 143 and 0.10 mL of concentrated nitric acid were added, and the solution was 144 stirred for another 2 h. Further, the contents of both beakers were mixed and 145 stirred for 20 min. The homogeneous mixture was transferred to an autoclave 146 and subjected to a thermal treatment as described in our previous work (Ariza-147 Tarazona et al., 2020).

The crystalline structure of the photocatalysts was investigated by X-ray
diffraction (XRD), using a Siemens D5000 diffractometer with Cu Kα radiation
(accelerating voltage of 40 kV, current of 40 mA, and 2θ-step of 0.023°). The
scan was collected between 5 and 80 degrees. The functional groups at the
photocatalysts' surface were detected with attenuated total Reflectance-Fourier
transform infrared (ATR-FTIR) spectroscopy. The ATR-FTIR spectra were

154 collected using a Vertex 70 ATR-FTIR Bruker spectrometer by averaging 32 scans between 4000 and 400 cm⁻¹ with a 4 cm⁻¹ spectral resolution. The surface 155 156 area was estimated by nitrogen adsorption using a Micromeritics TriStar II Plus 157 3.01 surface area analyzer (analysis bath temperature of 77.3 K, equilibration 158 interval of 5 s, and degasification at 110 °C). The photocatalysts' band gap (E_q) 159 values were calculated from their Diffuse reflectance spectra (DRS). Measures 160 were carried out in the 300-800 nm range on a Jasco V-670 UV-Vis/NIR 161 spectrophotometer equipped with an ILN-725 integration sphere. X-ray 162 photoelectron spectroscopy (XPS) was used to investigate the surface 163 chemistry of the photocatalysts. XPS analyses were carried out using a Thermo 164 Scientific Escalab 250Xi, maintaining a base analysis pressure of ~10-10 mbar; 165 the excitation of the analyzed photoelectrons was performed with a 166 monochromatic X-ray source of Al K α (1486.68 eV) with a analysis area radius 167 of 650µm. The analysis conditions for the high-resolution zones were step 168 energy of 20 eV and 45° take-off angle with a step of 0.1eV. Microstructural 169 characterization was carried out by field emission gun scanning electron 170 microscopy (FEG-SEM) using an FEI Nova Nano SEM 450 instrument. 171 1.2. PET MPs preparation and characterization 172 Secondary PET MPs were obtained by grinding a PET food container with a 173 coffee grinder. These particles were then sifted to obtain particles of size ≤500 174 µm. Their size and morphology were analyzed by optical microscopy (OM) 175 using a Leica EZ4 stereomicroscope. To determine the type of polymer present 176 in the sample, an attenuated total reflectance Fourier transformed infrared 177 spectroscopy (ATR-FTIR) in a Vertex 70 ATR-FTIR Bruker spectrometer was 178 used.

179 **1.3.** Photocatalytic degradation experiments

180 Photocatalytic degradation experiments were performed at room temperature in 181 a closed reaction chamber, where a 50W LED visible light lamp illuminated the 182 sample from the top. The sample consisted of a Batch-type glass container 183 containing 200 mg of PET microplastics and 200 mg of TS-ME or TS-MG in 50 184 mL of buffer solution of pH 6 or 8. The light irradiance of the lamp was 500 185 W/m² throughout the experiments. The solution was kept under continuous 186 stirring at 350 rpm for 120 h of irradiation. After irradiation, the solution was 187 filtered with a Whatman 40 filter, dried at room temperature for 12 h, and then 188 incubated at 50 °C for 2 h in an oven. The heterogeneous mixture of 189 photocatalyst/PET microplastics was weighed, and the weight of the 190 semiconductor was extracted to obtain the remanent weight of the 191 microplastics. The photolysis tests followed the same process without the 192 photocatalysts. The degradation was followed by measuring the % of mass loss 193 of the PET microplastics according to the following equation: % Mass loss = $\frac{\text{Initial mass of PET MPs-Mass of PET MPs at 120 h}}{\text{Initial mass of PET MPs}} x 100 (Eq.1)$ 194

195 ATR-FTIR analysis was used to determine the photocatalytic degradation by 196 identifying functional groups formed after 120 h of visible light irradiation. The 197 carbonyl index (CI) was used to assess the degradation of PET MPs. According 198 to the literature, CI was calculated by dividing the absorption of the carbonyl 199 peak (1710 cm⁻¹) by the absorption of the reference band. Reference bands for 200 PET were read at 1504 cm⁻¹, which corresponds to the aromatic ring C=C bond 201 (Piccardo et al., 2020), and at 871 cm⁻¹, which corresponds to the C-H 202 vibrations of the aromatic ring (Janczak et al., 2020).

The crystallinity of PET MPs was studied by Differential scanning calorimetry
(DSC) to detect alterations in the PET macromolecule because of photocatalytic
degradation. The analysis was carried out in a TA Instruments DSC 2010
Differential Scanning Calorimeter in a nitrogen atmosphere. The temperature
ranged from 20 to 300 °C with a heating rate of 10 °C/min. The glass transition
(
$$T_g$$
), cold crystallization (T_{cc}), and melting temperatures (T_m) were determined.
The T_g was obtained at the inflexion point of the heat capacity. The exothermic
peak minimum and endothermic peak maximum correspond to T_{cc} and T_m ,
respectively. The cold crystallization and melting enthalpies were obtained from
the area under the curve of their corresponding peaks. The degree of
crystallinity (X_c) during the DSC heating stage of the PET MPs was calculated
according to the following equation (Mendiburu-Valor et al., 2022):

215
$$Xc = \frac{\Delta H_m - \Delta H_{cc}}{\Delta H_0} \times 100$$
(Eq.2)

216 Where X_c is the degree of crystallinity (%), ΔH_m is the melting enthalpy (J/g),

 ΔH_{cc} is the cold crystallization enthalpy (J/g), and ΔH_0 is the melting enthalpy for

a 100% crystalline PET (140 J/g) (Loaeza et al., 2021).

The thermal behavior of PET MPs before and after photocatalytic degradation
was investigated by thermogravimetric analysis (TGA). TGA measurements
were performed in an STA 429 CD Simultaneous thermal analyzer. The
measurement temperature range was 50–600 °C, the heating rate was 10
°C/min, and the mass of the samples was around 15 mg. The tests were
performed in a helium atmosphere at 50 ml/min.

2. Results and discussion

2.1. C,N-TiO₂/SiO₂ Characterization

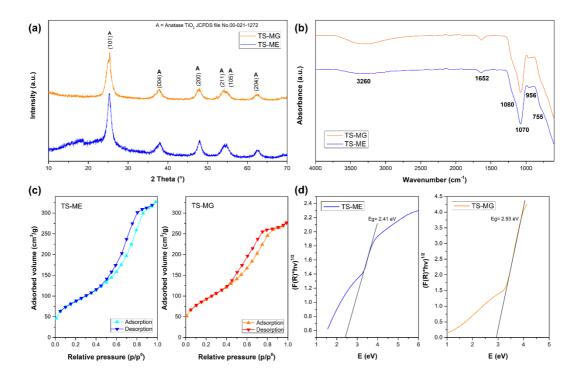


Figure 1. (a) XRD patterns, (b) FTIR-ATR spectra, (c) nitrogen adsorption
isotherms, and (d) Eg calculation for TS-MG and TS-ME samples.

Figure 1(a) shows the XRD patterns for the C,N-TiO₂/SiO₂ powders. All the peaks correspond to the anatase TiO₂ and are in accordance with the JCPDS file No. 00-021-1272. The incorporation of EPF proteins in the synthesis makes the materials amorphous, as reported earlier (Ariza-Tarazona et al., 2020, 2019; Zeng et al., 2015). It can be noted that the crystalline structure of TiO_2 is not affected by the incorporation of SiO₂, as the last is in a lower percentage than the detection limit of the instrument, and SiO₂ possibly exists in an amorphous state (Zhang et al., 2019). The FTIR spectrum of the sample is shown in Figure 1(b). The absorption peaks at 755 and 438 cm⁻¹ were attributed to Ti–O and Ti–

245	O-Ti bonds. The broad band at 3500-3200 cm ⁻¹ was assigned to the stretching
246	vibrations of hydroxyl groups and the bending vibration of the adsorbed H_2O .
247	The absorption peak around 1652 cm^{-1} was referred to the O–H and N–H
248	bonds. The presence of Si in the samples is confirmed by the characteristic
249	peak at 1070 cm ^{-1} , attributed to the antisymmetric stretching vibration of the Si–
250	O–Si bond (Zhang et al., 2019). It is also possible that this band overlaps with
251	the band at 1080 cm ^{-1} , corresponding to the Ti–N bond. FTIR spectroscopy
252	also detected the symmetric stretching vibration of the Ti-O-Si bond at 956
253	cm^{-1} . It has been reported that the existence of this bond can improve the
254	photocatalytic activity of the semiconductor since it may lead to the increase of
255	surface defects, resulting in a higher capture of photogenerated carriers and an
256	enhanced activity of the hydroxyl radical (Guo et al., 2014).

Table 1. Properties of the C,N-TiO₂/SiO₂ photocatalysts using the extrapallial
fluid of different species of mussels as a doping source.

Sample	E_g (Ev)	Activation wavelength (nm)	S _{BET} (m²/g)
TS-ME	2.41	514	313
TS-MG	2.93	423	332

259

The properties of the C,N-TiO₂/SiO₂ photocatalysts are presented in Table 1. The optical properties of the photocatalyst were investigated by UV-Vis diffuse reflectance spectroscopy. The absorbance spectrum is shown in Figure 1S (Supplementary material). Using the Kubelka-Munk theory, the E_g values were obtained from the intersection of a straight line from the linear region with the abscissa axis of a plot the $[F(R)hv]^{1/2}$ vs. *E*. The bandgap energy was found to be 2.41 eV and 2.93 for TS-ME and TS-MG, respectively. This result suggests
that C,N-TiO₂/SiO₂ could have an enhanced photocatalytic activity under visible
light.

269 The N₂ adsorption-desorption isotherm is presented in Figure 1(d). The samples 270 showed a type IV isotherm according to IUPAC, characteristic of mesoporous 271 materials. The hysteresis loop was attributed to H1, indicating that the 272 semiconductor is composed of agglomerates of spherical particles. These 273 textural properties of the material explain the high obtained BET surface area of 274 313 m²/g for TS-ME and 332 mg²/g for TS-MG. Zeng et al. (Zeng et al., 2015) 275 synthesized different N-TiO₂ photocatalysts using different concentrations of 276 EPF-derived proteins. Among them, the one with the best physical properties 277 had a 1200 µg/mL concentration. This material had a surface area value three 278 times that of Degussa P25, indicating the positive effect of implementing the 279 EPF from mussels. As they stated, the proteins in this fluid form a 3D network 280 with sites or channels where TiO₂ nuclei can be generated after being mixed 281 with the TiO₂ precursor (titanium tetrabutoxide) and then kept at rest for 4 h. 282 After being subjected to thermal treatment, the proteins were removed, leaving 283 a porous structure of N-TiO₂ with a high surface area. For the C,N-TiO₂ material 284 that was synthesized in our previous work using Mytilus Edulis mussels (Ariza-285 Tarazona et al., 2020), the superficial area value is four times than the one of 286 Degussa P25, while for the C,N-TiO₂/SiO₂ synthesized in this work is up to five 287 times the superficial area of Degussa P25. The differences in the increase in 288 surface area between the semiconductors synthesized using *Mytilus Edulis* (TS-289 ME) and *Mytilus Galloprovincialis* (TS-MG) mussels and the semiconductor

synthesized by Zeng et al. (Zeng et al., 2015) can be attributed to the differentproperties of the species of mussels used.

292 The photocatalyst surface was also analyzed using X-ray photoelectron 293 spectroscopy (XPS). The XPS survey in Figures 2(a) and 2(e) confirms the 294 presence of O, Ti, Si, N, and C elements. The O 1s spectrum exhibits two 295 peaks for TS-ME (Figure 2(b) and one peak for TS-MG (Figure 2(h)). The peak 296 at 532.96 eV may be attributed to Ti-O-Si species (Aman et al., 2016) that had 297 been already identified according to the FT-IR spectrum of the sample, while 298 the peaks at around 530 eV are ascribed to the O in the Ti-O-Ti lattice. The 299 spectrum for Ti 2p presents two peaks at around 458 and 464 eV for both 300 samples, corresponding to Ti $2p_{3/2}$ and Ti $2p_{1/2}$, respectively (Zeng et al., 2015). 301 N 1s XPS spectra of TS-ME (Figure 2b) show a broad peak at 400 eV, 302 suggesting an interstitial nitrogen dopant (Ang et al., 2009), while for TS-MG 303 (Figure 2(f)), an N 1s peak is detected at 396 eV, suggesting that the nitrogen is 304 found in a substitutional position (Zeng et al., 2015). In the spectrum of C 305 element, for TS-ME (Figure 2(c)), two peaks were identified at 288.75 and 306 284.88 eV, which are attributed to sp2 hybridization in CNOs or C of Ti–O–C 307 bonds and the C=O in CNOs, respectively (Zhang et al., 2019), while TS-MG 308 (Figure 2(g)) presents a broad peak from 288.80 eV to 285.26 eV, suggesting 309 the two peaks mentioned above are overlapped. In the XPS spectrum of Si 2p 310 for both samples (Figure 2(d) and 2(h)), one peak was observed at around 103 311 eV, which may be attributed to Si-O-Si (Zhang et al., 2019).

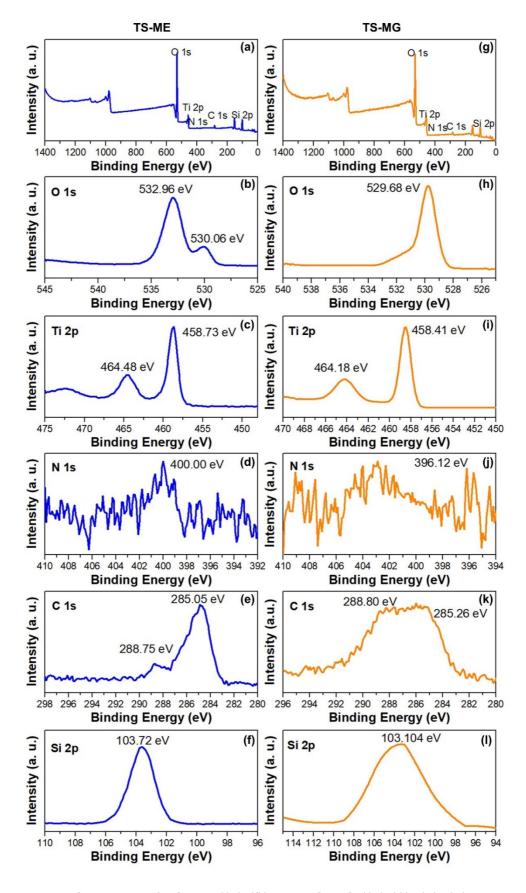
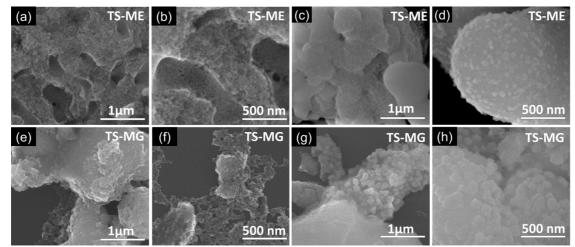




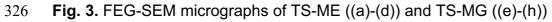
Figure 2. XPS spectra of TS-ME ((a)-(f)) and TS-MG ((g)-(l)): (a), (g) the survey
XPS spectra; (b), (h) O 1s; (d), (j) Ti 2p; (e), (k) C 1s; (f), (l) Si 2p.

315

316 FEG-SEM micrographs for both photocatalysts are presented in Figure 3. It can 317 be noted that the photocatalysts are composed of two phases. The first is a 3D 318 network (Figure 3(a)-(b) and Figure 3(e)-(f)) of TiO_2 , in which the proteins 319 served as templates to generate porosity. The second one is formed by 320 agglomerates of spherical particles that also contribute to increasing the 321 porosity of the C,N-TiO₂/SiO₂ photocatalyst. From Figure 3, it can be inferred 322 that the 3D network is composed of macropores of diameters between 0.5 - 1.6323 µm. In addition, the materials also exhibit agglomerates of 1-3 µm constituted of 324 spherical particles with an average radius of 20-50 nm.



325



327 photocatalysts.

328

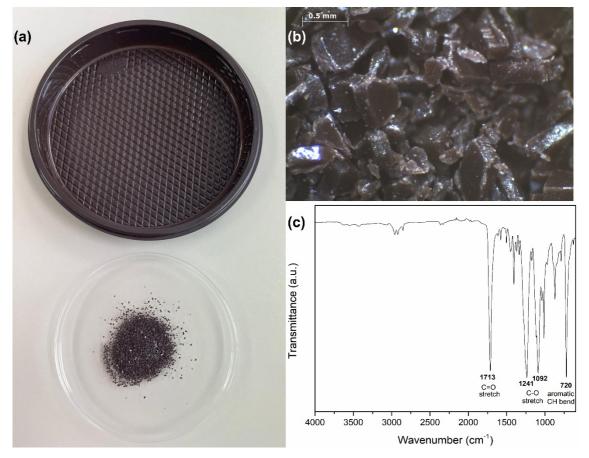
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3.3. MPs characterization

330 The original food container used to obtain the MPs and the characterization of

- the samples of microplastics obtained after grinding are presented in Figure 4.
- 332 The optical microscopy in Figure 4(b) shows that the sample comprises
- 333 approximately 200-500 µm sharp microplastic fragments. Figure 4(c) presents
- the FTIR spectrum of the obtained fragments. The sample presents all main

335 characteristic bands of PET according to the literature (Piccardo et al., 2020). 336 The band located at 2953 cm⁻¹ is assigned to the C-H asymmetric stretching 337 vibrations, while the following bands at 2917 and 2846 cm⁻¹ are attributed to the 338 asymmetrical and symmetrical vibrations of CH₂, respectively (Aljourna and 339 Abboudi, 2016). The absorption band at 1713 cm⁻¹ corresponds to the C=O 340 stretch vibration of the carboxylic ester group, the bands around 1238 to 1092 341 cm⁻¹ are attributed to C-O stretch vibrations, while the bands at 872 and 720 342 cm⁻¹ correspond to the aromatic CH out-of-plane bend (Mendiburu-Valor et al., 343 2022)



345 **Figure 4.** (a) Original PET container, (b) Optical microscopy, and (c) FTIR

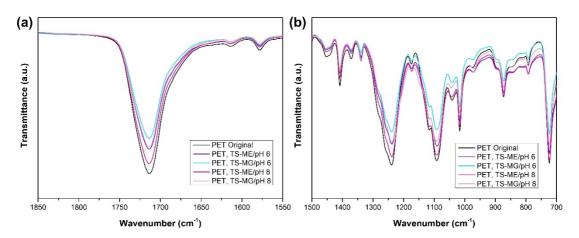
- 346 spectrum of PET MPs.
- 347
- 348 **3.4.** Photocatalytic degradation tests

349 The mass loss of PET samples after photocatalysis and photolysis tests with 350 the calculation of the carbonyl index are presented in Table 2. The mass loss 351 obtained after 120 h of visible light irradiation indicates that the C,N-TiO₂/SiO₂ 352 photocatalysts can degrade PET microplastics at room temperature and pH 6 or 353 8, achieving mass loss values ranging from 9.35 to 16.22%. Meanwhile, it was 354 proven that the experiments' conditions do not influence the % of mass loss 355 according to the photolysis experiments. Zhou et al. (Zhou et al., 2022b) 356 obtained a 10.23% degradation of PET fiber-based MPs under AM 1.5 357 irradiation for 48 h. However, a slightly more alkaline pH of 9 was necessary. In 358 another work, Zhou et al. (Zhou et al., 2022a) reported an approximate 29% 359 mass loss of PET fiber-based MPs under AM 1.5 irradiation for 48 h. 360 Nonetheless, a hydrothermal pretreatment at 180 °C for 12 h was necessary to 361 induce the initial rough appearance and molecular weight reduction. 362 The analysis of the bands in the FTIR-ATR spectrum is generally used to study 363 PET degradation as it allows monitoring of chemical changes even at the early 364 stages of degradation (Ferreira et al., 2021). Figure 5(a) shows the carbonyl 365 band, while Figure 5(b) shows the spectra from 1500 cm⁻¹ to 700 cm⁻¹ of PET 366 MPs before and after photocatalytic degradation. The decrease of the band at 367 1714 cm⁻¹ shown in Figure 5a demonstrated the chain scission of the ester 368 group because of photocatalytic degradation. Figure 5(b) presents the ATR-369 FTIR spectra of original and degraded PET samples from 1500 to 700 cm⁻¹. The bands at 1453 and 1407 cm⁻¹ are associated with the C-H deformation and the 370 371 vibration of the phenylene ring, respectively (Aljoumaa and Abboudi, 2016). The 372 bands at 1370 and 1340 cm⁻¹ refer to C-H crystalline gauche and trans conformations, respectively. Meanwhile, the band at 1238 cm⁻¹ is designated for 373

374 asymmetric C-C-O stretching (Mendiburu-Valor et al., 2022). A band at 1220 375 cm⁻¹ developed after irradiation and is associated with the carboxylic acid dimer, 376 as also reported by Horne et al. (Horne et al., 2020). Meanwhile, the bands at 377 1117 and 1092 cm⁻¹ correspond to the *gauche* and *trans* conformation of the 378 ester C-O-C stretching, respectively (Aljournaa and Abboudi, 2016). A decrease 379 in the bands at 1238 and 1092 cm⁻¹ correlated with C-O degradation (Piccardo 380 et al., 2020). Furthermore, the band at 1015 cm⁻¹, which is attributed to the in-381 plane bending of C-H in the benzene ring, is also directly related to PET 382 crystallinity, according to literature (Bertoldo et al., 2010; Cole et al., 2002). The bands at 970 and 898 cm⁻¹ correspond to the C-H bond stretching of the *trans* 383 384 and cis isomers, respectively (Horne et al., 2020). The bands around 872 and 725 cm⁻¹ are attributed to the aromatic C-H out-of-plane bending and wagging 385 386 (Aljoumaa and Abboudi, 2016). However, there is no development of new 387 bands in the stretching vibration of the hydroxyl group or in the region 388 associated with the O-H stretching vibration of ethylene end groups (3615-3115 389 cm⁻¹), as presented in Figure 2S. The development of these bands is 390 associated with the formation of carboxylic and alcoholic end groups due to the 391 photochemical and hydrolytic degradation of the polymer (loakeimidis et al., 392 2016; Mendiburu-Valor et al., 2022; Miranda et al., 2021). Also, at later stages 393 of degradation, additional bands start to form in the region from 1705 to 1685 394 cm⁻¹ assigned to the formation of a carboxylic acid dimer, end groups, 395 terephthalic acid monomer, and quinone species (Horne et al., 2020). This 396 indicates that after 120 h of visible light irradiation, the PET samples are in the 397 early stages of degradation.

398 The carbonyl index, an indicator of the degree of MPs' degradation, tends to reduce as the carbonyl band at 1714 cm⁻¹ decreases after 120 h of 399 photocatalytic degradation when 1504 cm⁻¹ and 871 cm⁻¹ are set as reference 400 401 bands. When using 1504 cm⁻¹ as a reference band, it can be noted that the CI 402 values of all samples are up to 10% lower than the PET original except for PET, 403 TS-MG/pH. This could be attributed to this method's limitations in detecting the 404 degradation of polymers at the early stages of degradation (Rouillon et al., 405 2016). However, when using 871 cm⁻¹ as a reference band, the CI values of the 406 irradiated samples are 12 to 25% lower than the original PET sample, according

407 to the mass loss results obtained.



409 Figure 5. FTIR-ATR spectra of original PET MPs in the regions between (a)
410 1850-1550 cm⁻¹ (carbonyl region) and (b) 1500 - 700 cm⁻¹.

411

408

Table 2.% of mass loss and CI calculation for PET samples

Sample	Mass loss (%)	Cl(A1714/A1504) (Piccardo et al.,	CI(A1714/A871) (Janczak et al.,	
		2020)	2020)	
PET Original	-	14.12	3.00	
PET, TS-ME/pH 6	13.65 ± 0.90	12.77	2.49	
PET, TS-MG/pH 6	16.22 ± 7.52	13.86	2.24	
PET, TS-ME/ pH 8	9.35 ± 0.24	13.98	2.43	
PET, TS-MG/ pH 8	13.83 ± 0.55	15.05	2.62	
PET, Photolysis/pH 6	0.27 ± 0.31	-	-	
PET, Photolysis/ pH 8	1.39 ± 0.47	-	-	

413

414 DSC analysis was used to investigate the response of PET MPs to heating 415 before and after photocatalytic degradation. In particular, the glass transition 416 temperature (T_g) , the cold crystallization temperature (T_{cc}) , and the melting 417 temperature were obtained for the heating stage. In contrast, the crystallization 418 temperature (T_c) was obtained for the cooling stage. In addition, PET MPs' 419 degree of crystallinity (X_c) was also determined for both stages as it can give 420 information on a polymer's mechanical properties, stability, and chemical 421 resistance (Kong and Hay, 2002). 422 The heating stage of DSC analysis for PET MPs is presented in Figure 6(a). As 423 can be noted, there are differences in the values of the glass transition 424 temperature (T_g) , the cold crystallization temperature (T_{cc}) , and the melting 425 temperature (T_m) (values are displayed in Table 3. According to Figure 6(a), the 426 T_g values of PET MPs after photocatalytic degradation are significantly higher, 427 and their associated peaks become more intense than the T_g values of original 428 PET MPs. This phenomenon has been reported earlier and is attributed to the 429 relaxation process of the amorphous region of PET MPs resulting from the 430 mobility of the molecules segments (Farhoodi et al., 2012). Meanwhile, the cold 431 crystallization peaks, associated with the rearrangement of the amorphous 432 phase into a crystal phase during DSC heating (Fechine et al., 2002), are 433 slightly lower after photocatalytic degradation of PET MPs. The second thermal 434 transition observed corresponds to the melting temperature. The obtained T_m 435 values after photocatalytic degradation are slightly lower than the one of the 436 original PET, which could also be attributed to the relaxation process (Aljoumaa 437 and Abboudi, 2016).

In the cooling scan depicted in Figure 6(b), the original PET MPs showed a pronounced peak associated with the crystallization process of PET. However, the peak of the degraded MPs was considerably less intense and appeared at lower temperatures than the original MPs, as exhibited in Table 3. The decrease in T_c can be associated with branching and crosslinking reactions (Horne et al., 2020).

444 The degree of crystallinity calculated from Eq. 2 is also presented in Table 3. 445 The degree of crystallinity, also known as the crystallinity index, is often used as 446 an indicator of the degradation of PET (Hagihara et al., 2014). The degree of 447 crystallinity obtained for the samples subjected to photocatalytic degradation at 448 pH 6 and 8 under visible irradiation shows a marked decrease compared to 449 original PET MPs. This effect has been reported before in literature (Fechine et 450 al., 2002; Miranda et al., 2021; Nguyen-Tri and Prud'homme, 2019; Niaounakis 451 et al., 2019) but is also in disagreement with other works which have reported 452 the opposite effect (Arhant et al., 2019; Farhoodi et al., 2012). To understand 453 this phenomenon, it is important to identify the degradation process that the 454 polymer is going through. The most relevant degradation mechanisms of PET 455 include photo-oxidation, thermal and hydrolytic degradation (Fotopoulou and 456 Karapanagioti, 2017).

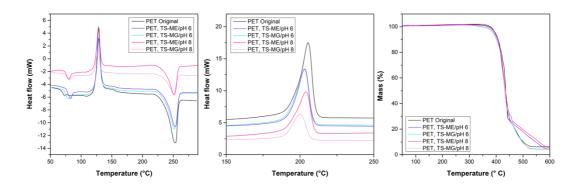
In this work, the main pathways for degradation of PET MPs are hydrolysis, as the experiments are performed in an aquatic environment and visible lightinduced photo-oxidation. During the photo-oxidation of PET, it is expected that a chemi-crystallization process will occur. This process consists of the reorganization of the amorphous phase into a crystalline phase because of the chain scission during degradation, leading to an increase in crystallinity (Arhant

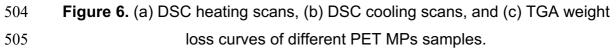
463 et al., 2019; Fechine et al., 2002). Since the rearrangement of the molecules 464 depends on their mobility, it is expected that the chemi-crystallization process 465 occurs only at temperatures higher than the glass transition temperature (T_q) 466 (Fechine et al., 2002). However, it has been proven that the chemi-467 crystallization process can occur even at temperatures below the T_g if the 468 environment contains moisture, as the water can act as a plasticizer, promoting 469 molecular mobility (Fechine et al., 2002). This process is known as hydrolysis. 470 During the hydrolysis process, carboxylic and hydroxyl end groups are formed, 471 leading to the breakdown of water-activated ester bonds, resulting in the 472 cleavage of polymer chains (Mendiburu-Valor et al., 2022). Hydrolysis occurs 473 when the water diffuses into the amorphous regions of the polymer, leaving the 474 crystalline regions untouched because of its impermeability to water 475 (Fotopoulou and Karapanagioti, 2017; Sang et al., 2020) and therefore 476 increasing the crystallinity of PET. Nonetheless, as indicated earlier, a decrease 477 in crystallinity in which photolysis and hydrolysis were the main degradation 478 mechanisms, as investigated in this work, has been reported by several works 479 (Gardette et al., 2014; Miranda et al., 2021; Nguyen-Tri and Prud'homme, 2019; 480 Niaounakis et al., 2019). An explanation could be that PET absorbs light 481 causing chain scission within its crystalline regions, resulting in a decrease of 482 crystallinity, but further research is needed to determine the factors that 483 promote one degradation pathway over the other when hydrolytic degradation 484 and photo-oxidation are combined (Sang et al., 2020). 485 Thermogravimetric analysis was used to study the effect of visible light 486 photocatalytic degradation on the thermal stability of PET MPs at different pH. 487 The thermal curves are presented in Figure 6(c). and the T_{onset} temperatures for

488	each condition are shown in Table 3. The original PET MPs present good
489	thermostability until around 350 °C when thermal degradation occurs. However,
490	PET MPs that underwent a photocatalytic degradation process showed a
491	decrease in the degradation temperature and T_{onset} . This is attributed to the
492	chain scission resulting from photocatalytic degradation, causing a decline in
493	the molar mass and embrittlement of the polymer chain, reducing the thermal
494	stability of the PET MPs (Lee et al., 2012; Pires et al., 2015). All the samples
495	exhibited a two-step thermal degradation. The primary step occurs between 300
496	- 450 °C and corresponds to the ester linkage decomposition due to chain
497	scission, generating vinyl- and carboxy-terminated chain fragments (Achagri et
498	al., 2020). The secondary step starts at 600 $^\circ$ C and corresponds to C-C bonds
499	cleavage, forming volatile compounds (Achagri et al., 2020).
-	

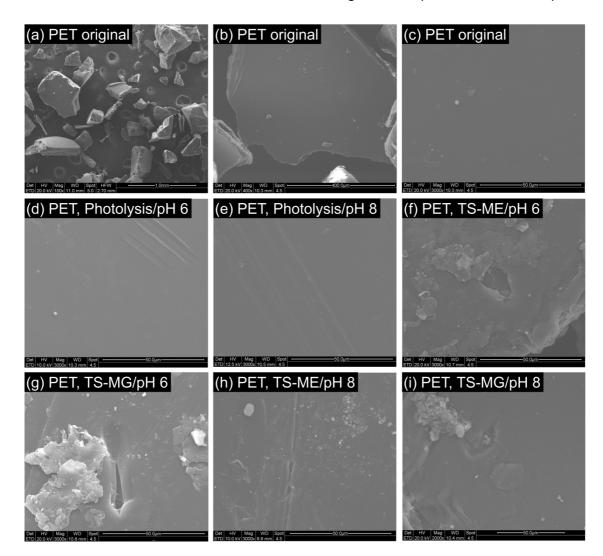
- **Table 3.** Thermal properties of PET MPs before and after photocatalytic
- 501 degradation

Comple	Heating			Cooling		Tonset	
Sample	<i>T_g</i> (°C)	<i>T_{cc}</i> (°C)	<i>T_m</i> (°C)	X _c (%)	<i>T</i> _c (°C)	X _c (%)	(°C)
PET Original	73.0	129.0	254.0	9.1	206.0	28.5	418.0
PET, TS-ME/pH 6	83.0	129.1	253.1	5.3	203.1	21.3	407.6
PET, TS-MG/pH 6	79.0	128.8	251.6	5.7	203.2	22.8	403.4
PET, TS-ME/pH 8	79.8	128.3	251.1	4.9	204.0	17.3	406.4
PET, TS-MG/pH 8	82.0	127.8	250.4	2.8	200.1	10.8	405.1





506 SEM micrographs of the PET MPs were taken to investigate the changes in 507 morphology due to photocatalytic degradation. For this study, PET MPs were 508 obtained from the grinding of a PET food container, leaving cuts on the surface 509 of the samples because of the use of blades in the fragmentation process, 510 which would limit the analysis of the samples using SEM. Therefore, we analyzed the samples from the surfaces that were not cut by the blades and 511 512 remained intact. As can be noted from Figure 7, the original PET MPs exhibit a 513 uniform flat surface, which was not significantly altered after the photolysis. 514 However, after the PET microplastics were exposed to a photocatalytic 515 treatment using TS-ME and TS-MG photocatalysts and pH 6 and 8, holes and 516 cracks were formed on their surface due to degradation (Horne et al., 2020).



- Figure 7. SEM micrographs of PET MPs (a)-(c) before photocatalytic
 degradation; (d)-(e) after photolysis; (f)-(i) after 120 h of irradiation with different
 photocatalysts and pH.
- 521

522 **4.** Conclusions

523 This work conducted a photocatalytic degradation using novel protein-based 524 C,N-TiO₂/SiO₂ photocatalysts. It was demonstrated that the photocatalysts have 525 optimal photon absorption and surface area properties, making them suitable to 526 work in a wide range of the electromagnetic spectrum and degrade various 527 contaminants, including PET MPs. Analysis of the conducted experiments' 528 mass loss and carbonyl index revealed the possibility of degrading fragments of 529 PET size \leq 500 µm in ambient temperature conditions and pH 6 and 8. 530 According to the obtained results, there is no significant difference in the 531 photocatalytic activity of C,N-TiO₂/SiO₂ photocatalysts if they are synthesized using Mytilus edulis or Mytilus galloprovincialis extrapallial fluid as a source of 532 533 carbon and nitrogen doping. This is an outstanding achievement as this 534 synthesis can be performed in different parts of the world where these species 535 are found, and the results would be similar. Also, results indicate that the main 536 degradation mechanisms of PET MPs were photo-oxidation and hydrolysis. As 537 PET MPs were predominantly amorphous, the degradation was improved as 538 the hydrolysis occurred in the amorphous region of the polymer, which was also 539 boosted by the slightly acidic and alkaline conditions. Nonetheless, further 540 research is needed to thoroughly understand the mechanism of hydrolytic 541 degradation and the changes in crystallinity when the main mechanisms are 542 photo-oxidation and hydrolysis.

543 **CRediT authorship contribution statement**

544 Maria Camila Ariza-Tarazona: Methodology, Investigation, Formal Analysis, 545 Writing - Original Draft, Writing - Review & Editing Cristina Siligardi: Writing -546 Review & Editing, Supervision, Resources Hugo Alejandro Carreón-López: Investigation José Enrique Valdéz-Cerda: Supervision, Paolo Pozzi: 547 548 Resources Garima Kaushik: Resources, Writing - Review & Editing Juan 549 Francisco Villarreal-Chiu: Methodology, Review & Editing, Supervision Erika 550 Iveth Cedillo-González: Conceptualization, Methodology, Investigation, Writing - Original Draft, Writing - Review & Editing, Supervision, Project administration, 551 552 Funding acquisition.

553

554 **Declaration of Competing Interest**

- 555 The authors declare no competing financial interest.
- 556 Data Availability
- 557 The data that support the findings of this study are available from the
- 558 corresponding authors (MCAT, EICG), upon reasonable request.
- 559

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- 562

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775 776	Supplementary Information
777	Low Environmental Impact Remediation of Microplastics: Visible-Light
778	Photocatalytic Degradation of PET Microplastics Using Bio-Inspired C,N-
779	TiO ₂ /SiO ₂ Photocatalysts
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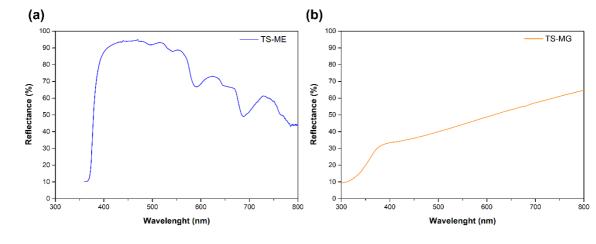
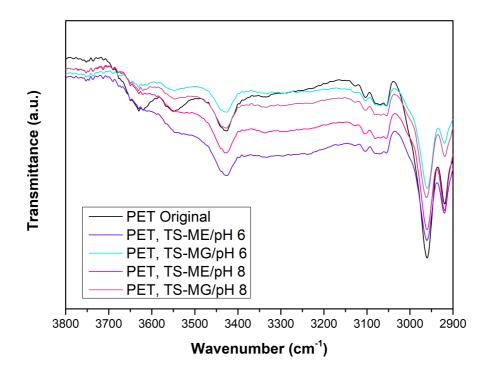




Fig 1S. Absorbance spectra of C,N-TiO₂/SiO₂ samples.



802	Fig 2S. ATR-FTIR of PET MPs in the range from 3800 to 2900 cm ⁻¹ .
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